

**AN EXPERIMENTAL AND THEORETICAL STUDY OF THE EFFECT OF
TEMPERATURE ON THE MECHANICAL BEHAVIOR OF NANOCLAY
REINFORCED POLYMERS**

by

Selen Bayar

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01/26/2012

Date

Prof. Feridun Delale

Chair of Examining Committee

01/26/2012

Date

Prof. Mumtaz Kassir

Executive Offer

Prof. Benjamin Liaw

Prof. Jackie Jie Li

Prof. Ali Sadegh

Dr. Jerry Chung

Supervisory Committee

THE CITY UNIVERSITY OF NEW YORK

Abstract

AN EXPERIMENTAL AND THEORETICAL STUDY OF THE EFFECT OF TEMPERATURE ON THE MECHANICAL BEHAVIOR OF NANOCCLAY REINFORCED POLYMERS

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Adviser: Professor Feridun Delale

The goals of this study are to investigate the tensile loading and low velocity impact response of nanoclay reinforced polymers at various temperatures. Three types of polypropylene (PP 3371, Borealis and TP 3868) and epoxy with various nanoclay reinforcement percentages were considered. Tensile tests were conducted on ASTM Type I specimens instrumented with strain gauges using an MTS testing machine equipped with an environmental chamber. Low velocity impact tests were also performed using an Instron-Dynatup 8250 impact test machine equipped with an environmental chamber. Tensile test results were used to determine the effect of nanoclay reinforcement and different resins on the mechanical properties at various temperatures.

The tensile tests results indicate that the Young's modulus of the nanocomposite increases with increasing nanoclay reinforcement percentage. The temperature has even a more significant

effect. It was observed that as the temperature decreases the material becomes brittle, has higher stiffness and fails at lower strains. High temperatures have the opposite effect, in that, as the temperature increases the material loses stiffness and becomes more ductile. Temperature and nanoclay reinforcement affect the Poisson's ratio also, but this effect is less significant. In general, as the temperature increases the Poisson's ratio also increases. However, an increase in nanoclay reinforcement generally reduces the Poisson's ratio.

The mechanical properties of polymer/clay nanocomposites were also calculated using the Mori-Tanaka formulation and the finite element method. Furthermore, the Mori-Tanaka model was modified to include the effect of temperature and voids. In the Mori-Tanaka formulation three types of nanoclay particle distribution was assumed: oriented nanoclay particles parallel to the direction of tensile loading, 2-D randomly distributed particles and 3-D randomly distributed particles. The finite element calculations were performed on a representative volume element having the same reinforcement percentage as the test specimens, with the nanoclay particles placed in a plane parallel to the loading direction.

The comparison of theoretical and experimental results shows that both the Mori-Tanaka formulation and the finite element method provide effective tools to predict the mechanical properties of nanoclay reinforced composites. Furthermore, including the effect of temperature and voids provided a better match with experimental results.

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1. INTRODUCTION

Since the 1960's composites have been studied extensively due to their high strength to weight and stiffness to weight ratios and tailorability. Conventional composites consist of a matrix (in many cases a polymeric resin) reinforced with fibers or fabrics. Composites have enjoyed a spectacular success as evidenced by their widespread use in military and commercial aircraft, the automotive industry, sporting goods and even health care products. However, the new trend in structural material design is to make the material multifunctional, i.e. to impart to it thermal, electrical, optical properties in addition to the conventional properties of strength, stiffness and impact and fatigue resistance. One way of making the material multifunctional is with nanoadditives. With the recent advances in nanotechnology resin systems have been developed with carbon nanotubes (CNTs), nanoclay particles or flakes, carbon black, etc. to achieve new properties for multifunctionality. In this research, we propose to study the properties of nanoclay reinforced polymers through an experimental and analytical/numerical program.

The polymer/nanoclay composites will not be strong enough by themselves to be used as structure materials. Thus, they need to be reinforced with fibers or fabrics. It is believed that the reinforcement of polymers with nanoclay may improve the strength, stiffness and high temperature resistance of the polymer which in turn may help to reduce its flammability and also toxicity. Thus the main application of using nanoclay additives in polymeric composite systems is to improve their fire retardancy properties, and also reduce toxicity which may ensure as a consequence of fire.

Research in reinforced polymeric nanocomposites has been increasing at a fast pace due to their potential for improved mechanical and chemical properties compared to conventionally filled

composites [1-5]. These improvements may include increased strength [6], higher modulus [7-12], thermal stability [13-15] and decreased flammability [16-20]. After reinforced with fibers or fabrics a structural material able to resist static and dynamic loads can be obtained.

The polymer/nanoclay composite specimens used in this research were manufactured by injection molding Novus Technologies Corporation as ASTM standard Type I dog-bone shaped specimens with varying weight fractions of nanoclay (0%, 0.2%, 1%, 3%, 6% and 10%). Three different polypropylenes (PP) and one type of epoxy namely, EPON 828 were also used as resin. For most specimens PP 3371 was used as the resin. However, to study the effect of different types of PP, a limited number of specimens were manufactured with Borealis and Total Petrochemical 3868 polymers. Also 4"x4"x(1/8) " nanoclay reinforced PP specimens were subjected to low velocity drop weight tests.

Since the main objective of this thesis is to study the effect of temperature on nanoclay reinforced polymers, the specimens were subjected to tensile and low velocity impact loads at temperatures varying from -65°F to 160°F. To predict the experimentally determined nanocomposite properties a numerical/analytical model was developed.

1.1 OBJECTIVES OF THIS STUDY

The main objective of this thesis is to study the effect of temperature on nanoclay reinforced polymers. The research plan calls for subjecting the specimens to tensile and low velocity impact loads at temperatures varying from -65°F to 160°F. Also specimens with various nanoclay reinforcement percentage are used to ascertain the effect of nanoclay reinforcement on the properties of the nanocomposite. By using different polypropylene resins and epoxy, the effect of different resins is also studied. To predict the experimentally determined nanocomposite

properties, micromechanics based models supplemented with 3-D finite element calculations are developed.

1.2 CONTRIBUTIONS OF THIS STUDY

The most important contributions of this thesis are: a) determining the effect of nanoclay reinforcement on the mechanical properties of various resins b) determining the effect of temperature on the mechanical properties of the nanocomposites c) using different resins to ascertain the effect of resin on the nanocomposites and d) developing analytical/numerical methods which include the effect of temperature to predict the observed properties of the nanocomposite without conducting expensive and time consuming experiments.

2. LITERATURE REVIEW

The literature on composite material systems under mechanical and hygrothermal loadings is extensive. Accordingly, we will restrict our literature review to nanoclay reinforced polymers and related topics.

Since the 1980's, research in polymer based nanocomposites has increased dramatically with the production of nylon-clay nanocomposites by Toyota [1,21] and their actual or potential applications in diverse areas, such as aerospace components, automotive structures, military equipment, etc.

As stated in [22] “two main types of polymer-clay morphology can be obtained depending on the interaction between the clay and the polymer matrix: namely, intercalated and exfoliated. Usually exfoliated nanocomposites are preferred because of showing the best property improvements [23]”. The improvement of the material properties in nylon/clay nanocomposites was demonstrated by the Toyota research group [24]. Also many researchers investigated numerous other polymers. Some of the polymers include polypropylene [25-70], polyethylene[71-80], polystyrene [81-87], poly(ethylene oxide) [88-92], polycaprolactone [93,94], polyimides [95-112], polycarbonate [113,114], and epoxy resin [115-119].

The usage of nanoscale fillers in polymers has provided an opportunity to design new materials with significantly improved performance and multifunctionality.

Ongoing research and studies have shown that, in general, polymer based nanocomposites exhibit great improvement in mechanical properties such as strength and stiffness compared to pure polymers with the addition of minimal amount of nanosize clay particles [120-128].

The nanoclay related studies found in the literature are grouped around a common characteristic. The review is given below.

2.1. STUDIES ON THE CHARACTERIZATION OF POLYMER/CLAY NANOCOMPOSITES

Recent research shows that processing variables have a significant effect on the mechanical properties of polymer/ clay nanocomposites (PCNs). There are several studies on the effects of processing variables and characterization of nanocomposites.

The basic physical and chemical properties of (PCNs) are crucial in engineering studies. Characterization tools also are extremely important in obtaining these properties. Several techniques have been used to characterize the nanocomposites [129]. Wide-angle X-ray diffraction (WAXD), small-angle X-ray scattering (SAXS), scanning electron microscopy (SEM) and transmission electron microscopy (TEM) are the commonly used techniques in characterization of nanocomposite [130, 131]. TEM allows a qualitative understanding of the internal structure, spatial distribution of the various phases and views of the defect structure through direct visualization [132].

Nam [133] focused on advanced preparation methods for PCNs by following previous research which dealt with the complete exfoliation of silicate layers in the polymer matrix [134, 135] and succeeded in preparing well-ordered intercalated polypropylene (PP)/clay nanocomposites (PPCNs) with melt extrusion process [136]. Identification of the hierarchical structure was conducted by using wide-angle X-ray diffraction (XRD), small-angle X-ray scattering, transmission electron microscopy (TEM), polarizing optical microscopy and light scattering to discuss the structure-property relationships of PPCNs having various clay particles [136]. The

limitation of the exfoliation toward the individual silicate layers is observed as an effect of the increment of intercalation.

Wu [137] prepared nanoparticles filled polypropylene nanocomposites with pre-grafted nanoparticles by some polymers using irradiation. Transmission electron microscopy (TEM) was used to observe the appearance of the irradiation grafted nanocomposites. Sharma [138] studied polypropylene/surface modified clay nanocomposites and followed a different preparation method. Transmission electron microscopy (TEM), X-ray diffraction and scanning electron microscopy (SEM) were used to understand exfoliation structure and morphology, respectively.

Rohlman [139] studied the preparation of polypropylene (PP) based nanocomposites by melt mixing using modified polypropylene and various organophilic montmorillonites. Scanning electron microscopy (SEM) was used for examination of the morphology and X-ray diffraction (XRD) was used to analyze the structure of the material. It was found that organophilic montmorillonites have strong effect on the morphology of the nanocomposite for given processing and composition conditions.

Epoxy is another polymer used with nanoclay and there are many ongoing studies on epoxy-clay nanocomposites. Wang [140] prepared epoxy-clay nanocomposites using two organoclays which were cured with various chemicals at different temperatures. TEM and X-ray diffraction (XRD) were used to understand the interlayer distance of the clay layers. In this study Wang showed that the relative curing speeds of extralayer and interlayer played a big role in determining clay exfoliation. It was also shown that the curing speed of inter and extralayer intercalated epoxy/clay is the key factor of exfoliation. Jiankun [141] focused on the exfoliation behavior of organoclays in epoxy resin. X-ray diffraction was used to obtain exfoliation and intercalation

properties of organoclay. Also the nature of the clays and the curing agent used to determine the exfoliating ability of the organoclays was also investigated. Xu [142] also studied the intercalation and exfoliation behavior of epoxy resin in the presence of organophilic montmorillonite. It was observed that exfoliation and intercalation properties are not affected by curing condition. Kornmann [143] explored a different processing method to obtain epoxy-clay nanocomposites. The nanocomposites were obtained using different cation-exchange capacities (CEC) and two montmorillonite clays (MMT). X-Ray diffraction (XRD), TEM and SEM were used to investigate the dispersion of clay. The most important conclusion of this study is that the cation-exchange capacity of the clays affects the intercalation between the layers. SEM observations showed that a better dispersion of the clay particles in the nanocomposite compared to conventional nanoclay based composites. Sarathi [144] studied on the exfoliation characteristics in epoxy nanocomposites with wide-angle X-Ray diffraction (WAXD); Saber-Samandari [145] used TEM to investigate the effect of different types of clay with various processing conditions on intercalation/exfoliation of clay/epoxy nanocomposites; and Yasmin [146] determined the exfoliation and intercalation characteristic of epoxy based nanocomposites with different type of clays using TEM. As a result a classification was obtained as to which modified clay and which processing method produced better intercalation/exfoliation.

In some of these studies mechanical properties were discussed in light of better intercalation/exfoliation of nanoclay.

Several other studies have been reported on the property and structure of inorganic particle-polymer combinations which exhibited superior modulus and yield strength in association with high toughness [147-170].

2.2 STUDIES ON THE MECHANICAL PROPERTIES OF POLYMER/CLAY NANOCOMPOSITES

2.2.1 Experimental Studies

There is a large number of studies dealing with the mechanical properties of polymer/clay nanocomposites. Here we present the studies more related to our research.

Wu [137] performed tensile testing on low nanoparticles loaded polymer composites at five different cross-head speeds at room temperature. The effect of nanoparticles on stiffness, strength and toughness was determined. Simultaneous improvement in Young's modulus strength and elongation was observed. Sharma [138] measured mechanical properties such as tensile strength, tensile modulus and elongation at break in addition to studying the effects of organically modified clay on the physical, thermal and morphological properties of nanocomposites. Differential scanning calorimetry (DSC), thermo gravimetric analysis (TGA) and melt flow index (MFI) methods were used to determine the properties of polymer nanocomposites prepared with polypropylene/modified clays. Thermal and mechanical test results showed that the interaction of polymer and clay in nanocomposites have high effect on mechanical and thermal properties because of the favorable interface of nanoclay and matrix. Different manufacturing machinery, clay modifiers and conditions can significantly affect the resulting polymer nanocomposites [22]. In Dolgovskij [171], the effect of different mixer types on the exfoliation of polypropylene(PP) nanocomposites was explicitly demonstrated. In the same vein, the importance of extruder types on the delimitation and dispersion of layered silicate nanocomposites was shown by Dennis et al [172]. Another tried dispersion methods is the usage of supercritical carbon dioxide as a processing aid by Quang T. Nquen [173]. Galgali [174]

worked on the effect of clay orientation on the tensile modulus of polypropylene nanoclay composites and the relationship between clay orientation and shear rate. The test results indicate that for compatibilized nanocomposites the orientation of the clay increased with extrusion shear rate and for uncompatibilized nanocomposites the clay orientation was independent of the shear rate. Bureau [175] and Drozdov [176] performed similar studies on tensile testing of polypropylene-clay nanocomposites. Bureau used a polypropylene matrix with organo-modified clays. The nanocomposites showed significant improvement in Young's modulus and yield stress. Drozdov performed also tensile tests with different strain rates, creep tests with different stresses and relaxation tests at different strains at room temperature. Test results showed the effect of nanoclay reinforcement on mechanical properties and creep resistance. Clay types can have critical effect on the physical properties of the nanocomposites. To understand the effect of different clays, polypropylenes and preparation methods, Dong [177] performed tensile tests with PP/organoclay nanocomposites which were prepared by using three types of organomodified montmorillonite clay, three grades of PP and a twice direct compounding manufacturing method. The results showed that the PP grade plays the most significant role in the overall mechanical properties and clay content is the second most important factor. Santos [178] optimized the mechanical properties of polypropylene-clay nanocomposites with combination of two different modified clays. Tensile test results showed that the combination of two different clays caused an increase in flexural modulus and higher impact strength.

Beside studies on polypropylene/clay nanocomposites, there are also several studies on epoxy/clay nanocomposites. Yasmin [146] studied the mechanical and thermal behavior of epoxy/clay nanocomposites. Epoxy/clay nanocomposites were prepared by shear mixing using different types and concentrations of clay particles and the mechanical and viscoelastic

properties of the nanocomposites were obtained. Furthermore, the thermal expansion coefficient (CTE) was measured using a thermomechanical analyzer. Test results showed that the elastic modulus and storage modulus were improved and the thermal expansion coefficient was reduced by the addition of clay particles. Sarathi [144] studied the thermal, mechanical and electrical properties of epoxy/clay nanocomposites. Nanoclay reinforcement was increased to up to 5% wt to ascertain the effect of clay reinforcement. Short time breakdown voltage test, accelerated aging test, tracking test were used to analyze the electrical insulation characteristics of the nanocomposite. The thermal properties were obtained by thermogravimetric-differential thermal analysis (TG-DTA). The storage modulus of nanocomposites was obtained by dynamic mechanical analysis (DMA). All the test results showed that the breakdown strength, mechanical properties and the tracking time of epoxy nanocomposites were higher than those for the epoxy resin.

Saber-Samandari [145] manufactured clay/epoxy nanocomposites and performed several experiments including different types of clay with various processing methods such as different centrifuge rotor speeds and curing temperatures. The elastic modulus of epoxy-clay nanocomposites increased with the amount of clay up to a maximum of 6%. The tensile strength and energy to failure increased with clay percentage of up to 10%. Slower rotor speed gave the best results for the properties. On the other hand lower curing temperature led to better results while higher curing temperature caused specimen brittleness and deterioration of properties. Zhao [179] focused on the toughening mechanism of epoxy resin mixed with micro/nano particles and studied the effects of particle size on the maximum stress. The critical particle size for the material system was determined to be $0.95\mu\text{m}$. Thus it was deduced that particle size should not be more than $0.95\mu\text{m}$ to reduce stress concentrations in the material.

2.2.2 Theoretical Studies

Compared to experimental studies, the number of publications on polymer/clay nanocomposites is relatively small. Fornes [180] applied the Halpin-Tsai and Mori-Tanaka re-enforcement theories to predict the modulus of nylon based nanocomposites. The modulus obtained using Mori-Tanaka calculation increased with nanoclay reinforcement as predicted. The Halpin-Tsai gave higher values for the modulus but still could be used to predict the value of modulus. Sheng [181] performed a more detailed study on theoretical analysis of nanocomposites. A micromechanical model was developed to account for the morphology of the nanocomposite, particle volume fraction, particle aspect ratio, mechanical properties of matrix, exfoliated clay layer thickness and layer spacing. For each case the Halpin-Tsai and Mori-Tanaka models were applied to calculate the variation of the normalized Young's modulus of the nanocomposite with volume fraction, particle aspect ratio, layer thickness and layer spacing. In addition some calculated results were compared with tensile test data. Comparing the Halpin-Tsai and Mori-Tanaka results it was observed that the Mori-Tanaka calculations give closer results to the tensile test data.

Drozdov [176] studied the viscoelasticity and viscoplasticity of polypropylene/clay nanocomposites. New constitutive equations were developed taking into account the viscoelasticity and viscoplasticity of the nanocomposite and these equations were used to analysis of the time-dependent response in long-term creep tests. The results show the time to failure of nanocomposites exceeds that of polypropylene for all stresses.

Sheng also presented in [181] a FEM model to estimate the Young's modulus of polymer/clay nanocomposites with the representative volume element constructed using TEM results. The

plane strain FEM simulation results for the normalized Young's modulus of the nanocomposite were compared with those obtained from the Mori-Tanaka model. The main reason for comparing the FE results with the Mori-Tanaka model solution is that it gives the most accurate estimates among analytical models [182]. Sheng [181] also compared the experimental data with Halpin-Tsai, Mori-Tanaka and FEM results. The comparison shows that the Mori-Tanaka model and FE simulation are in good agreement with experimental data, while there is more deviation with the Halpin-Tsai calculations.

2.3. STUDIES ON THE LOW VELOCITY IMPACT RESPONSE OF POLYMER/CLAY NANOCOMPOSITES

There are not many studies on the impact response of polymer nanocomposites. Generally impact tests are performed on fiber (nano)composites which include polymer/clay mixture as a matrix and fibers for reinforcement. Because of their brittleness under impact polymer/clay nanocomposites are more prone to damage than conventional fiber or fabric reinforced composites. There are few studies on low velocity impact test of polymer/clay nanocomposites and many on fiber reinforced polymer/clay nanocomposites.

Yuan [183] studied and analyzed the micromechanism of deformation during impact loading of polypropylene-clay nanocomposites. The temperature range was 0°C to +70°C during impact loading and in this range the addition of clay to polypropylene increased the impact strength for low nanoclay percentage. Furthermore, Yuan used differential scanning calorimetry (DSC), dynamic mechanical analysis (DMA), transmission electron microscopy (TEM), wide-angle X-ray diffraction (WAXD) and scanning electron microscopy (SEM) techniques to understand the behavior of polypropylene/clay nanocomposites. The crystal structure and dispersion of nanoclay

particles affected the toughness of nanoclay reinforced PP composites. Lin [184], performed low velocity impact tests on polymer and two different nanoparticle reinforced composites. The impact strength of the nanocomposites was improved somewhat by the filler content. The impact energy and the corresponding initial velocity for the two nanocomposites were determined and showed to have a linear relationship with the impact results. Also it was noted that the impact strength of nanocomposites is extensively affected by the temperature, geometry, type of filler content, volume and dispersion of the nanoparticles. Iqbal [185] investigated the impact damage resistance of carbon fiber nanoclay filled epoxy composites. Up to 3 % wt inclusion of nanoclay in the matrix helped to lower the impact damage size, and improve the damage tolerance. Ye [186] used epoxy-based nanocomposites with natural nanotubes which were manufactured with a process similar to solution intercalation of layered silicate into polymers [187, 188]. The test results showed that blending epoxy with 2.3 wt % nanotubes caused to obtain higher impact strength.

2.4. STUDIES ON THERMAL RESPONSE OF POLYMER/CLAY NANOCOMPOSITES

The main objective of this thesis is to ascertain the temperature effect on polymer/clay nanocomposites. Literature search showed that there are not many studies on the temperature effect.

Most of the temperature studies are related to curing temperature, temperature of resin and hardener during processing, thermal stability of nanocomposites etc. Studies do not include the temperature effect on material properties, except for [146] which deals with epoxy based nanocomposites.

Yasmin [146] studied the temperature effect on the elastic modulus of epoxy based nanocomposites. Epoxy/ clay nanocomposites with 1%-10% clay reinforcement were considered. The elastic modulus of the nanocomposites was measured at various temperatures. The nanocomposites exhibited higher elastic modulus with higher clay reinforcement percentage but lower stiffness at higher temperatures. Yasmin [146] also studied on the thermal expansion coefficients and showed that the thermal expansion coefficient was reduced with incorporation of clay particles.

Other studies focus on the thermal stability of nanocomposites because it is expected that the inclusion of clay flakes increase thermal stability while promoting flame retardancy. Blumstein [189] studied the thermal stability of polymethyl methacrylate (PMMA). He showed that the thermal stability of PMMA with montmorillonite clay was higher than pure PMMA. Guo [190] worked on the thermal behavior of nanocomposites made of polypropylene and organically modified clays. Thermal gravimetric method was used to obtain thermal behavior of the nanocomposites. The results showed that thermal stability increased with the addition of nanoclay into the pure polypropylene matrix. Kashiwagi [191] studied the thermal stability of polyamide 6/clay nanocomposites and samples were tested from 30°C to 800°C. The results showed that the inclusion of clay flakes caused a reduction in the peak heat release rate. Qin [192] also studied the thermal stability of polypropylene/montmorillonite nanocomposites. The results showed that the addition of clay particles increased the decomposition temperature of PP resin. Also the peak heat release rate of combustion reduced with addition of clay particles. Modesti [193] used the thermogravimetric analysis with a heating rate from room temperature to 900°C to predict the thermal stability of polypropylene nanocomposites. It was claimed that the composition of the nanocomposites plays an important role in determining thermal stability. The

results of thermogravimetric analyses showed that the inclusion of clay increased thermal stability.

2.5. STUDIES ON FIRE RETARDATION AND TOXICITY RESPONSE OF POLYMER/CLAY NANOCOMPOSITES

Polymer/clay nanocomposites may be used in structural composites for vehicles. Beyond structural integrity, fire retardation or flame retardancy and toxicity are considered as critical issues. Gilman [194] reviewed the flame retardation properties of nanocomposites. Heat release rate, peak of heat release rate, heat of combustion, etc. are important parameters measured by cone calorimetry in the flame retardant behavior of nanocomposites. The main process to test fire retardation is to expose the sample to a heat flux and record the heat release rate as a function of time. In clay reinforced nanocomposites a significant reduction in the peak of heat release rate is observed. Gilman [195] showed that the effect of the exfoliated and/or intercalated structures on the flame retardancy. Guo [190] studied the flame retardancy behavior of neat polypropylene and polypropylene/clay nanocomposites. Neat polypropylene burned easily, yielded a sharp heat release rate and produced a large amount of smoke, while polypropylene/clay nanocomposites had reduced values of heat release rate. Furthermore, the dispersion of clays in the matrix affect the flame retardancy, such that if the clay flakes are well dispersed in the polypropylene nanocomposite, then high rate of flame retardancy is observed. Kashiwagi [191] performed a study on flammability properties of polyamide 6/clay nanocomposites with 2% and 5% weight fraction of clay. The results showed that, increasing the weight fraction of clay had the effect of increasing the ignition delay time and reducing the peak heat release rate. An important result obtained from this study is that the level of clay in the nanocomposite does not significantly affect the reduction in total heat release. Qin [192] investigated the flammability properties of

polypropylene/montmorillonite (PP/MMT) nanocomposites and observed reduction of flammability of the nanocomposites. To study the flammability properties of polypropylene (PP) nanocomposites, Modesti [193] used the glow wire test and concluded that the dispersion of nanoclays in PP resin with compatibilizer leads to an enhancement of fire resistance.

All the studies show that adding nanoclay to the polymeric resin greatly improves the fire behavior of nanocomposites.

3. EXPERIMENTAL STUDY ON POLYMER/CLAY NANOCOMPOSITES

3.1 THE SPECIMENS AND MATERIALS

Novus Technologies Corporation, manufactured nanoclay reinforced Polypropylene (PP) and epoxy Type I dog-bone shaped specimens for tensile testing, following ASTM D638-08: Standard Test Method for Tensile Properties of Plastics using injection molding. The main resins used were PP 3371 and epoxy with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement by weight. Also additional specimens using two different resins, namely Borealis and Total Petrochemical 3868, were manufactured with 3% nanoclay reinforcement to ascertain the effect of resin on mechanical properties. Figure 3.1 shows the specimens for a) neat PP 3371, b) EPON 828 epoxy with 3% nanoclay reinforcement, c) Borealis and d) Total Petrochemical (TP) 3868 with 3% nanoclay reinforcement. Figure 3.2 shows the specimens which were instrumented with strain gages.

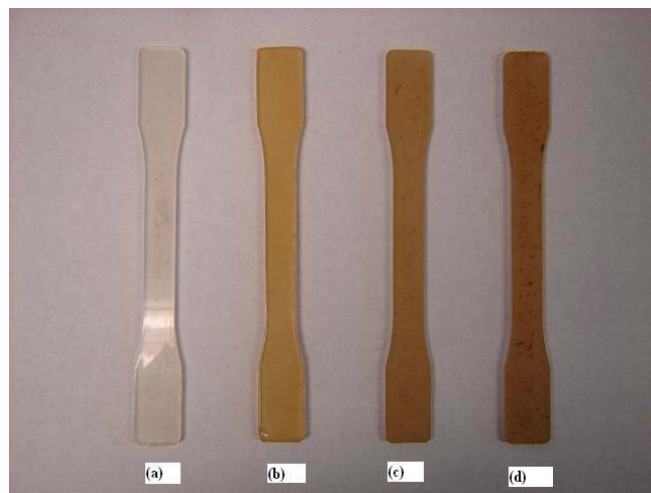


Figure 3.1 Type I dog-bone shaped tensile specimens a) PP 3371, b) EPON 828, c) Borealis and d) TP 3868

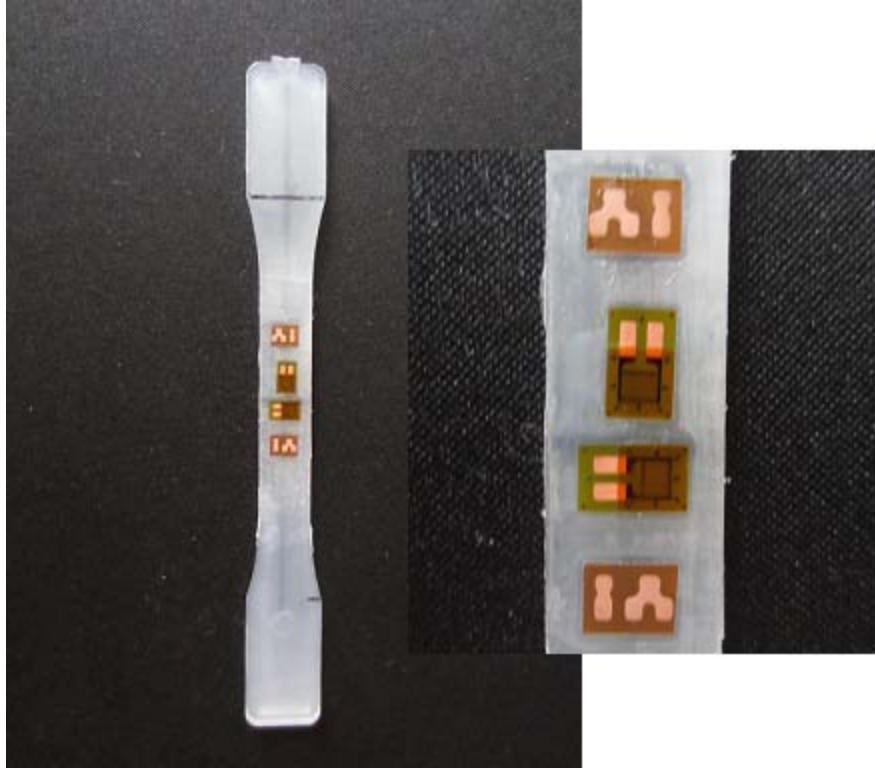


Figure 3.2 PP 3371 specimen with attached strain gages

After tensile testing all the specimens at low, room and high temperatures, low impact drop-weight tests were performed at various temperatures on nanoclay reinforced Polypropylene (PP) 4"x4"x(1/8)" square specimens which were also manufactured by Novus Technologies Corporation. The main resin used was PP 3371 with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement by weight. In Figure 3.3, the low velocity impact test specimen instrumented with strain gages is shown.

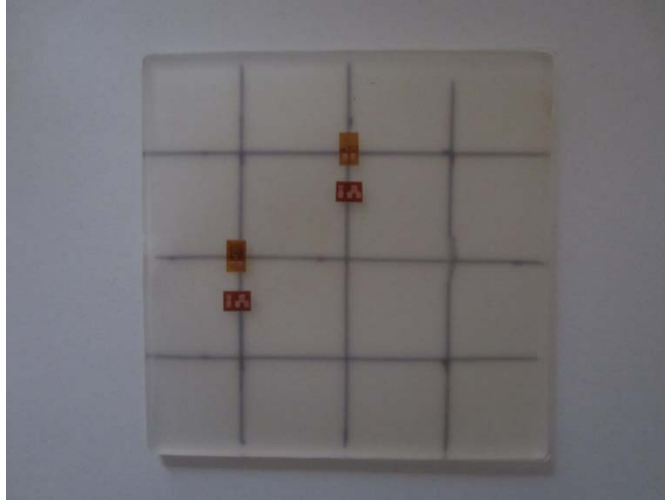


Figure 3.3 Low velocity impact test specimen (PP 3371)

3.2 TENSILE TESTING

3.2.1 Effect of Nanoclay Reinforcement

The PP 3371 specimens were first subjected to room-temperature tensile loading at a crosshead speed of 5 mm/min in an MTS machine. Both loading and elongation histories were recorded through a National Instruments (NI)-DAQ based Lab View data acquisition system. The room-temperature stress-strain curves of PP 3371 specimens with various nanoclay reinforcements were then obtained using time-history readings from load-cell and crosshead motion. Stress - strain curves were plotted for each PP 3371 specimen with different nanoclay reinforcement with the strains calculated from the crosshead displacements. For each case 5 to 10 specimens were used and the stress - strain curves obtained are shown in Figures 3.4 - 3.9. One should note that the specimens did not fail and testing was stopped at a strain level of approximately 95%.

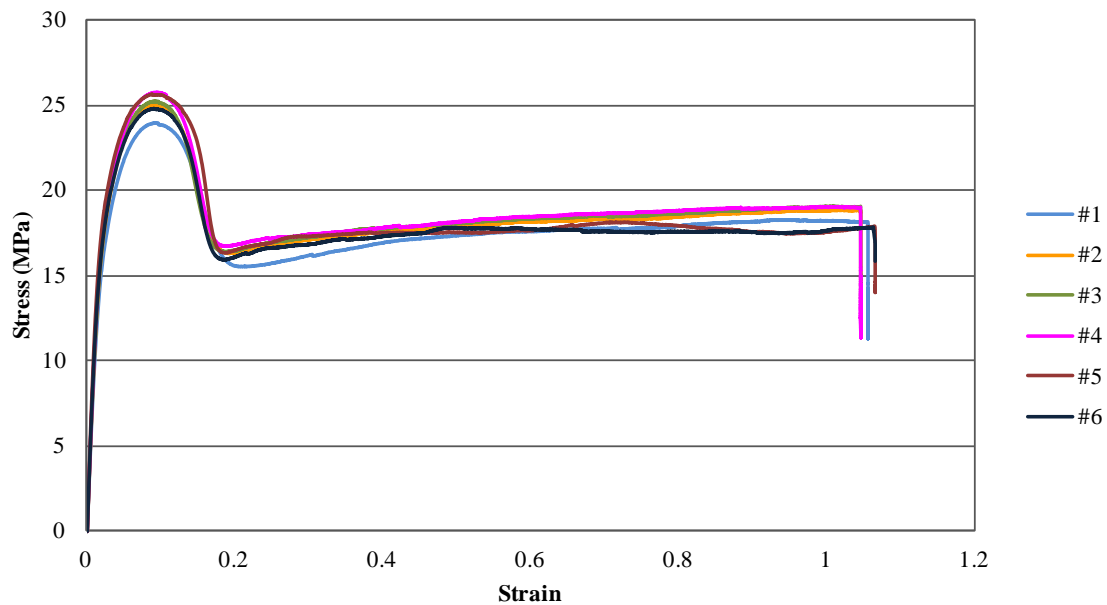


Figure 3.4 Stress-strain curves of neat PP 3371 specimens

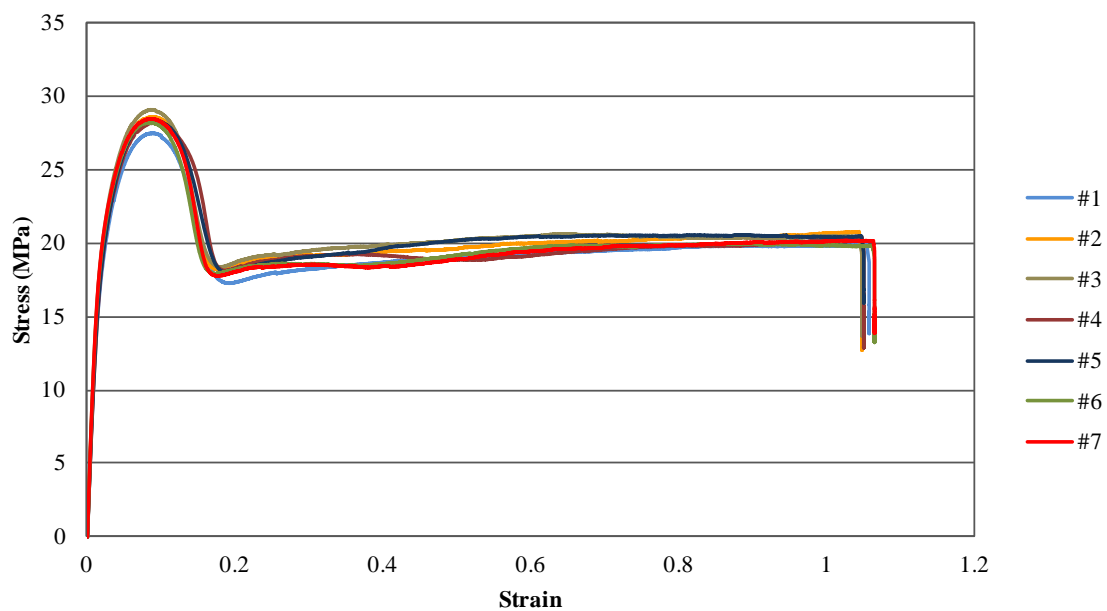


Figure 3.5 Stress - strain curves of PP 3371 specimens with 0.2% nanoclay reinforcement

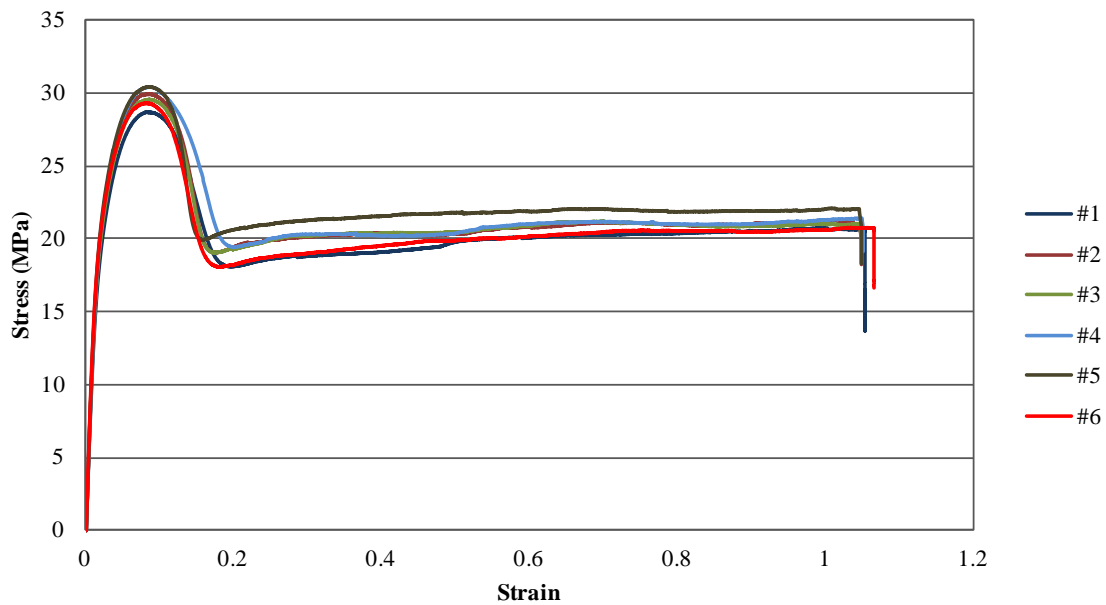


Figure 3.6 Stress - strain curves of PP 3371 specimens with 1% nanoclay reinforcement

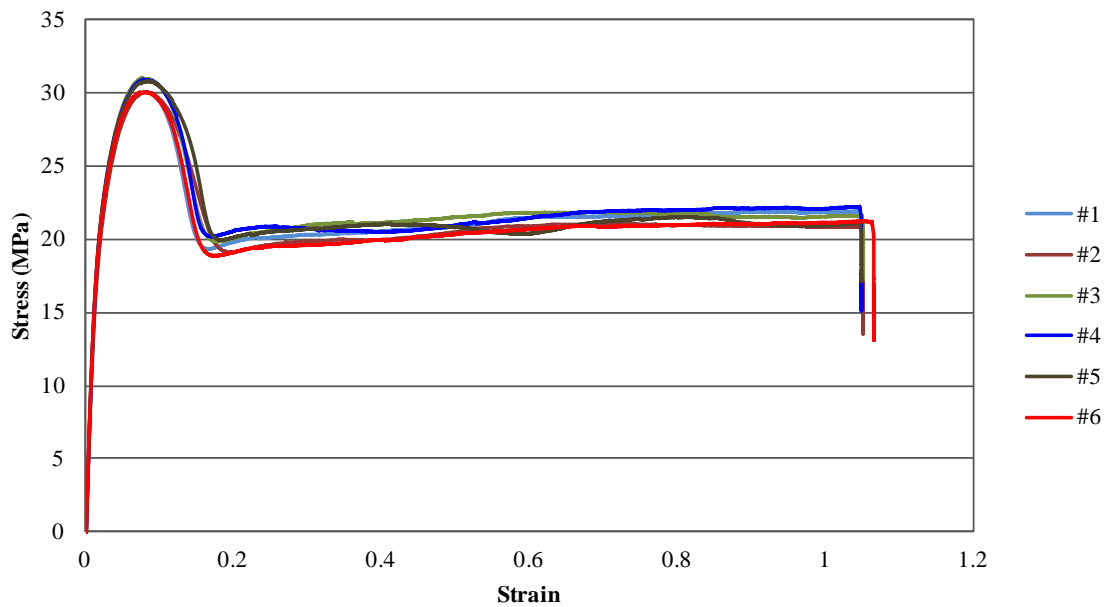


Figure 3.7 Stress - strain curves of PP 3371 specimens with 3% nanoclay reinforcement

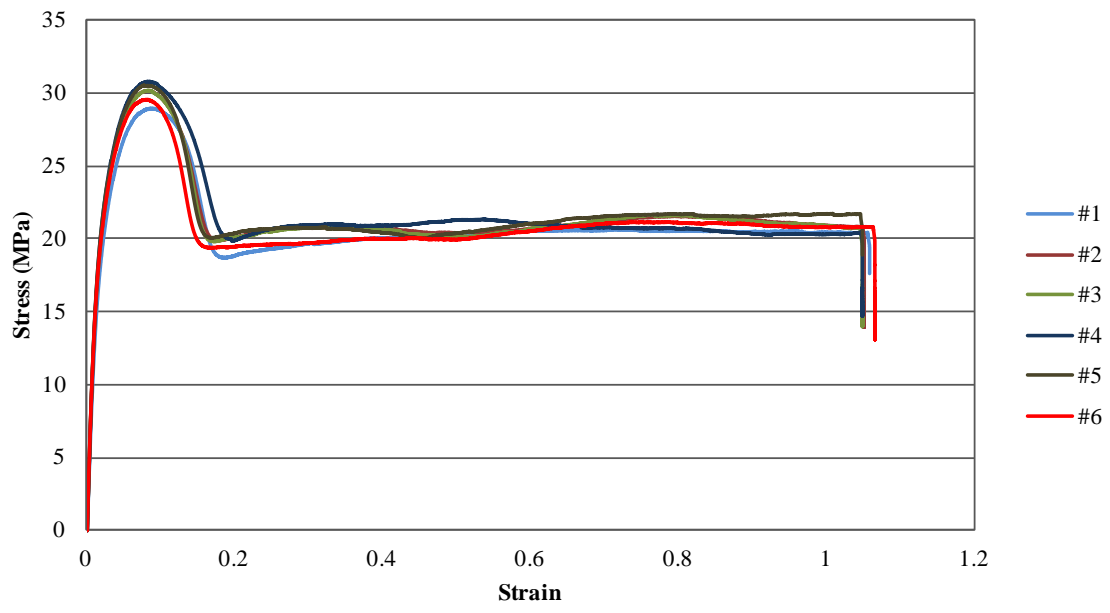


Figure 3.8 Stress - strain curves of PP 3371 specimens with 6% nanoclay reinforcement

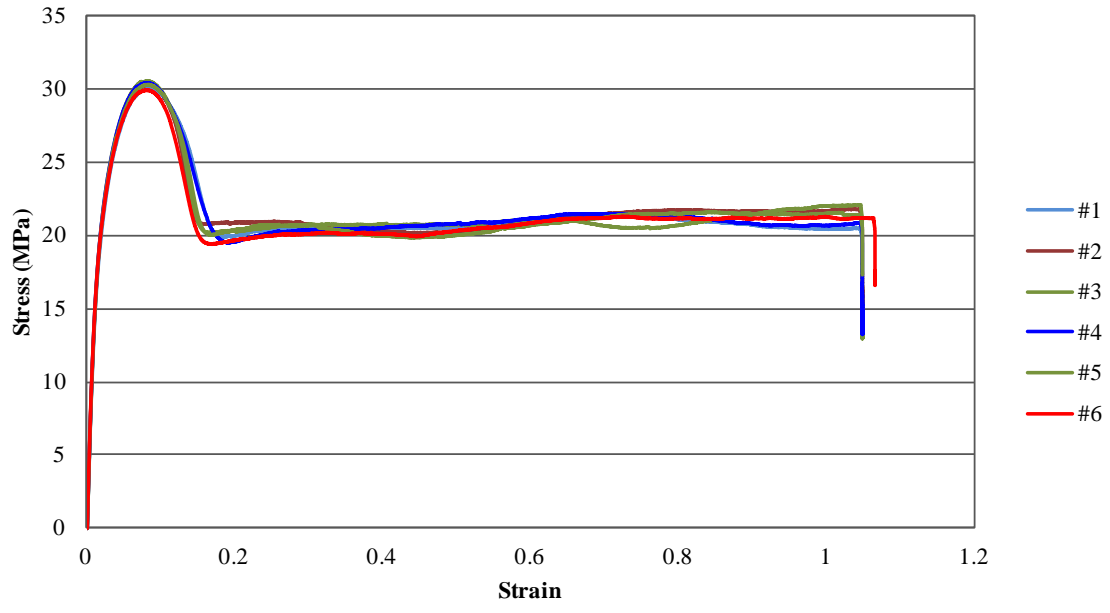


Figure 3.9 Stress - strain curves of PP 3371 specimens with 10% nanoclay reinforcement

As shown in Figures 3.4 - 3.9, all PP 3371 specimens with various nanoclay reinforcements show the same behavior under tensile loading at room temperature. After the stress reaches its peak value necking occurs and none of the specimens failed at 95% strain.

Various percentages of nanoclay reinforcement were used to ascertain its effect on the mechanical properties of the resulting nanocomposites. Figure 3.10 shows the comparison of the stress - strain curves of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement.

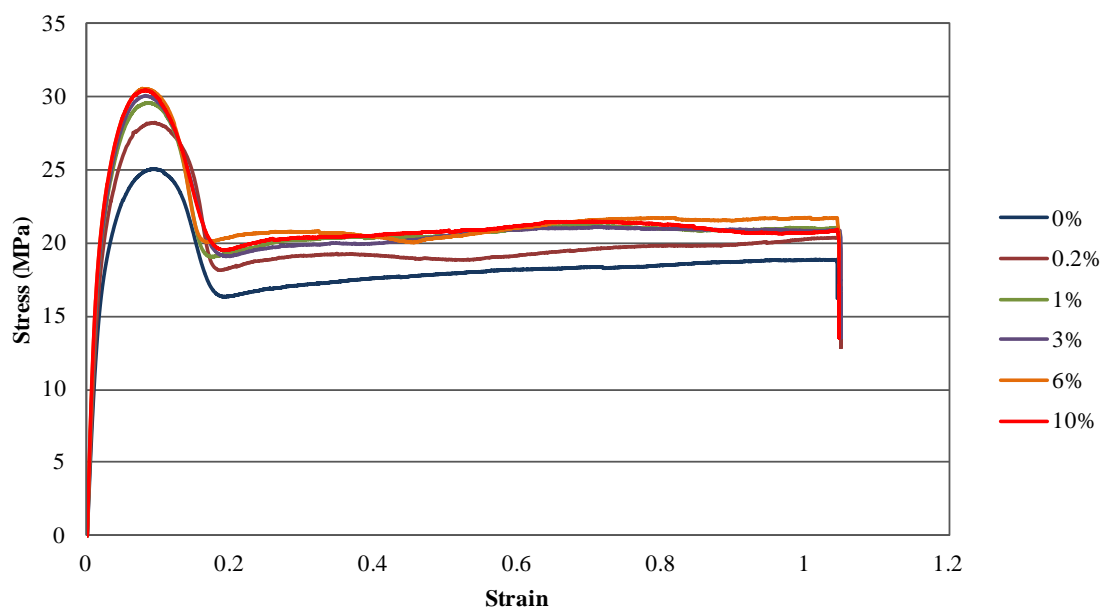


Figure 3.10 Comparison of stress - strain curves of PP 3371 specimens with various nanoclay reinforcement percentages

The results in Figure 3.10 show that, as the percentage of nanoclay reinforcement increases the Young's modulus and the ultimate tensile stress of the nanocomposite also increase. Thus, the nanocomposite becomes stiffer and stronger with higher percentages of nanoclay. However, this effect becomes less pronounced for percentages above 3%. Average value of key mechanical properties, such as Young's modulus, ultimate tensile strength, strain at failure, etc., are given in Table 3.1.

Table 3.1 Average values of material properties of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at room temperature

PP 3371 with	Young's Modulus (GPa) (Average)	Poisson's Ratio (Average)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
0%	1.200	0.3947	25.122	0.0745	1.0233
0.2%	1.317	0.4167	27.826	0.0668	1.0258
1%	1.508	0.4056	29.277	0.0661	1.0268
3%	1.628	0.3852	29.900	0.0653	1.0397
6%	1.646	0.3722	30.009	0.0659	1.0332
10%	1.742	0.3725	30.073	0.0658	1.0331

In addition to using time-history readings from load-cell and crosshead motion, a digital camera was used to capture the surface strain progression. The main reason for using the digital camera was to prove the reliability of the test results obtained from other methods. During the test a picture of the marked specimen was taken every 15 seconds and the strains were calculated by using the displacements obtained from the taken pictures. Figure 3.11 shows a marked specimen mounted in the MTS machine. The optical method was used for pure PP 3371 specimens and PP 3371 specimens with 10% nanoclay reinforcement. The stress - strain curves obtained using the crosshead displacements and the optical method for neat and 10% clay reinforced specimens are given in Figures 3.12 and 3.13 respectively and show remarkable consistency. Thus, one can use the strains calculated from the crosshead motion with some confidence.

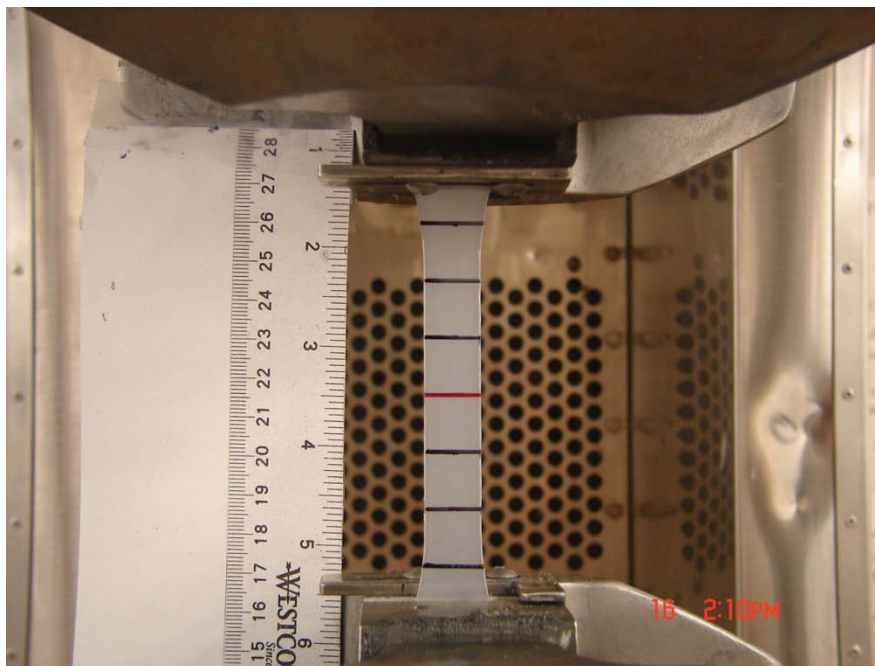


Figure 3.11 A ruler and a marked specimen under tensile load

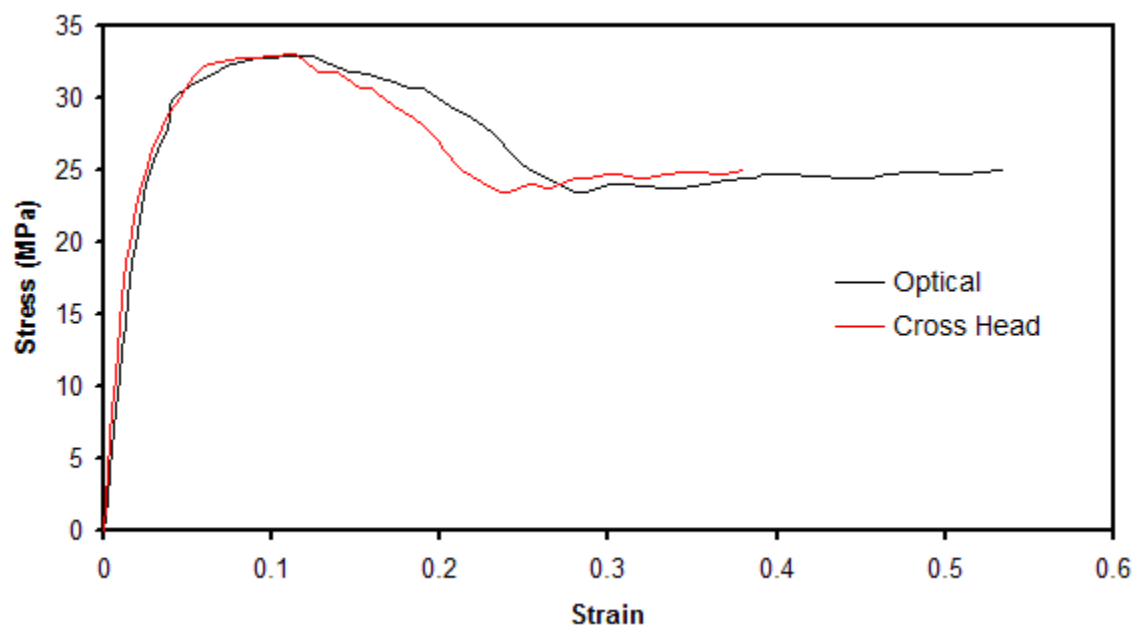


Figure 3.12 Comparison of crosshead and optical stress-strain curves for neat PP3371 specimens

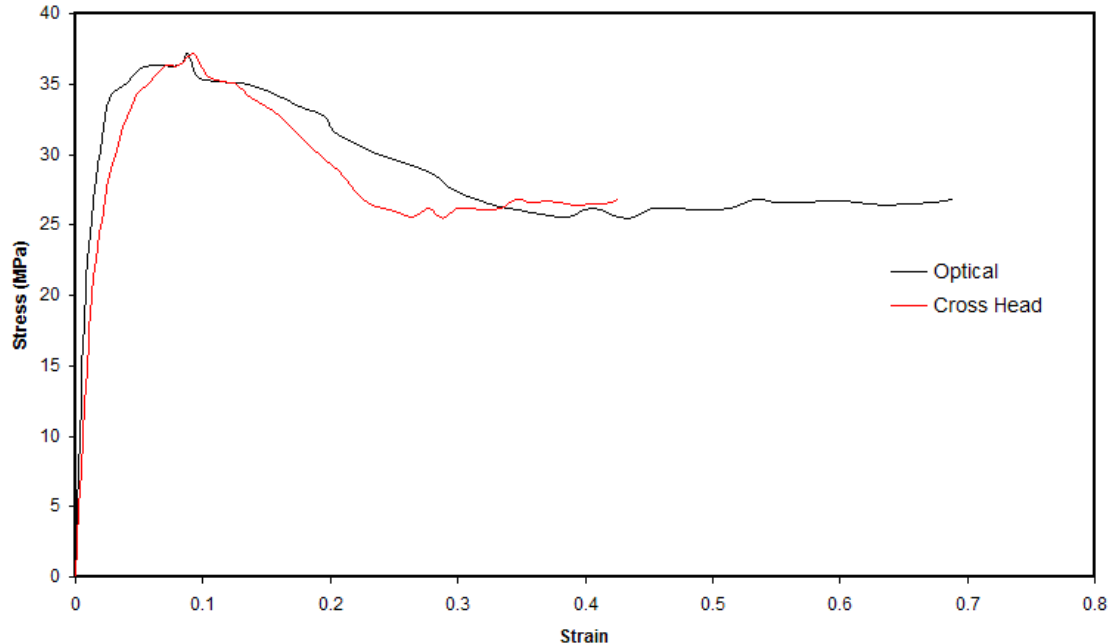


Figure 3.13 Comparison of crosshead and optical stress-strain curves for PP 3371 specimens reinforced with 10% nanoclay

3.2.2 Effect of Temperature

The PP 3371 specimens with various nanoclay reinforcements were then subjected to tensile loading at different temperatures in an MTS machine with a crosshead speed of 5 mm/min. At high temperatures, the environmental chamber was used to perform the tensile tests at 160°F (71°C) and at 120°F (49°C). At low temperatures, the environmental chamber was cooled with liquid nitrogen to reach -4°F (-20°C) and then -65°F (-54°C). The stress-strain curves were plotted for each specimen with the strains calculated from the crosshead displacements. Poisson's ratio values were calculated from strain gage data.

3.2.2-a) Tensile test of PP 3371 at 160°F (71°C)

To obtain the properties of PP 3371 specimen with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at high temperatures the specimens were first subjected to tensile test at 160°F.

The stress - strain curves of PP 3371 specimens with various nanoclay reinforcement percentages are shown Figures 3.14 – 3.19.

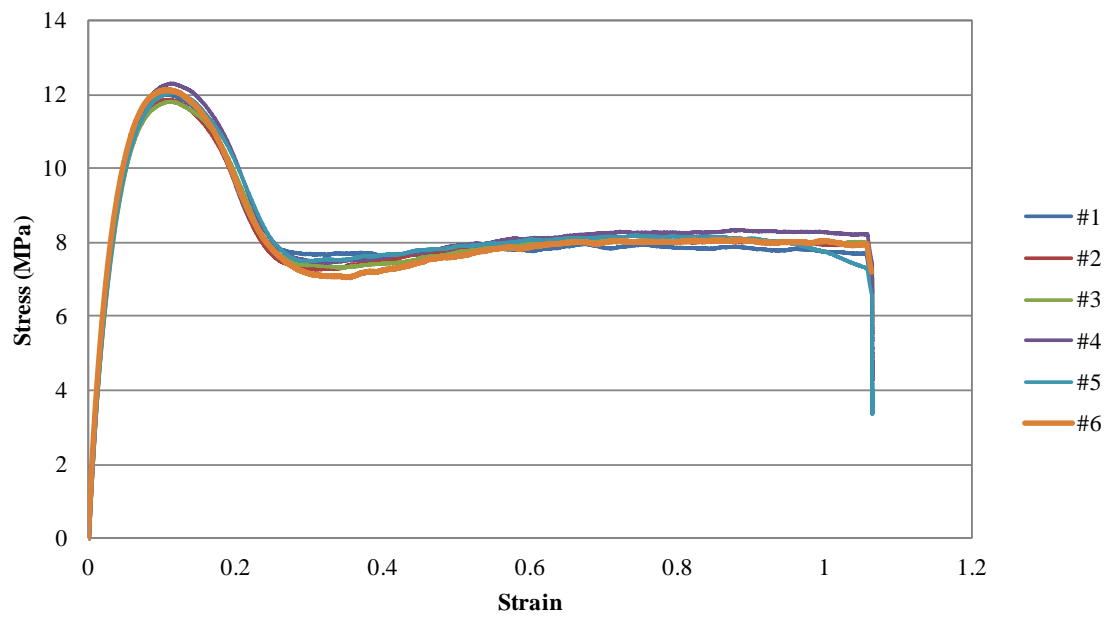


Figure 3.14 Stress - strain curves of neat PP 3371 specimens at 160°F (71°C)

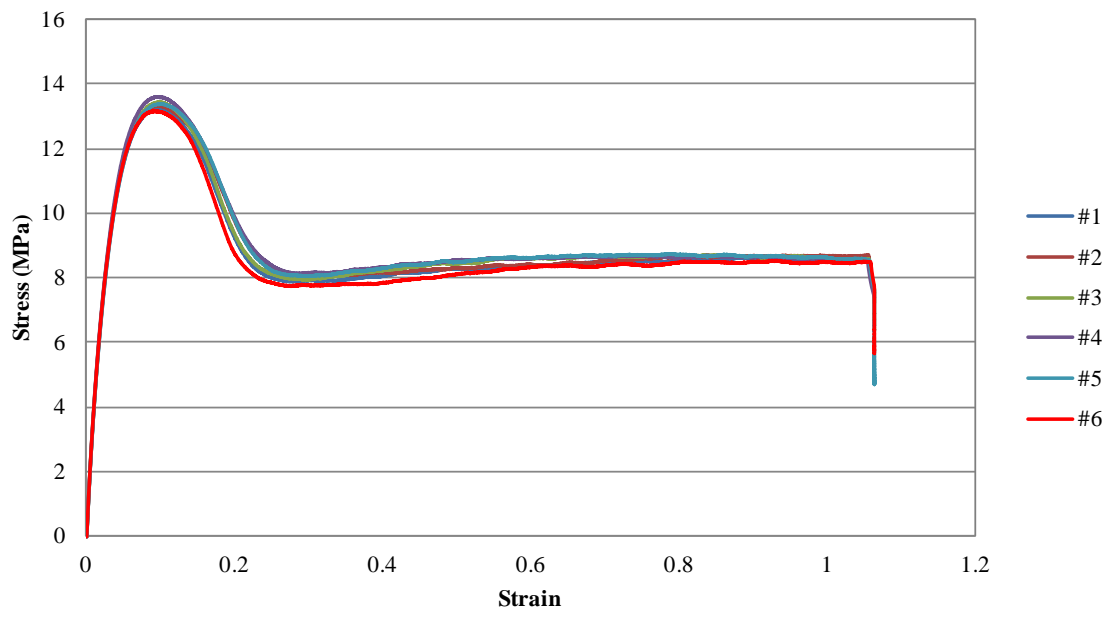


Figure 3.15 Stress - strain curves of PP 3371 specimens with 0.2% nanoclay reinforcement at 160°F (71°C)

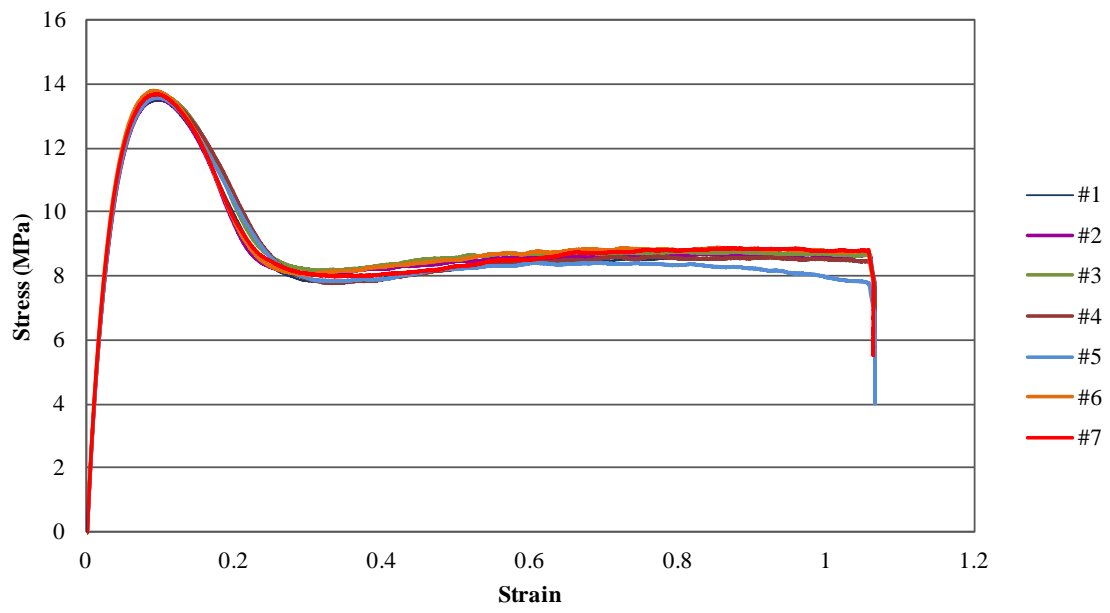


Figure 3.16 Stress - strain curves of PP 3371 specimens with 1% nanoclay reinforcement at 160°F (71°C)

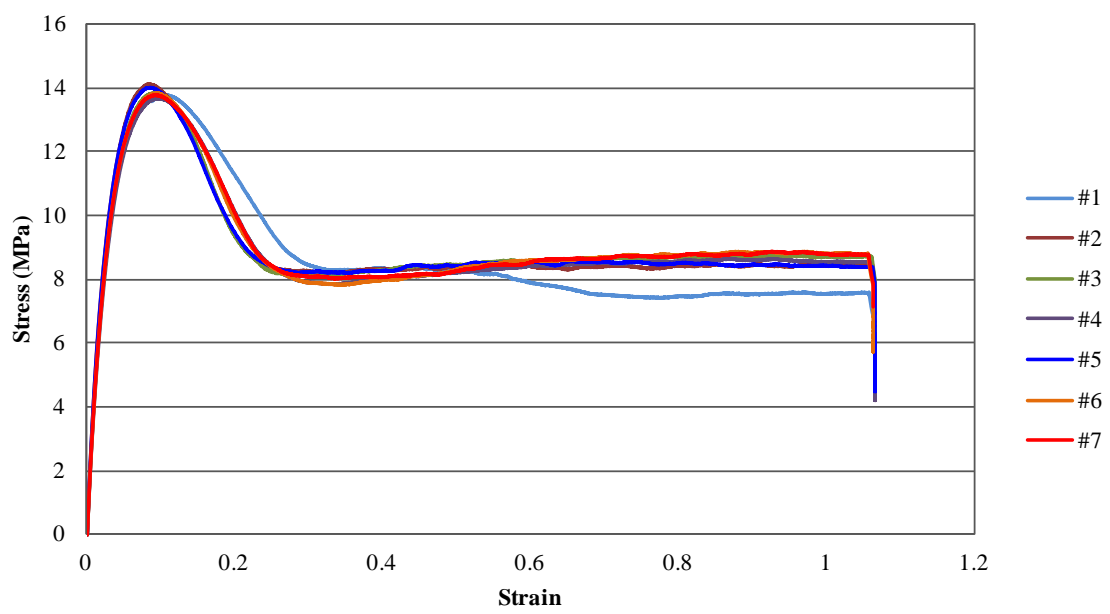


Figure 3.17 Stress – strain curves of PP 3371 specimens with 3% nanoclay reinforcement at 160°F (71°C)

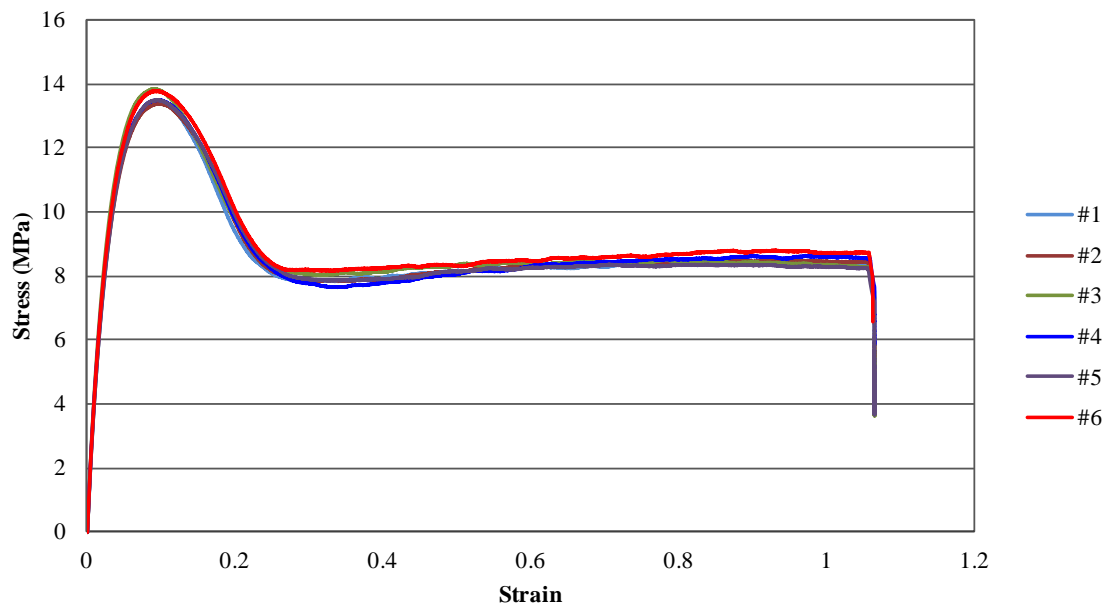


Figure 3.18 Stress - strain curves of PP 3371 specimens with 6% nanoclay reinforcement at 160°F (71°C)

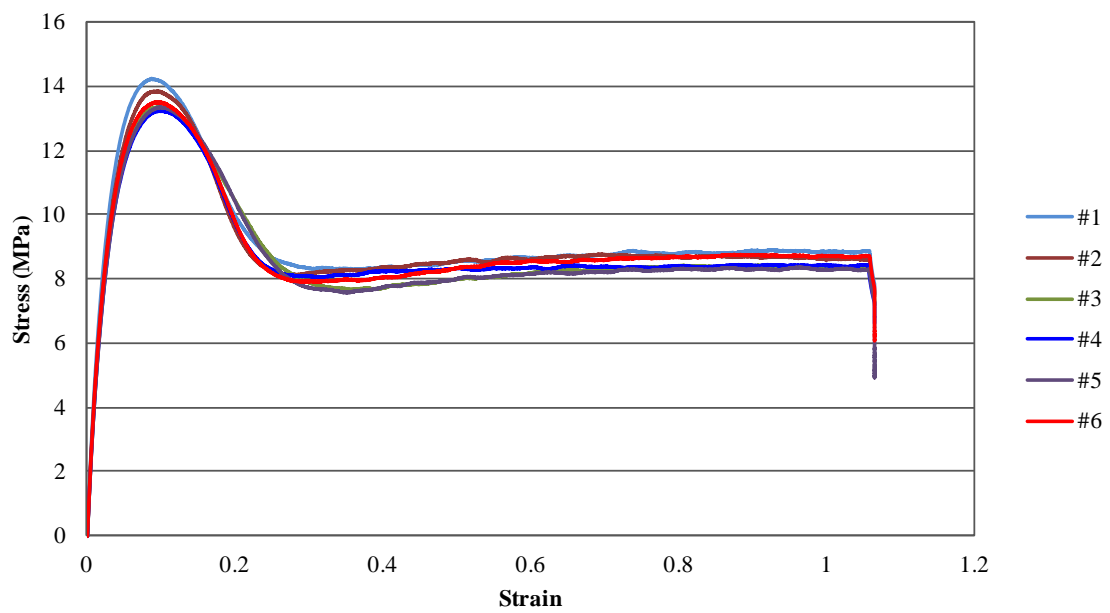


Figure 3.19 Stress - strain curves of PP 3371 specimens with 10% nanoclay reinforcement at 160°F (71°C)

The results in Figures 3.14 – 3.19 are summarized in Figure 3.20 which depicts the effect of nanoclay reinforcement at 160°F (71°C).

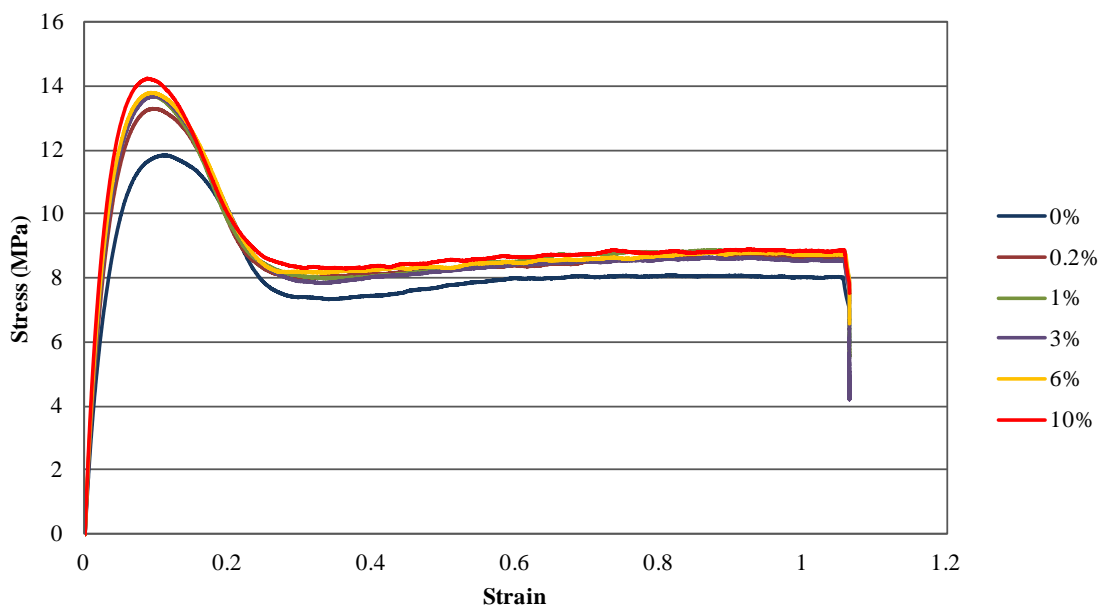


Figure 3.20 Comparison of stress-strain curves of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at 160°F (71°C)

From the results shown in Figure 3.20, it is seen that at 160°F (71°C), an increase in nanoclay reinforcement leads to a higher Young's modulus and ultimate stress as was the case at room temperature.

Mechanical properties, such as Young's modulus, ultimate stress, etc. obtained at 160°F (71°C) are given in Table 3.2.

Table 3.2 Average values of material properties of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at 160°F (71°C)

PP 3371 with	Young's Modulus (GPa) (Average)	Poisson's Ratio (Average)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
0%	0.392	0.4256	11.730	0.0819	1.0360
0.2%	0.449	0.4782	12.997	0.0725	1.0308
1%	0.465	0.4715	13.303	0.0732	1.0341
3%	0.497	0.4523	13.567	0.0713	1.0354
6%	0.503	0.5077	13.396	0.0774	1.0374
10%	0.515	0.4896	13.431	0.0791	1.0338

3.2.2-b) Tensile test of PP 3371 at 120°F (49°C)

Next the specimens were tested at 120°F (49°C). The stress - strain curves of PP 3371 specimens with various nanoclay reinforcement percentages are plotted with strains calculated from the crosshead displacements and shown Figures 3.21 – 3.26.

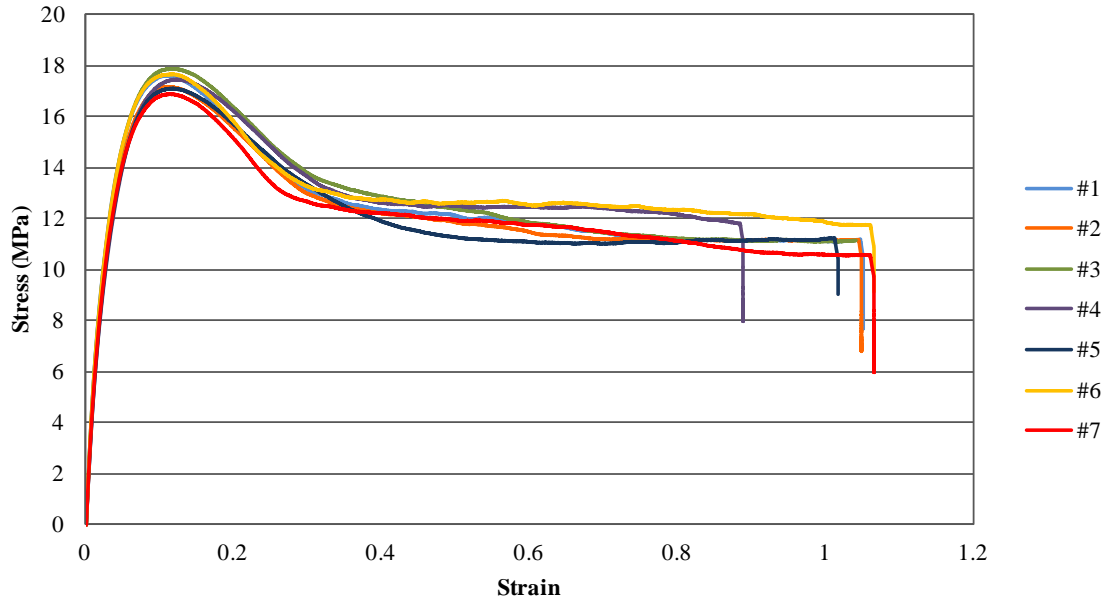


Figure 3.21 Stress - strain curves of neat PP 3371 specimens at 120°F (49°C)

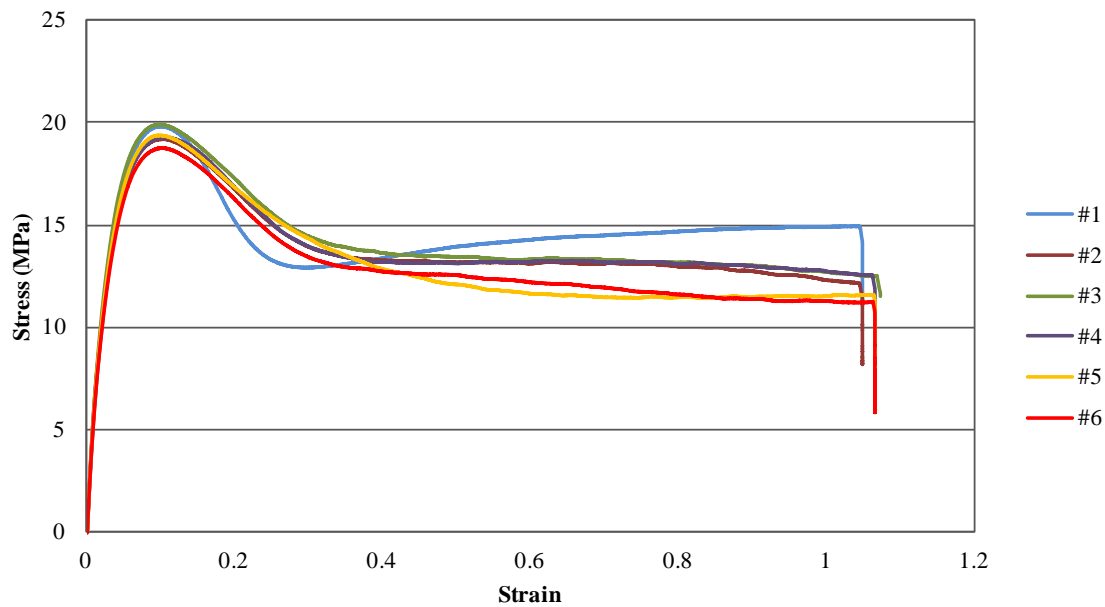


Figure 3.22 Stress - strain curves of PP 3371 specimens with 0.2% nanoclay reinforcement at 120°F (49°C)

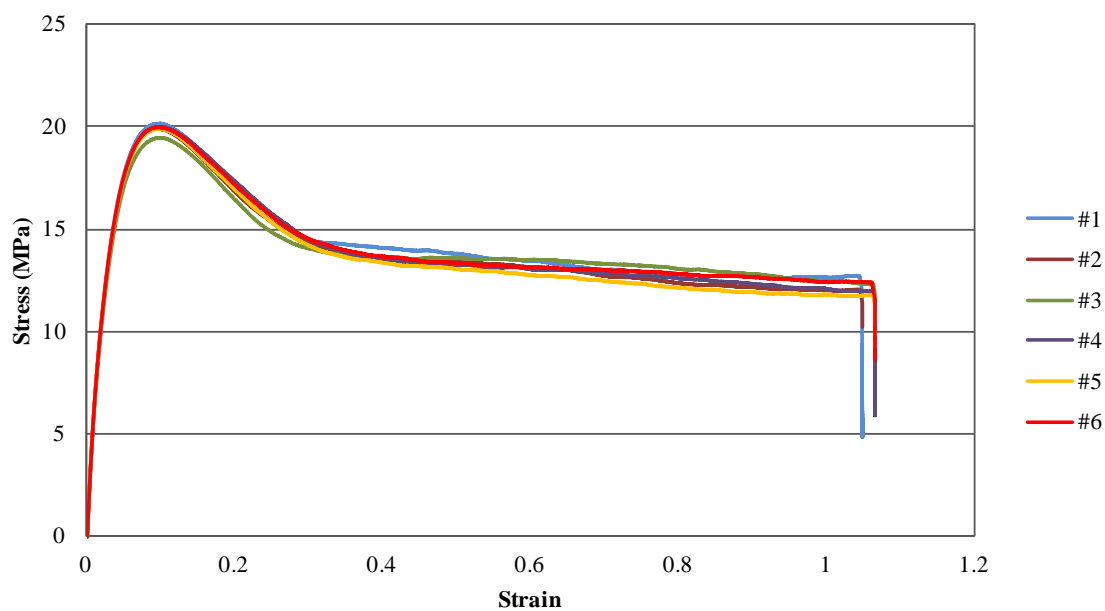


Figure 3.23 Stress - strain curves of PP 3371 specimens with 1% nanoclay reinforcement at 120°F (49°C)

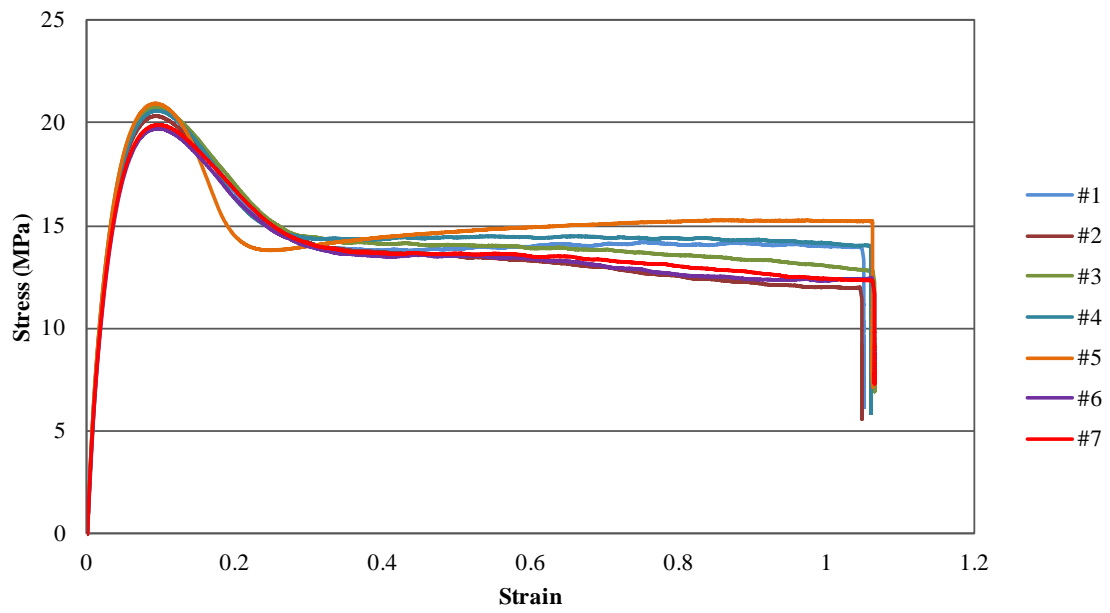


Figure 3.24 Stress - strain curves of PP 3371 specimens with 3% nanoclay reinforcement at 120°F (49°C)

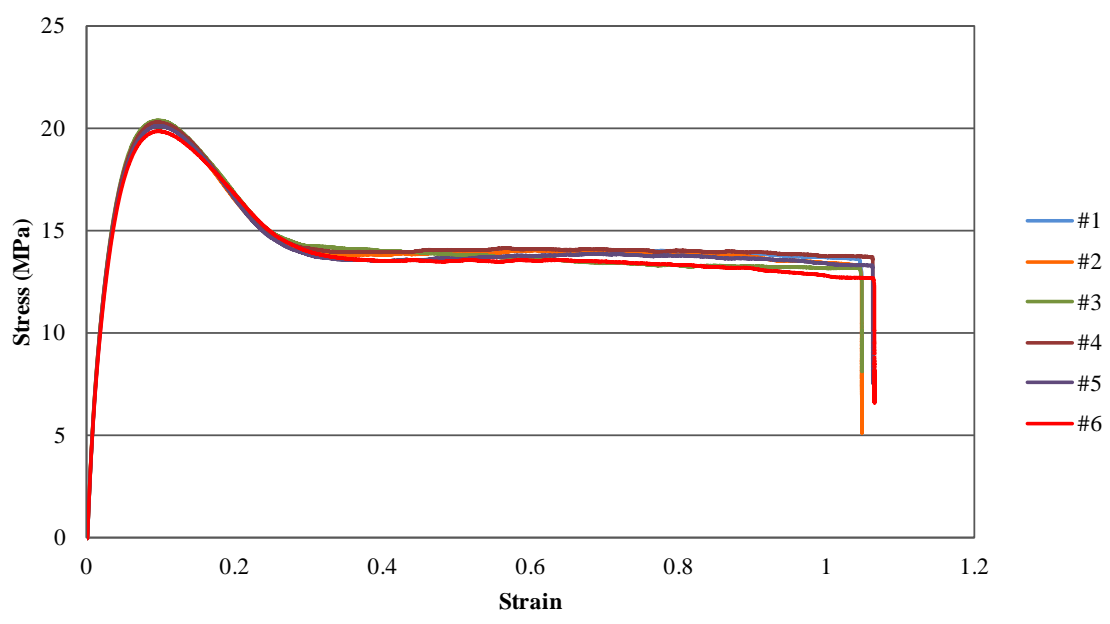


Figure 3.25 Stress-strain curves of PP 3371 specimens with 6% nanoclay reinforcement at 120°F (49°C)

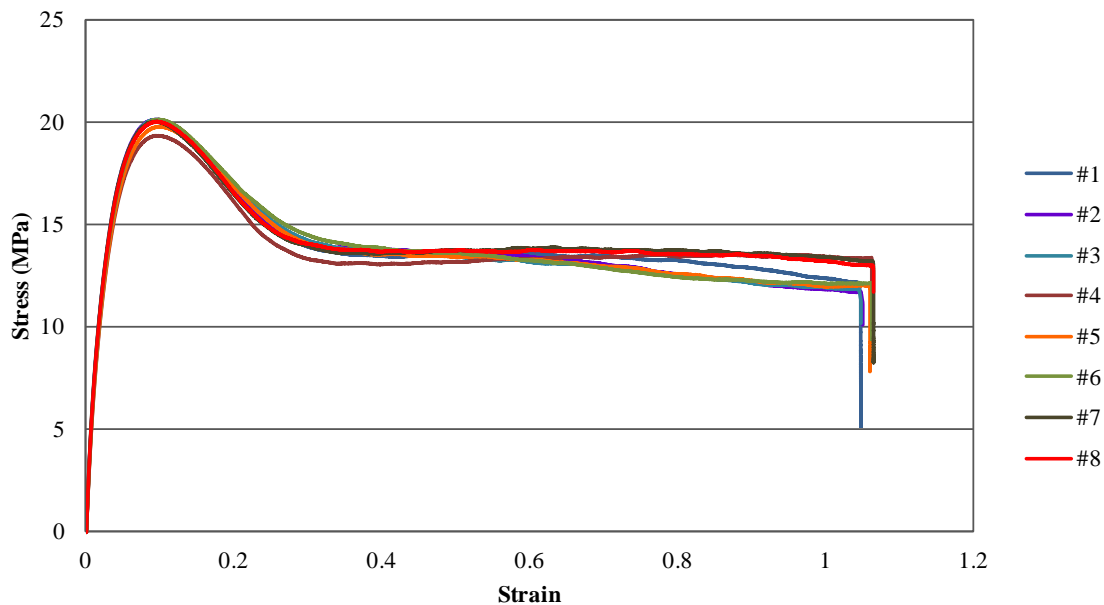


Figure 3.26 Stress-strain curves of PP 3371 specimens with 10% nanoclay reinforcement at 120°F (49°C)

The results from Figures 3.21 – 3.26 are consolidated in Figure 3.27.

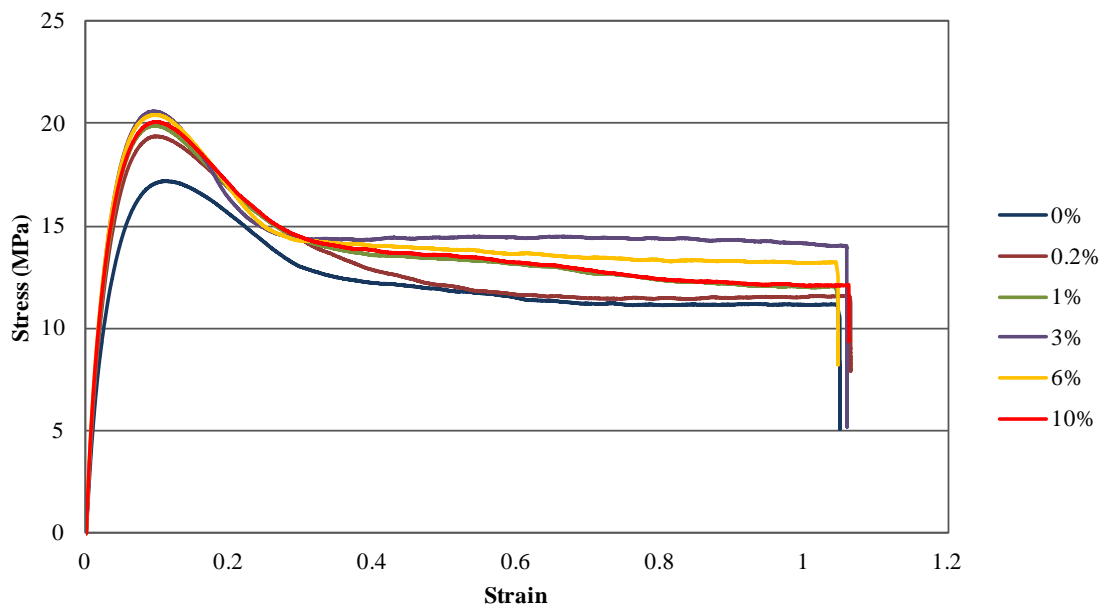


Figure 3.27 Comparison of stress - strain curves of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at 120°F (49°C)

From the results shown in Figure 3.27, it is seen that at 120°F also an increase in nanoclay reinforcement leads to a higher Young's modulus and ultimate stress as was the case at room temperature. Furthermore, for PP 3371 specimens with 1%, 3% and 6% nanoclay reinforcement the spread between the curves is very small. Average value of mechanical properties, such as Young's modulus, ultimate stress, etc. obtained at 120°F (49°C) are given in Table 3.3.

Table 3.3. Average values of material properties of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at 120°F (49°C)

PP 3371 with	Young's Modulus (GPa) (Average)	Poisson's Ratio (Average)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
0%	0.616	0.4038	17.166	0.0919	1.0021
0.2%	0.728	0.4350	19.087	0.0788	1.0355
1%	0.765	0.4631	19.583	0.0772	1.0345
3%	0.828	0.4355	19.954	0.0789	1.0336
6%	0.833	0.4334	19.863	0.0759	1.0248
10%	0.866	0.4111	19.981	0.0759	1.0353

3.2.2-c) Tensile test of PP 3371 at -4°F (-20°C)

The nanoclay reinforced PP 3371 specimens were also tested at low temperatures starting with -4°F (-20°C). The stress - strain curves of PP 3371 specimens with various percentage of nanoclay reinforcement are plotted in Figures 3.28-3.33.

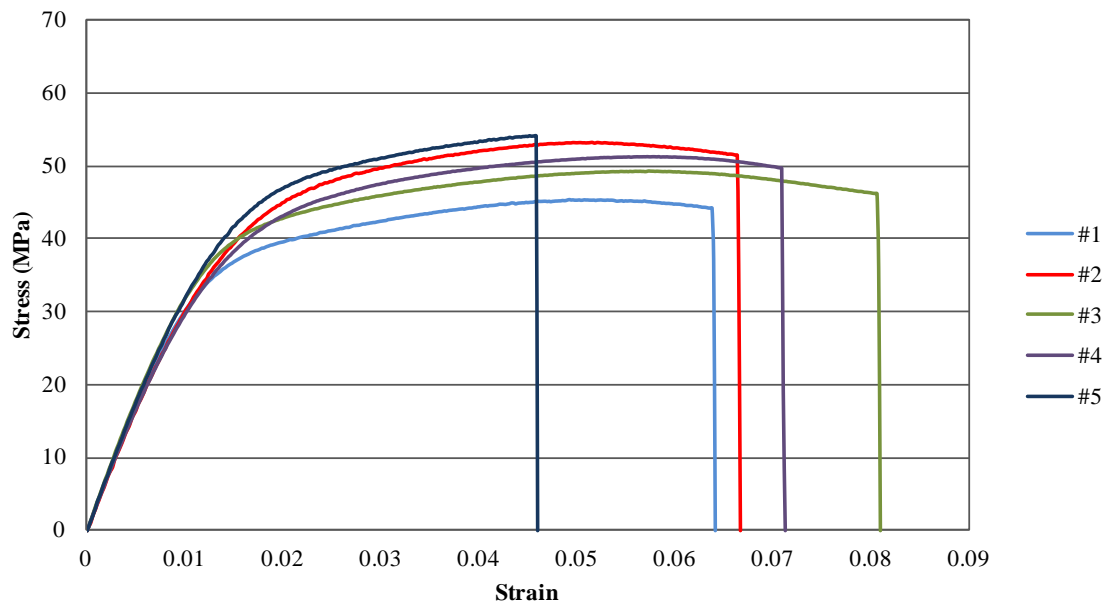


Figure 3.28 Stress - strain curves of PP 3371 specimens with 0% nanoclay reinforcement at -4°F (-20°C)

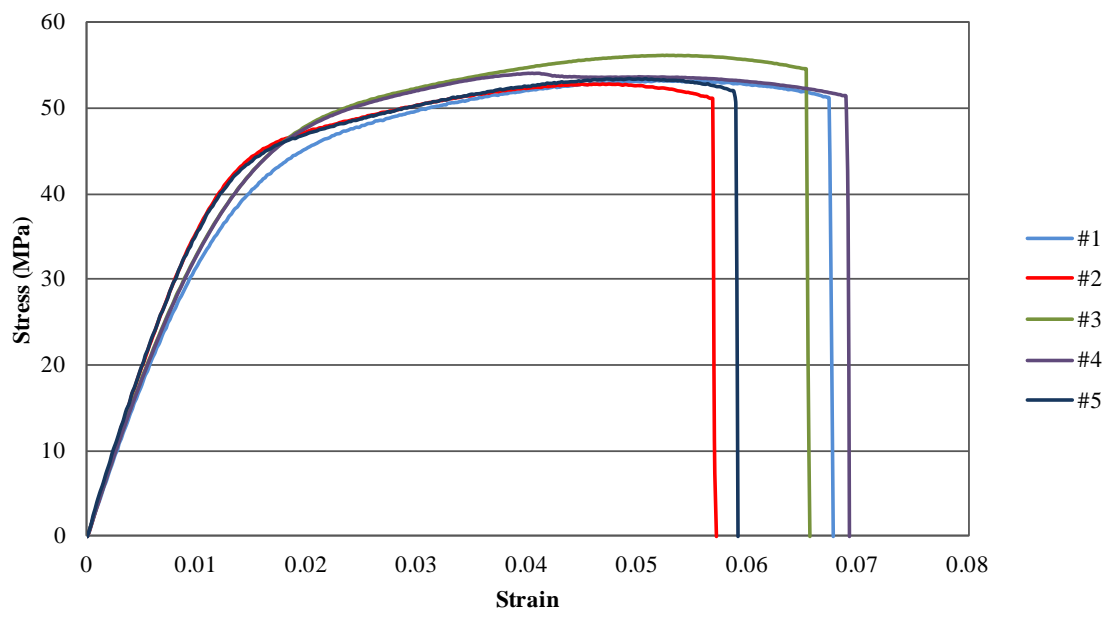


Figure 3.29 Stress - strain curves of PP 3371 specimens with 0.2% nanoclay reinforcement at -4°F (-20°C)

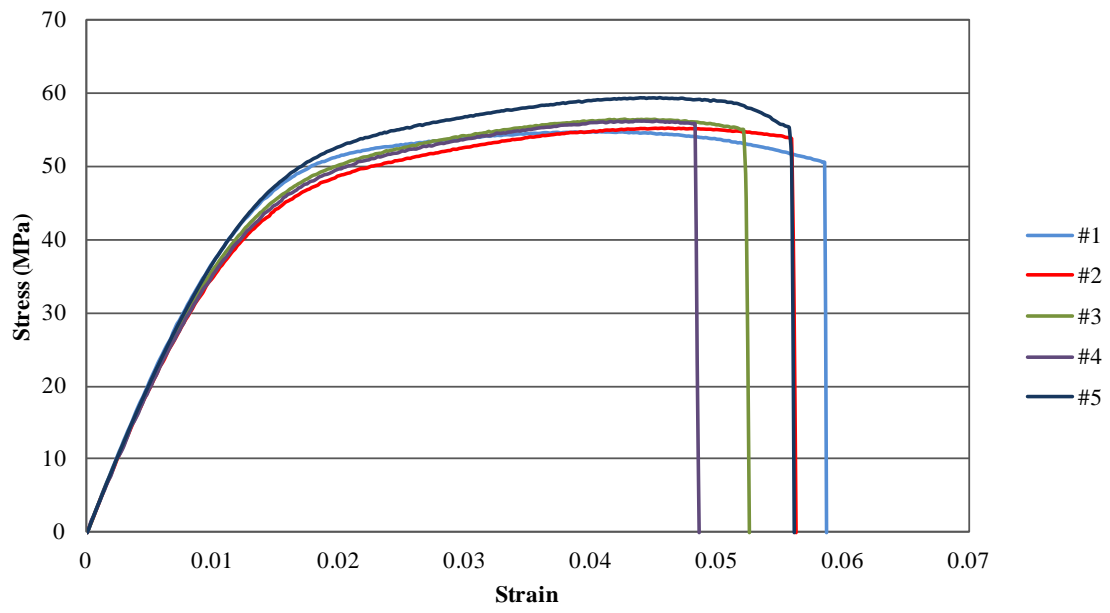


Figure 3.30 Stress - strain curves of PP 3371 specimens with 1% nanoclay reinforcement at -4°F (-20°C)

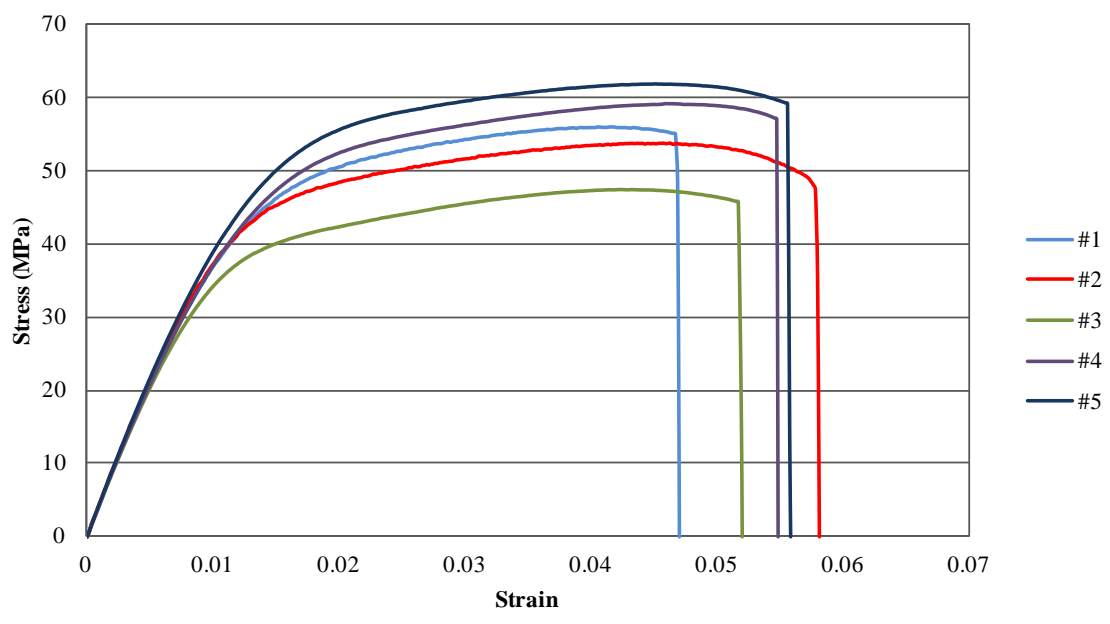


Figure 3.31 Stress - strain curves of PP 3371 specimens with 3% nanoclay reinforcement at -4°F (-20°C)

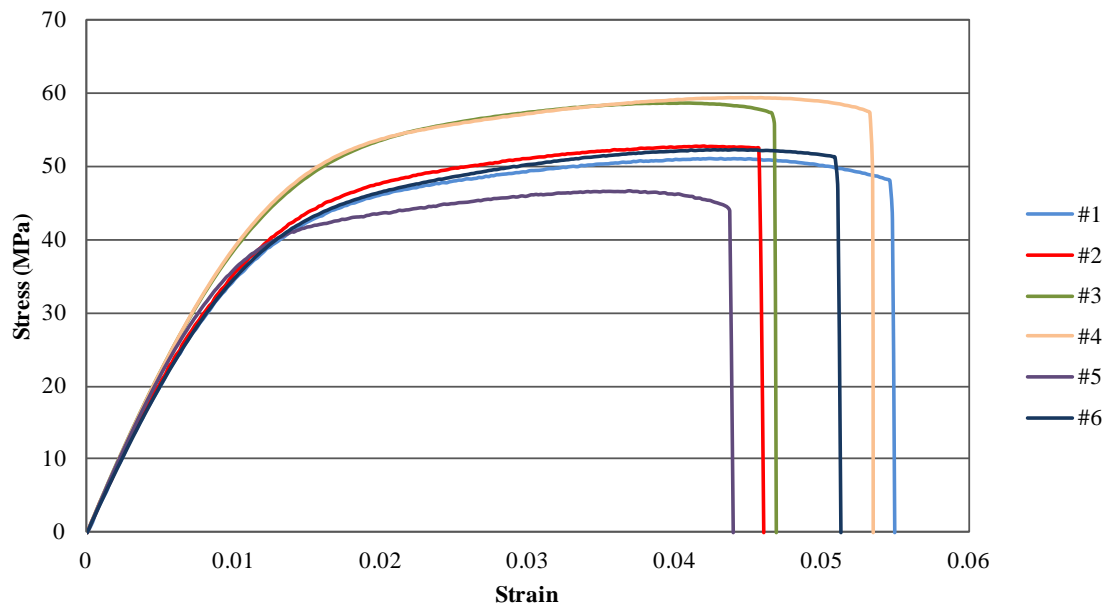


Figure 3.32 Stress - strain curves of PP 3371 specimens with 6% nanoclay reinforcement at -4°F (-20°C)

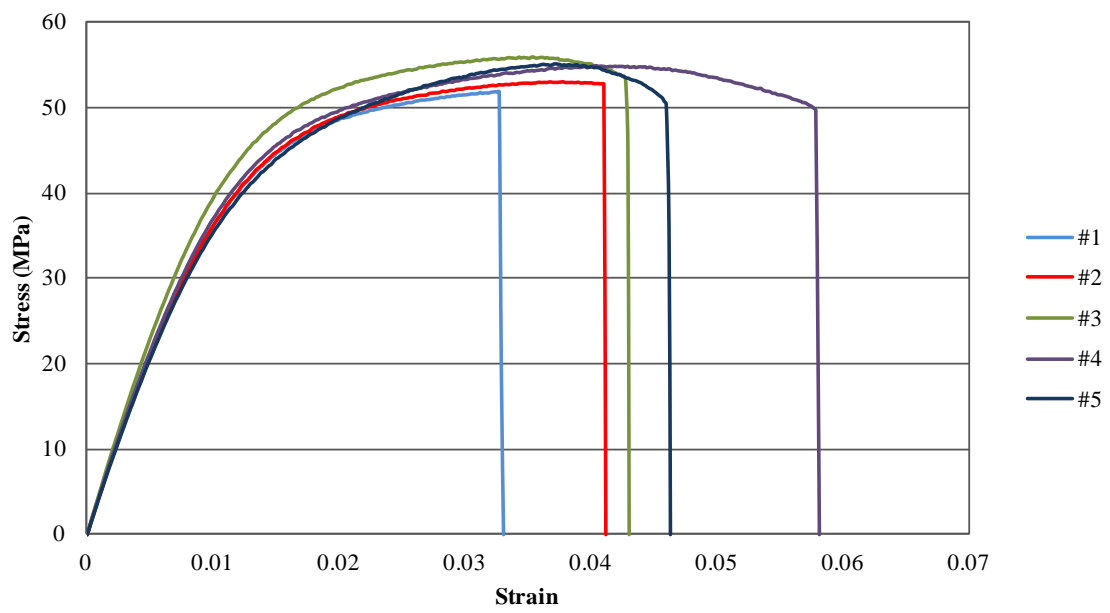


Figure 3.33 Stress - strain curves of PP 3371 specimens with 10% nanoclay reinforcement at -4°F (-20°C)

Again the results for various nanoclay reinforcement percentages at -4°F (-20°C) are summarized in Figure 3.34.

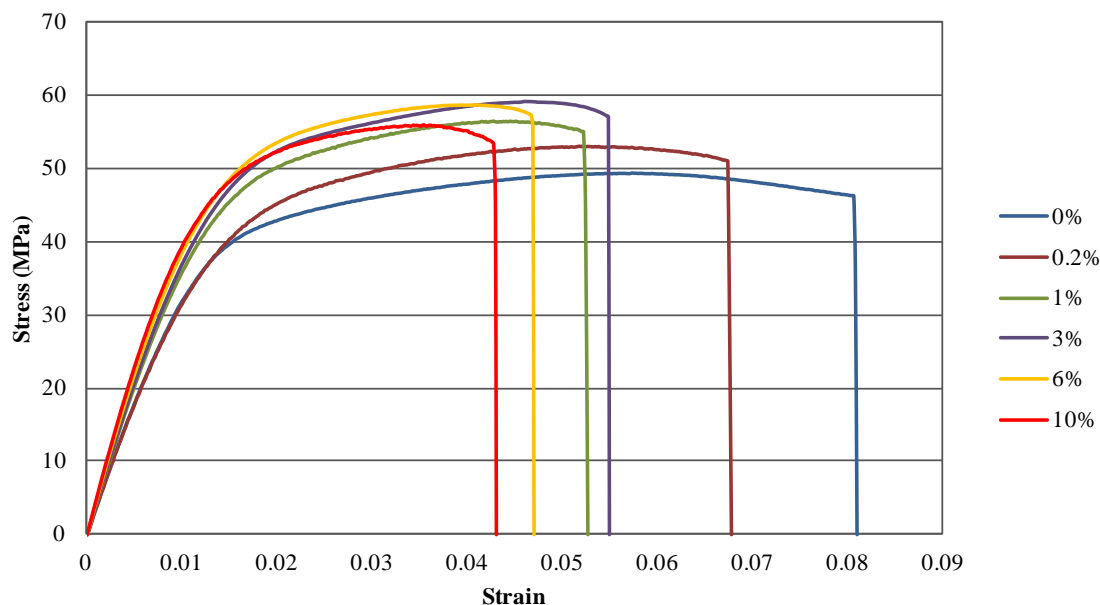


Figure 3.34 Comparison of stress-strain curves of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at -4°F (-20°C)

As seen in Figure 3.34, at low temperature all the specimens failed at relatively low strains (4% to 9%) compared to failure strains at room and high temperatures. In general, specimens with a higher percentage of nanoclay reinforcement failed at lower strains. Average values of mechanical properties, such as Young's modulus, ultimate stress, etc. obtained at -4°F (-20°C) are given in Table 3.4.

Table 3.4. Average of values of material properties of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at -4°F (-20°C)

PP 3371 with	Young's Modulus (GPa) (Average)	Poisson's Ratio (Average)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
0%	3.346	0.3438	50.607	0.0517	0.0642
0.2%	3.713	0.3307	53.8378	0.0491	0.0619
1%	4.008	0.3156	56.3626	0.0431	0.0523
3%	4.179	0.3114	55.5831	0.0433	0.0518
6%	4.212	0.3056	52.3178	0.0409	0.0471
10%	4.333	0.2928	54.0557	0.0367	0.0427

3.2.2-d) Tensile test of PP 3371 at -65°F (-54°C)

The tensile tests were also performed at the lowest temperature value of -65°F (-54°C). The stress vs. strain curves are shown Figures 3.35 – 3.40.

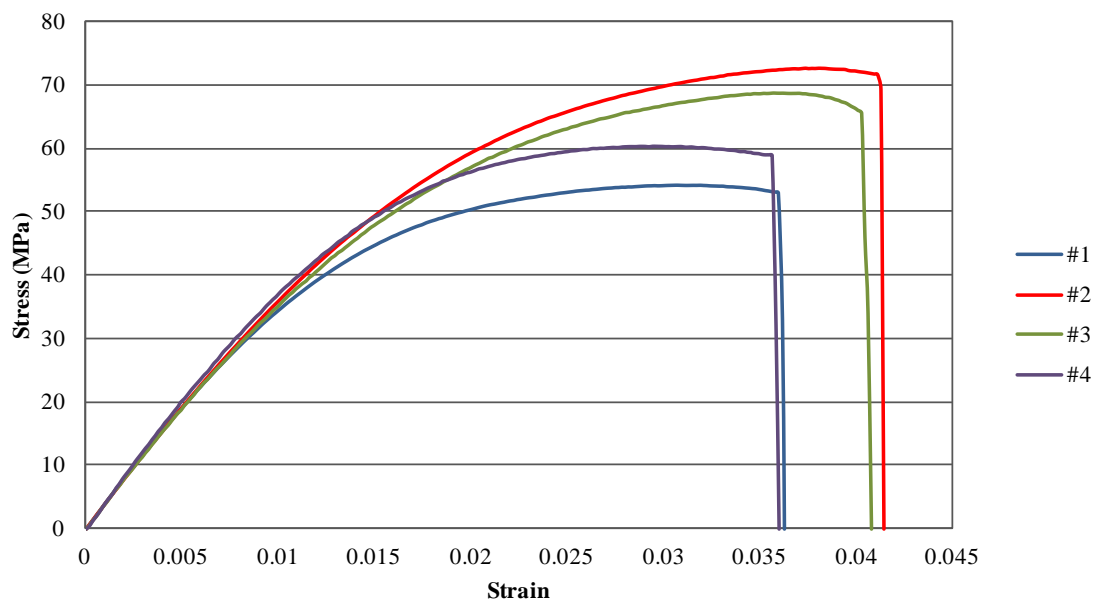


Figure 3.35 Stress - strain curves of neat PP 3371 specimens at -65°F (-54°C)

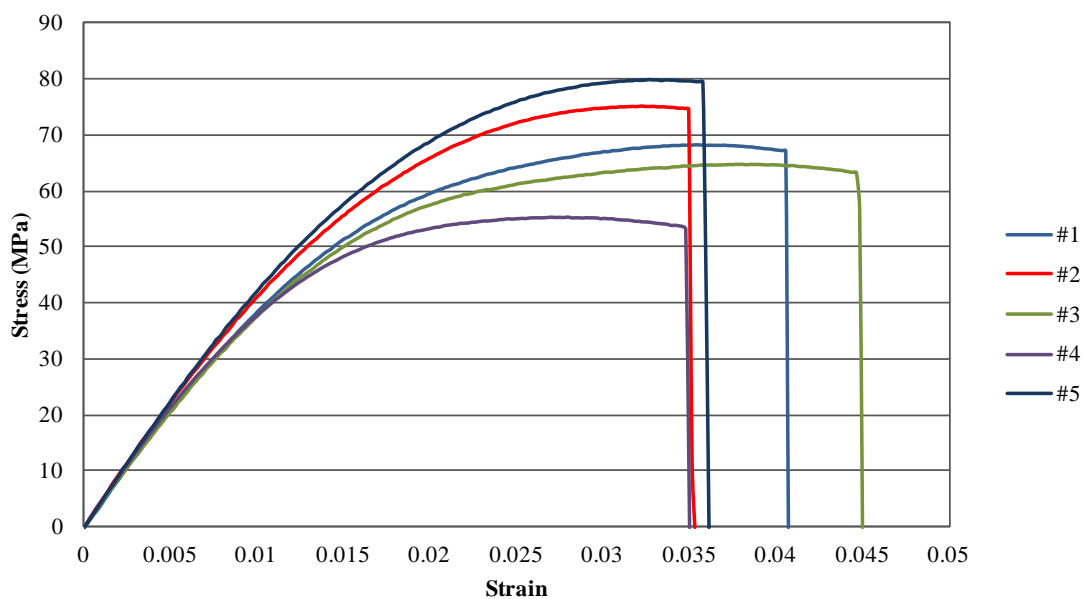


Figure 3.36 Stress - strain curves of PP 3371 specimens with 0.2% nanoclay reinforcement at -65°F (-54°C)

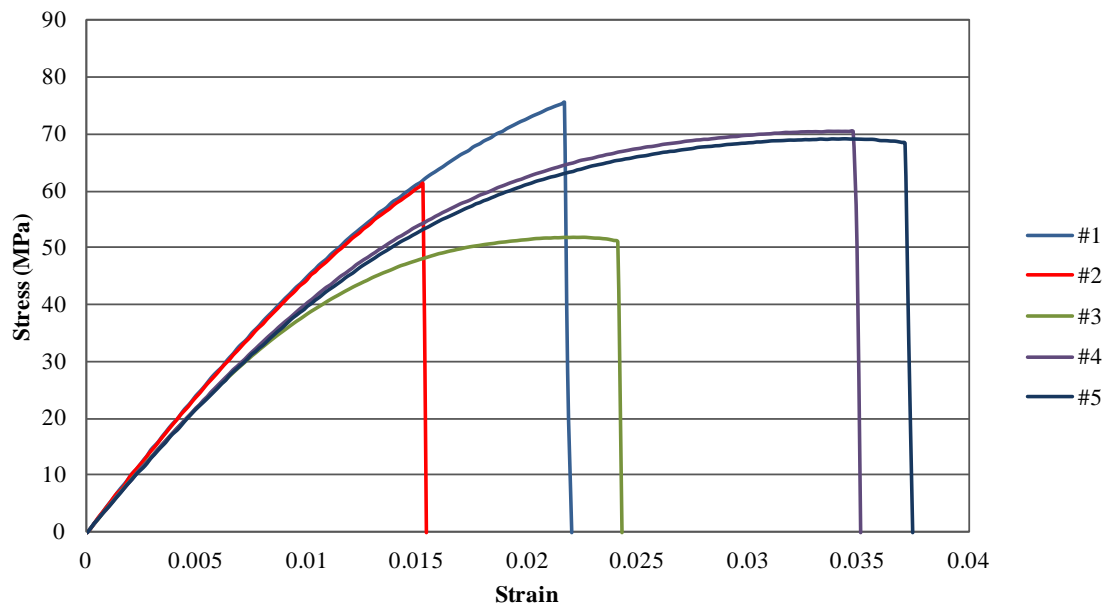


Figure 3.37 Stress - strain curves of PP 3371 specimens with 1% nanoclay reinforcement at -65°F (-54°C)

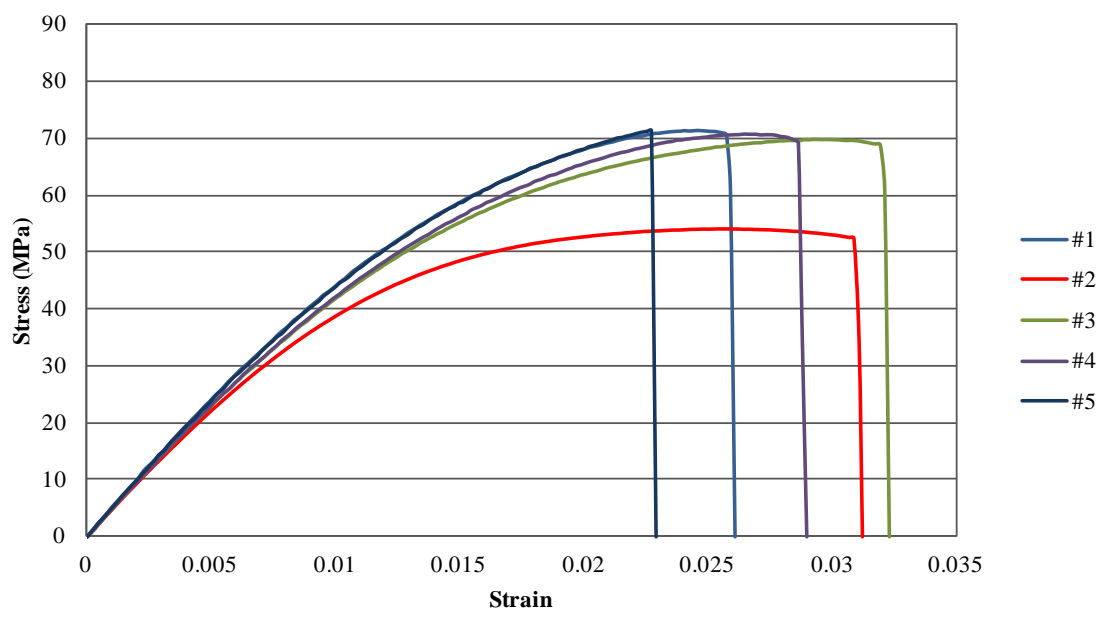


Figure 3.38 Stress - strain curves of PP 3371 specimens with 3% nanoclay reinforcement at -65°F (-54°C)

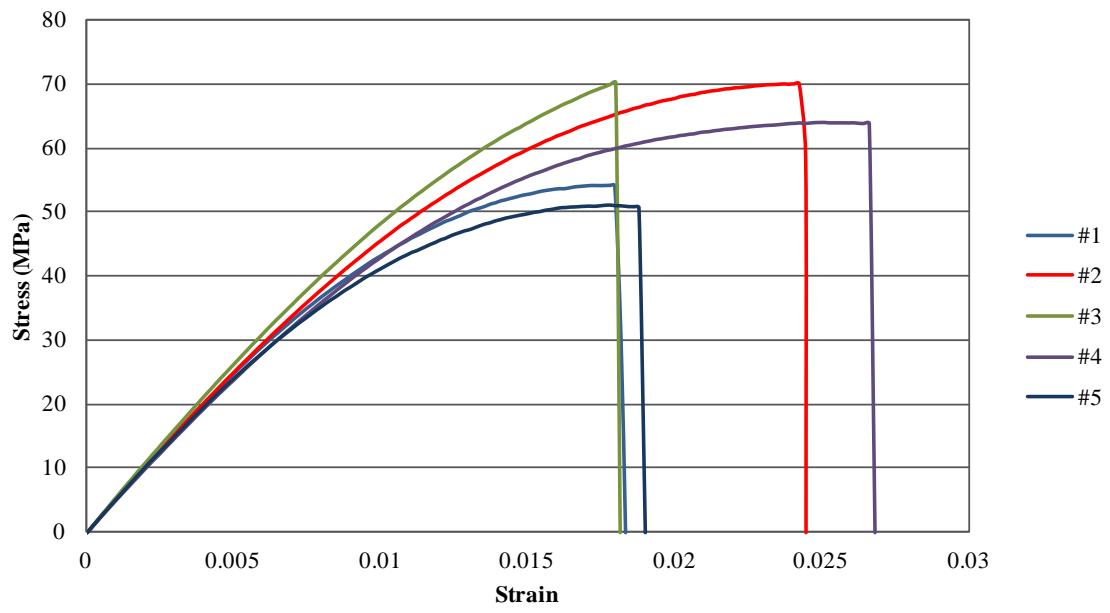


Figure 3.39 Stress - strain curves of PP 3371 specimens with 6% nanoclay reinforcement at -65°F (-54°C)

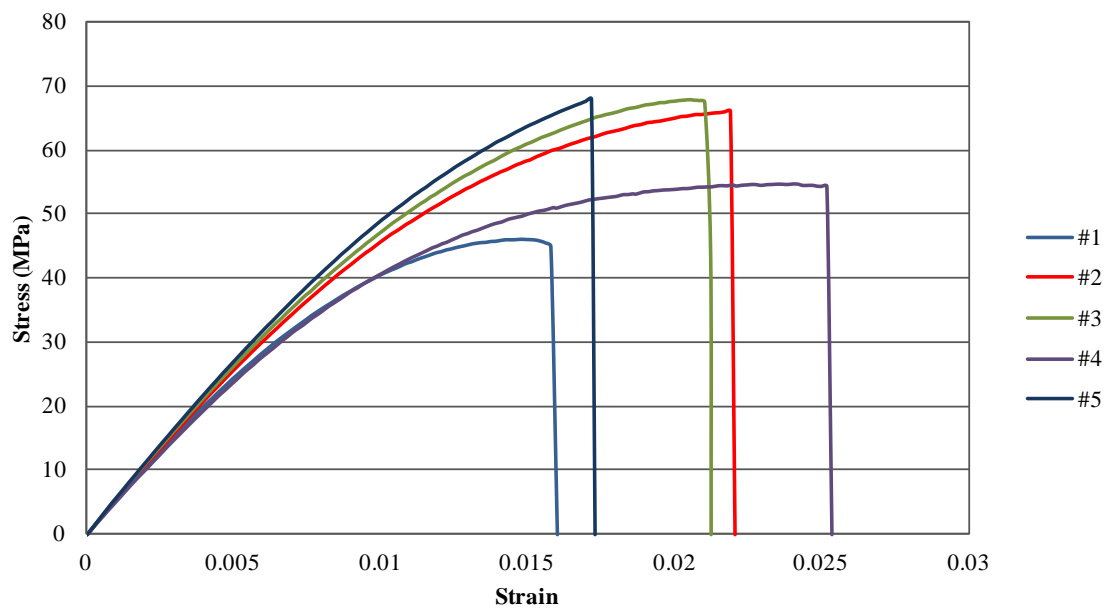


Figure 3.40 Stress - strain curves of PP 3371 specimens with 10% nanoclay reinforcement at -65°F (-54°C)

The results in Figures 3.35 – 3.40 are summarized in Figure 3.41 which depicts the effect of nanoclay reinforcement at -65°F (-54°C).

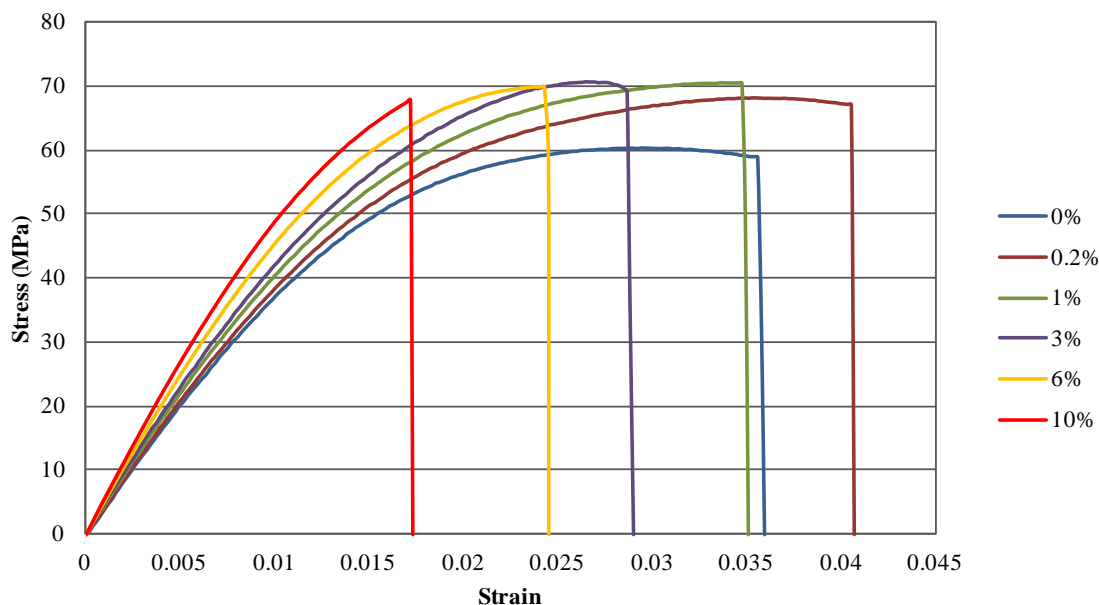


Figure 3.41 Comparison of stress-strain curves of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at -65°F (-54°C)

From the results shown in Figure 3.41, it is seen that at -65°F (-54°C) an increase in nanoclay reinforcement leads to a higher Young's modulus and ultimate stress as was the case at room temperature. At -65°F the specimens failed at relatively low strains (1.5% to 4%) in contrast to the specimens tested at room and high temperatures. Mechanical properties, such as Young's modulus, ultimate stress, etc. obtained at -65°F (-54°C) are given in Table 3.5.

Table 3.5 Average values of material properties of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement at -65°F (-54°C)

PP 3371 with	Young's Modulus (GPa) (Average)	Poisson's Ratio (Average)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
0%	3.866	0.3429	63.9034	0.0327	0.0357
0.2%	4.284	0.3283	68.5570	0.0325	0.0372
1%	4.603	0.3193	65.1837	0.0254	0.0259
3%	4.694	0.3076	67.0177	0.0247	0.0272
6%	4.973	0.3039	61.5911	0.0203	0.0203
10%	5.092	0.2983	60.2389	0.0192	0.0194

3.2.2-e) Comparison of stress-strain curves of PP 3371 specimens at various temperatures for a given nanoclay reinforcement percentage

The stress-strain curves for a given nanoclay reinforcement percentage obtained at $-65^{\circ}\text{F}(-54^{\circ}\text{C})$, $-4^{\circ}\text{F}(-20^{\circ}\text{C})$, room temperature, $120^{\circ}\text{F}(49^{\circ}\text{C})$ and $160^{\circ}\text{F}(71^{\circ}\text{C})$ are plotted on the same graph to showcase the effect of temperature on the nanocomposite. In Figures 3.42-3.47, the stress-strain curves of PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement are shown, respectively.

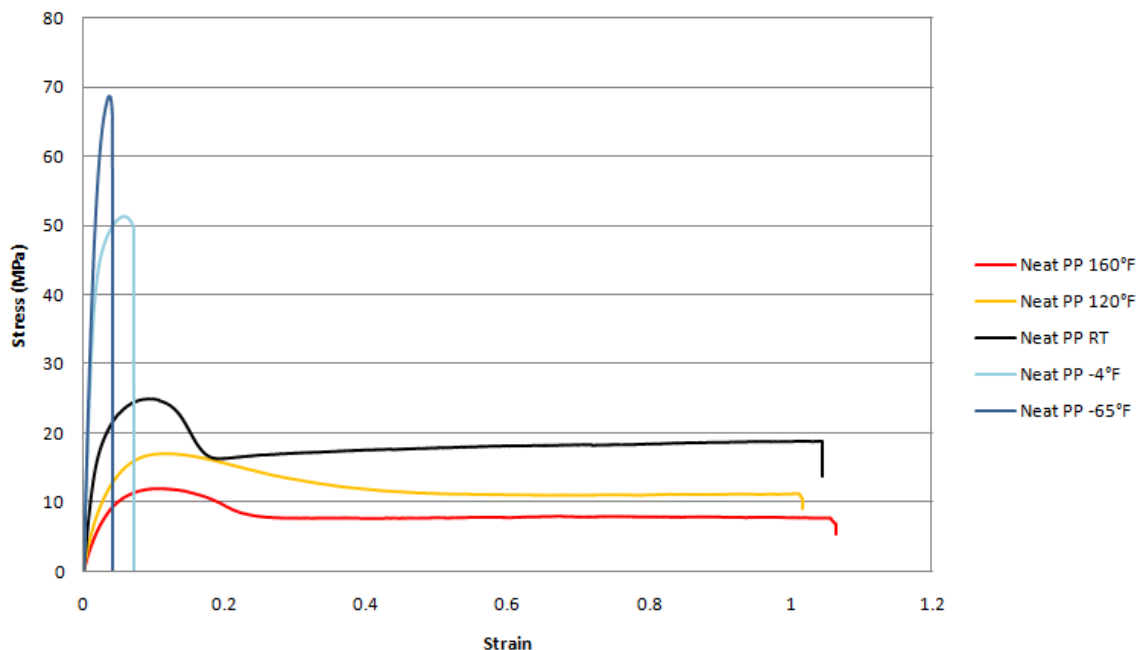


Figure 3.42 Comparison of stress - strain curves of neat PP 3371 specimens obtained at $-65^{\circ}\text{F}(-54^{\circ}\text{C})$, $-4^{\circ}\text{F}(-20^{\circ}\text{C})$, RT, $120^{\circ}\text{F}(49^{\circ}\text{C})$ and $160^{\circ}\text{F}(71^{\circ}\text{C})$

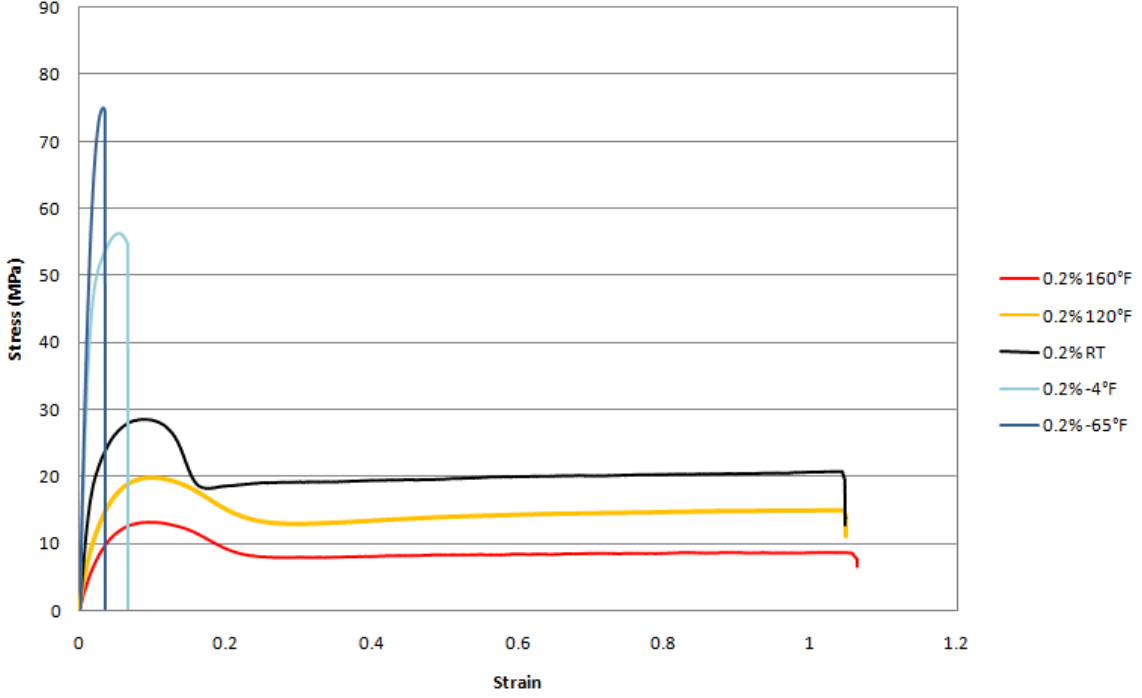


Figure 3.43 Comparison of stress - strain curves of PP 3371 specimens with 0.2% nanoclay reinforcement obtained at -65°F (-54°C), -4°F (-20°C), RT, 120°F (49°C)

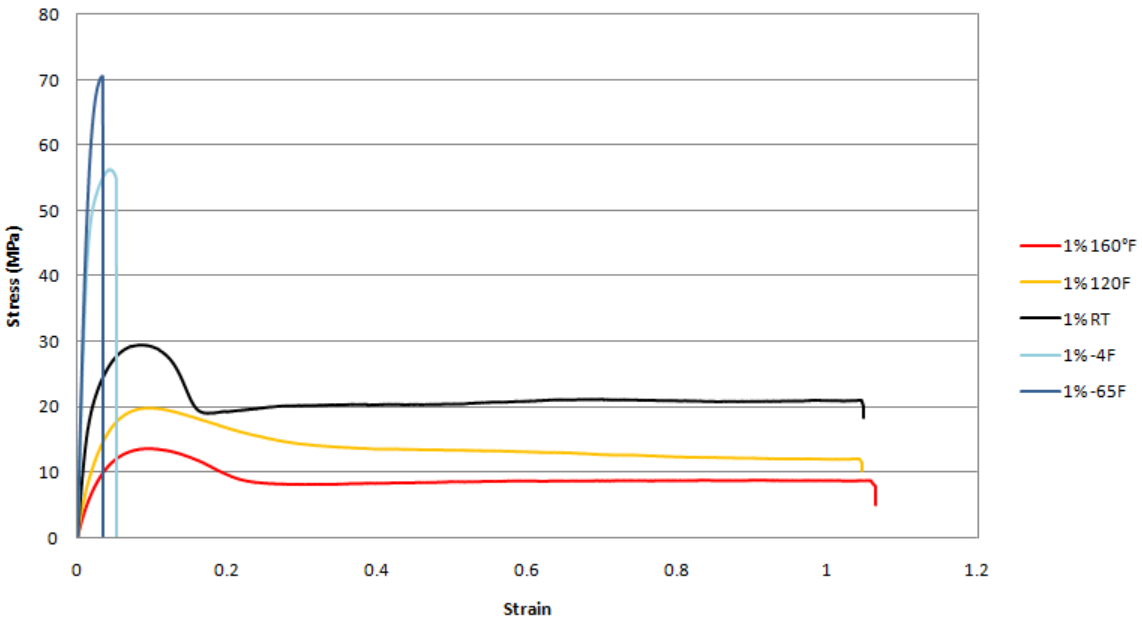


Figure 3.44 Comparison of stress - strain curves of PP 3371 specimens with 1% nanoclay reinforcement obtained at -65°F (-54°C), -4°F (-20°C), RT, 120°F (49°C)

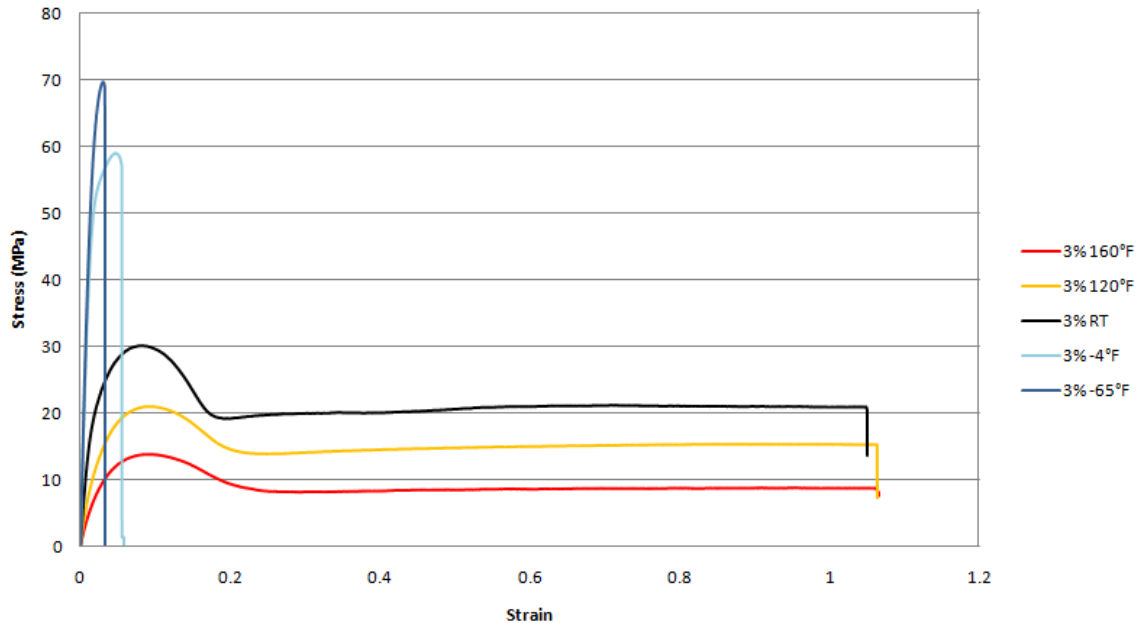


Figure 3.45 Comparison of stress-strain curves of PP 3371 specimens with 3% nanoclay reinforcement obtained at -65°F (-54°C), -4°F (-20°C), RT, 120°F (49°C)

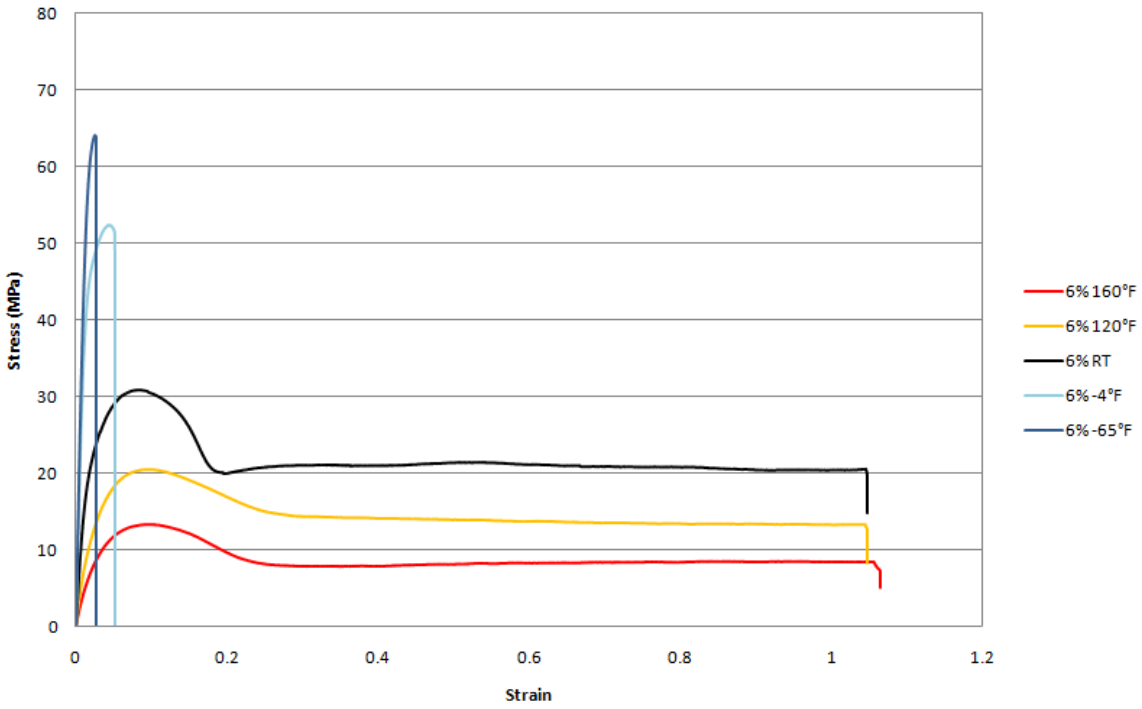


Figure 3.46 Comparison of stress-strain curves of PP 3371 specimens with 6% nanoclay reinforcement obtained at -65°F (-54°C), -4°F (-20°C), RT, 120°F (49°C)

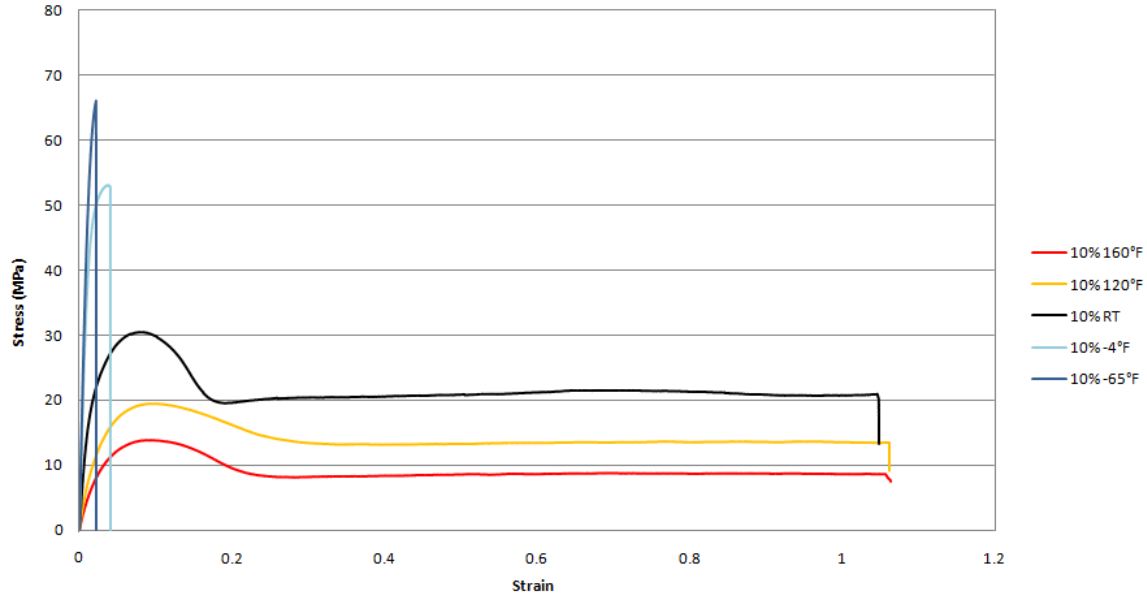


Figure 3.47 Comparison of stress-strain curves of PP 3371 specimens with 10% nanoclay reinforcement obtained at -65°F (-54°C), -4°F (-20°C), RT, 120°F (49°C)

As can be deduced from the graphs, high temperature has a significant and deleterious effect on the material properties. As the temperature increases, the Young's modulus and the ultimate tensile stress of the nanocomposite decrease. However, as the temperature decreases, the Young's modulus and ultimate tensile stress increase. At low temperatures the material becomes brittle and fails at relatively low strains. The results are summarized in Table 3.6.

Table 3.6 Material properties of PP 3371 specimens with 0%, 0.2% 1%, 3%, 6% and 10% nanoclay reinforcement at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C)

PP 3371	Temperature	Young's Modulus (GPa)	Poisson's Ratio	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
0%	-65°F (-54°C)	3.866	0.3429	63.9034	0.0327	0.0357
	-4 °F (-20°C)	3.346	0.3438	50.607	0.0517	0.0642
	Room Temp.	1.200	0.3947	25.122	0.0745	1.0233
	120 °F (49°C)	0.616	0.4038	17.166	0.0919	1.0021
	160 °F (71°C)	0.392	0.4256	11.730	0.0819	1.0360
0.2%	-65°F (-54°C)	4.284	0.3283	68.5570	0.0325	0.0372
	-4 °F (-20°C)	3.713	0.3307	53.8378	0.0491	0.0619
	Room Temp.	1.317	0.4167	27.826	0.0668	1.0258
	120 °F (49°C)	0.728	0.4350	19.087	0.0788	1.0355
	160 °F (71°C)	0.449	0.4782	12.997	0.0725	1.0308
1%	-65°F (-54°C)	4.603	0.3193	65.1837	0.0254	0.0259
	-4 °F (-20°C)	4.008	0.3156	56.3626	0.0431	0.0523
	Room Temp.	1.508	0.4056	29.277	0.0661	1.0268
	120 °F (49°C)	0.765	0.4631	19.583	0.0772	1.0345
	160 °F (71°C)	0.465	0.4715	13.303	0.0732	1.0341
3%	-65°F (-54°C)	4.694	0.3076	67.0177	0.0247	0.0272
	-4 °F (-20°C)	4.179	0.3114	55.5831	0.0433	0.0518
	Room Temp.	1.628	0.3852	29.900	0.0653	1.0397
	120 °F (49°C)	0.828	0.4355	19.954	0.0789	1.0336
	160 °F (71°C)	0.497	0.4523	13.567	0.0713	1.0354
6%	-65°F (-54°C)	4.973	0.3039	61.5911	0.0203	0.0203
	-4 °F (-20°C)	4.212	0.3056	52.3178	0.0409	0.0471
	Room Temp.	1.646	0.3722	30.009	0.0659	1.0332
	120 °F (49°C)	0.833	0.4334	19.863	0.0759	1.0248
	160 °F (71°C)	0.503	0.5077	13.396	0.0774	1.0374
10%	-65°F (-54°C)	5.092	0.2983	60.2389	0.0192	0.0194
	-4 °F (-20°C)	4.333	0.2928	54.0557	0.0367	0.0427
	Room Temp.	1.742	0.3725	30.073	0.0658	1.0331
	120 °F (49°C)	0.866	0.4111	19.981	0.0759	1.0353
	160 °F (71°C)	0.515	0.4896	13.431	0.0791	1.0338

3.2.2-f) Variation of Young's Modulus & Poisson's Ratio with Temperature

After analyzing all tensile test results for each nanoclay reinforcement value and temperature, the variation of Young's modulus with temperature is presented for all reinforcement percentages to obtain a better understanding of the behavior of nanoclay reinforced composites. Additionally, the variation of the Poisson's ratio with temperature is presented. Figures 3.48 and 3.49 show the variation of Young's modulus and Poisson's ratio respectively with temperature for each reinforcement percentage.

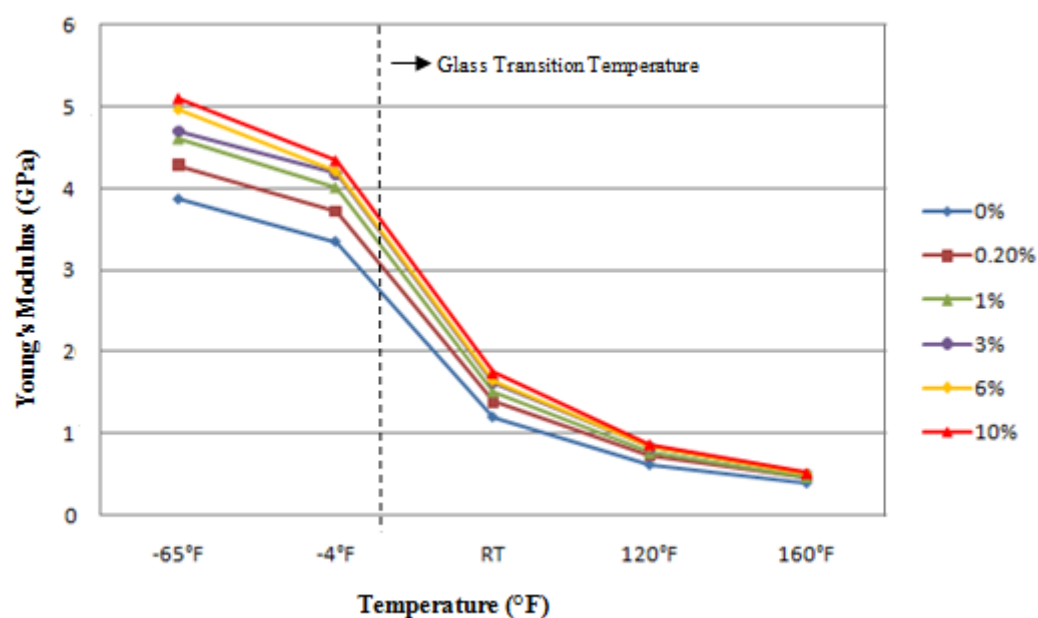


Figure 3.48 Variation of Young's modulus with temperature

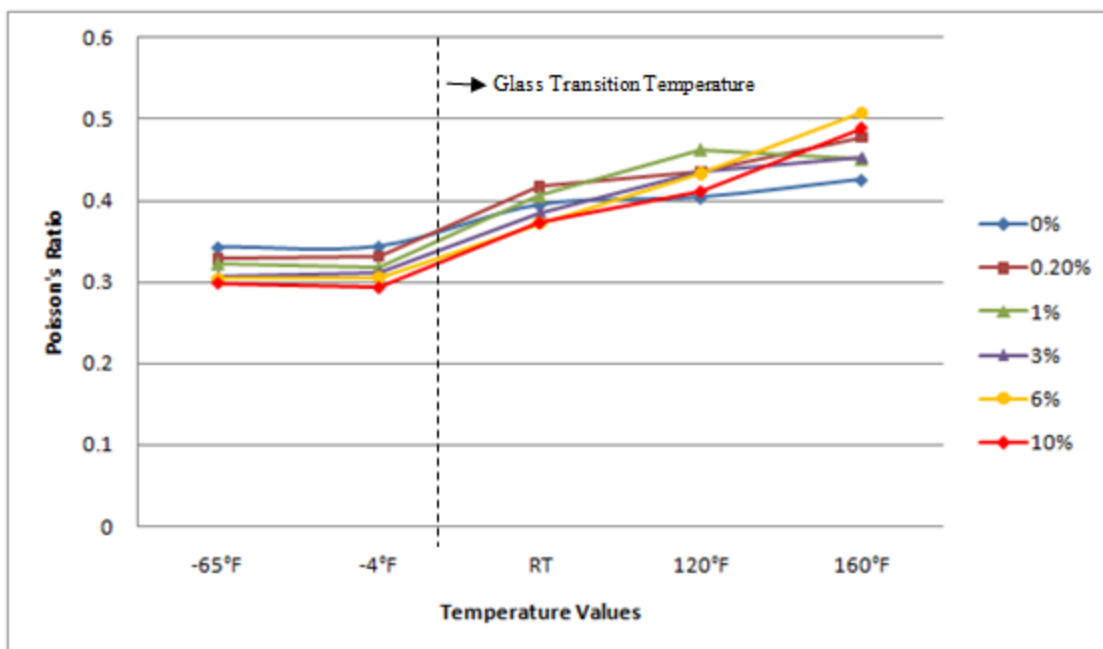


Figure 3.49 Variation of Poisson's ratio with temperature

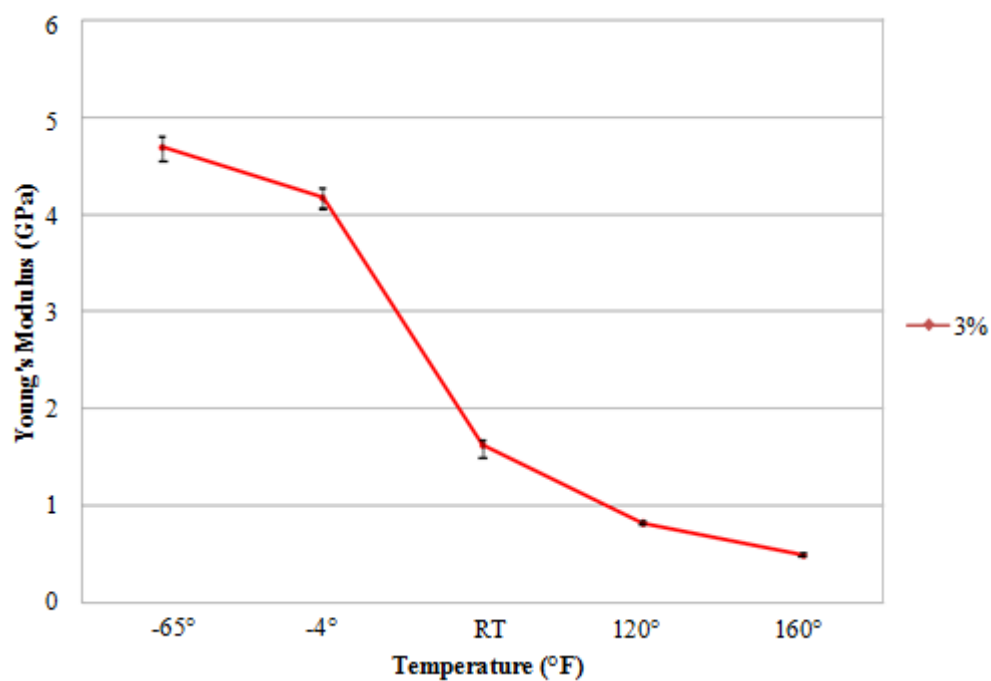


Figure 3.50 Range of Young's modulus for PP 3371 with 3% nanoclay reinforcement

Figure 3.48 clearly shows that the Young's modulus increases with increasing nanoclay reinforcement percentage and decreases as the temperature increases. However, for the Poisson's ratio Figure 3.49 indicates that while it increases with increasing temperature, its variation with reinforcement percentage is not so obvious.

The same figures also show that as the temperature crosses the glass transition temperature of PP there is significant decrease in the Young's modulus for all test specimens indicating large and rubbery deformations. The jump in Poisson's ratio does not appear to be as significant.

The results are also summarized in Table 3.7 which shows the average values of Young's modulus for each nanoclay reinforcement percentage at each temperature and Table 3.8 which shows the average values of Poisson's ratio for each nanoclay reinforcement percentage at each temperature.

Table 3.7 Average Young's Moduli (GPa)

	-65F	-4F	RT	120F	160F
0%	3.866	3.346	1.200	0.616	0.392
0.2%	4.284	3.713	1.317	0.728	0.449
1%	4.603	4.008	1.508	0.765	0.465
3%	4.694	4.179	1.628	0.828	0.497
6%	4.973	4.212	1.646	0.833	0.503
10%	5.092	4.333	1.742	0.866	0.515

Table 3.8 Average Poisson's Ratios

	-65F	-4F	RT	120F	160F
0%	0.3429	0.3438	0.3947	0.4038	0.4256
0.2%	0.3283	0.3307	0.4167	0.435	0.4782
1%	0.3224	0.3176	0.4056	0.4631	0.4516
3%	0.3076	0.3114	0.3852	0.4355	0.4523
6%	0.3039	0.3056	0.3722	0.4334	0.5077
10%	0.2983	0.2928	0.3725	0.4111	0.4896

3.2.2-g) Variation of Young's Modulus & Poisson's Ratio with Nanoclay Reinforcement Percentage

The results are also plotted with varying nanoclay reinforcement percentage for each test temperature (Figure 3.51 and 3.52).

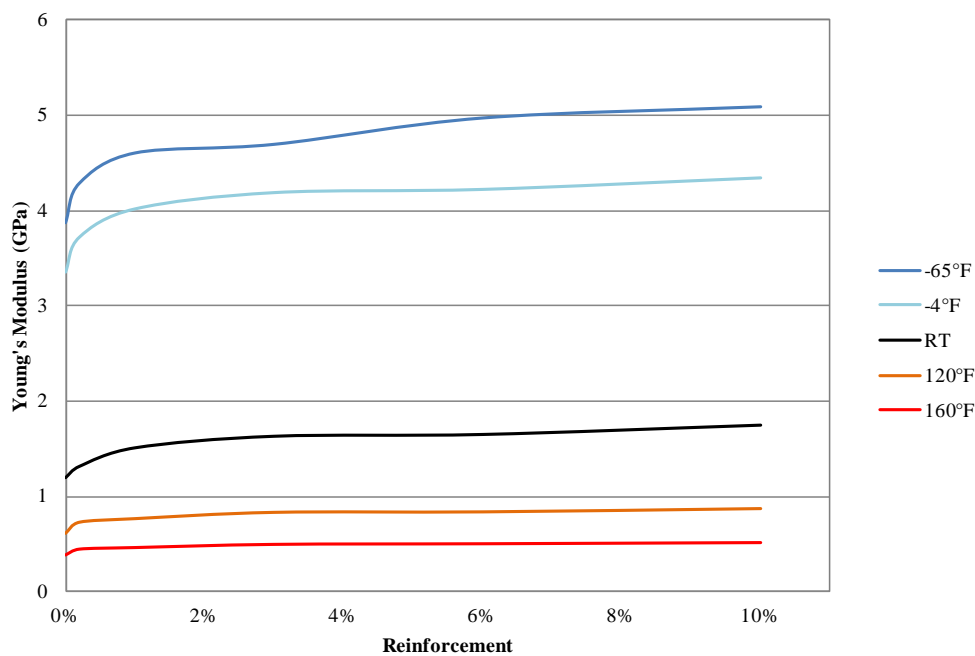


Figure 3.51 Variation of Young's modulus with nanoclay reinforcement percentage at various test temperatures

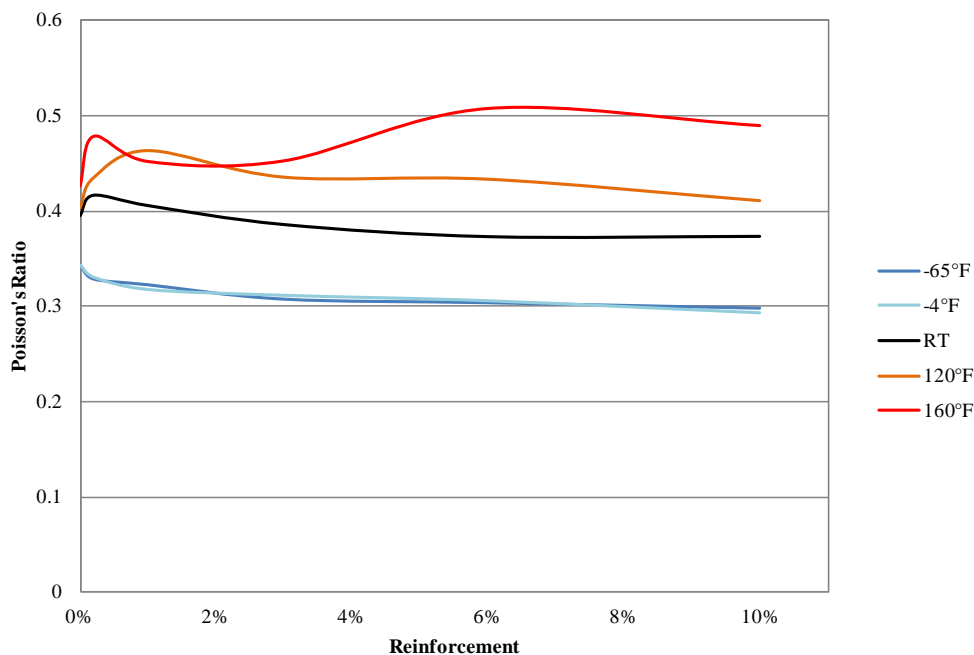


Figure 3.52 Variation of Poisson's ratio with nanoclay reinforcement percentage at various test temperatures

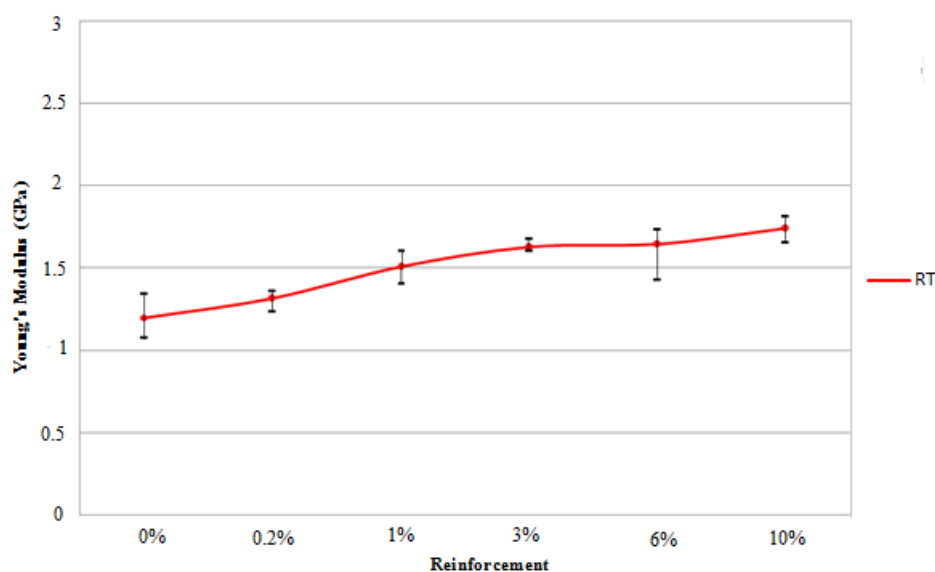


Figure 3.53 Range of Young's modulus test results vs. reinforcement at room temperature

In Figure 3.51, we can observe the drastic effect of temperature on Young's modulus. When the temperature decreases the Young's modulus shows a dramatic rise at any given reinforcement percentage. However, as was mentioned before, partly because of the glass transition effect and

partly because of the difficulties in obtaining the Poisson's ratio experimentally, as can be seen in Figure 3.52 a consistent trend is difficult to discern.

3.2.3 Effect of different resins

In addition to Polypropylene (PP) 3371 resin, we also tested 3% nanoclay reinforced specimens with three different resins, namely, Borealis, Total Petrochemical (TP) 3868 and EPON 828 Epoxy. These specimens were also manufactured by Novus Technologies Corporation.

3.2.3-a) Testing of Borealis specimens with 3% nanoclay reinforcement at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C)

Borealis specimens with 3% nanoclay reinforcement were subjected to tensile test at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C) and the resulting stress-strain curves are plotted in Figures 3.54 – 3.58 and also summarized in Table 3.9. Borealis specimens with 3% nanoclay reinforcement show large deformation similar to PP 3371 specimens with 3% nanoclay reinforcement at room temperature, 120°F (49°C) and 160°F (71°C). Furthermore, these specimens failed too at relatively low strains at low temperatures. The shapes of the curves are also remarkably similar. However, Borealis specimens with 3% nanoclay have higher ultimate tensile stress than PP 3371 specimens.

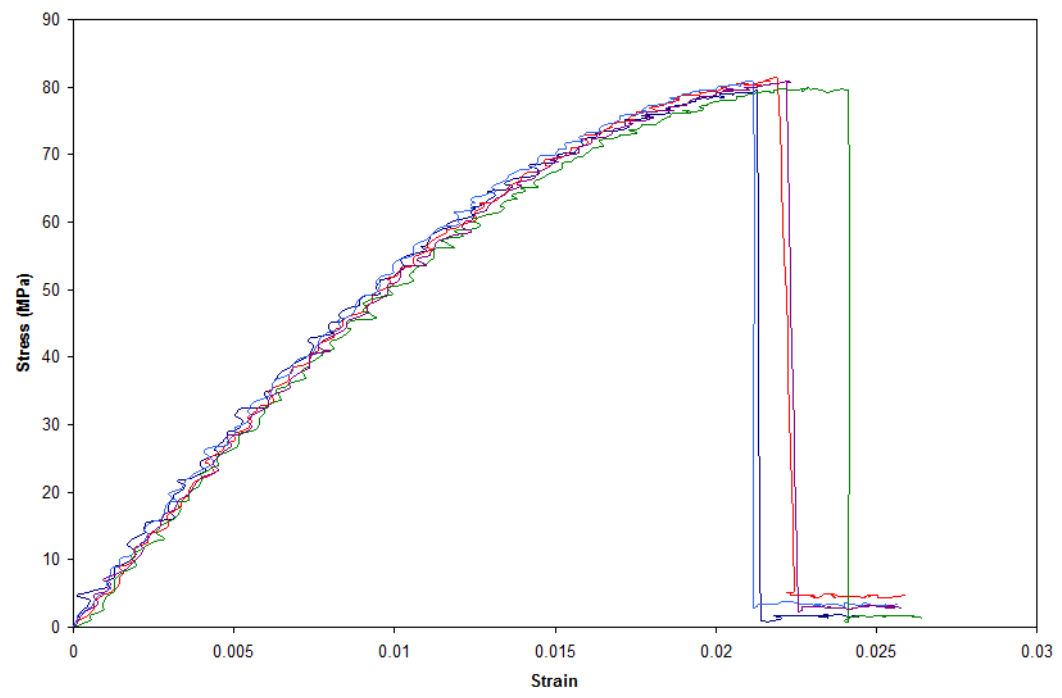


Figure 3.54 Stress - strain curves of Borealis resin specimens with 3% nanoclay reinforcement at -65°F (-54°C)

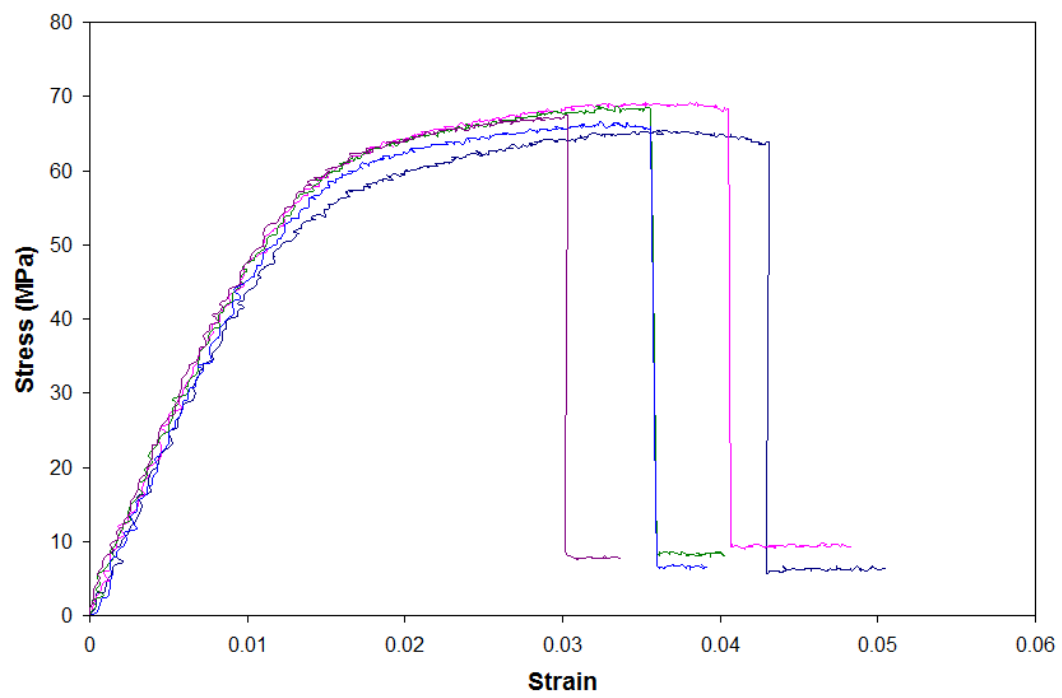


Figure 3.55 Stress - strain curves of Borealis resin specimens with 3% nanoclay reinforcement at -4°F (-20°C)

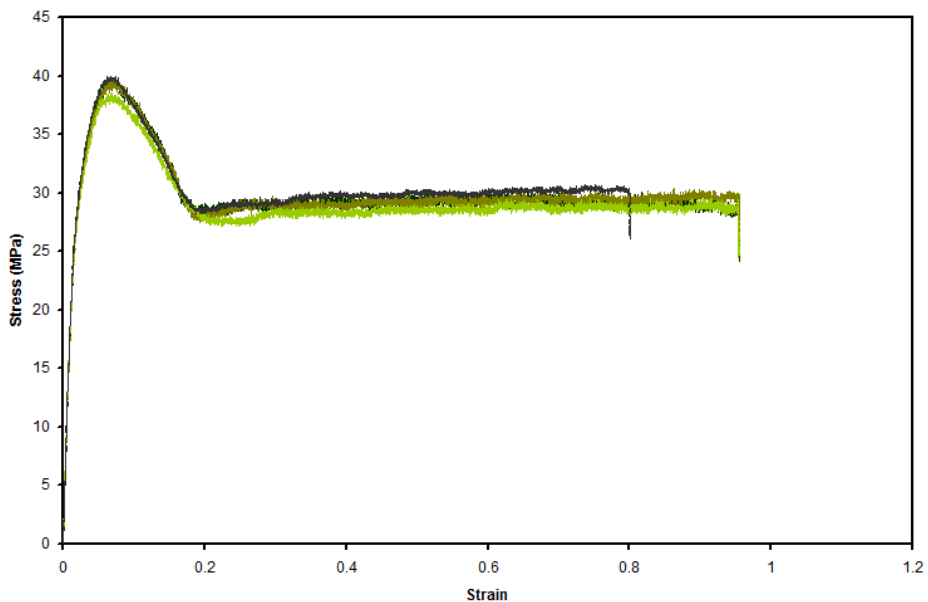


Figure 3.56 Stress - strain curves of Borealis resin specimens with 3% nanoclay reinforcement at room temperature

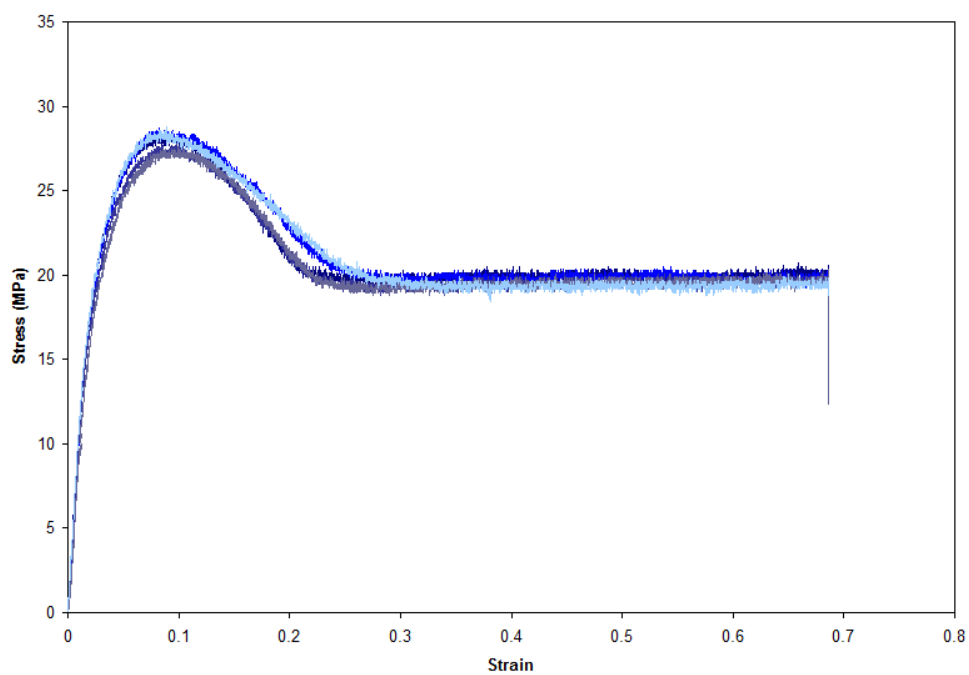


Figure 3.57 Stress - strain curves of Borealis resin specimens with 3% nanoclay reinforcement at 120°F (49°C)

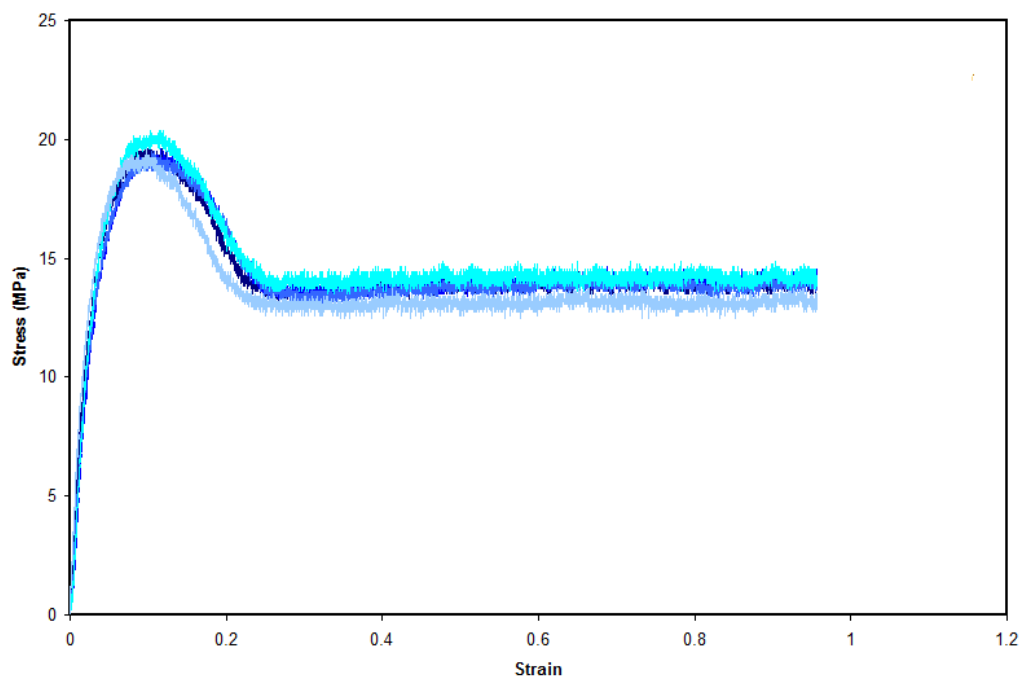


Figure 3.58 Stress - strain curves of Borealis resin specimens with 3% nanoclay reinforcement at 160°F (71°C)

Table 3.9 Average material properties of Borealis specimens with 3% nanoclay reinforcement at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C)

Borealis Specimens	Temperature	Young's Modulus (GPa)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
3%	-65°F	5.704	80.140	0.022	0.022
	-4°F	5.109	67.173	0.033	0.037
	Room Temp.	1.965	38.944	0.073	0.954
	120°F	1.020	27.914	0.091	0.951
	160°F	0.569	19.405	0.106	0.951

3.2.3-b) Testing of Total Petrochemical 3868 specimens with 3% nanoclay reinforcement at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C)

TP 3868 specimens with 3% nanoclay reinforcement were subjected to tensile test to obtain the properties at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C). The stress - strain curves of TP 3868 specimens are shown in Figures 3.59 – 3.63 and the results are summarized in Table 3.10. The TP 3868 specimens with 3% nanoclay reinforcement failed at relatively low strains compared to PP 3371 and Borealis specimens at low and room temperatures. However at high temperatures TP 3868 loses its brittleness and behaves like a ductile material.

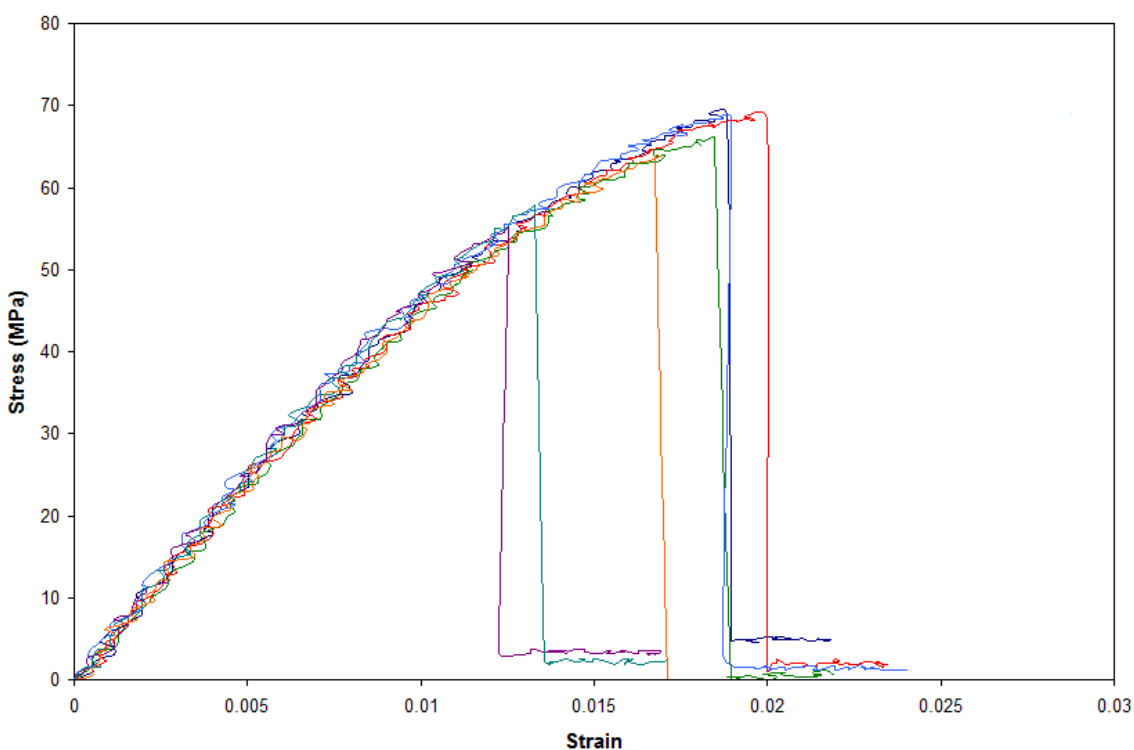


Figure 3.59 Stress - strain curves of TP 3868 specimens with 3% nanoclay reinforcement at -65°F (-54°C)

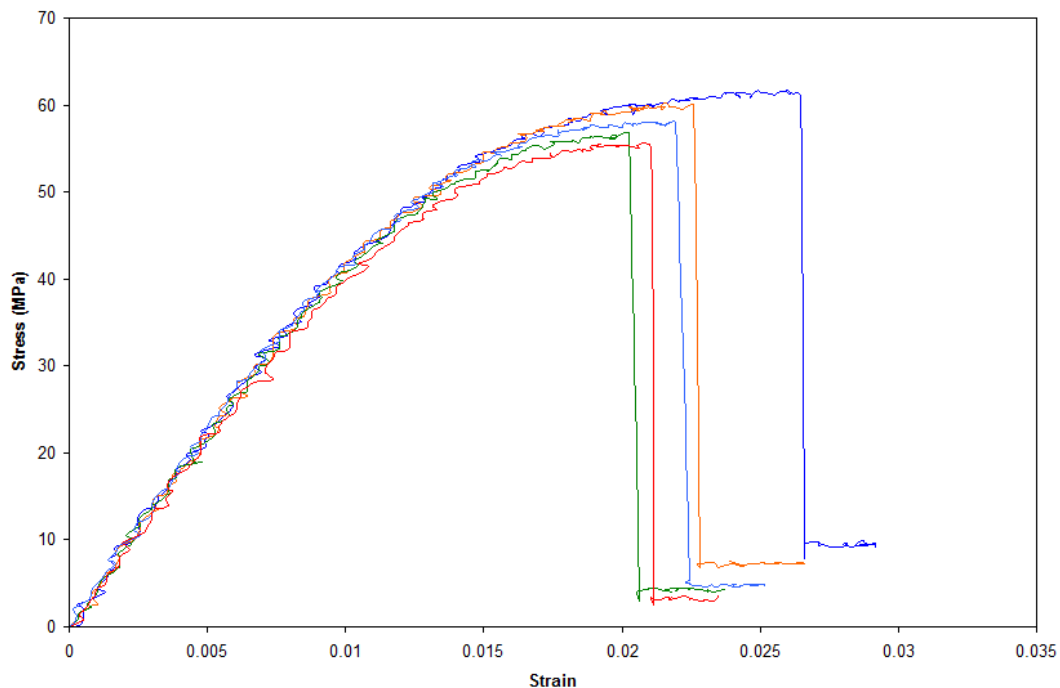


Figure 3.60 Stress - strain curves of TP 3868 specimens with 3% nanoclay reinforcement at -4°F (-20°C)

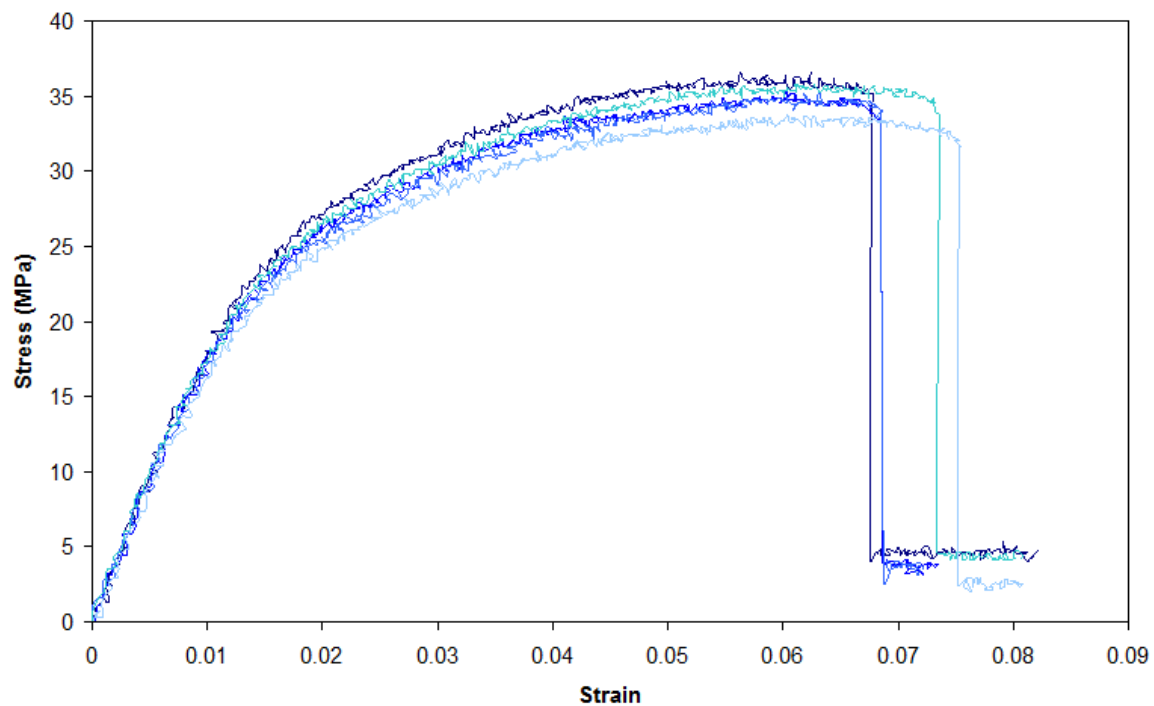


Figure 3.61 Stress - strain curves of TP 3868 specimens with 3% nanoclay reinforcement at room temperature

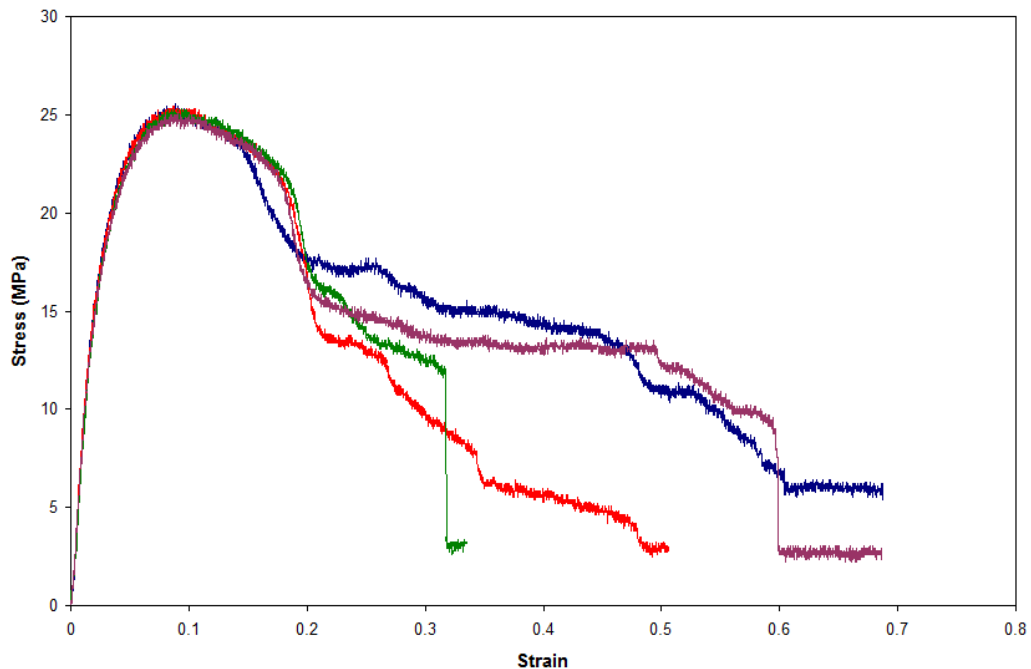


Figure 3.62 Stress - strain curves of TP 3868 specimens with 3% nanoclay reinforcement at 120°F (49°C)

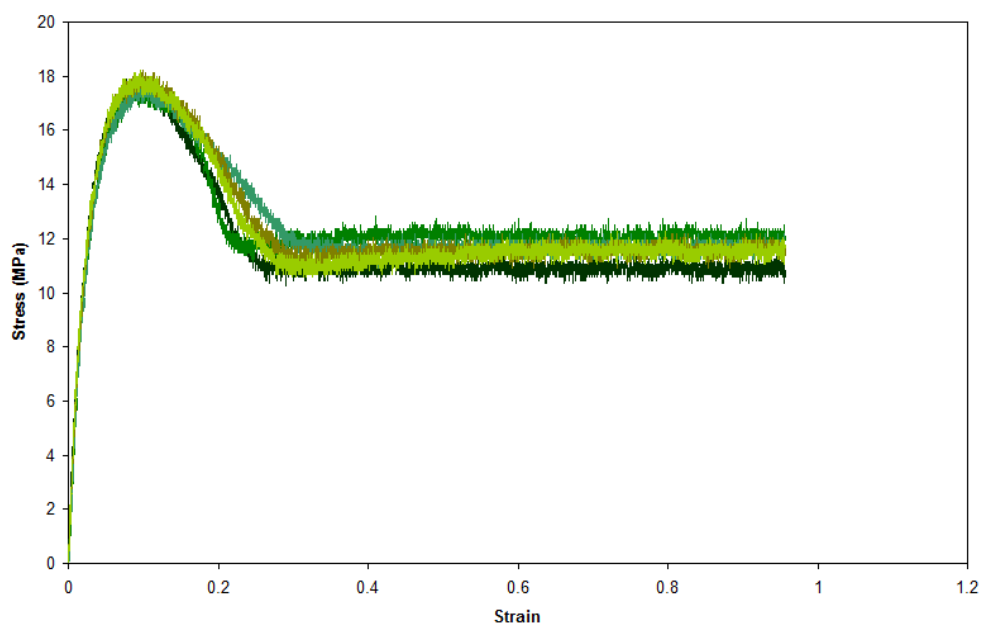


Figure 3.63 Stress - strain curves of TP 3868 specimens with 3% nanoclay reinforcement at 160°F (71°C)

Table 3.10 Average material properties of TP 3868 specimens with 3% nanoclay reinforcement at -65°F (-54°C), -4°F (-20°C), room temperature, 120°F (49°C) and 160°F (71°C)

TP 3868 Specimens	Temperature	Young's Modulus (GPa)	Ultimate Stress (MPa)	Strain at Ultimate Stress	End of Test Strain
3%	-65°F	4.858	64.157	0.017	0.017
	-4°F	4.423	58.192	0.022	0.022
	Room Temp.	1.875	34.783	0.061	0.071
	120°F	0.830	25.0997	0.093	0.526
	160°F	0.587	17.671	0.099	0.945

3.2.3-c) Testing of EPON 828 Epoxy specimens with 0%, 1%, 3%, 6% and 10% nanoclay reinforcement at room temperature

The EPON 828 epoxy specimens with various nanoclay reinforcements were subjected to room temperature tensile loading at a crosshead speed of 1 mm/min in an MTS machine. In addition, some specimens were instrumented with strain gages to obtain the Poisson's ratio. The room-temperature stress-strain curves of EPON 828 specimens with various nanoclay reinforcements were then obtained using time-history readings from load-cell for the stresses and crosshead motion, extensometer and strain gages for the strains. The stress-strain curves were then plotted for each EPON 828 specimen. Key mechanical properties, such as Young's modulus, ultimate tensile strength, strain at ultimate stress, Poisson's ratio etc., were then obtained from these curves and tabulated. For tensile tests five specimens were used for each case and the stress-strain curves are given in Figures 3.64- 3.68.

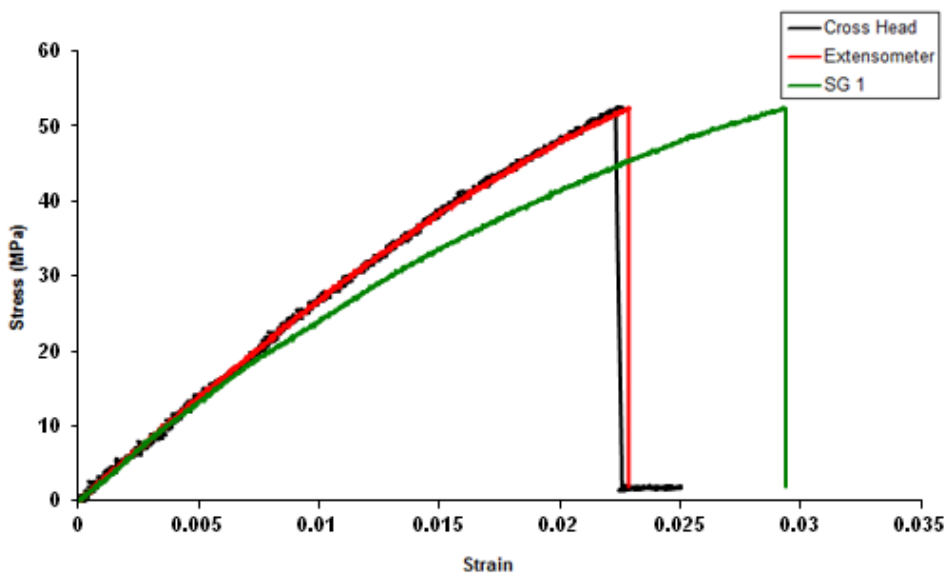


Figure 3.64 Stress - strain curves of neat EPON 828 specimens at room temperature

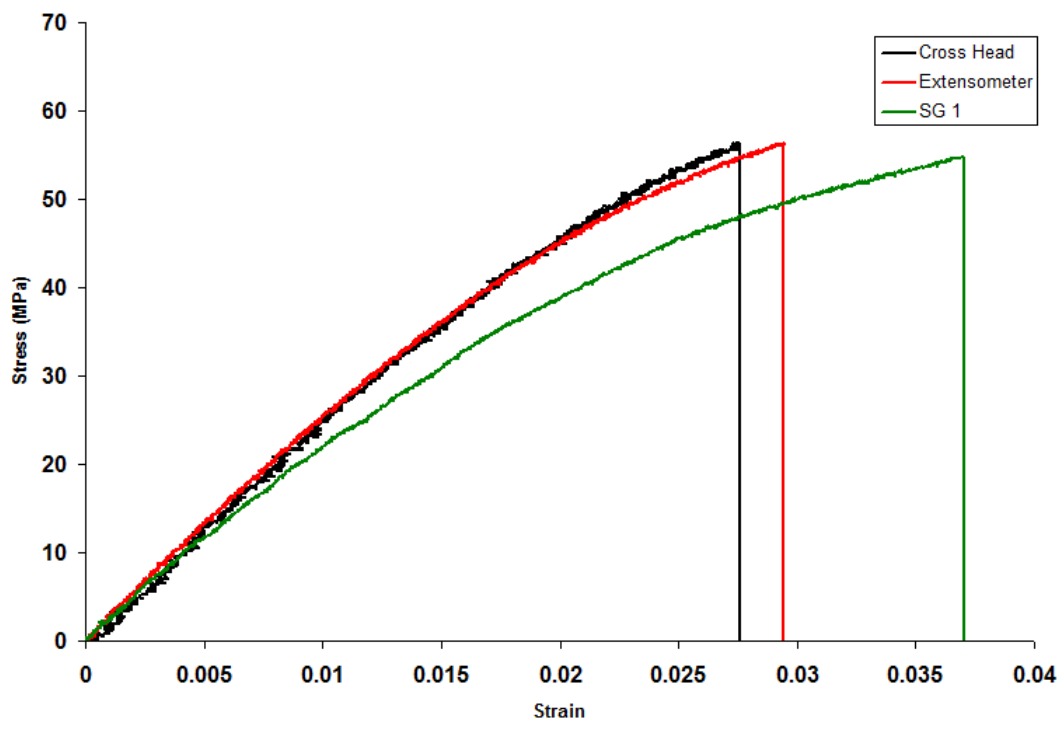


Figure 3.65 Stress - strain curves of EPON 828 specimens with 1% nanoclay reinforcement

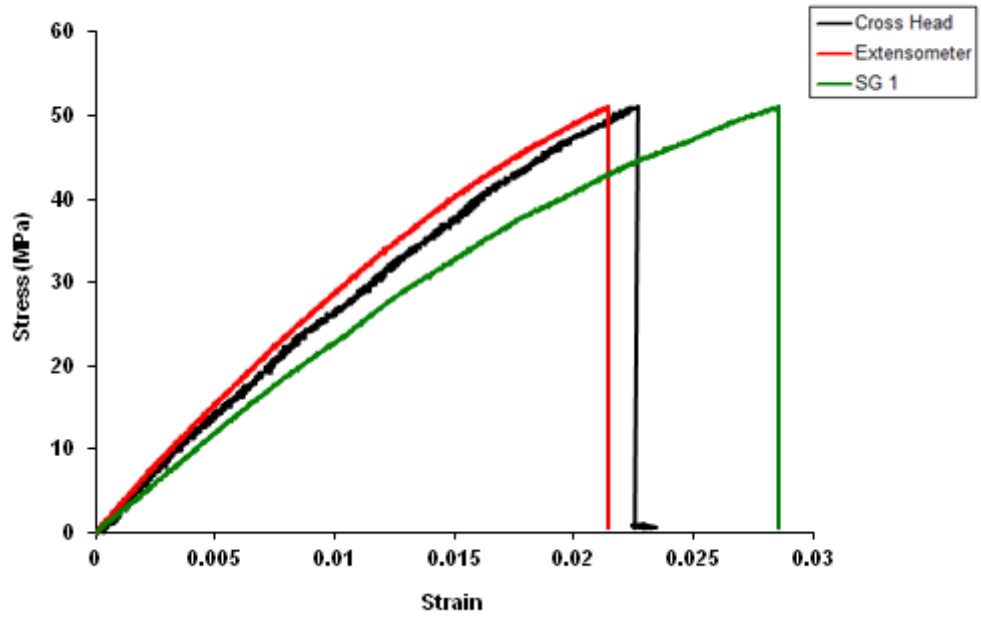


Figure 3.66 Stress - strain curves of EPON 828 specimens with 3% nanoclay reinforcement

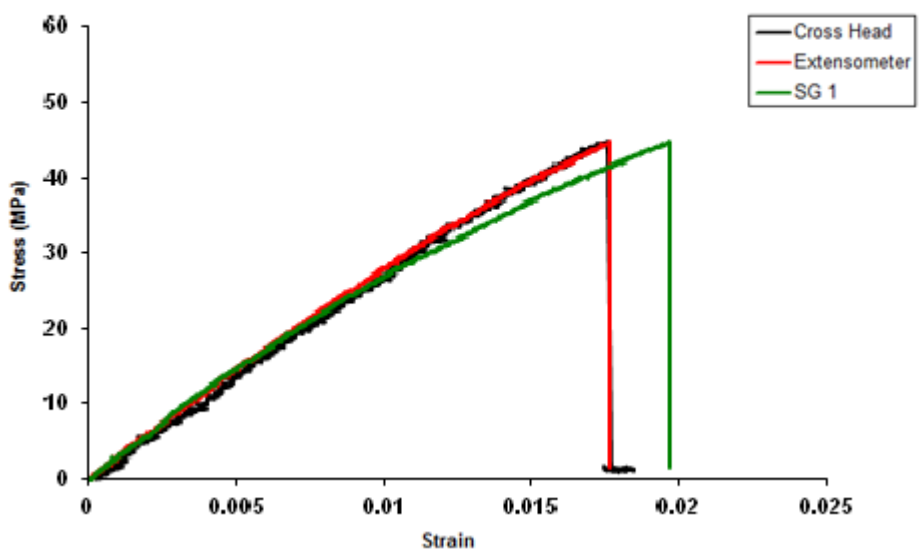


Figure 3.67 Stress - strain curves of EPON 828 specimens with 6% nanoclay reinforcement

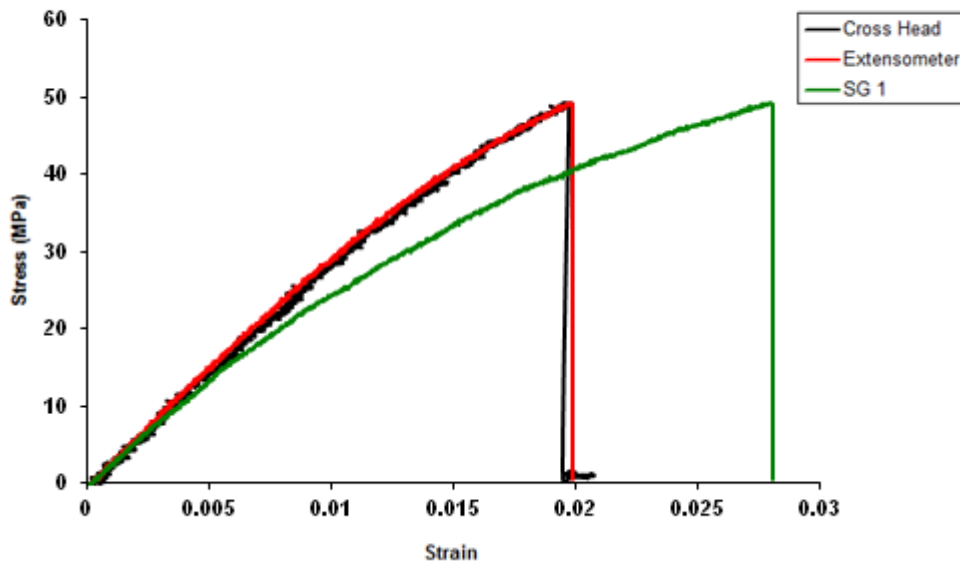


Figure 3.68 Stress - strain curves of EPON 828 specimens with 10% nanoclay reinforcement

Various percentages of nanoclay reinforcement were used to ascertain its effect on the mechanical properties of the nanocomposite. Figure 3.69 shows the comparison of the stress - strain curves of EPON 828 epoxy specimens with 0%, 1%, 3%, 6% and 10% nanoclay reinforcement. Since the curves obtained with the extensometer were more consistent, they are used in Figure 3.69.

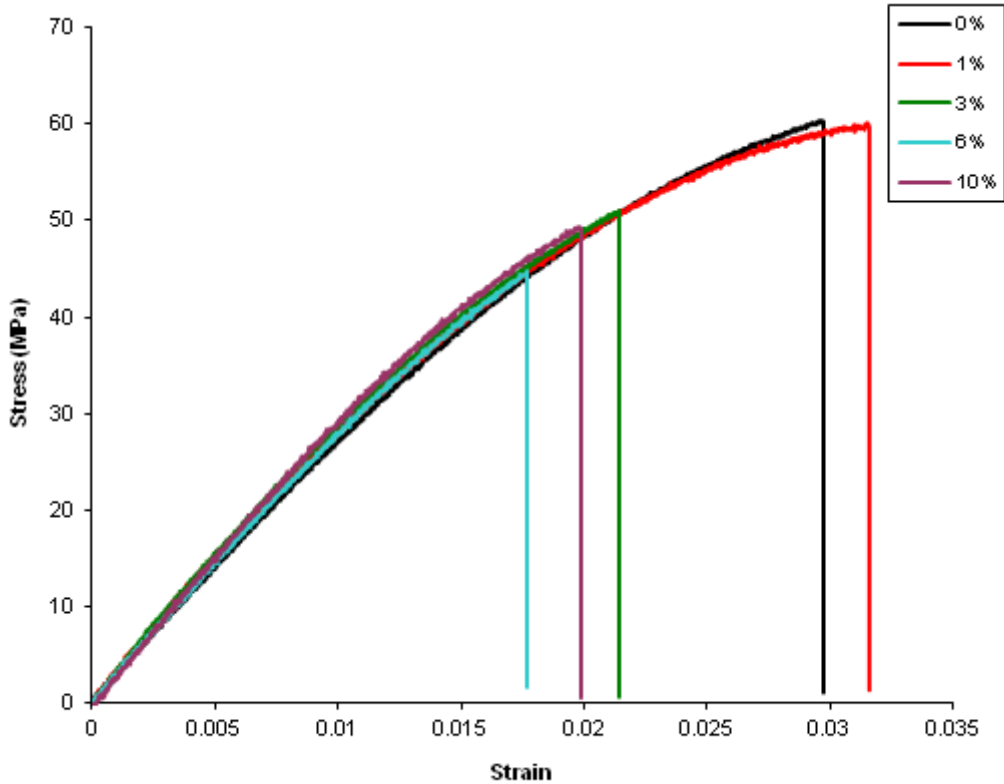


Figure 3.69 Comparison of stress-strain curves of EPON 828 specimens with various nanoclay reinforcement percentages

The results from Figure 3.69 indicate that in general with increased nanoclay reinforcement, both the failure strain and the failure stress of the nanocomposites decrease.

All the test results and the average material properties are given in Table 3.11 and Table 3.12, respectively. Table 3.12 also includes data available from the manufacturer of the epoxy. As seen in Table 3.11, the average Young's modulus obtained from crosshead, extensometer and strain gage techniques are close to each other, as well as to the manufacturer's data. But since the test results obtained with extensometer are more reliable, they are used in calculating the average Young's modulus for each case.

Table 3.11 Material properties of EPON 828 specimens with 0%, 1%, 3%, 6% and 10% nanoclay reinforcement at room temperature

EPON 828 with	#	Young's Modulus (GPa) – Cross Head	Young's Modulus (GPa) – Extensometer	Young's Modulus (GPa) – Strain Gage	NOVUS Test Result (GPa)	Young's Modulus (GPa) (Manufacturer)	Poisson's Ratio
0%	1	2.673	-	-	2.289	2.758	-
	2	2.743	2.789	2.589			0.2915
	3	2.760	2.917	2.837			0.3097
	4	2.725	2.747	2.695			0.3303
1%	1	2.672	-	-	-	-	-
	2	2.617	2.918	-	-	-	-
	3	2.986	2.986	-	-	-	-
	4	2.650	3.199	2.613	-	-	0.3459
	5	2.478	2.755	2.361	-	-	0.3237
	6	2.258	2.683	2.420	-	-	0.3235
	7	2.869	2.935	2.327	-	-	0.3020
3%	1	2.890	3.235	2.393	-	-	0.3407
	2	2.674	-	2.508	-	-	0.3285
	3	2.614	2.995	2.391	-	-	0.3374
	4	2.292	2.410	2.256	-	-	0.3087
	5	2.838	3.296	-	-	-	-
6%	1	2.464	2.458	-	-	-	-
	2	2.540	2.935	-	-	-	-
	3	2.215	3.426	2.403	-	-	-
	4	2.645	3.020	3.063	-	-	0.3488
	5	3.064	3.383	2.630	-	-	0.3316
	6	3.290	3.357	2.613	-	-	0.3317
10%	1	3.070	3.440	-	-	-	-
	2	2.768	-	-	-	-	-
	3	2.644	2.832	-	-	-	-
	4	3.012	3.446	-	-	-	-
	5	3.147	3.438	-	-	-	-
	6	2.841	4.289	2.779	-	-	0.3538
	7	3.621	3.562	2.922	-	-	0.3288
	8	2.811	2.736	2.607	-	-	0.3074

Table 3.12 Average material properties of EPON 828 specimens with 0%, 1%, 3%, 6% and 10% nanoclay reinforcement at room temperature

EPON 828	Young's Modulus (GPa) (Average) - Extensometer	Maximum Strength (MPa)	Failure Strain (ϵ_f)	Young's Modulus NOVUS Test Result (GPa)	Young's Modulus (GPa) (Manufacturer)	Poisson's Ratio ν_{12}
0%	2.8177	59.06	0.032	2.289	2.758	0.3105
1%	2.9127	58.84	0.032	-	-	0.3238
3%	2.9840	50.99	0.022	-	-	0.3288
6%	3.0965	38.97	0.014	-	-	0.3340
10%	3.3427	37.74	0.014	-	-	0.3348

3.2.4 Comparison of Test Results for Various Resins

We have the test results for Borealis and TP 3868 specimens with 3% nanoclay reinforcement at various temperatures and also the test results for EPON 828 at room temperature at different nanoclay reinforcement percentages. Figure 3.70 shows the comparison of stress vs. strain curves for the four different resins with 3% nanoclay reinforcement at room temperature.

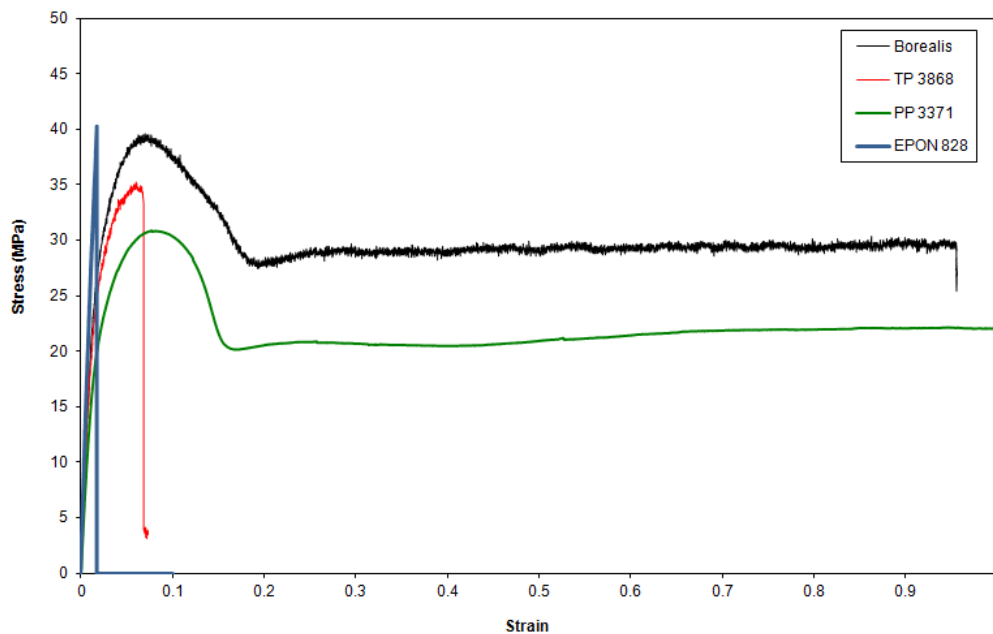


Figure 3.70 Comparison of stress-strain curves of PP 3371, Borealis, TP 3868 and EPON 828 specimens with 3% nanoclay reinforcement at room temperature

Figure 3.70 indicates that our main resin, PP 3371, is the most ductile material at room temperature when compared with the other resins. Epoxy EPON 828 shows the most brittle behavior at room temperature.

The comparison of the stress-strain curves for the additional PP based resins, namely Borealis and TP 3868, at various temperatures are displayed in Figures 3.71 and 3.72 respectively. It is noted that at low temperatures both resins become brittle and at higher temperatures are more or less ductile. It is also observed that at the same temperature Borealis is relatively more ductile than TP3868.

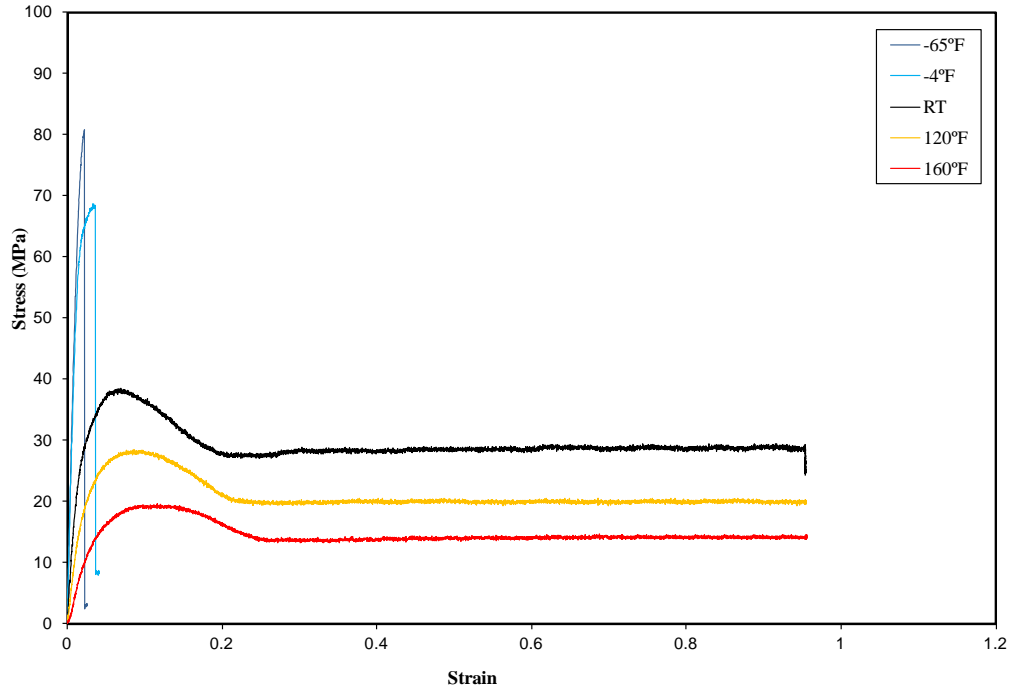


Figure 3.71 Comparison of stress-strain curves of Borealis specimens with 3% nanoclay reinforcement at various temperatures

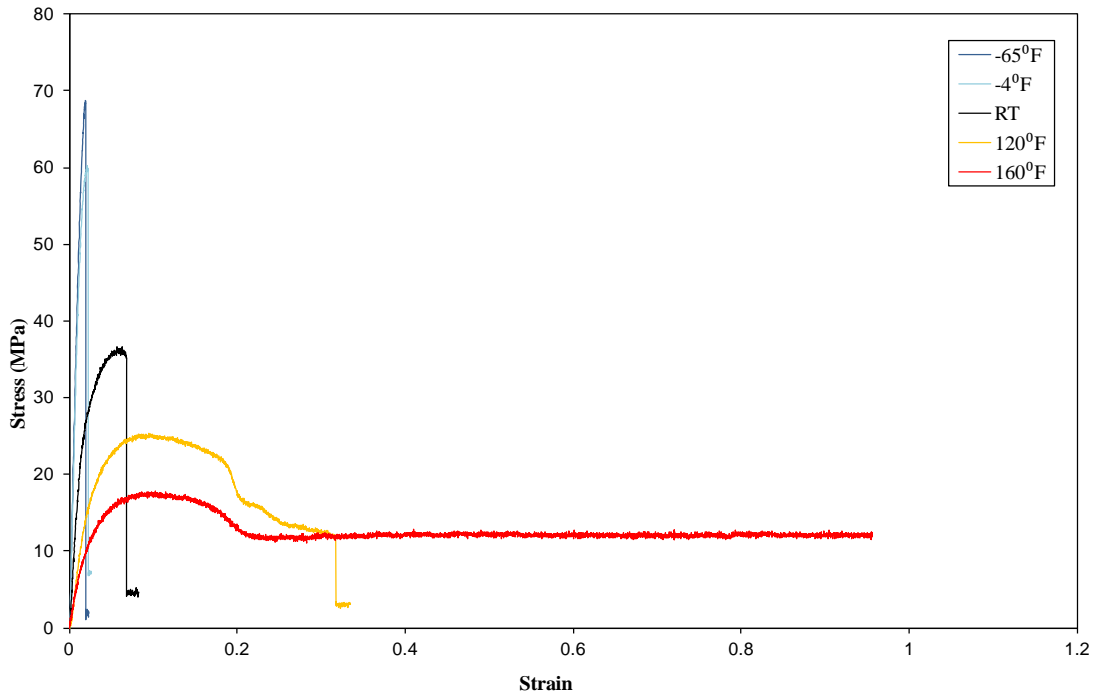


Figure 3.72 Comparison of stress-strain curves of TP 3868 specimens with 3% nanoclay reinforcement at various temperatures

3.3. LOW VELOCITY IMPACT TESTS

Next, low velocity (drop weight) impact tests were carried out at an energy level appropriate for reinforced plastics and various temperatures. PP 3371 specimens with 0%, 0.2%, 1%, 3%, 6% and 10% nanoclay reinforcement were subjected to low velocity impact tests. All specimens had dimensions of 4"x4"×(1/8)". Tests were carried out using an instrumented Instron-Dynatup 8520 drop-weight impact tester.

The results in Table 3.13 show the behavior of specimens at one energy level and -65°F, -4°F, room temperature, 120°F and 160°F. Depending on the drop height, the weight used and the

percentage of nanoclay reinforcement, the following was observed: some specimens were penetrated (P), some cracked (F) and some visually remained virtually intact (NP), that is they were not penetrated. The drop height was 6” and the weight used was 4.126 kg.

Table 3.13 Drop weight tests results at -65°F, -4°F, room temperature, 120°F and 160°F

Drop Height = ~6”

Drop Weight = ~4.126 kg

Nanoclay Reinforcement Percentage	Specimen	Test Results (at RT)	Test Results (at 120°F)	Test Results (at 160°F)	Test Results (at -4°F)	Test Results (at -65°F)
0%	<i>1</i>	NP	NP	NP	P	P
	<i>2</i>	NP	NP	NP	P	P
	<i>3</i>	NP		NP		
	<i>4</i>	NP				
	<i>5</i>	NP				
	<i>6</i>	NP				
	<i>7</i>	NP				
	<i>8</i>	NP				
	<i>9</i>	NP				
	<i>10</i>	NP				
0.2%	<i>1</i>	NP	NP	NP	P	P
	<i>2</i>	F	NP	NP	P	P
	<i>3</i>	P	NP	NP		
	<i>4</i>	P	NP			
	<i>5</i>	NP				
	<i>6</i>	NP				
	<i>7</i>	F				
	<i>8</i>	NP				
	<i>9</i>	P				
	<i>10</i>	NP				

Table 3.13 Continued

1%	1	P	NP	NP	P	P
	2	F	NP	NP	P	P
	3	P				
	4	P				
	5	P				
	6	F				
3%	1	P	NP	NP	<i>Not performed; Penetration assumed</i>	<i>Not performed; Penetration assumed</i>
	2	P	NP	NP		
	3	P		NP		
	4	P				
	5	F				
6%	1	F	NP	NP	<i>Not performed; Penetration assumed</i>	<i>Not performed; Penetration assumed</i>
	2	F	NP	NP		
	3	P				
	4	P				
10%	1	<i>Not performed; Penetration assumed</i>	NP	NP	<i>Not performed- Penetration assumed</i>	<i>Not performed; Penetration assumed</i>
	2		NP	NP		

P: Penetrated

NP: Virtually intact

F: Some cracked

3.3.1 Low Velocity Drop Weight Test Results at Various Temperatures

In Figures 3.73-3.76 we compare the behavior of neat PP specimens at -4°F , -65°F , room temperature, 120°F and 160°F . Neat PP specimens were not penetrated at room temperature and at higher temperatures. However, all were penetrated at the lower temperatures. In case of penetration, the effect is clearly seen in the Force - Displacement and Force - Time (Figures 3.73 and 3.76) curves.

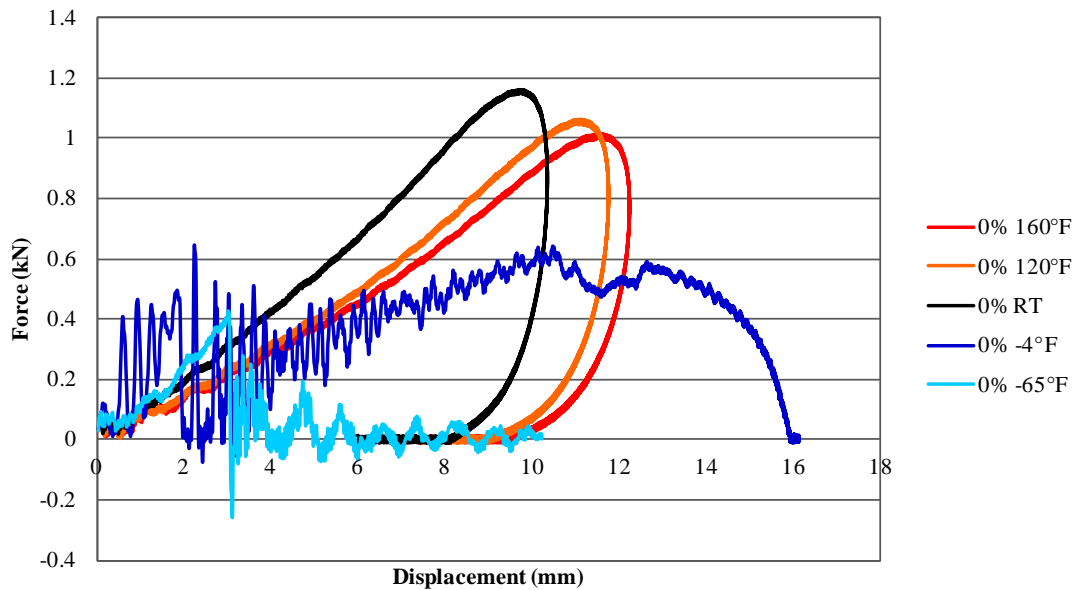


Figure 3.73 Force vs. displacement curves for neat PP specimens at -4°F , -65°F , room temperature, 120°F and 160°F

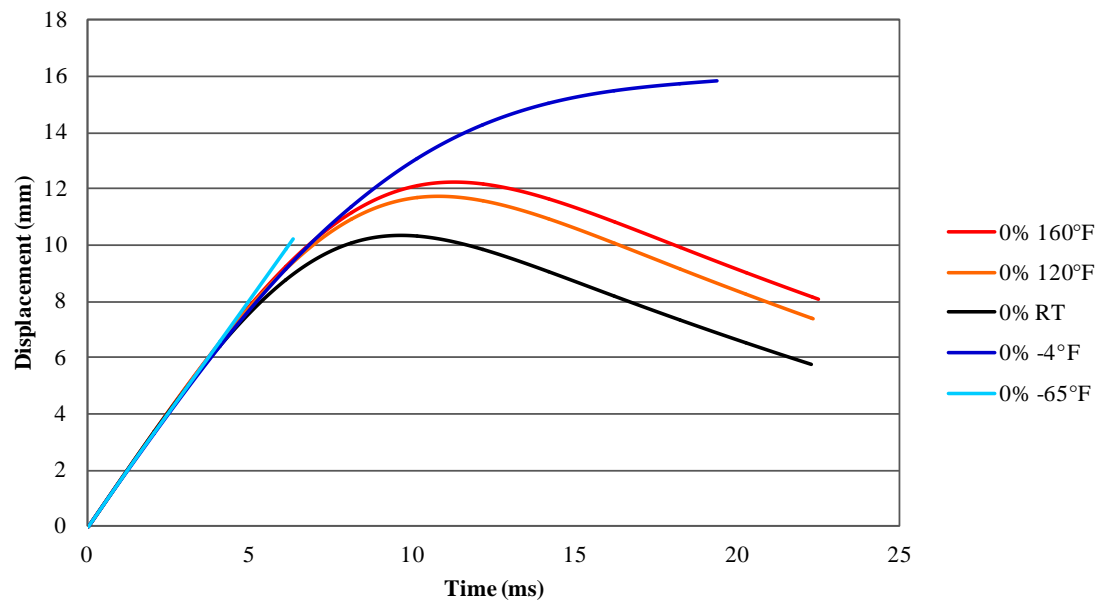


Figure 3.74 Displacement vs. time curves for neat PP specimens at -4°F, -65°F, room temperature, 120°F and 160°F

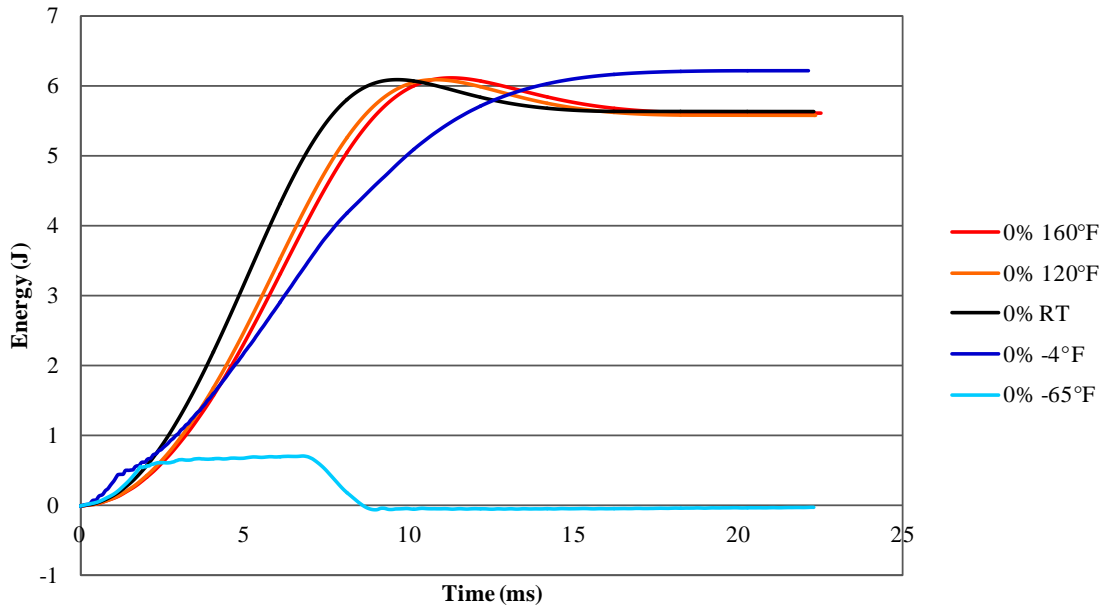


Figure 3.75 Energy vs. time curves for neat PP specimens at -4°F, -65°F, room temperature, 120°F and 160°F

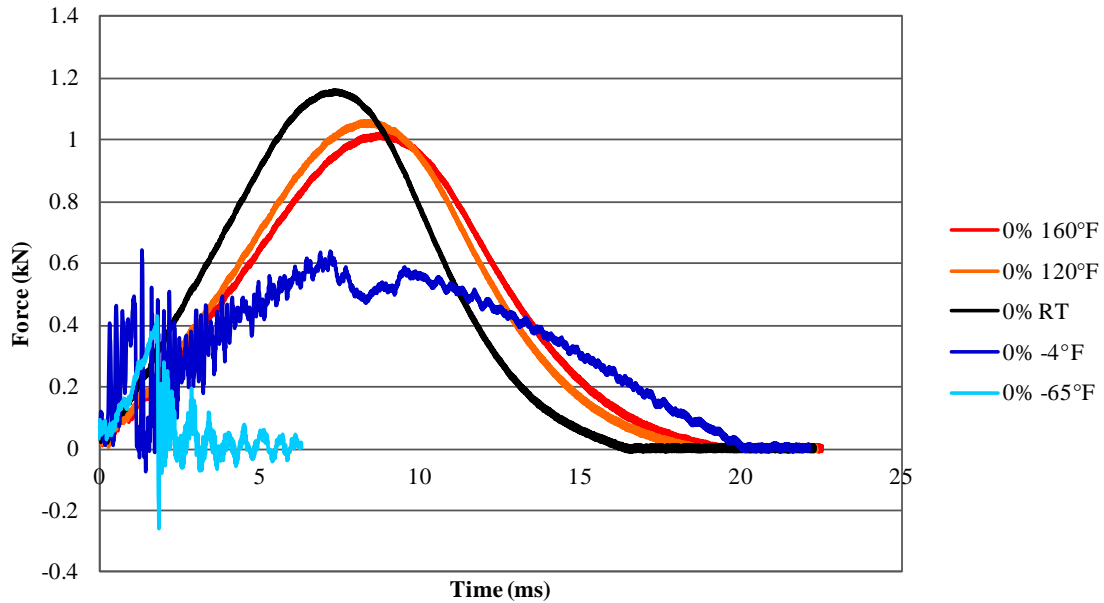


Figure 3.76 Force vs. time curves for neat PP specimens at -4°F , -65°F , room temperature, 120°F and 160°F

Similarly, in Figures 3.77-3.80 the results for 0.2% nanoclay reinforced specimens at various temperatures are compared. The specimens showed the same behavior at room and high temperatures. However, at low temperatures all specimens were penetrated. Again the evidence of penetration can be seen in Figures 3.77 and 3.80.

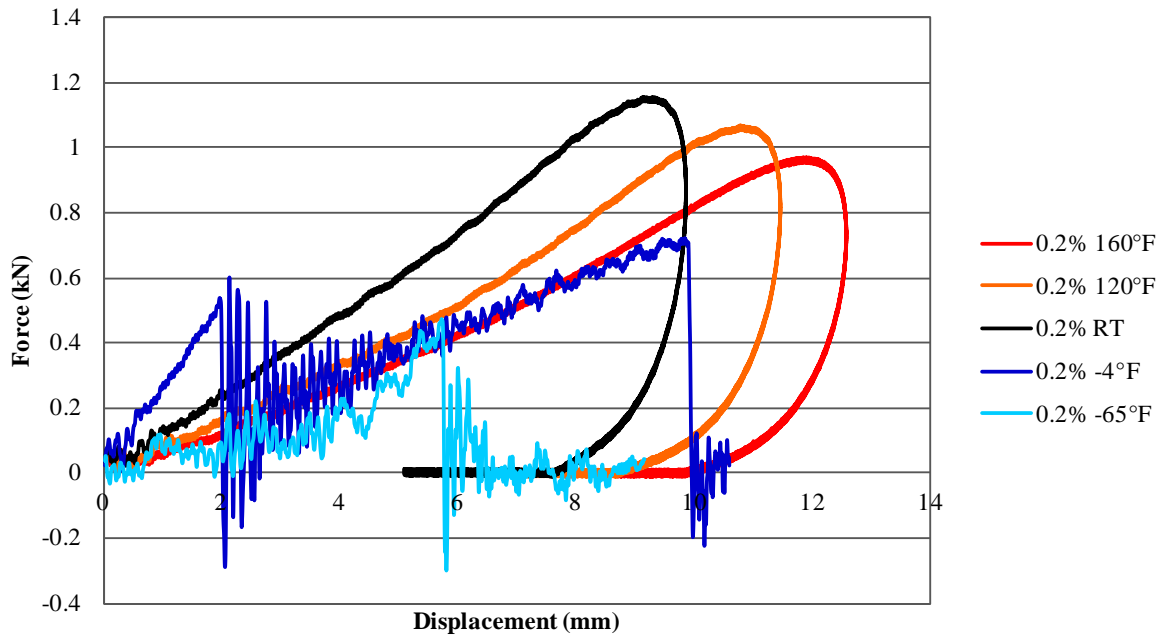


Figure 3.77 Force vs. displacement curves for PP with 0.2% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

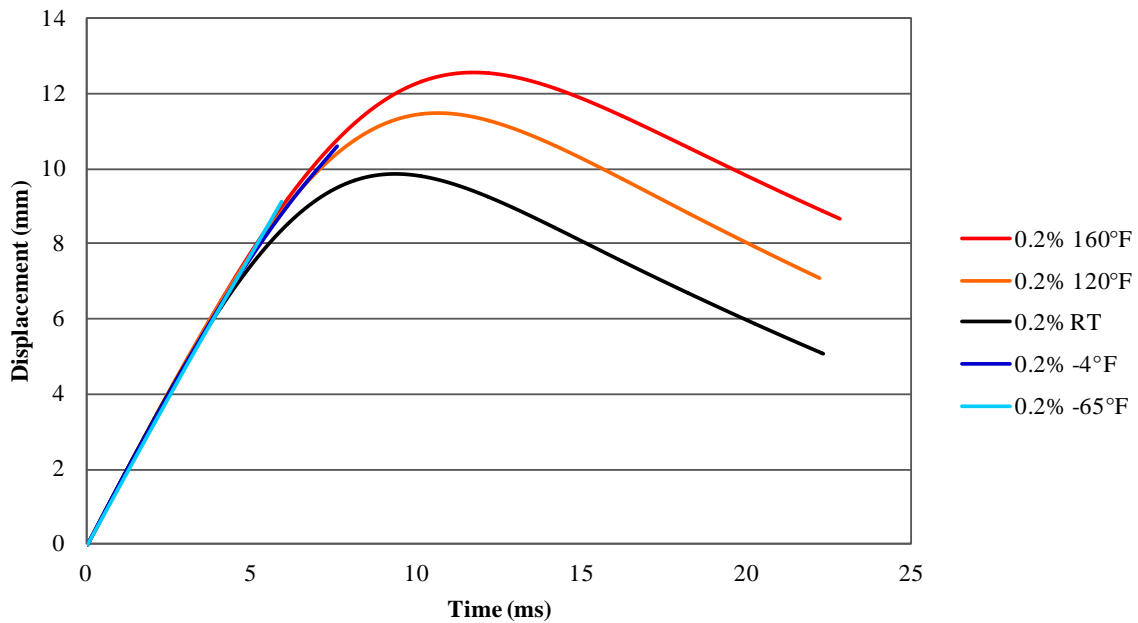


Figure 3.78 Displacement vs. time curves for PP with 0.2% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

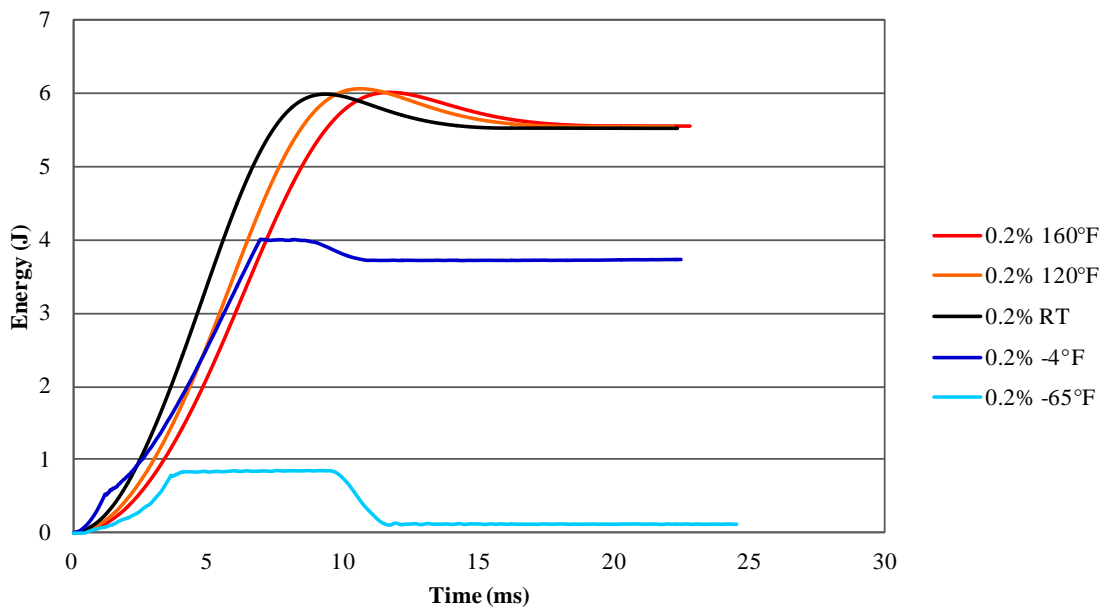


Figure 3.79 Energy vs. time curves for PP with 0.2% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

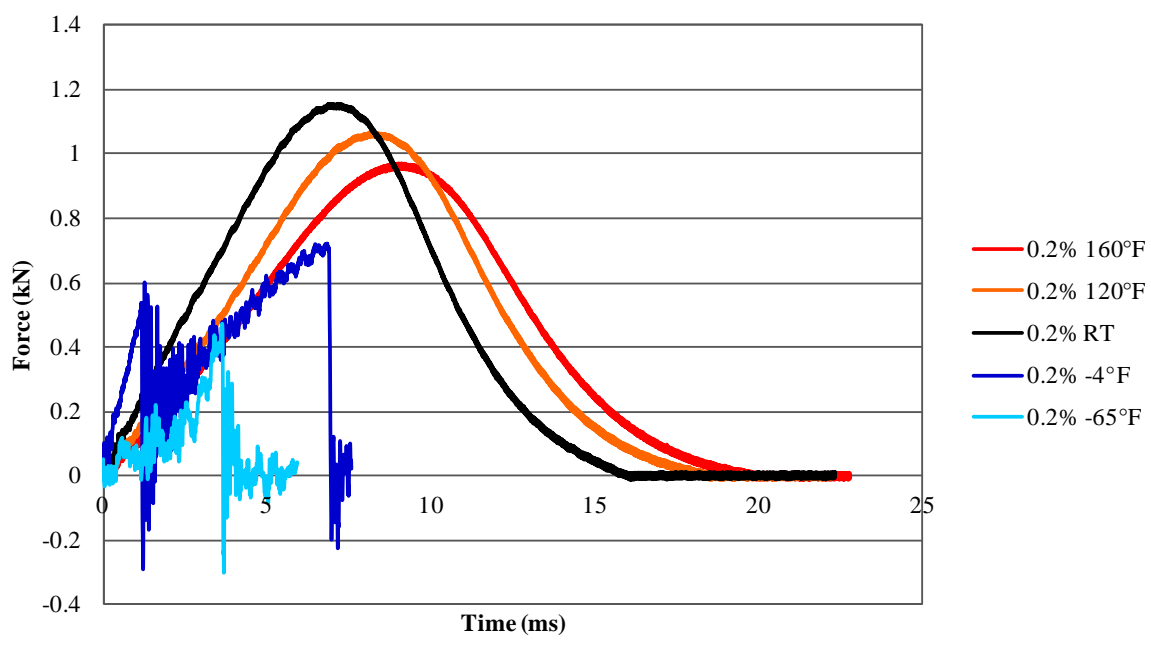


Figure 3.80 Force vs. time curves for PP with 0.2% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

The results for 1% nanoclay reinforced PP specimens at various temperatures are summarized in Figures 3.81-3.84. Again at low temperatures all the specimens were penetrated. In addition, at room temperature specimens were penetrated and two cracked. This is due to the fact that the increase in nanoclay makes the material more brittle.

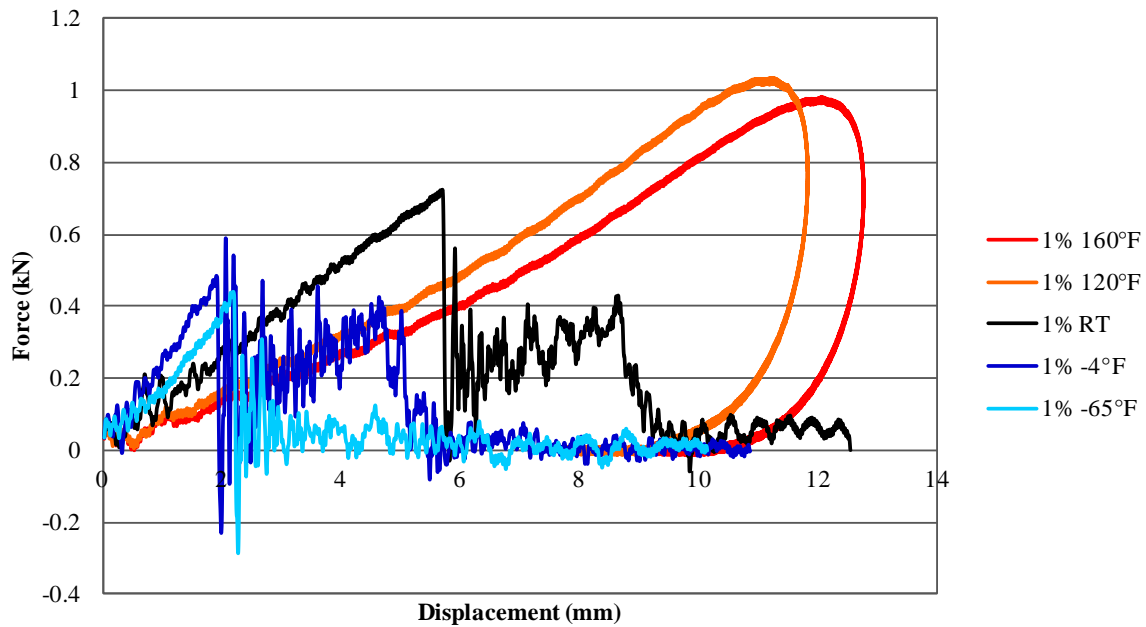


Figure 3.81 Force vs. displacement curves for PP with 1% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

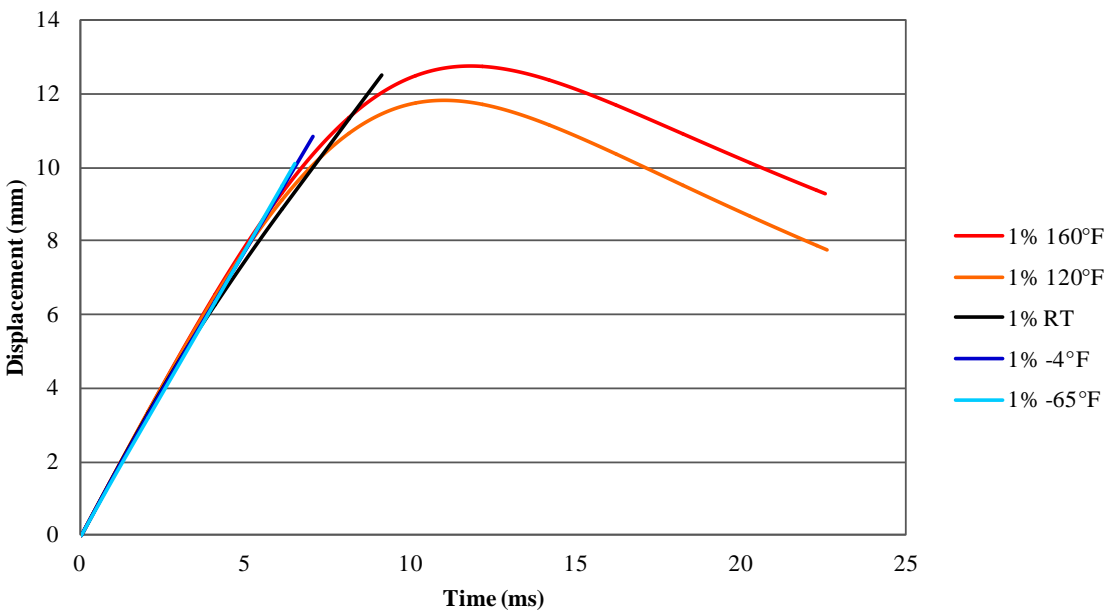


Figure 3.82 Displacement vs. time curves for PP with 1% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

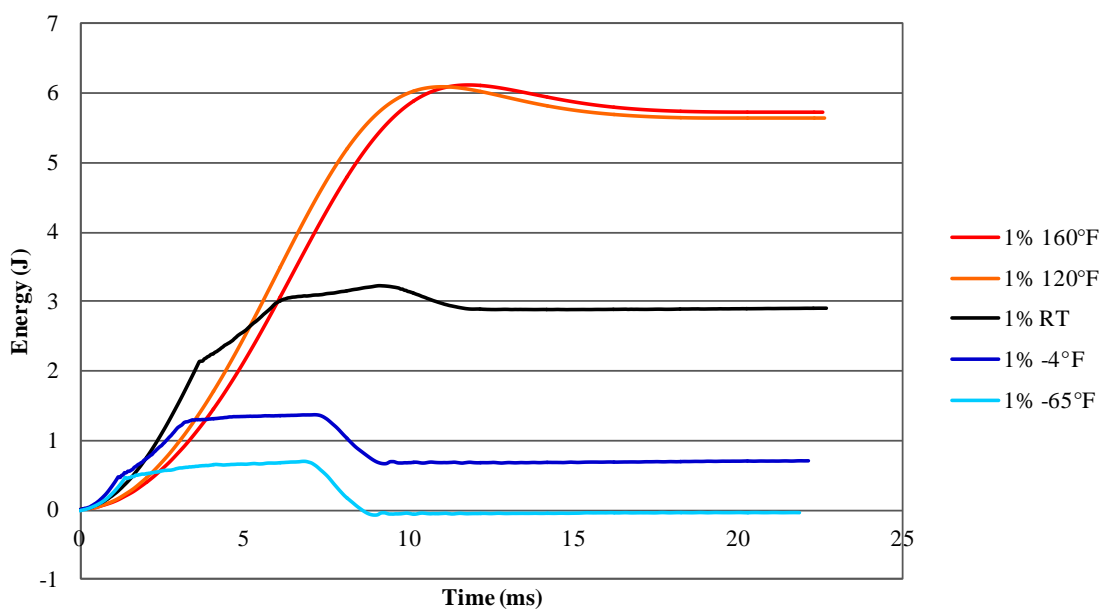


Figure 3.83 Energy vs. time curves for PP with 1% nanoclay reinforced specimens at -4°F, -65°F, room temperature, 120°F and 160°F

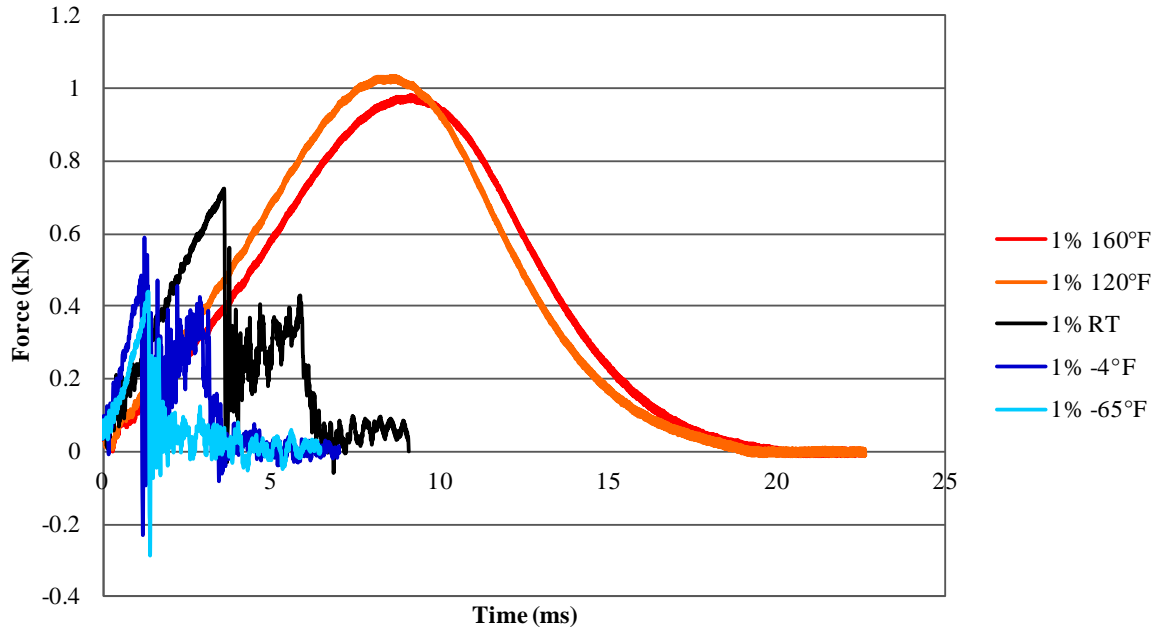


Figure 3.84 Force vs. time curves for PP with 1% nanoclay reinforced specimens at -4°F , -65°F , room temperature, 120°F and 160°F

The remaining specimens with 3%, 6% and 10% nanoclay reinforcement were not subjected to low velocity drop weight tests at low temperatures, because all were expected to be penetrated since a higher percentage of nanoclay makes the material more brittle.

Figures 3.85-3.88 show present the results for 3% nanoclay reinforced specimens at RT, 120°F and 160°F . At this reinforcement level at room temperature 4 specimens were penetrated and one cracked. At 120°F and 160°F the specimens were not penetrated.

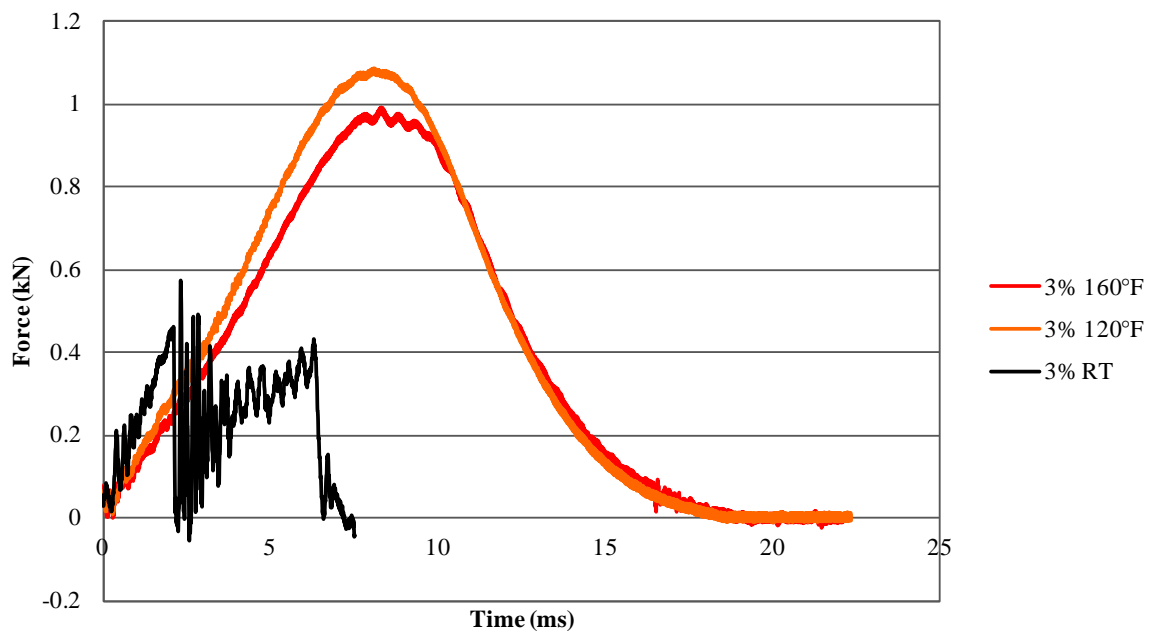


Figure 3.85 Force vs. displacement curves for PP with 3% nanoclay reinforced specimens at RT, 120°F and 160°F

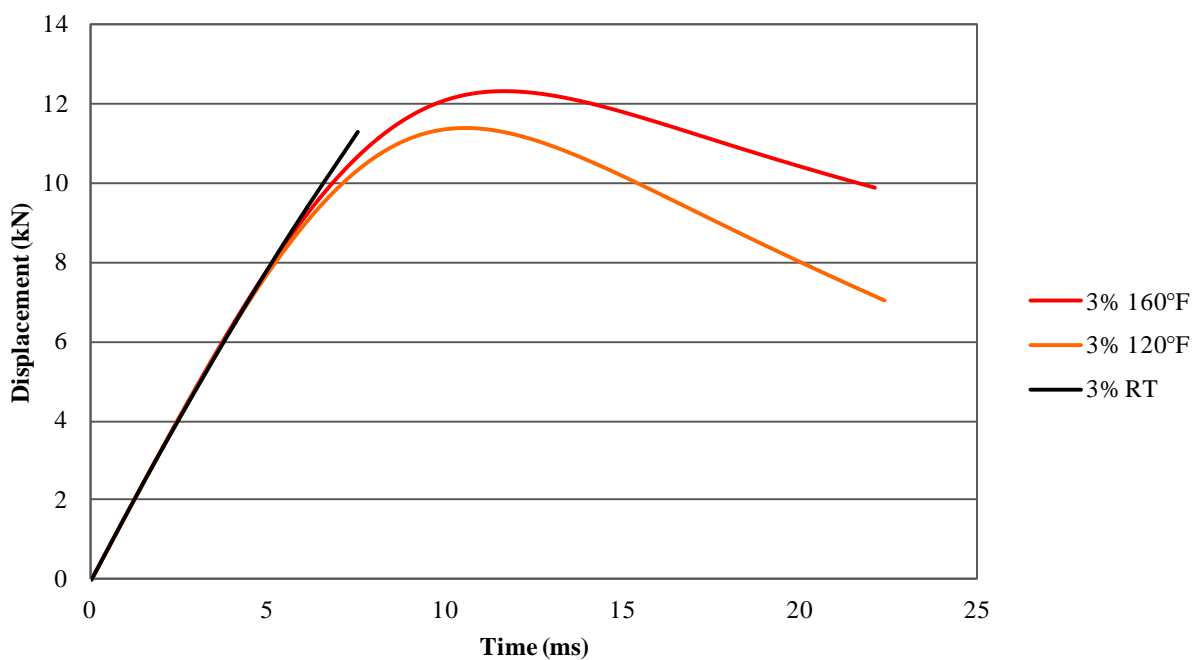


Figure 3.86 Displacement vs. time curves for PP with 3% nanoclay reinforced specimens at RT, 120°F and 160°F

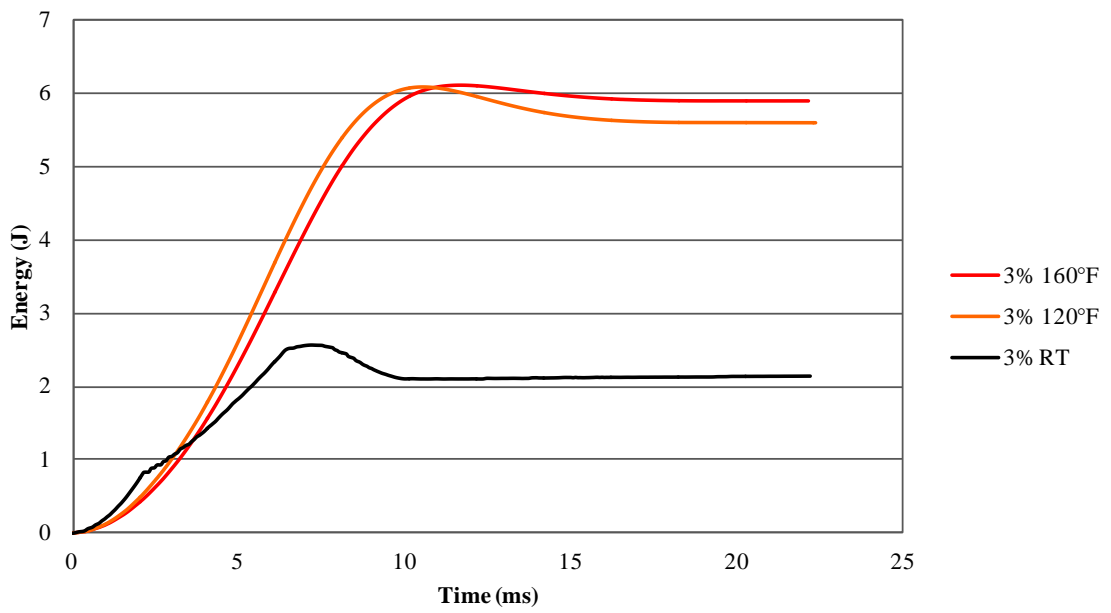


Figure 3.87 Energy vs. time curves for PP with 3% nanoclay reinforced specimens at RT, 120°F and 160°F

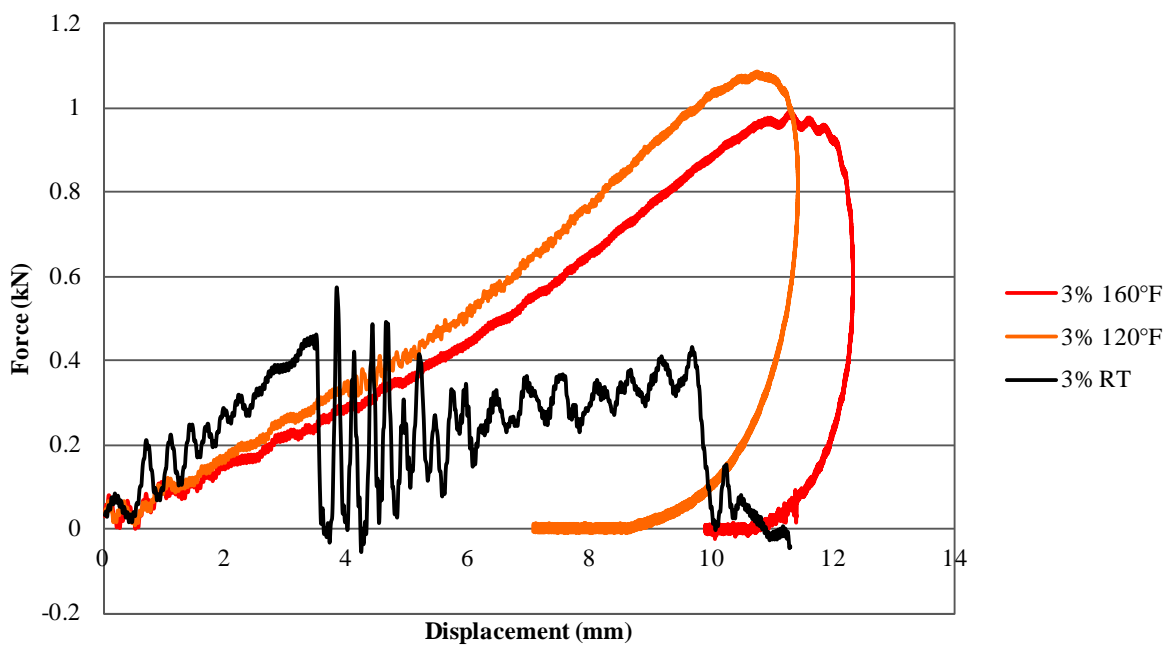


Figure 3.88 Force vs. displacement curves for PP with 3% nanoclay reinforced specimens at RT, 120°F and 160°F

The low velocity drop weight test results for PP 3371 specimens with 6% nanoclay reinforcement are depicted in Figures 3.89 –3.93. The behavior was similar to that observed for the 3% specimens. That is at room temperature two specimens were penetrated and two cracked. However, at higher temperatures they were not penetrated.

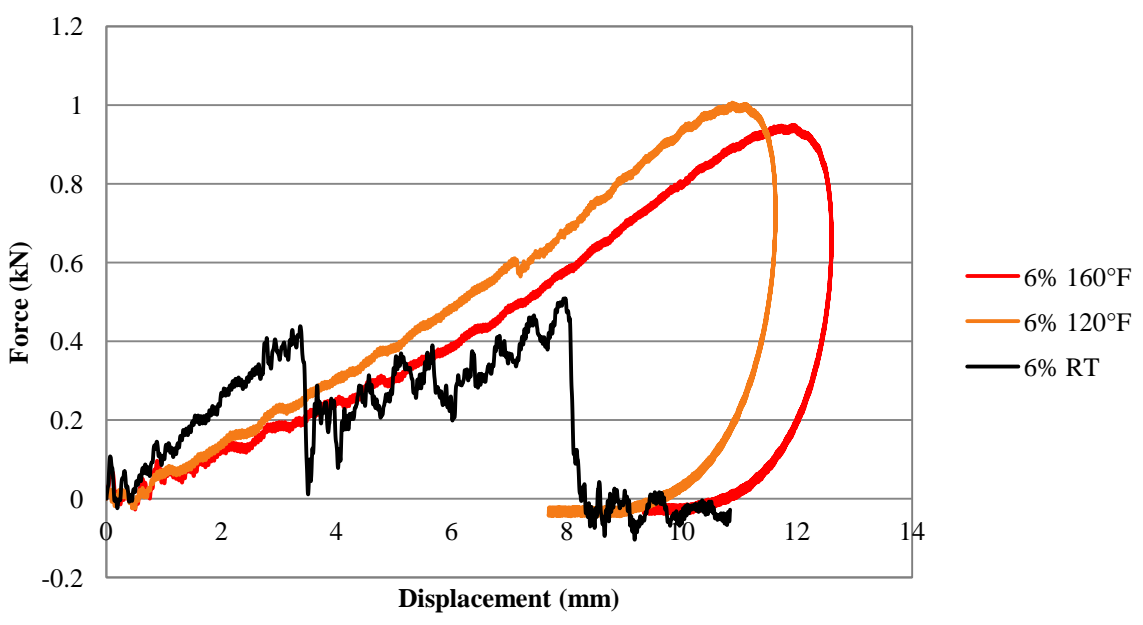


Figure 3.89 Force vs. displacement curves for PP with 6% nanoclay reinforced specimens at RT, 120°F and 160°F

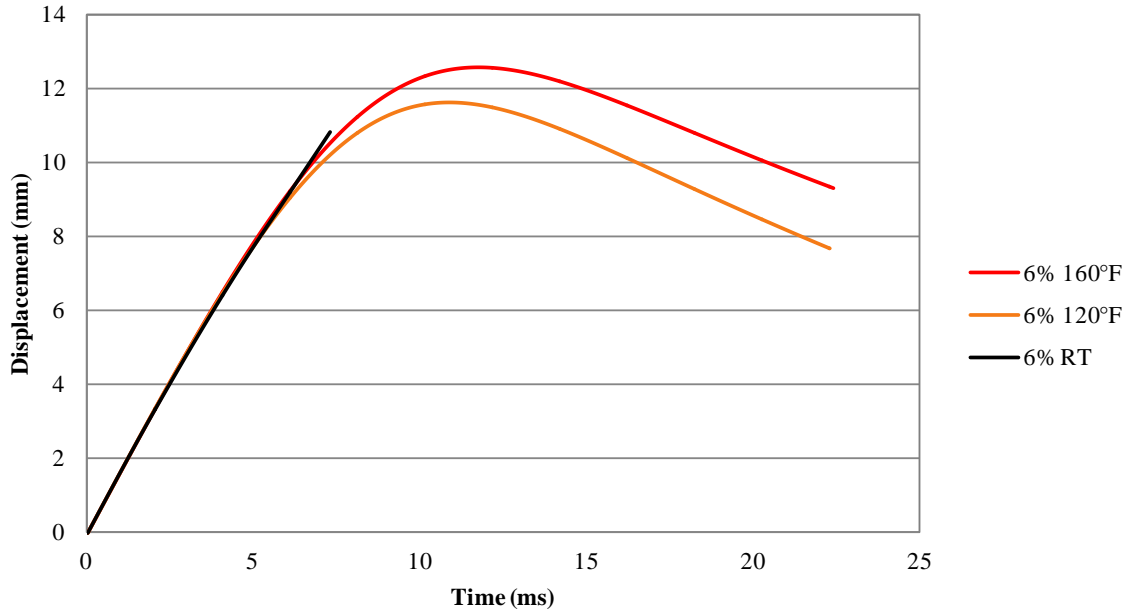


Figure 3.90 Displacement vs. time curves for PP with 6% nanoclay reinforced specimens at RT, 120°F and 160°F

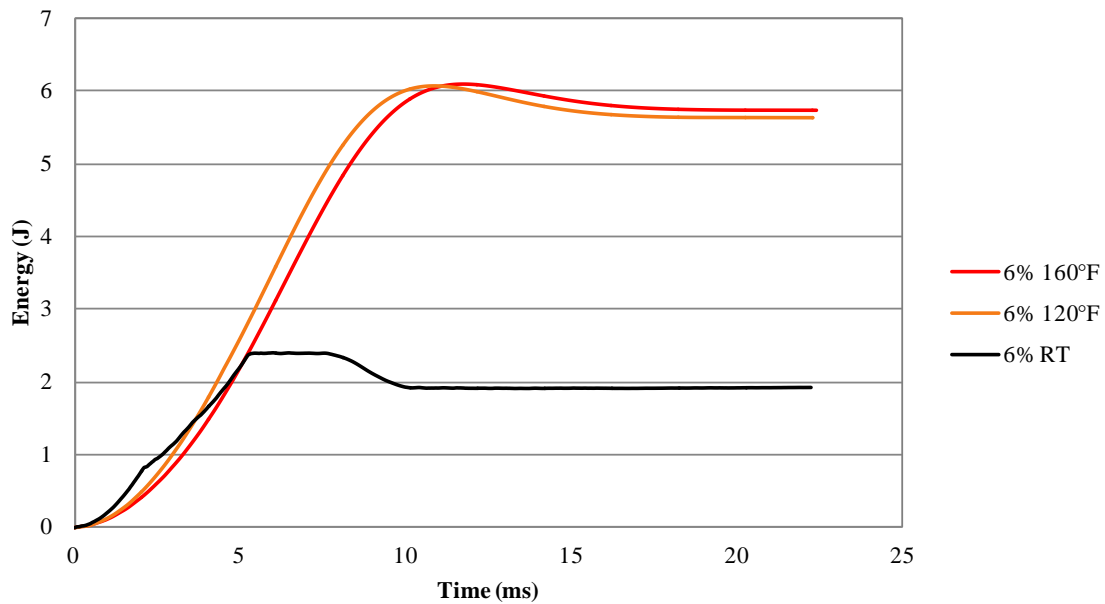


Figure 3.91 Energy vs. time curves for PP with 6% nanoclay reinforced specimens at RT, 120F and 160F

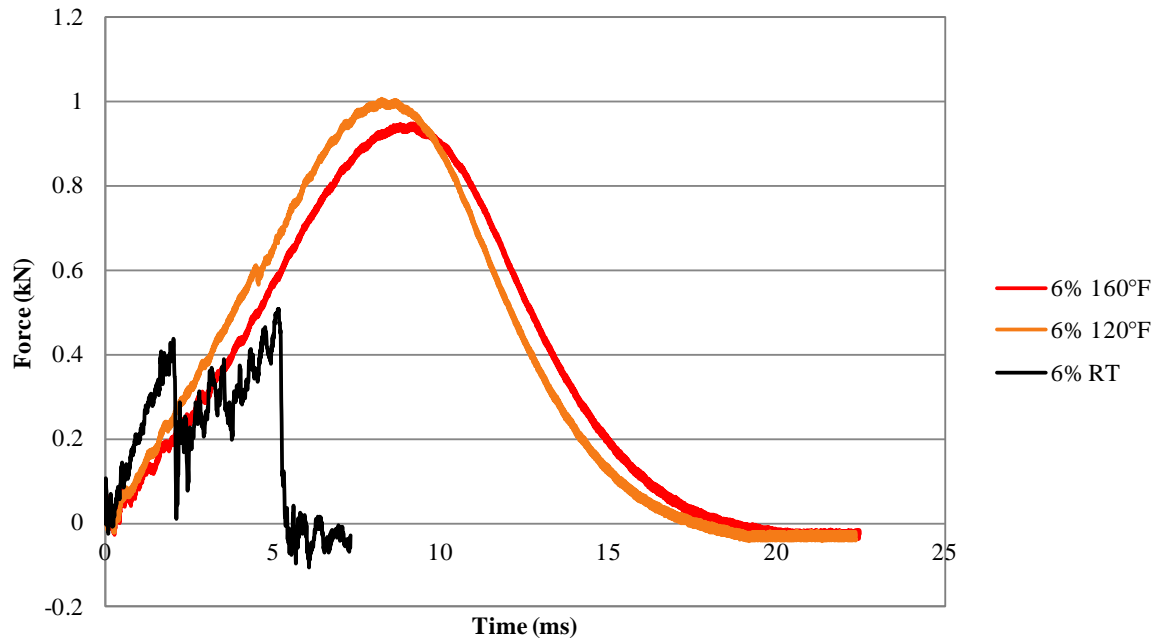


Figure 3.92 Force vs. time curves for PP with 6% nanoclay reinforced specimens at RT, 120F and 160F

Finally, the 10% nanoclay reinforced PP 3371 specimens were tested at 120°F and 160°F only, because even for 6% reinforcement most of the specimens were penetrated at room temperature.

The test results are given at Figures 3.93-3.96.

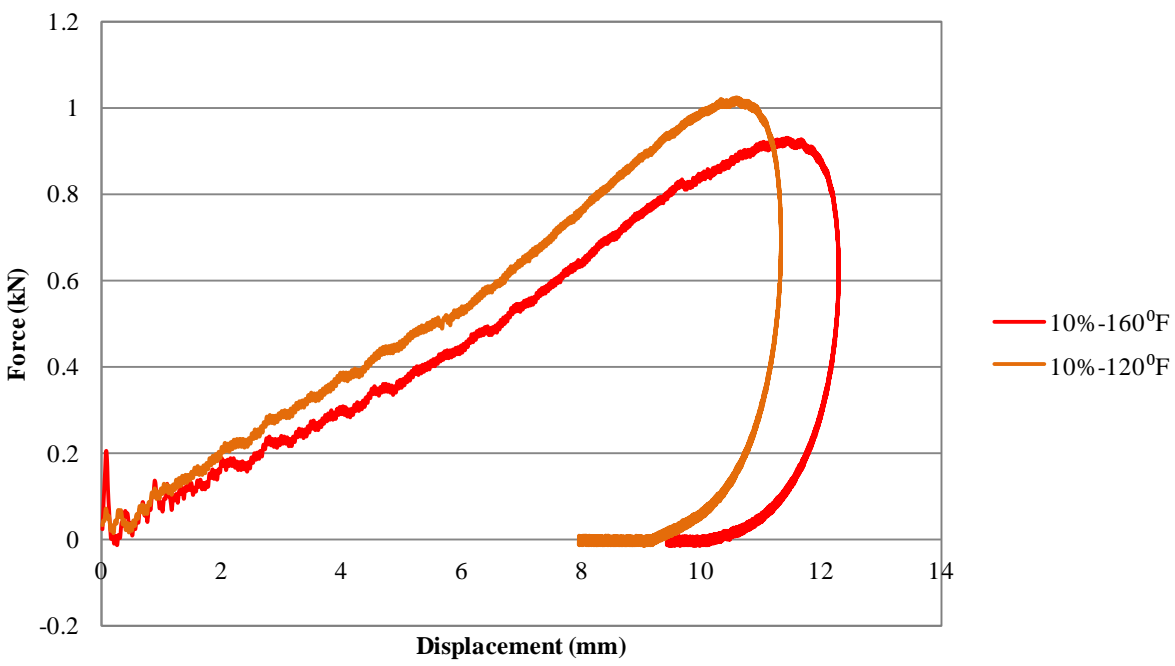


Figure 3.93 Force vs. displacement curves for PP with 10% nanoclay reinforced specimens at 120°F and 160°F

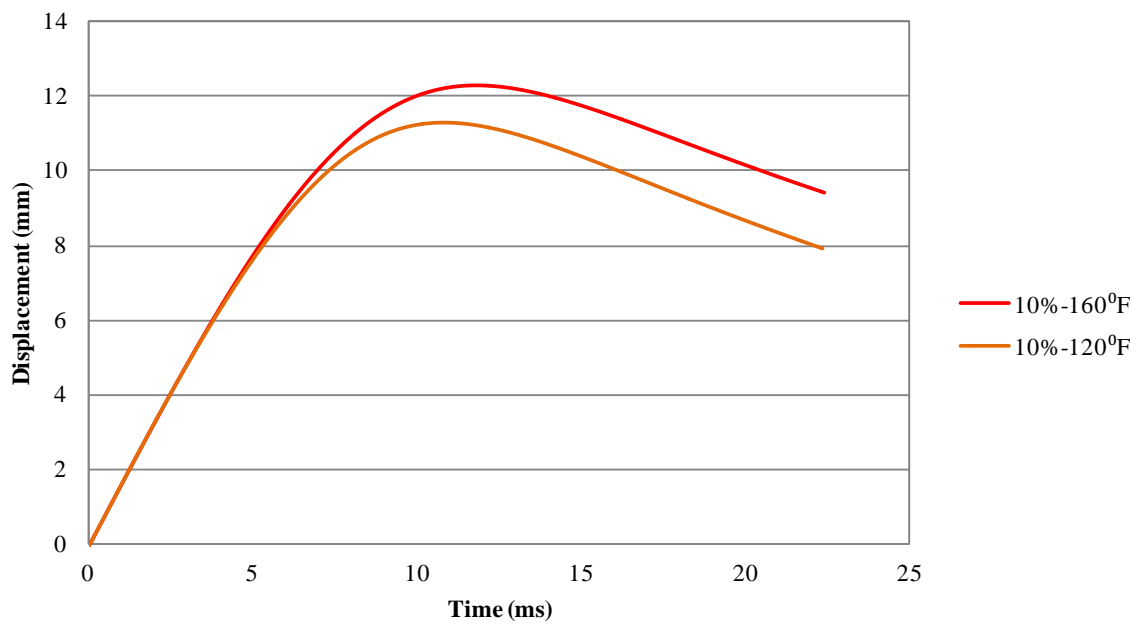


Figure 3.94 Displacement vs. time curves for PP with 10% nanoclay reinforced specimens at 120°F and 160°F

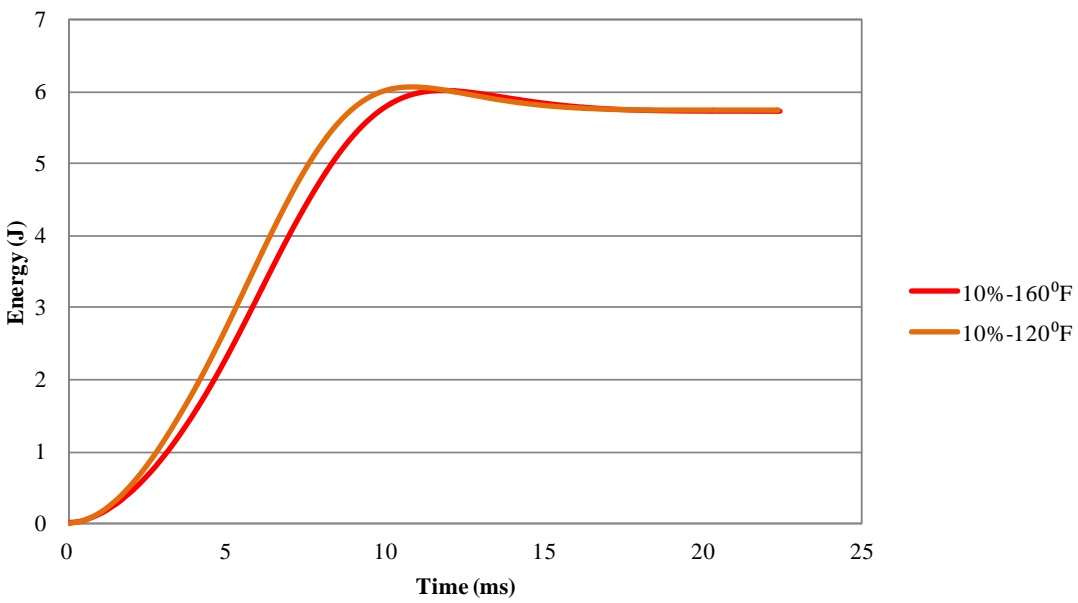


Figure 3.95 Energy vs. time curves for PP with 10% nanoclay reinforced specimens at 120°F and 160°F

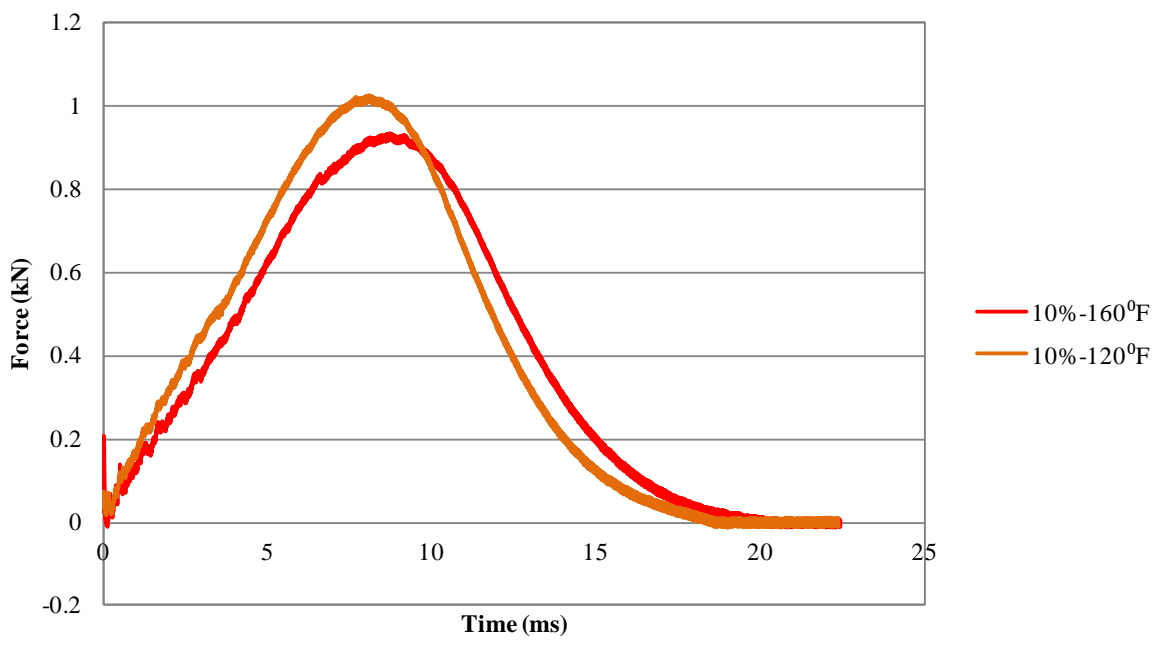


Figure 3.96 Force vs. time curves for PP with 10% nanoclay reinforced specimens at 120°F and 160°F

3.3.2. Comparison of Low Velocity Drop Weight Test Results at a Given Temperature

The test results are replotted at a given temperature to ascertain the effect of nanoclay reinforcement percentage. Figures 3.97-3.100 show the comparison of the results for various nanoclay reinforcement percentages at 160°F. The results for 120°F, RT, -4°F and -65°F are similarly displayed in Figures 3.101-3.104, Figures 3.105-3.108, Figures 3.109-3.112 and Figures 3.113-3.116 respectively.

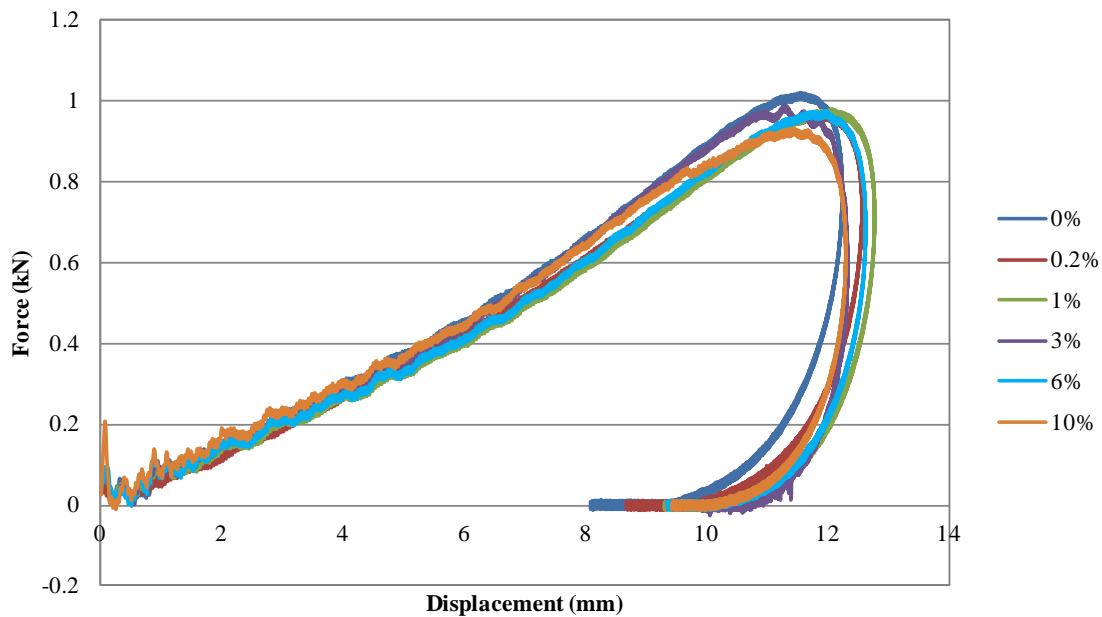


Figure 3.97 Force vs. displacement curves for PP specimens reinforced with different nanoclay percentages at 160°F

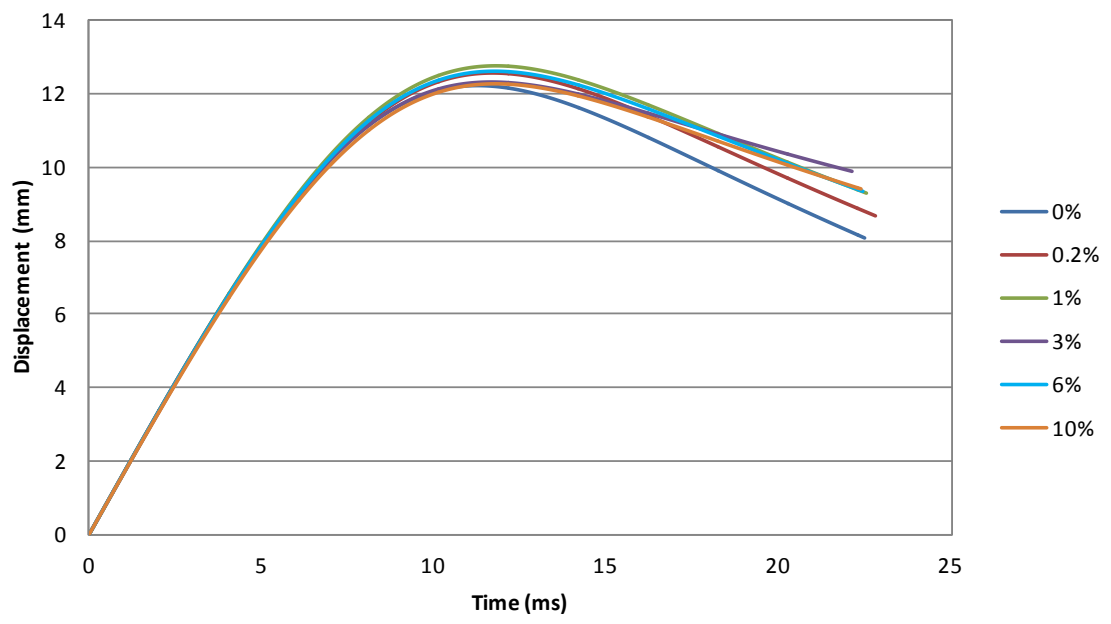


Figure 3.98 Displacement vs. time curves for PP specimens reinforced with different nanoclay percentages at 160°F

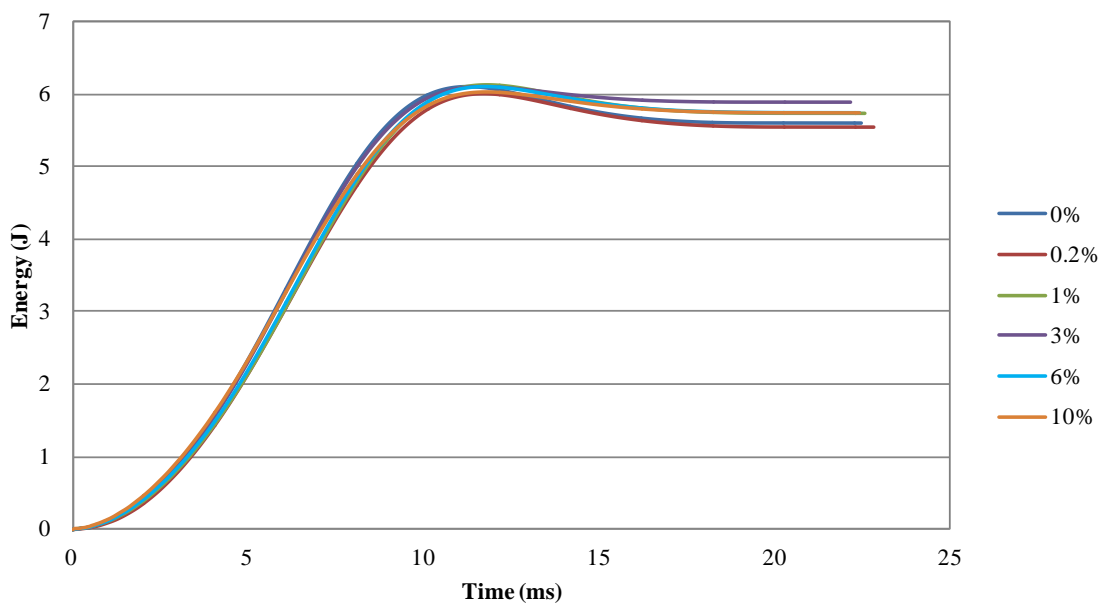


Figure 3.99 Energy vs. time curves for PP specimens reinforced with different nanoclay percentages at 160°F

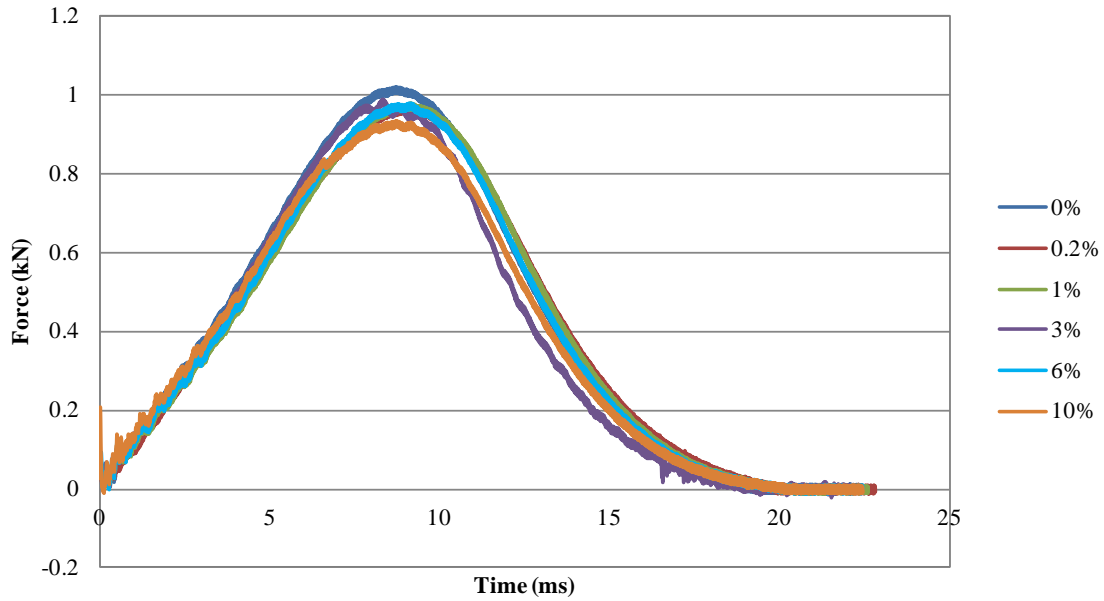


Figure 3.100 Force vs. time curves for PP specimens reinforced with different nanoclay percentages at 160°F

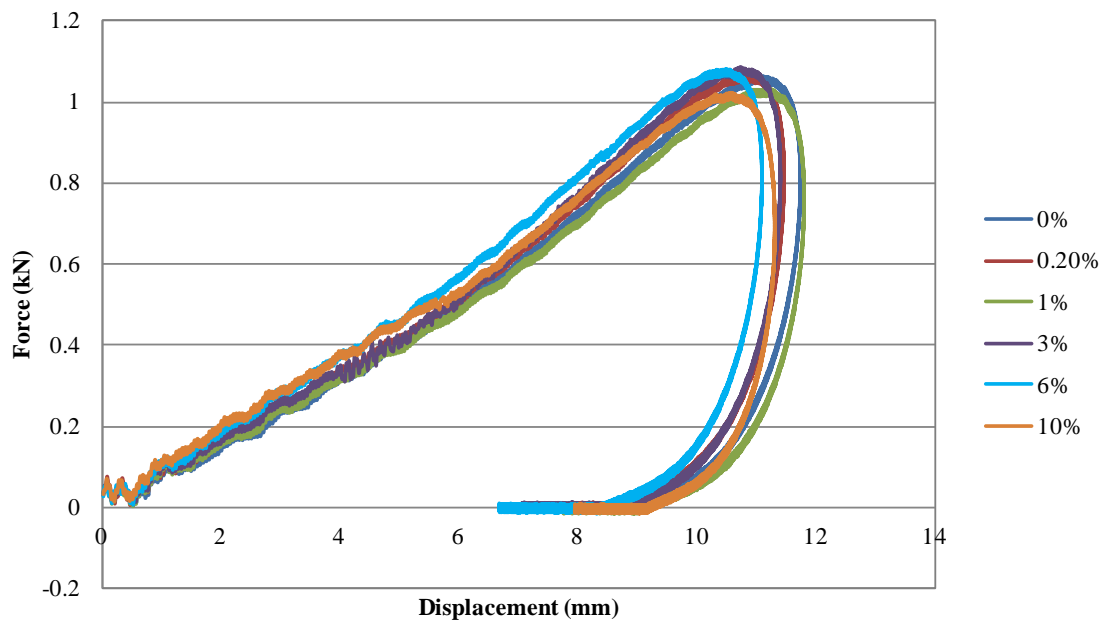


Figure 3.101 Force vs. displacement curves for PP specimens reinforced with different nanoclay percentages at 120°F

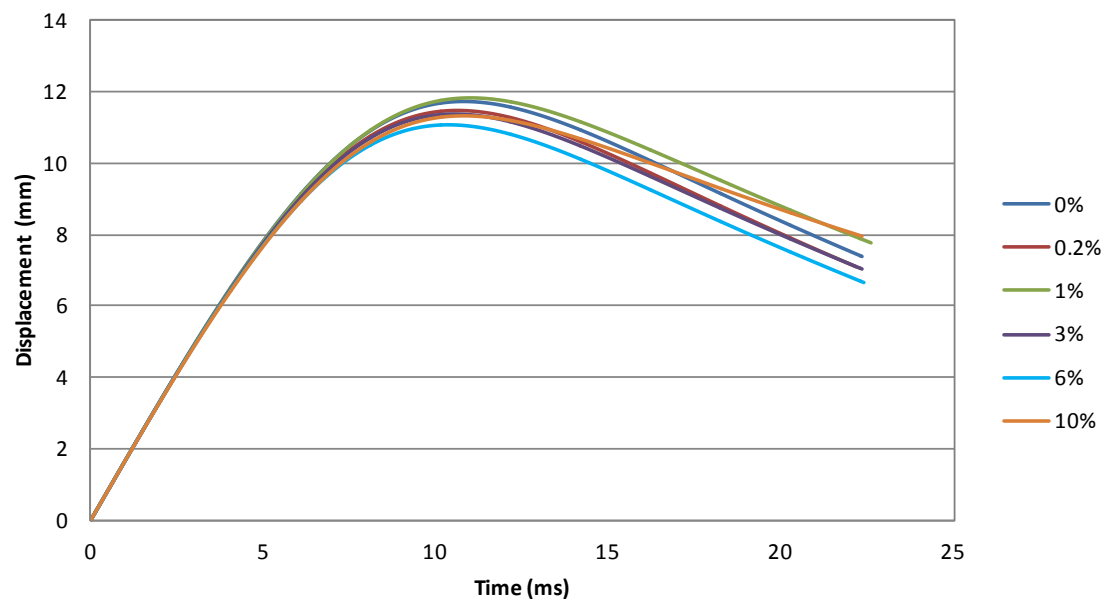


Figure 3.102 Displacement vs. time curves for PP specimens reinforced with different nanoclay percentages at 120°F

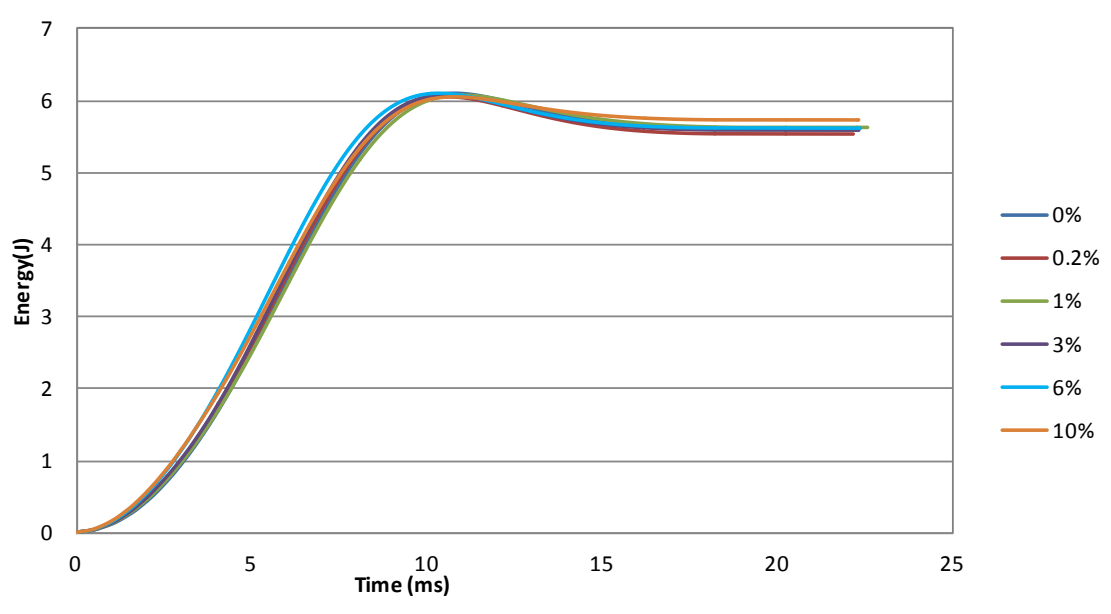


Figure 3.103 Energy vs. time curves for PP specimens reinforced with different nanoclay percentages at 120°F

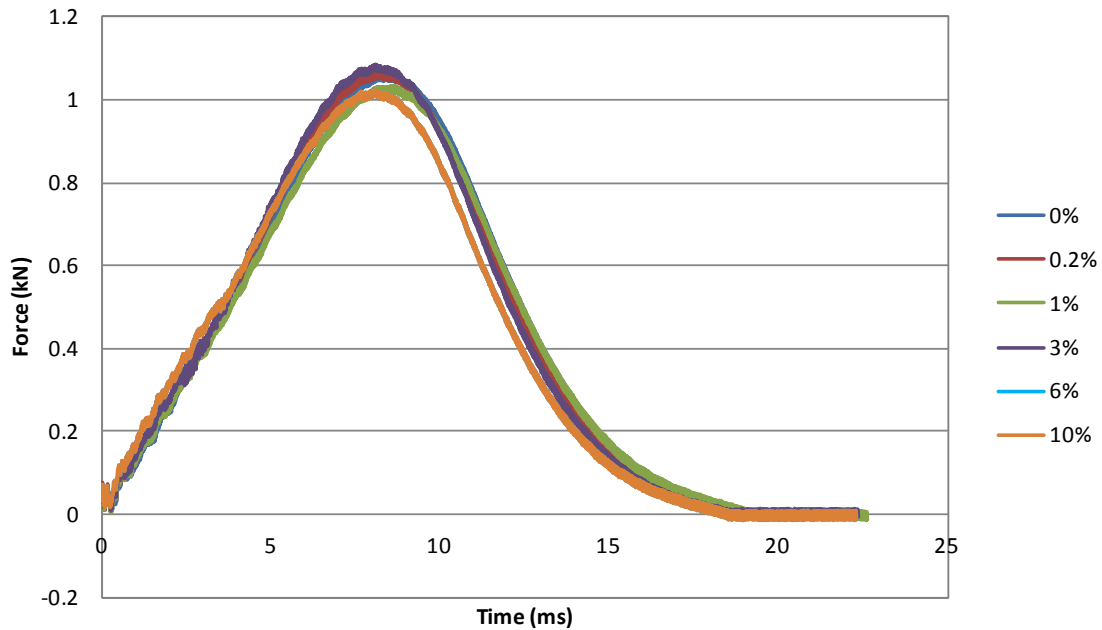


Figure 3.104 Force vs. time curves for PP specimens reinforced with different nanoclay percentages at 120°F

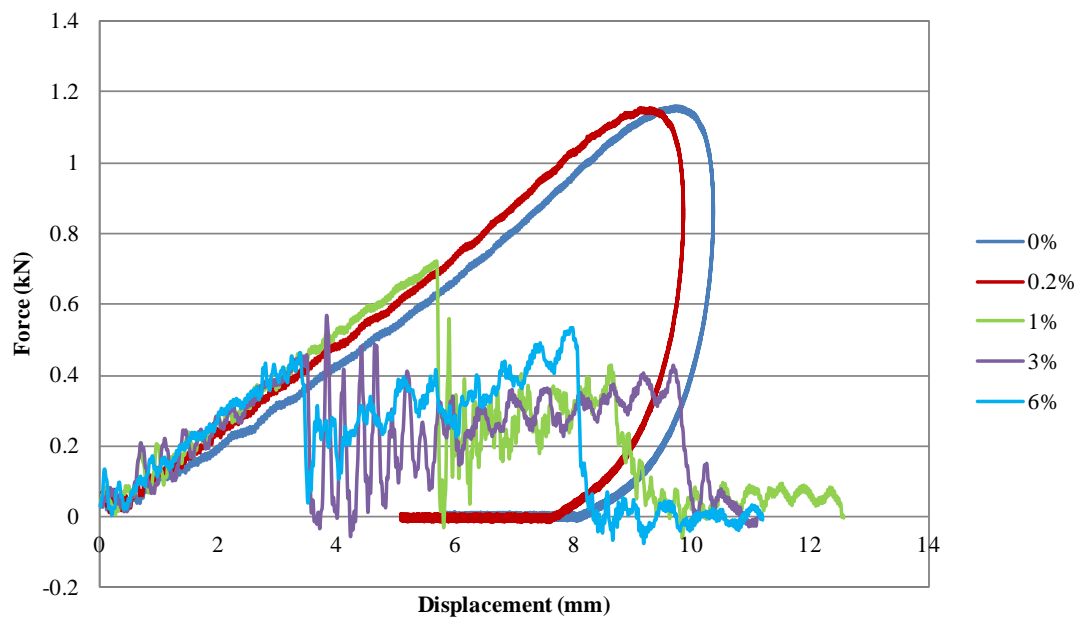


Figure 3.105 Force vs. displacement curves for PP specimens reinforced with different nanoclay percentages at RT

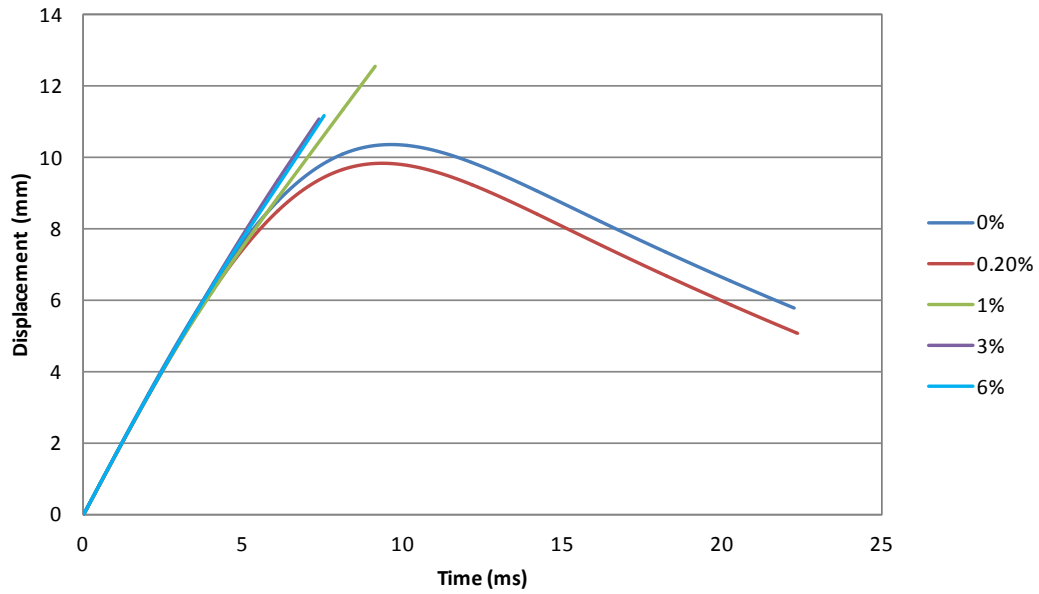


Figure 3.106 Displacement vs. time curves for PP specimens reinforced with different nanoclay percentages at RT

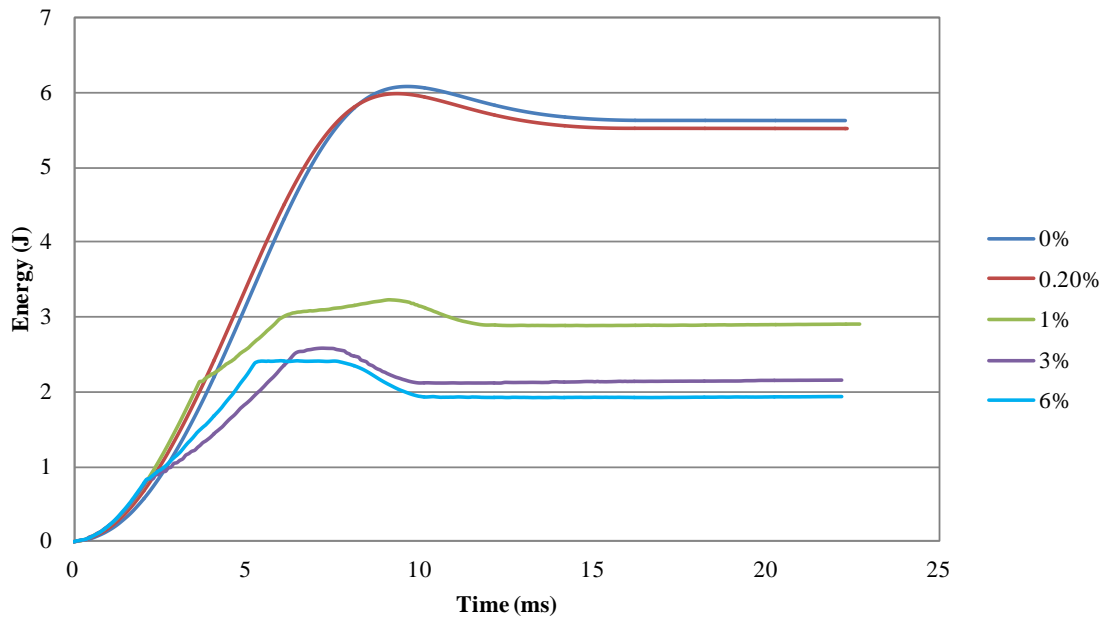


Figure 3.107 Energy vs. time curves for PP specimens reinforced with different nanoclay percentages at RT

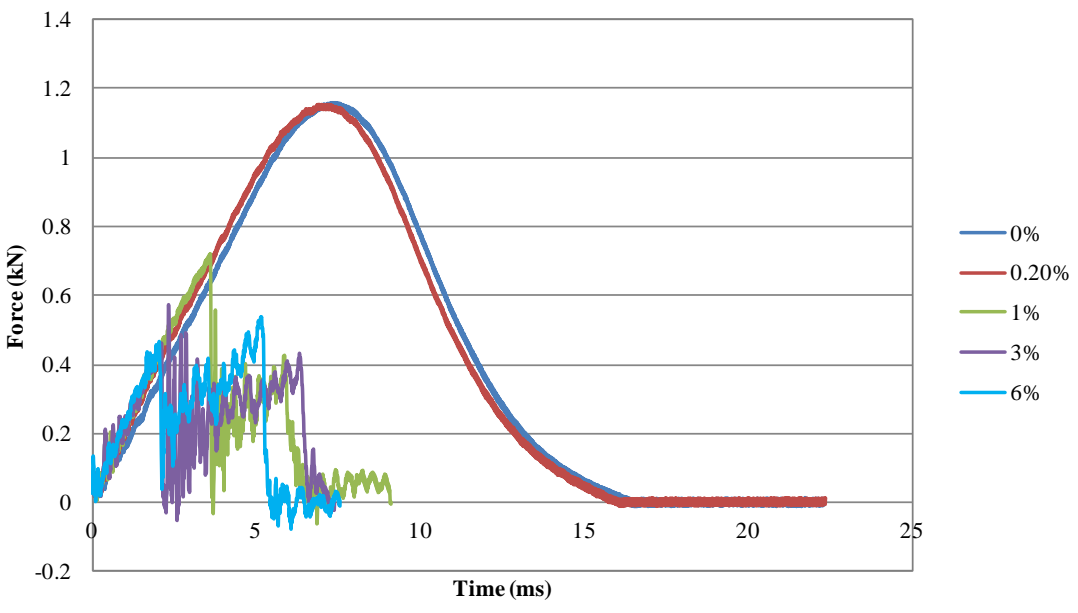


Figure 3.108 Force vs. time curves for PP specimens reinforced with different nanoclay percentages at RT

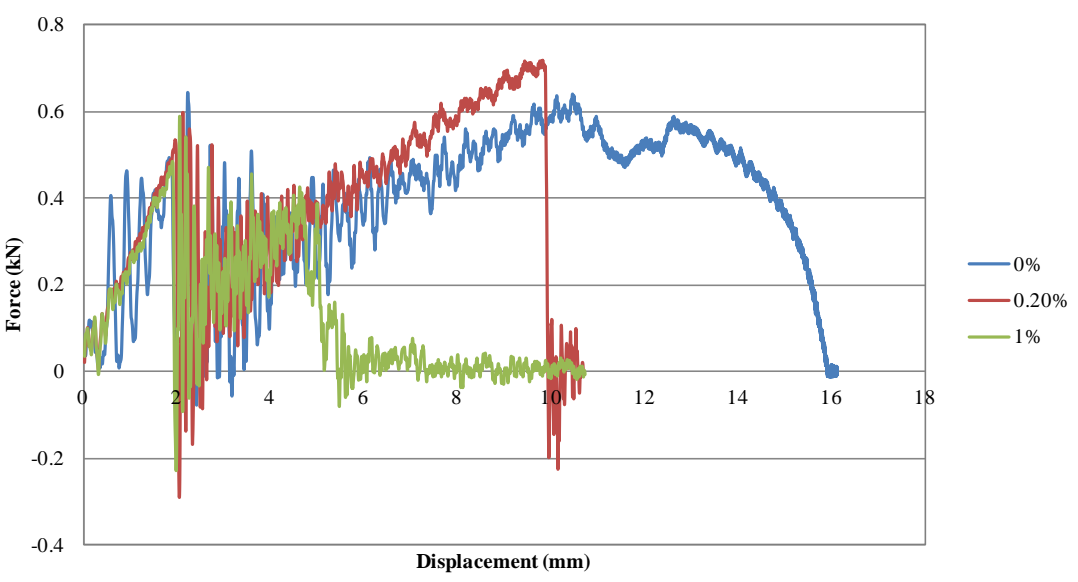


Figure 3.109 Force vs. displacement curves for PP specimens reinforced with different nanoclay percentages at -4°F

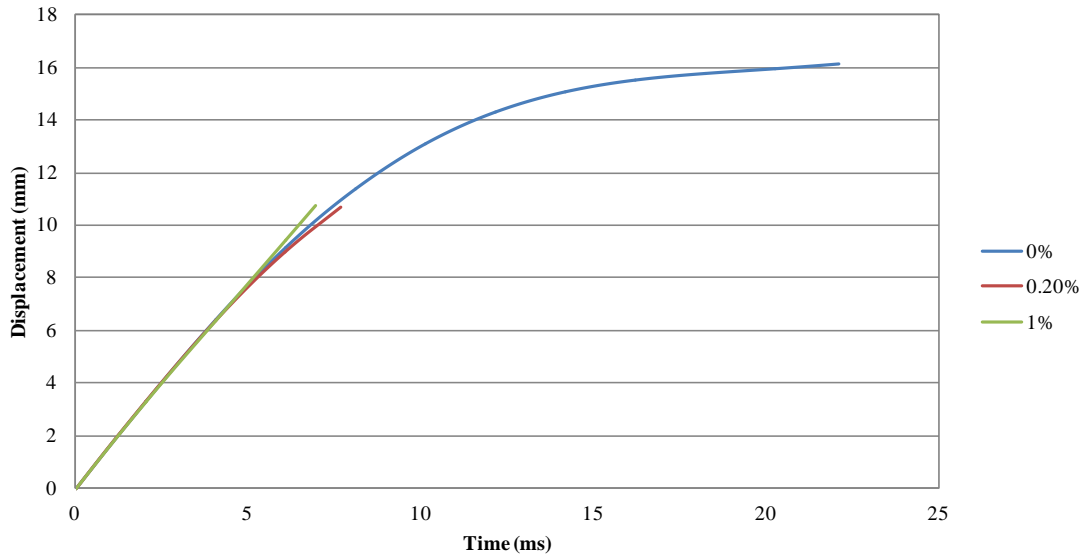


Figure 3.110 Displacement vs. time curves for PP specimens reinforced with different nanoclay percentages at -4°F

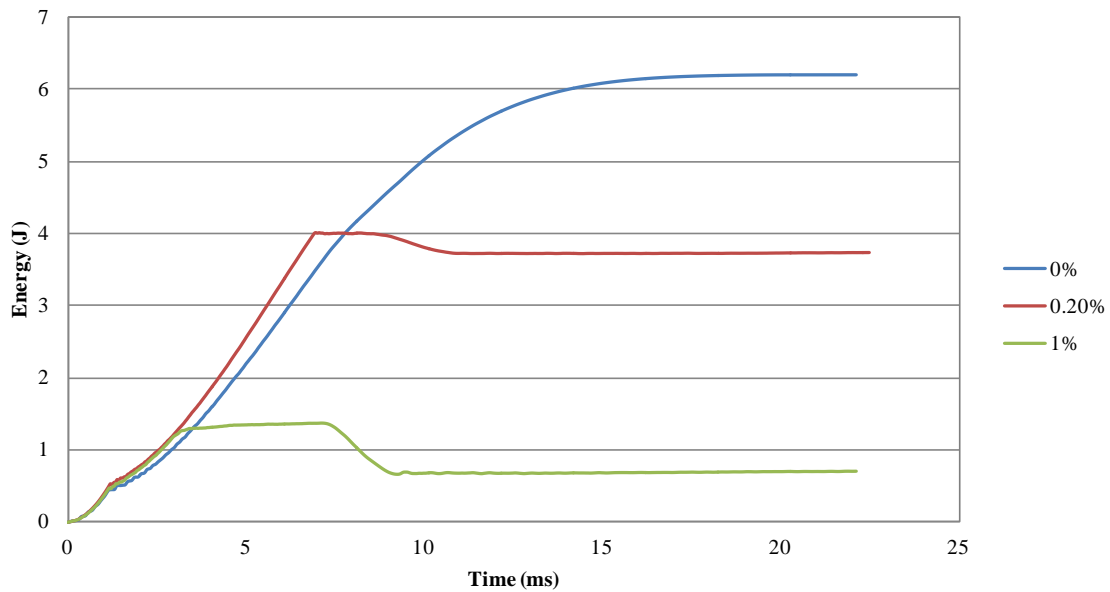


Figure 3.111 Energy vs. time curves for PP specimens reinforced with different nanoclay percentages at -4°F

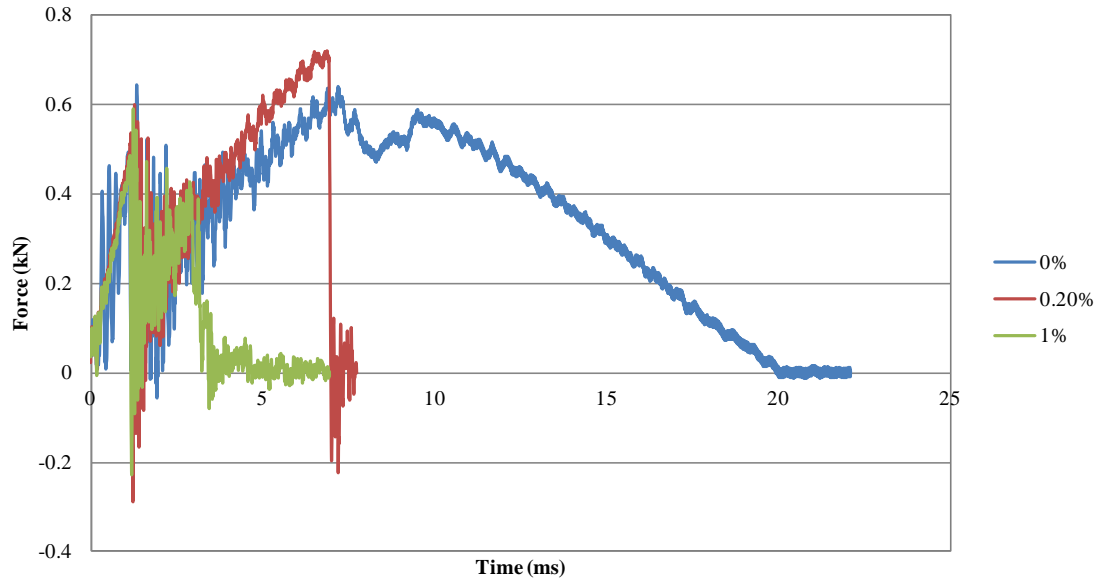


Figure 3.112 Force vs. time curves for PP specimens reinforced with different nanoclay percentages at -4°F

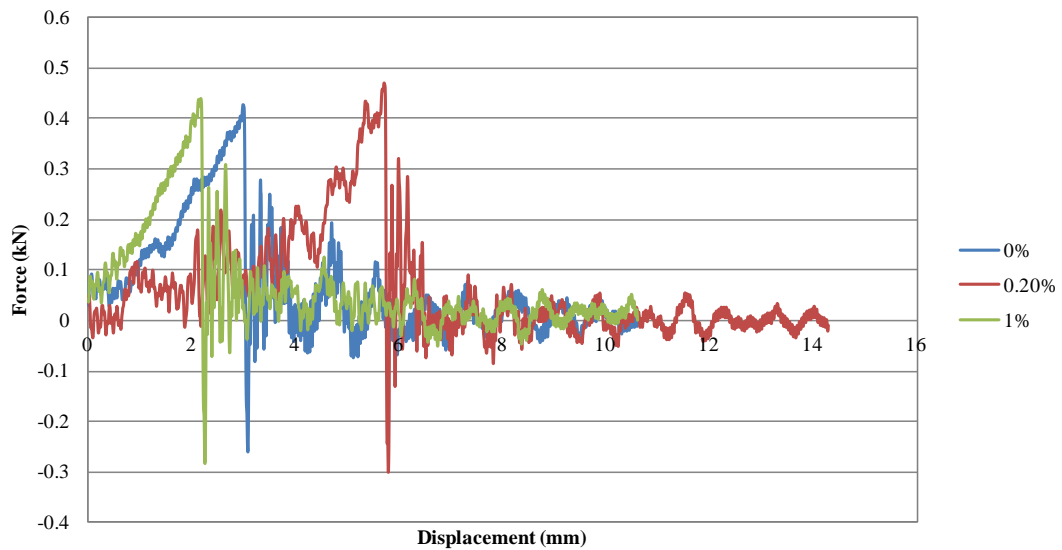


Figure 3.113 Force vs. displacement curves for PP specimens reinforced with different nanoclay percentages at -65°F

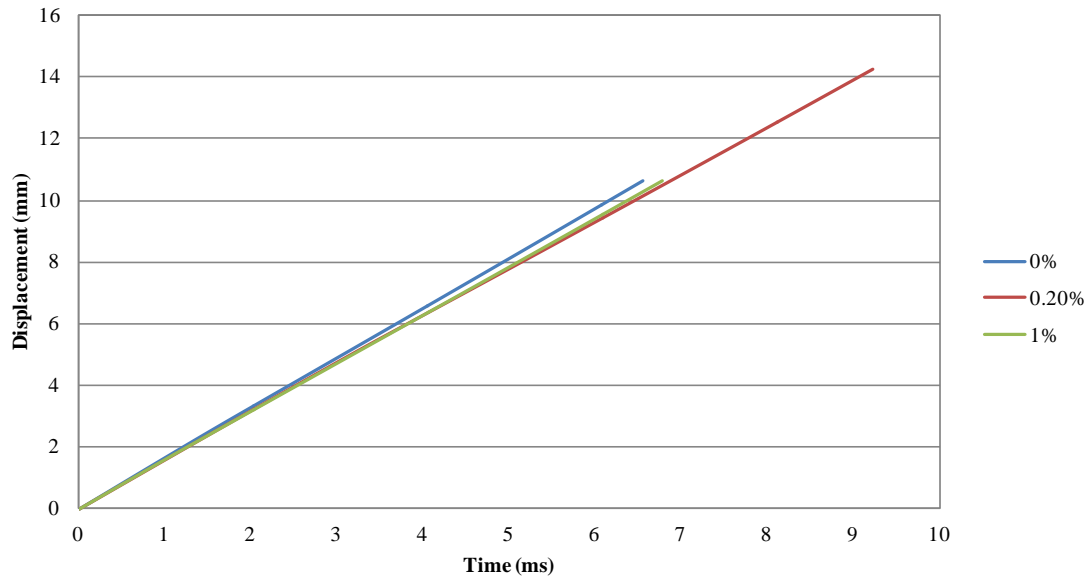


Figure 3.114 Displacement vs. time curves for PP specimens reinforced with different nanoclay percentages at -65°F

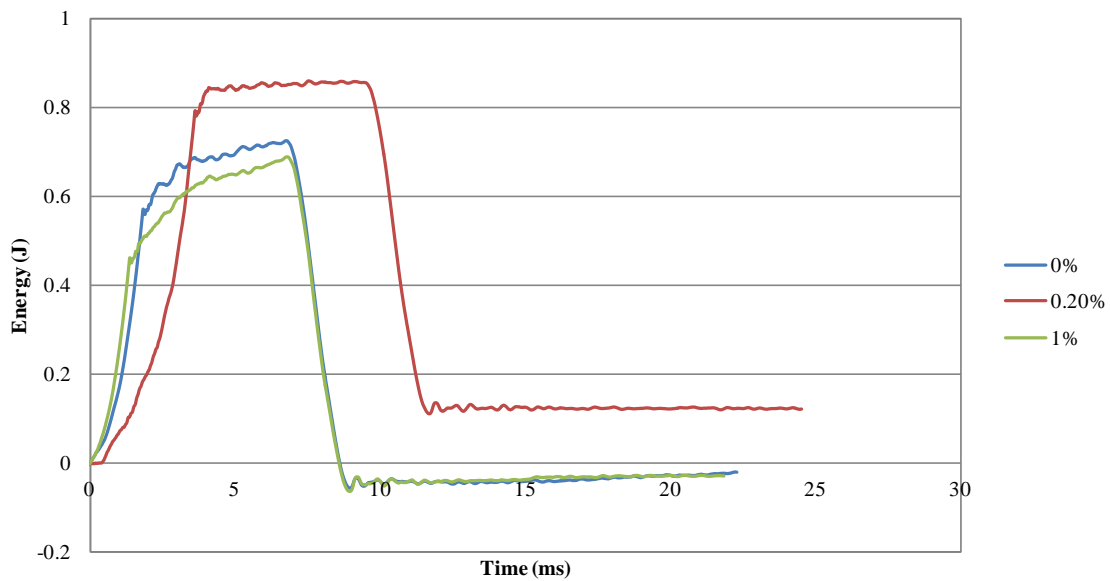


Figure 3.115 Energy vs. time curves for PP specimens reinforced with different nanoclay percentages at -65°F

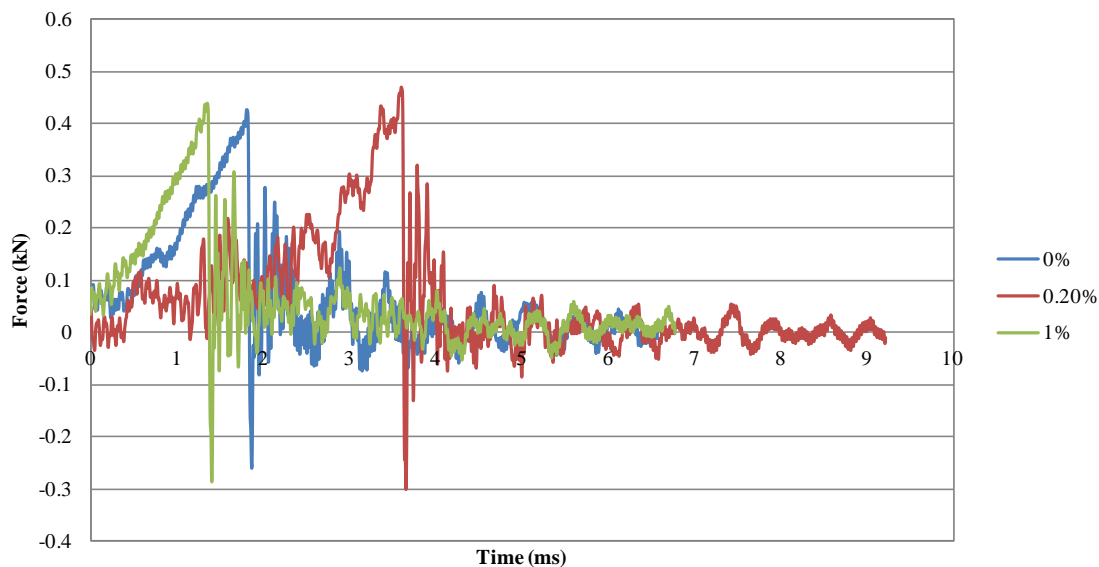


Figure 3.116 Force vs. time curves for PP specimens reinforced with different nanoclay percentages at -65°F

As a result of the low velocity impact tests, we can state that high temperatures affect polypropylene significantly and practically eliminate the effect of nanoclay reinforcement. The specimens show ductile behavior. However, at room and low temperatures the effect of nanoclay reinforcement is discernable and addition of nanoclay makes the specimens more brittle.

3.4. ELECTRON MICROSCOPY

As limited number of the specimens were studied under TEM and SEM. The purpose of this task is to observe the dispersion of nanoclay in the polypropylene matrix.

A scanning electron microscope (SEM) was used to obtain the images presented below. We started working with 10% nanoclay reinforced PP specimens because they have the highest nanoclay percentage and it would be easier to observe nanoclay particles inside PP. We obtained

some good images using SEM. We also tried to get images with a transmission electron microscope (TEM), but since SEM gave better images we continued our work with SEM.

To obtain the SEM images, we had to cut a very thin section of PP specimen. We used a diamond knife and cut a section of 200nm thickness and placed it inside the SEM. Figures 3.117-3.119 depict the images obtained. Red marks show the nanoclay particles.

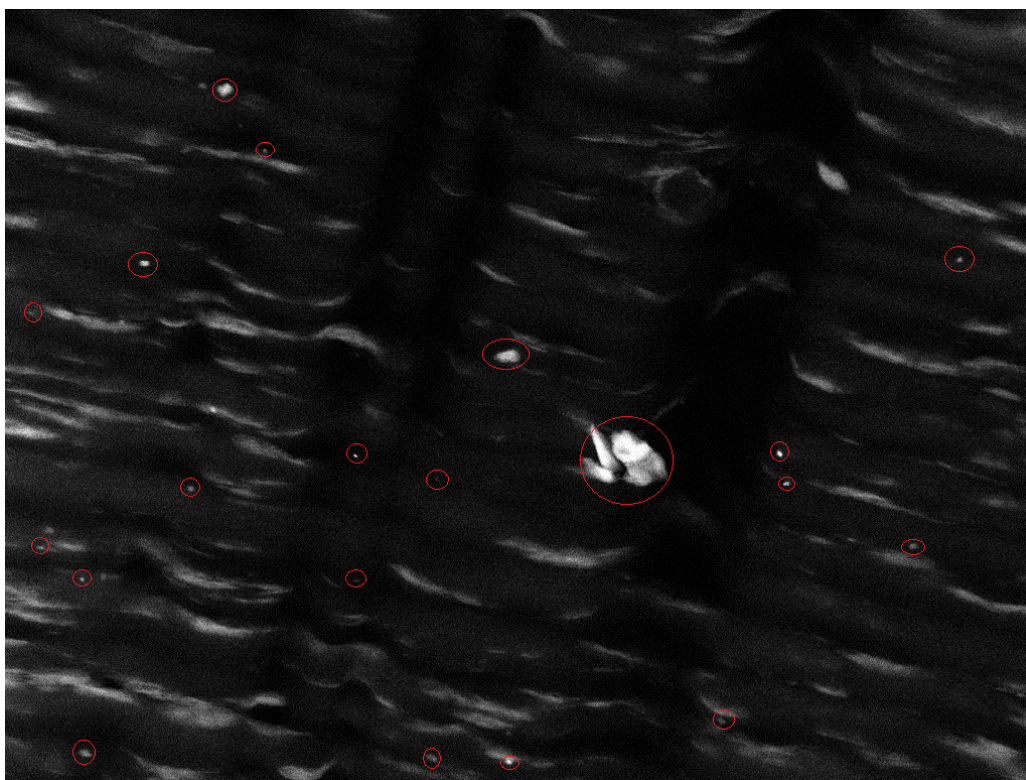


Figure 3.117 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen

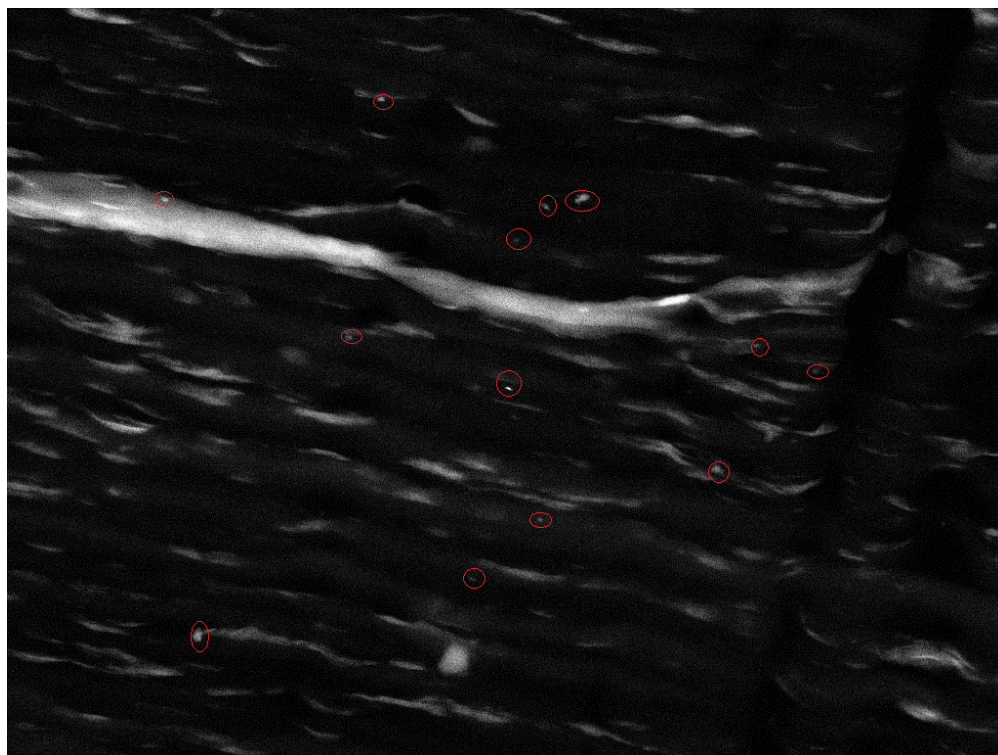


Figure 3.118 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen

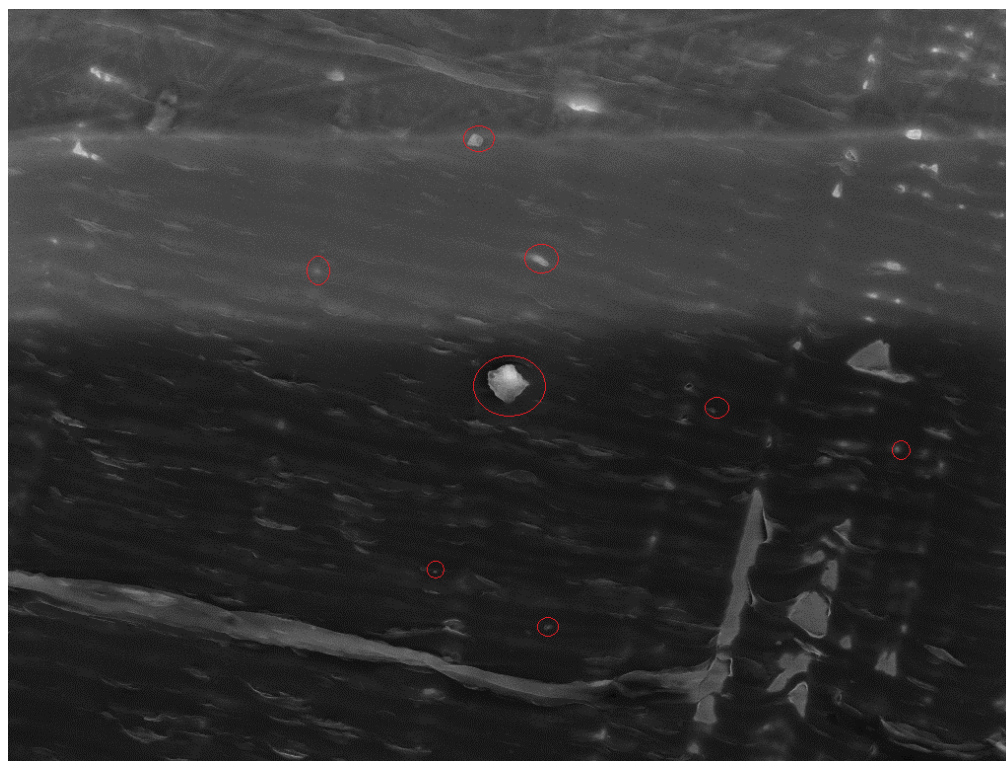


Figure 3.119 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen

Figure 3.120 shows a chunk of nanoclay particle inside the 10% nanoclay reinforced specimen.

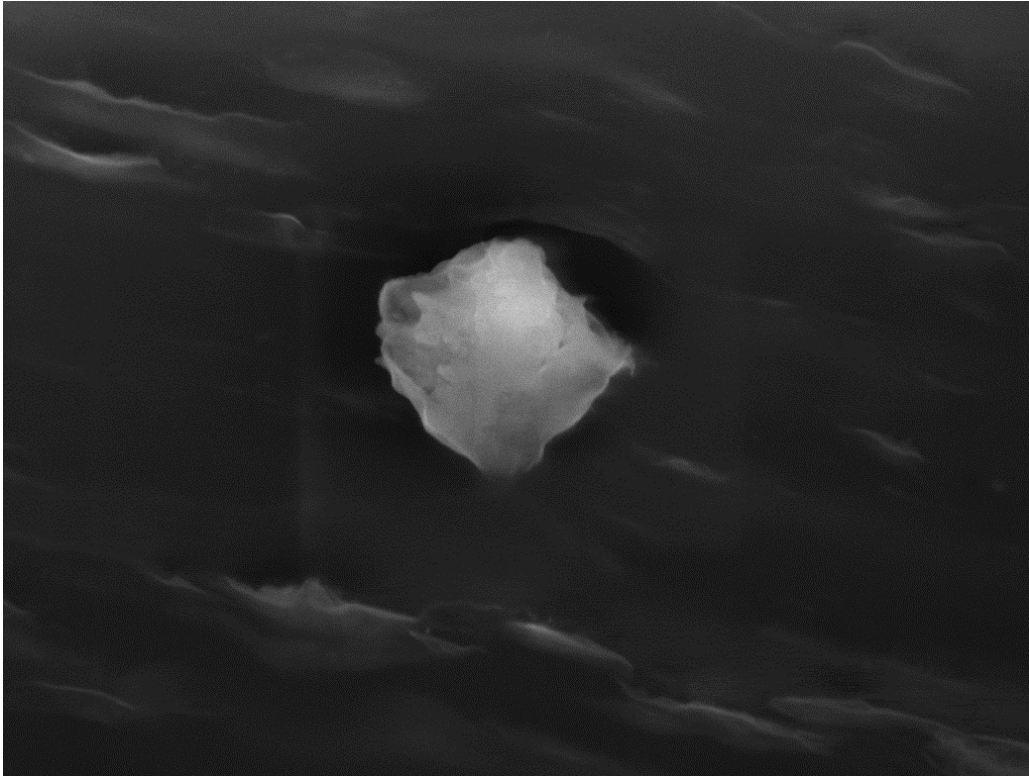


Figure3.120 Nanoclay particle showing agglomeration of nanoclay flakes.

Additional SEM images of 10% nanoclay reinforced PP specimens (Figures 3.121-3.124) are presented below.

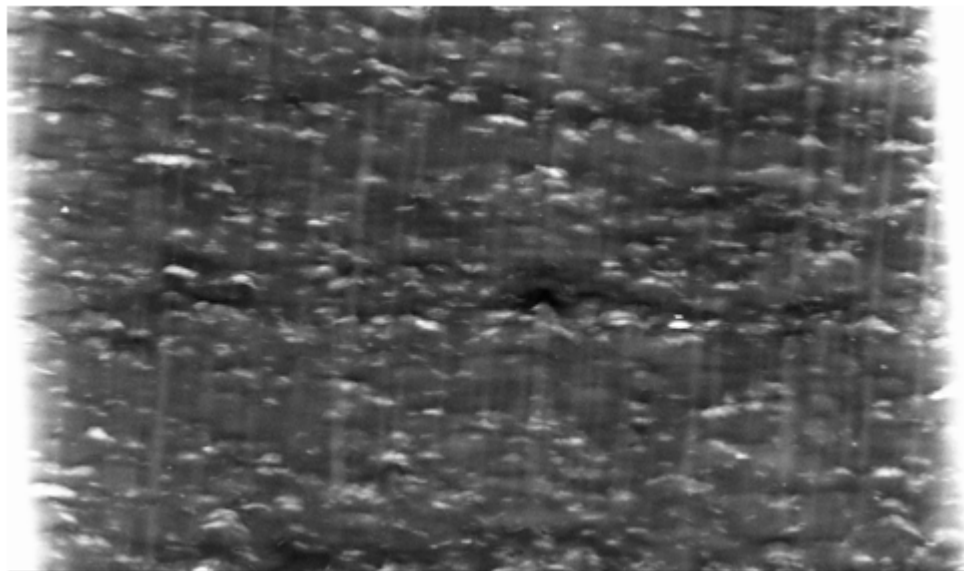


Figure 3.121 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen

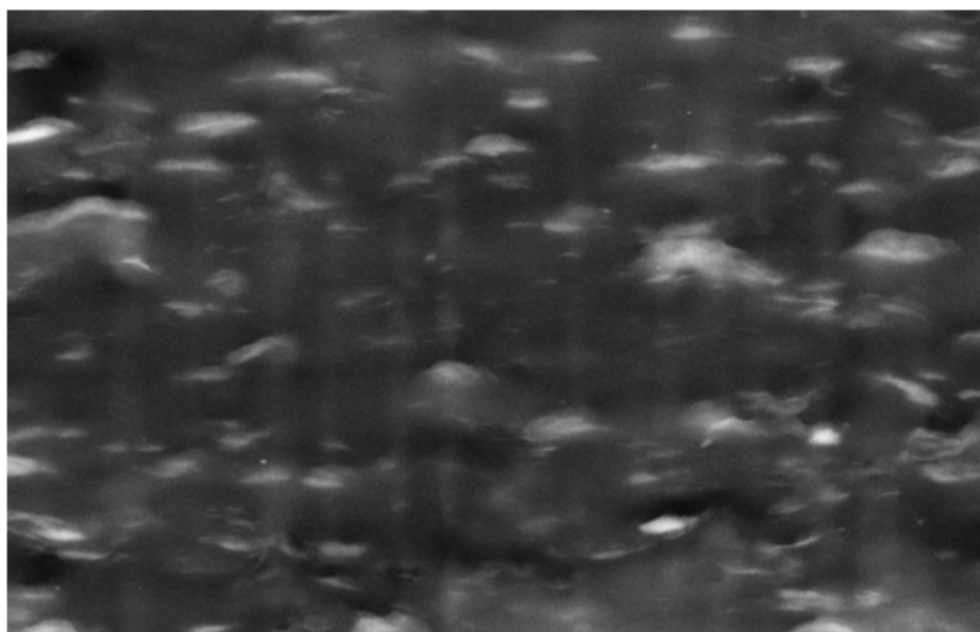


Figure 3.122 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen

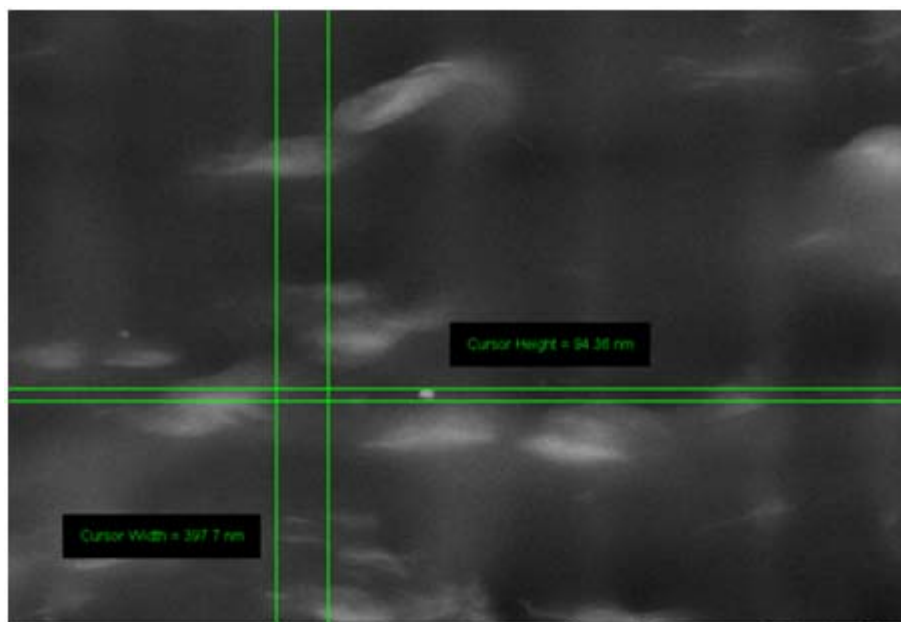


Figure 3.123 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen

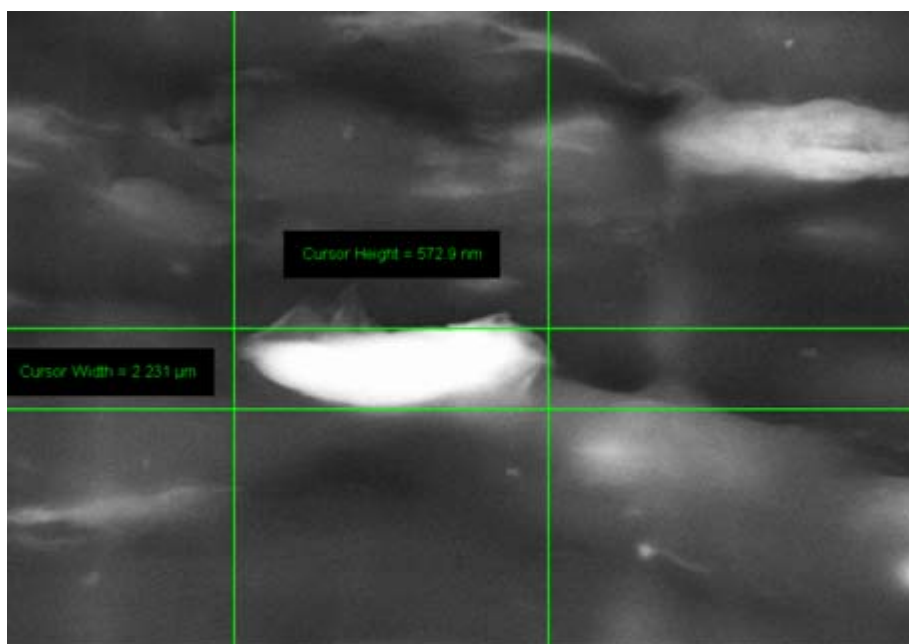


Figure 3.124 Dispersion of nanoclay particles in 10% nanoclay reinforced PP specimen.

Figure 3.123 and Figure 3.124 show two clay particles with different dimensions.

The SEM images for 1% nanoclay reinforced PP specimens are given below in Figures 3.123-3.125.

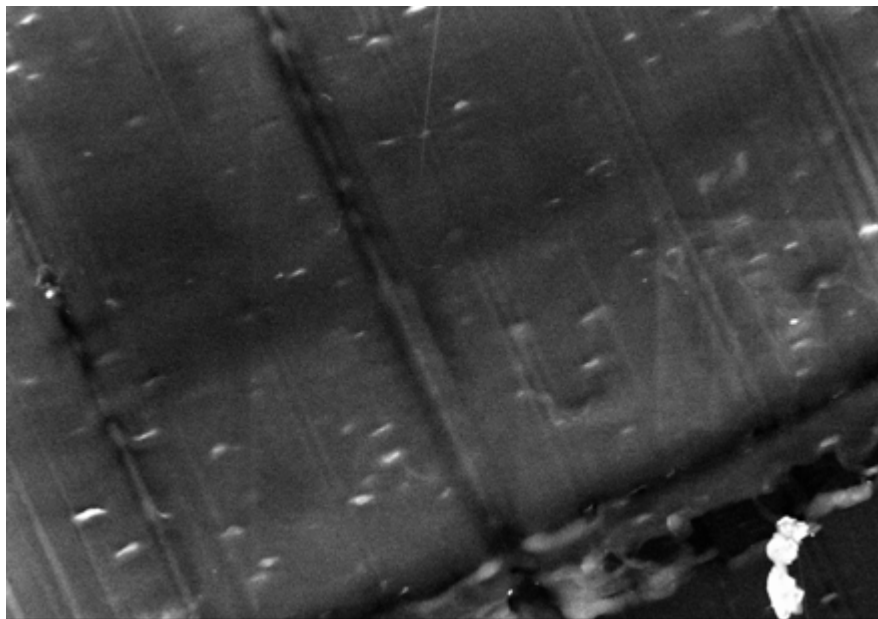


Figure 3.125 Dispersion of nanoclay particles in 1% nanoclay reinforced PP specimen

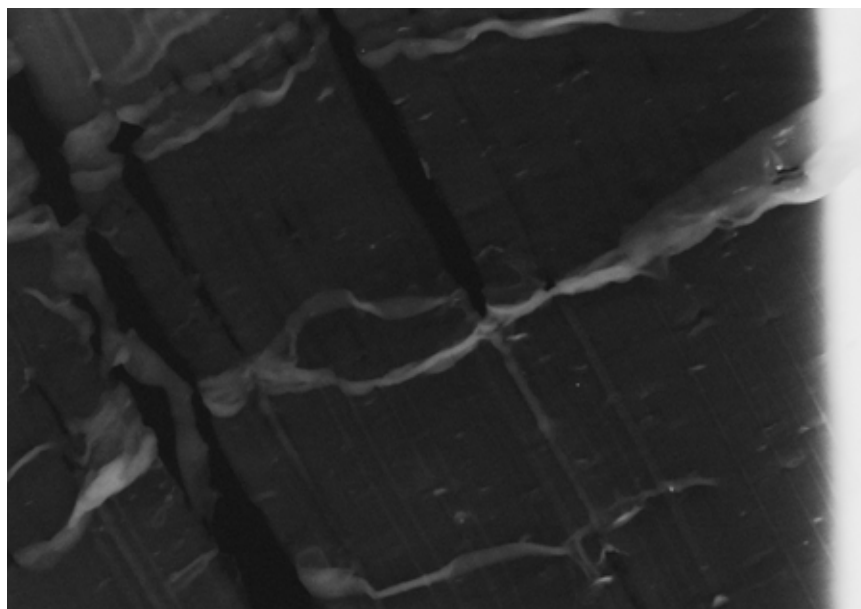


Figure 3.126 Dispersion of nanoclay particles in 1% nanoclay reinforced PP specimen

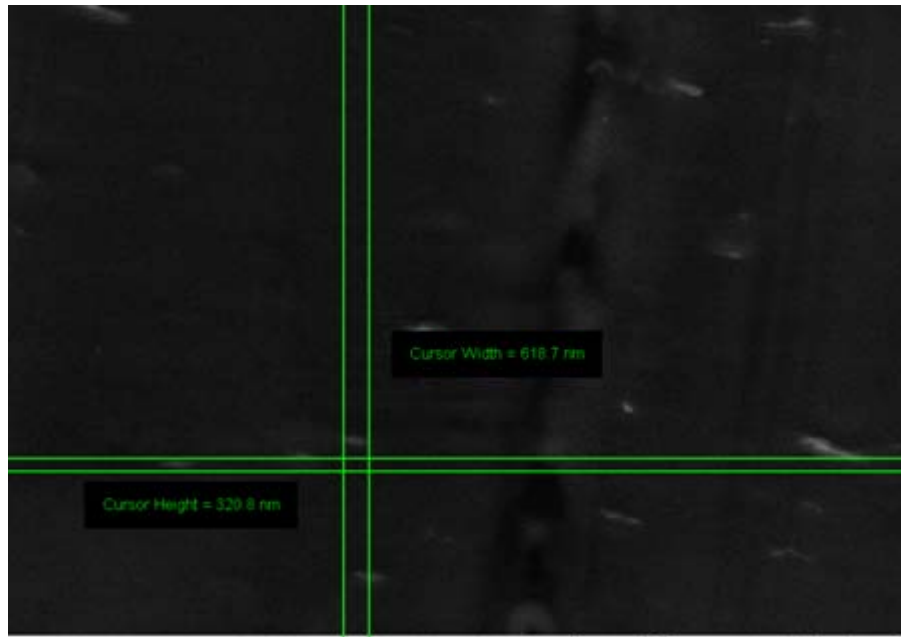


Figure 3.127 Dispersion of nanoclay particles in 1% nanoclay reinforced PP specimen

Figure 3.127 show the dimension of clay particle in a 1% nanoclay reinforced PP specimen.

The SEM images show that proper nanoclay particle dispersion at high percentages is difficult to achieve. Bad dispersion results in big chunks of nanoclay particles and possible debonding between particle and resin. Also it is observed that the dispersion of particles is more random than oriented.

4. MODELING AND SIMULATION

Some modeling and simulation studies also were carried out for this thesis. Testing is expensive and time consuming. Thus, the purpose of modeling and simulation studies is to develop models to predict the properties. If the models are successful, then they can be used to predict the properties for cases where test data are not available. Accordingly, micromechanics modeling and 3-D finite element was carried out to predict some of the mechanical properties obtained through the experimental program.

4.1 MICROMECHANICS MODELING

The purpose of micromechanics modeling of polymer/clay nanocomposites is to predict their elastic properties. For multi-scale modeling, intercalated and exfoliated clay systems were studied. A representative volume element consisting of N layers of intercalated or exfoliated clay platelets surrounded by polymer matrix was considered.

4.1.1 Application of Mori-Tanaka formulation for tensile testing

The Mori-Tanaka model, which is based on micromechanics and uses Eshelby's solution for inclusions embedded in an infinite matrix, is considered to simulate and predict the elastic properties of two-phase nanoclay reinforced nanocomposites. Here we approximate the nanoclay flakes as thin disks with the aspect ratio α calculated using their thickness and length. For analytical results, three different Mori Tanaka approaches are used. These approaches depend on particle orientation. The three approaches are "Oriented particles", "2-D Randomly distributed particles" and "3-D Randomly distributed particles".

4.1.1-a) Oriented Nanoclay Particles

Referring to [196], the normalized longitudinal Young's modulus (parallel to the tensile test direction), the normalized transverse Young's modulus, the normalized in plane shear modulus, the normalized out of plane shear modulus, the normalized plane strain bulk modulus and the major Poisson's ratio can be expressed as:

$$\frac{E_{11}}{E_m} = \frac{1}{1 + c[-2\nu_m A_3 + (1 - \nu_m)A_4 + (1 + \nu_m)A_5] / 2A} \quad (1)$$

$$\frac{E_{22}}{E_m} = \frac{1}{1 + c(A_1 + 2\nu_m A_2)A} \quad (2)$$

$$\frac{\mu_{21}}{\mu_m} = 1 + \frac{c}{\frac{\mu_m}{\mu_p - \mu_m} + 2(1 - c)S_{1212}} \quad (3)$$

$$\frac{\mu_{13}}{\mu_0} = 1 + \frac{c}{\frac{\mu_m}{\mu_p - \mu_m} + 2(1 - c)S_{2323}} \quad (4)$$

$$\frac{K_{13}}{K_m} = \frac{(1 + \nu_m)(1 - 2\nu_m)}{1 - \nu_m(1 + 2\nu_{21}) + c\{2(\nu_{21} - \nu_m)A_3 + [1 - \nu_m(1 + 2\nu_{21})]A_4\} / A} \quad (5)$$

where $K_m = \lambda_m + \mu_m$

$$\nu_{21}^2 = \frac{E_{22}}{E_{11}} - \frac{E_{22}}{4} \left(\frac{1}{\mu_{13}} + \frac{1}{K_{13}} \right) \quad (6)$$

and c is the volume fraction of nanoclay and the coefficients A_1, A_2, A_3, A_4, A_5 and A are given by:

$$A_1 = D_1(B_4 + B_5) - 2B_2 \quad (7)$$

$$A_2 = (1 + D_1)B_2 - (B_4 + B_5) \quad (8)$$

$$A_3 = B_1 - D_1B_3 \quad (9)$$

$$A_4 = (1 + D_1)B_1 - 2B_3 \quad (10)$$

$$A_5 = (1 - D_1)/(B_4 - B_5) \quad (11)$$

and

$$A = 2B_2B_3 - B_1(B_4 + B_5) \quad (11.a)$$

with

$$B_1 = cD_1 + D_2 + (1 - c)(D_1S_{1111} + 2S_{2211}) \quad (12)$$

$$B_2 = c + D_3 + (1 - c)(D_1S_{1122} + S_{2222} + S_{2233}) \quad (13)$$

$$B_3 = c + D_3 + (1 - c)[S_{1111} + (1 + D_1)S_{2211}] \quad (14)$$

$$B_4 = cD_1 + D_2 + (1 - c)(S_{1122} + D_1S_{2222} + S_{2233}) \quad (15)$$

$$B_5 = c + D_3 + (1 - c)(S_{1122} + S_{2222} + D_1S_{2233}) \quad (16)$$

and

$$D_1 = 1 + 2(\mu_p - \mu_m)/(\lambda_p - \lambda_m) \quad (17)$$

$$D_2 = (\lambda_m + 2\mu_m)/(\lambda_p - \lambda_m) \quad (18)$$

$$D_3 = \lambda_m/(\lambda_p - \lambda_m) \quad (19)$$

The D terms are defined by using μ_m, λ_m and μ_p, λ_p , which are the Lamé constants of the matrix and particles, respectively. In the B terms, the components of Eshelby's tensor S_{ijkl} given below are used:

$$S_{1111} = \frac{1}{2(1 - \nu_m)} \left\{ 1 - 2\nu_m + \frac{3\alpha^2 - 1}{\alpha^2 - 1} - \left[1 - 2\nu_m + \frac{3\alpha^2}{\alpha^2 - 1} \right] g \right\} \quad (20)$$

$$S_{2222} = S_{3333} = \frac{3}{8(1-\nu_m)} \frac{\alpha^2}{\alpha^2-1} + \frac{1}{4(1-\nu_m)} \left[1-2\nu_m - \frac{9}{4(\alpha^2-1)} \right] g \quad (21)$$

$$S_{2233} = S_{3322} = \frac{1}{4(1-\nu_m)} \left\{ \frac{\alpha^2}{2(\alpha^2-1)} - \left[1-2\nu_m + \frac{3}{4(\alpha^2-1)} \right] g \right\} \quad (22)$$

$$S_{2211} = S_{3311} = -\frac{1}{2(1-\nu_m)} \frac{\alpha^2}{\alpha^2-1} + \frac{1}{4(1-\nu_m)} \left\{ \frac{3\alpha^2}{\alpha^2-1} - (1-2\nu_m) \right\} g \quad (23)$$

$$S_{1122} = S_{1133} = -\frac{1}{2(1-\nu_m)} \left[1-2\nu_m + \frac{1}{\alpha^2-1} \right] + \frac{1}{2(1-\nu_m)} \left[1-2\nu_m + \frac{3}{2(\alpha^2-1)} \right] g \quad (24)$$

$$S_{2323} = S_{3232} = \frac{1}{4(1-\nu_m)} \left\{ \frac{\alpha^2}{2(\alpha^2-1)} + \left[1-2\nu_m - \frac{3}{4(\alpha^2-1)} \right] g \right\} \quad (25)$$

$$S_{1212} = S_{1313} = \frac{1}{4(1-\nu_m)} \left\{ 1-2\nu_m - \frac{\alpha^2+1}{\alpha^2-1} - \frac{1}{2} \left[1-2\nu_m - \frac{3(\alpha^2+1)}{\alpha^2-1} \right] g \right\} \quad (26)$$

Here ν_m is the Poisson's ratio of the matrix. In Eshelby's tensor, the g term has two different expressions depending on the aspect ratio of the inclusion:

$$g' = \frac{\alpha}{(\alpha^2-1)^{3/2}} \left\{ \alpha(\alpha^2-1)^{1/2} - \cosh^{-1} \alpha \right\} \quad \text{when } \alpha > 1 \quad (27.a)$$

$$g = \frac{\alpha}{(1-\alpha^2)^{3/2}} \left\{ \cos^{-1} \alpha - \alpha(1-\alpha^2)^{1/2} \right\} \quad \text{when } \alpha < 1 \quad (27.b)$$

Before we start the Mori-Tanaka calculations, first we have to determine some material and geometric properties of the constituents, namely matrix and nanoclay.

Matrix 1: PP 3371

E_m = average experimental result at each temperature value

ν_m = average experimental result at each temperature

V_m = volume fraction of matrix(calculated from weight fraction of matrix)

W_m = weight fraction of matrix

$\gamma_m = 8,829\text{N/m}^3$ (specific weight of matrix)

μ_m = calculated for average experimental result at each temperature value

λ_m = calculated for average experimental result at each temperature value

Matrix 2: EPON 828 Epoxy

$E_m = 2.8177$ GPa (average experimental result at room temperature)

$\nu_m = 0.3105$ (average experimental result at room temperature)

V_m = volume fraction of matrix (calculated from weight fraction of matrix)

W_m = weight fraction of matrix

$\gamma_m = 15,696\text{N/m}^3$ (specific weight of matrix)

$\mu_m = 1.075$ GPa (calculated)

$\lambda_m = 1.762$ GPa (calculated)

Particle: Nanoclay

$E_{nanoclay} = 300$ GPa (assumed from molecular dynamics calculations)

E_p = effective Young's modulus of particle = $E_{nanoclay}(N.d_s/t)$ (Referring to [181] and Figure 4.1)

$\nu_p = 0.2$ (assumed)

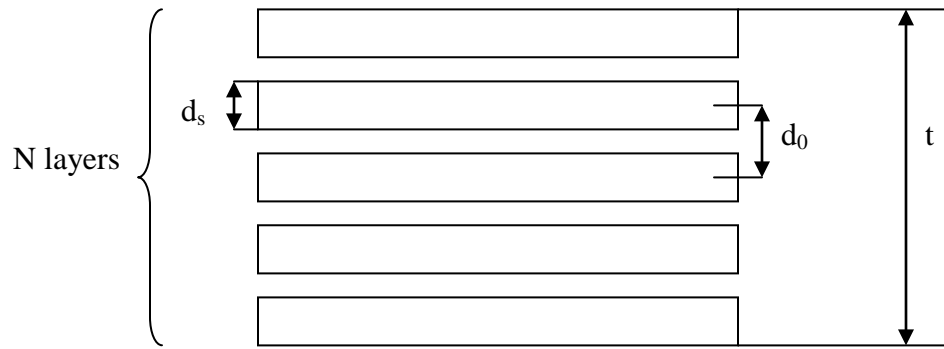
$V_p = c$ = volume fraction of nanoclay (calculated from weight fraction of nanoclay)

W_p = weight fraction of nanoclay

$\gamma_p = 18,639 \text{ N/m}^3$ (specific weight of nanoclay)

μ_p = effective shear modulus calculated for each particle geometry

λ_p = effective Lamè constant calculated for each particle geometry



$$\text{where } t = (N-1)d_0 + d_s$$

Figure 4.1 Model of nanoclay particle [181]

First, we calculate the volume fraction of nanoclay to be able to use the Mori-Tanaka formulation. The volume fraction in terms of the weight fraction can be written as:

$$c = \frac{W_p / \gamma_p}{W_p / \gamma_p + (1 - W_p) / \gamma_m} \quad (28)$$

For each nanoclay reinforcements, using Eq.28 the volume fraction of nanoclay c is obtained as (Table 4.1):

Table 4.1 Conversion of weight fractions of nanoclay to volume fractions

Weight fraction of nanoclay	Volume fraction of nanoclay
0.2%	0.095%
1%	0.48 %
3%	1.44 %
6%	2.93 %
10%	5 %

To determine the aspect ratio of nanoclay flakes, we use for thickness $t = (N-1)d_0 + d_s$ (referring to [181]) where N is the number of nanoclay flakes, $d_0 = 2.4$ nm, $d_s = 0.615$ nm and the diameter $D = 200$ nm. Thus, the aspect ratio of nanoclay can be calculated as: $\alpha = \frac{t}{D}$. For the parametric study, 6 different values of N are used and as a consequence 6 different aspect ratios are obtained. But for all cases the aspect ratio of nanoclay is smaller than 1 and thus we use g in Eshelby's tensor for the parametric study. The calculations for various weight fractions (converted to volume fractions), Young's moduli and thicknesses of nanoclay particles were performed using MATLAB R2007b.

4.1.1-b) Results of Parametric Study for Oriented Nanoclay Particles

To obtain the effect of the number of nanoclay flakes in a particle and the effect of the nanoclay's Young's modulus on nanocomposite properties a parametric study was performed. As a result of the parametric study, in the ensuing Mori-Tanaka calculations the number of flakes N in a nanoclay particle was varied from 1 to 6 and the nanoclay Young's modulus $E_{\text{nanoclay}} = 300$ GPa was used to calculate the particle thickness t and the effective particle Young's modulus E_p . For PP 3371 specimens with nanoclay reinforcement, the Mori-Tanaka calculations were performed using the different Young's modulus values obtained from the tensile tests at various

temperatures. For EPON 828 epoxy specimens with nanoclay reinforcement, the Mori-Tanaka calculations were carried out at room temperature only for $N = 6$ and $E_{\text{nanoclay}} = 300\text{GPa}$. The comparison of Mori-Tanaka calculations and experimental results for oriented particles is given in Tables 4.2 - 4.6. The comparison for the epoxy based nanocomposite is shown in Table 4.7.

Table 4.2 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for oriented particles
at 160°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E_{11} (GPa)	E_{22} (GPa)	Poisson's Ratio (Experimental)	ν_{21}
300	200	1	300	0.615	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.4586	0.4186	0.4782	0.4481
					1%	0.4654	0.7212	0.5081	0.4715	0.5019
					3%	0.4969	1.3800	0.6577	0.4523	0.5522
					6%	0.5031	2.4061	0.7884	0.5077	0.5767
					10%	0.5146	3.8646	0.8934	0.4896	0.5884
300	200	2	122.388	3.015	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.4077	0.3986	0.4782	0.4314
					1%	0.4654	0.4704	0.4238	0.4715	0.4515
					3%	0.4969	0.6287	0.4817	0.4523	0.4872
					6%	0.5031	0.8740	0.5574	0.5077	0.5195
					10%	0.5146	1.2213	0.6436	0.4896	0.5442
300	200	3	102.216	5.415	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.4013	0.3960	0.4782	0.4290
					1%	0.4654	0.4386	0.4116	0.4715	0.4418
					3%	0.4969	0.5333	0.4490	0.4523	0.4675
					6%	0.5031	0.6804	0.5020	0.5077	0.4952
					10%	0.5146	0.8887	0.5678	0.4896	0.5202
300	200	4	94.434	7.815	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3987	0.3949	0.4782	0.4281
					1%	0.4654	0.4257	0.4065	0.4715	0.4375
					3%	0.4969	0.4945	0.4349	0.4523	0.4576
					6%	0.5031	0.6015	0.4763	0.5077	0.4813
					10%	0.5146	0.7533	0.5300	0.4896	0.5046

Table 4.2 Continued

300	200	5	90.308	10.215	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3973	0.3944	0.4782	0.4275
					1%	0.4654	0.4187	0.4037	0.4715	0.4350
					3%	0.4969	0.4732	0.4269	0.4523	0.4516
					6%	0.5031	0.5583	0.4615	0.5077	0.4722
					10%	0.5146	0.6790	0.5073	0.4896	0.4936
300	200	6	87.753	12.6150	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3965	0.3940	0.4782	0.4272
					1%	0.4654	0.4143	0.4020	0.4715	0.4334
					3%	0.4969	0.4598	0.4218	0.4523	0.4476
					6%	0.5031	0.5309	0.4518	0.5077	0.4657
					10%	0.5146	0.6320	0.4921	0.4896	0.4854

Table 4.3 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/ clay nanocomposites for oriented particles at 120°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E_{11} (GPa)	E_{22} (GPa)	Poisson's Ratio (Experimental)	ν_{21}
300	200	1	300	0.615	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.7079	0.6491	0.4350	0.4212
					1%	0.7646	1.0721	0.7585	0.4631	0.4644
					3%	0.8283	1.9875	0.9348	0.4355	0.5070
					6%	0.8332	3.4107	1.0835	0.4334	0.5283
					10%	0.8664	5.4272	1.2016	0.4111	0.5385
300	200	2	122.388	3.015	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6386	0.6246	0.4350	0.4085
					1%	0.7646	0.7296	0.6576	0.4631	0.4249
					3%	0.8283	0.9597	0.7322	0.4355	0.4545
					6%	0.8332	1.3166	0.8274	0.4334	0.4817
					10%	0.8664	1.8211	0.9332	0.4111	0.5027

Table 4.3 Continued

300	200	3	102.216	5.415	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6297	0.6213	0.4350	0.4067
					1%	0.7646	0.6849	0.6422	0.4631	0.4172
					3%	0.8283	0.8251	0.6921	0.4355	0.4388
					6%	0.8332	1.0432	0.7613	0.4334	0.4622
					10%	0.8664	1.3518	0.8457	0.4111	0.4835
300	200	4	94.434	7.815	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6260	0.6199	0.4350	0.4059
					1%	0.7646	0.6663	0.6357	0.4631	0.4138
					3%	0.8283	0.7692	0.6741	0.4355	0.4309
					6%	0.8332	0.9295	0.7295	0.4334	0.4510
					10%	0.8664	1.1566	0.8002	0.4111	0.4709
300	200	5	90.308	10.215	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6239	0.6192	0.4350	0.4055
					1%	0.7646	0.6561	0.6321	0.4631	0.4118
					3%	0.8283	0.7382	0.6639	0.4355	0.4260
					6%	0.8332	0.8664	0.7107	0.4334	0.4435
					10%	0.8664	1.0482	0.7722	0.4111	0.4619
300	200	6	87.753	12.6150	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6226	0.6187	0.4350	0.4052
					1%	0.7646	0.6496	0.6297	0.4631	0.4105
					3%	0.8283	0.7184	0.6572	0.4355	0.4226
					6%	0.8332	0.8260	0.6983	0.4334	0.4382
					10%	0.8664	0.9789	0.7531	0.4111	0.4551

Table 4.4 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for oriented particles at room temperature

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E_{11} (GPa)	E_{22} (GPa)	Poisson's Ratio (Experimental)	ν_{21}
300	200	1	300	0.615	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.3379	1.2487	0.4167	0.4072
					1%	1.5085	1.8857	1.4140	0.4056	0.4411
					3%	1.6278	3.2622	1.6967	0.3852	0.4789
					6%	1.6463	5.3929	1.9512	0.3722	0.4995
					10%	1.7426	8.3927	2.1624	0.3725	0.5092
300	200	2	122.388	3.015	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2374	1.2139	0.4167	0.3984
					1%	1.5085	1.3871	1.2674	0.4056	0.4114
					3%	1.6278	1.7662	1.3893	0.3852	0.4358
					6%	1.6463	2.3531	1.5476	0.3722	0.4592
					10%	1.7426	3.1795	1.7270	0.3725	0.4778
300	200	3	102.216	5.415	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2236	1.2090	0.4167	0.3970
					1%	1.5085	1.3184	1.2443	0.4056	0.4058
					3%	1.6278	1.5592	1.3286	0.3852	0.4239
					6%	1.6463	1.9333	1.4464	0.3722	0.4440
					10%	1.7426	2.4611	1.5914	0.3725	0.4625
300	200	4	94.434	7.815	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2177	1.2069	0.4167	0.3965
					1%	1.5085	1.2887	1.2341	0.4056	0.4031
					3%	1.6278	1.4696	1.3005	0.3852	0.4177
					6%	1.6463	1.7513	1.3966	0.3722	0.4351
					10%	1.7426	2.1495	1.5198	0.3725	0.4525

Table 4.4 Continued

300	200	5	90.308	10.215	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2143	1.2057	0.4167	0.3961
					1%	1.5085	1.2718	1.2283	0.4056	0.4016
					3%	1.6278	1.4184	1.2841	0.3852	0.4138
					6%	1.6463	1.6472	1.3665	0.3722	0.4291
					10%	1.7426	1.9712	1.4747	0.3725	0.4452
300	200	6	87.753	12.6150	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2121	1.2050	0.4167	0.3959
					1%	1.5085	1.2608	1.2245	0.4056	0.4005
					3%	1.6278	1.3851	1.2733	0.3852	0.4111
					6%	1.6463	1.5793	1.3462	0.3722	0.4247
					10%	1.7426	1.8547	1.4435	0.3725	0.4397

Table 4.5 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for oriented particles at -4°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E_{11} (GPa)	E_{22} (GPa)	Poisson's Ratio (Experimental)	ν_{21}
300	200	1	300	0.615	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.5514	3.4050	0.3307	0.3481
					1%	4.0082	4.3722	3.6122	0.3156	0.3617
					3%	4.1786	6.4451	3.9978	0.3114	0.3811
					6%	4.2123	9.6416	4.3843	0.3056	0.3942
					10%	4.3331	14.1024	4.7373	0.2928	0.4011
300	200	2	122.388	3.015	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.4115	3.3667	0.3307	0.3453
					1%	4.0082	3.6732	3.4455	0.3156	0.3507
					3%	4.1786	4.3374	3.6283	0.3114	0.3617
					6%	4.2123	5.3652	3.8730	0.3056	0.3733
					10%	4.3331	6.8037	4.1611	0.2928	0.3833

Table 4.5 Continued

300	200	3	102.216	5.415	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3914	3.3610	0.3307	0.3448
					1%	4.0082	3.5726	3.4185	0.3156	0.3488
					3%	4.1786	4.0335	3.5568	0.3114	0.3574
					6%	4.2123	4.7488	3.7527	0.3056	0.3675
					10%	4.3331	5.7534	3.9981	0.2928	0.3772
300	200	4	94.434	7.815	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3822	3.3584	0.3307	0.3446
					1%	4.0082	3.5260	3.4057	0.3156	0.3479
					3%	4.1786	3.8925	3.5218	0.3114	0.3551
					6%	4.2123	4.4628	3.6906	0.3056	0.3641
					10%	4.3331	5.2656	3.9091	0.2928	0.3734
300	200	5	90.308	10.215	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3765	3.3568	0.3307	0.3445
					1%	4.0082	3.4978	3.3980	0.3156	0.3473
					3%	4.1786	3.8070	3.5001	0.3114	0.3536
					6%	4.2123	4.2890	3.6511	0.3056	0.3618
					10%	4.3331	4.9689	3.8505	0.2928	0.3706
300	200	6	87.753	12.6150	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3727	3.3557	0.3307	0.3444
					1%	4.0082	3.4784	3.3928	0.3156	0.3469
					3%	4.1786	3.7484	3.4851	0.3114	0.3525
					6%	4.2123	4.1697	3.6233	0.3056	0.3600
					10%	4.3331	4.7650	3.8085	0.2928	0.3683

Table 4.6 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for oriented particles at -65°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E_{11} (GPa)	E_{22} (GPa)	Poisson's Ratio (Experimental)	ν_{21}
300	200	1	300	0.615	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	4.0780	3.9263	0.3283	0.3467
					1%	4.6031	4.9307	4.1454	0.3193	0.3590
					3%	4.6944	7.0845	4.5646	0.3076	0.3772
					6%	4.9732	10.4035	4.9980	0.3039	0.3900
					10%	5.0920	15.0296	5.4032	0.2983	0.3970
300	200	2	122.388	3.015	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.9340	3.8868	0.3283	0.3442
					1%	4.6031	4.2111	3.9715	0.3193	0.3491
					3%	4.6944	4.9146	4.1697	0.3076	0.3591
					6%	4.9732	6.0026	4.4384	0.3039	0.3699
					10%	5.0920	7.5236	4.7590	0.2983	0.3792
300	200	3	102.216	5.415	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.9134	3.8809	0.3283	0.3438
					1%	4.6031	4.1078	3.9436	0.3193	0.3474
					3%	4.6944	4.6025	4.0950	0.3076	0.3553
					6%	4.9732	5.3699	4.3111	0.3039	0.3646
					10%	5.0920	6.4462	4.5842	0.2983	0.3737
300	200	4	94.434	7.815	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.9038	3.8782	0.3283	0.3437
					1%	4.6031	4.0597	3.9303	0.3193	0.3466
					3%	4.6944	4.4569	4.0584	0.3076	0.3533
					6%	4.9732	5.0743	4.2456	0.3039	0.3616
					10%	5.0920	5.9426	4.4893	0.2983	0.3703

Table 4.6 Continued

300	200	5	90.308	10.215	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8979	3.8765	0.3283	0.3436
					1%	4.6031	4.0302	3.9222	0.3193	0.3461
					3%	4.6944	4.3677	4.0355	0.3076	0.3520
					6%	4.9732	4.8933	4.2037	0.3039	0.3595
					10%	5.0920	5.6339	4.4268	0.2983	0.3677
300	200	6	87.753	12.6150	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8939	3.8754	0.3283	0.3435
					1%	4.6031	4.0099	3.9166	0.3193	0.3457
					3%	4.6944	4.3062	4.0196	0.3076	0.3510
					6%	4.9732	4.7682	4.1741	0.3039	0.3580
					10%	5.0920	5.4202	4.3817	0.2983	0.3657

Table 4.7 Comparison of Mori-Tanaka calculations and experimental results of EPON 828 epoxy / clay nanocomposites for oriented particles at room temperature

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E_{11} (GPa)	E_{22} (GPa)	Poisson's Ratio (Experimental)	ν_{21}
300	200	6	87.7525	12.6150	0%	2.8177	2.8177	2.8177	0.3105	0.3105
					1%	2.9127	2.8791	3.0247	0.3238	0.3150
					3%	2.9840	2.9977	3.4467	0.3288	0.3236
					6%	3.0965	3.1682	4.1018	0.3340	0.3316
					10%	3.3427	3.3880	5.0218	0.3348	0.3404

4.1.1-c) 2-D Randomly Distributed Nanoclay Particles

Here the nanoclay flakes are in a plane parallel to test direction, but are randomly distributed in the plane. Referring to [197], the normalized in plane Young's modulus, the normalized in plane shear modulus, the normalized out of plane shear modulus and the normalized out of plane Young's modulus for the case of 2-D randomly distributed nanoclay particles can be expressed as:

$$\frac{E_{11}}{E_m} = \left(\frac{1}{1 + cp_{11}} \right) \quad (29)$$

where c is the volume fraction given in equation (28) and p_{11} is:

$$p_{11} = \frac{1}{1 + c(b_1 - b_2)} \left\{ \frac{2(a_1 + a_2 - a_3) + a_4 + a_5 a}{16a} - \frac{1}{4[2S_{1212} + \mu_m / (\mu_p - \mu_m)]} \right\} \\ - \frac{(1 - \nu_m)(1 + cb_5) + 2c\nu_m b_3}{2c^2 b_3 b_4 - (1 + cb_5)[1 + c(b_1 + b_2)]} \frac{2(a_1 - a_2 + a_3) + a_4 + a_5 a}{8a} \quad (30) \\ + \frac{(1 - \nu_m)cb_4 + \nu_m [1 + c(b_1 + b_2)]}{2c^2 b_3 b_4 - (1 + cb_5)[1 + c(b_1 + b_2)]} \frac{-2a_2 + a_4 - a_5 a}{4a}$$

$$\frac{\mu_{12}}{\mu_m} = \left(\frac{1}{1 + cp_{12}} \right) \quad (31)$$

with p_{12} given as :

$$p_{12} = \left[\frac{2(a_1 + a_2 - a_3) + a_4 + a_5 a - \frac{4a}{2S_{1212} + \mu_m / (\mu_p - \mu_m)}}{8a + c[(S_{1122} - S_{2222} + 1)(2a_3 - a_4 - a_5 a) + 2(S_{1111} - S_{2211} - 1)(a_1 + a_2) + (S_{1122} - S_{2233})(2a_3 - a_4 + a_5 a) - \frac{4a(2S_{1212} - 1)}{2S_{1212} + \mu_m / (\mu_p - \mu_m)}]} \right] \quad (32)$$

$$\frac{\mu_{13}}{\mu_m} = \left(\frac{1}{1 + cp_{13}} \right) \quad (33)$$

with p_{13}

$$p_{13} = - \left[\frac{1}{2S_{1313} + \mu_m / (\mu_p - \mu_m)} + \frac{1}{2S_{2323} + \mu_m / (\mu_p - \mu_m)} \right] / \left\{ 2 - c \left[\frac{2S_{1313} - 1}{2S_{1313} + \mu_m / (\mu_p - \mu_m)} + \frac{2S_{2323} - 1}{2S_{2323} + \mu_m / (\mu_p - \mu_m)} \right] \right\} \quad (34)$$

$$\frac{E_{33}}{E_m} = \frac{1}{1 + cp_{33}} \quad (35)$$

where p_{33} is,

$$p_3 = \frac{\{ [v_m(1 + cb_5) + cb_3](2a_3 + a_4 - a_5a) - [1 + c(b_1 + b_2 + 2v_0b_4)](a_4 + a_5a) \}}{2a \{ 2c^2b_3b_4 - (1 + cb_5)[1 + c(b_1 + b_2)] \}} \quad (36)$$

The definition of the components of Eshelby's tensor and g were given in section 4.1.1-a). The other constants which are used to calculate the material properties of randomly distributed nanoclay particles are given below:

$$a_1 = 6(\kappa_p - \kappa_m)(\mu_p - \mu_m)(S_{2222} + S_{2233} - 1) - 2(\kappa_m\mu_p - \kappa_p\mu_m) + 6\kappa_p(\mu_p - \mu_m) \quad (37)$$

$$a_2 = 6(\kappa_p - \kappa_m)(\mu_p - \mu_m)S_{1133} + 2(\kappa_m\mu_p - \kappa_p\mu_m) \quad (38)$$

$$a_3 = -6(\kappa_p - \kappa_m)(\mu_p - \mu_m)S_{3311} - 2(\kappa_m\mu_p - \kappa_p\mu_m) \quad (39)$$

$$a_4 = 6(\kappa_p - \kappa_m)(\mu_p - \mu_m)(S_{1111} - 1) + 2(\kappa_m\mu_p - \kappa_p\mu_m) + 6\mu_p(\kappa_p - \kappa_m) \quad (40)$$

$$a_5 = 1 / [S_{3322} - S_{3333} + 1 - \mu_p / (\mu_p - \mu_m)] \quad (41)$$

$$\begin{aligned}
a = & 6(\kappa_p - \kappa_m)(\mu_p - \mu_m)[2S_{1133}S_{3311} - (S_{1111} - 1)(S_{3322} + S_{3333} - 1)] \\
& + 2(\kappa_m\mu_p - \kappa_p\mu_m)[2(S_{1133} + S_{3311}) + (S_{1111} - S_{3322} - S_{3333})] \\
& - 6\kappa_p(\mu_p - \mu_m)(S_{1111} - 1) - 6\mu_p(\kappa_p - \kappa_m)(S_{2222} + S_{2233} - 1) - 6\kappa_p\mu_p
\end{aligned} \tag{42}$$

$$\begin{aligned}
b_1 = & (1/16a)\{2a_3(6S_{1122} + S_{2222} + S_{2233} - 1) \\
& + a_4[3(S_{2222} + S_{2233} - 1) + 2S_{1122}] + 3a_5a(S_{2222} - S_{2233} - 1) \\
& + 2a_1[3(S_{1111} - 1) + S_{2211}] - 2a_2(S_{1111} + 3S_{2211} - 1) \\
& - 4a(2S_{1212} - 1)/[2S_{1212} + \mu_m/(\mu_p - \mu_m)]\}
\end{aligned} \tag{43}$$

$$\begin{aligned}
b_2 = & (1/16a)\{2a_3[2S_{1111} + 3(S_{2222} + S_{2233} - 1)] \\
& + a_4(6S_{1122} + S_{2222} + S_{2233} - 1) + a_5a(S_{2222} - S_{2233} - 1) \\
& + 2a_1(S_{1111} + 3S_{2211} - 1) - 2a_2[S_{2211} + 3(S_{1111} - 1)] \\
& + 4a(2S_{1212} - 1)/[2S_{1212} + \mu_m/(\mu_p - \mu_m)]\}
\end{aligned} \tag{44}$$

$$\begin{aligned}
b_3 = & (1/a)\{-2a_2(S_{1111} + S_{2211} - 1) + a_4(2S_{1122} + S_{2222} + S_{2233} - 1) \\
& - a_5a(S_{2222} - S_{2233} - 1)\}
\end{aligned} \tag{45}$$

$$\begin{aligned}
b_4 = & (1/4a)\{2(a_1 - a_2)S_{2211} + (2a_3 + a_4)(S_{2222} + S_{2233} - 1) \\
& - a_5a(S_{2222} - S_{2233} - 1)\}
\end{aligned} \tag{46}$$

$$\begin{aligned}
b_5 = & (1/2a)\{-2a_2S_{2211} + a_4(S_{2222} + S_{2233} - 1) \\
& + a_5a(S_{2222} - S_{2233} - 1)\}
\end{aligned} \tag{47}$$

4.1.1-d) Results of Parametric Study for 2-D Randomly Distributed Nanoclay Particles

For the parametric study, in the Mori-Tanaka calculations the number of flakes N in a nanoclay particle was varied from 1 to 6 and the nanoclay Young's modulus $E_{\text{nanoclay}}=300$ GPa was used to calculate the particle thickness t and the effective particle Young's modulus E_p . For PP 3371 specimens with nanoclay reinforcement, the Mori-Tanaka calculations were performed with the Young's modulus values obtained from the tensile tests at various temperatures. For EPON 828 epoxy specimens, the Mori-Tanaka calculations were carried out at room temperature only for

$N=6$ and $E_{\text{nanoclay}}=300\text{GPa}$. The comparison of Mori-Tanaka calculations and experimental results for 2-D randomly distributed particles in PP 3371 and EPON 828 epoxy is given in Tables 4.8 - 4.12 and Table 4.13 respectively.

Table 4.8 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 2-D randomly distributed particles at 160°F

E_{nanoclay} ν (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	$E_{11}=E_{22}$ (GPa)	E_{33} (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.4222	0.4590	0.4782	0.4493
					1%	0.4654	0.5342	0.7280	0.4715	0.4948
					3%	0.4969	0.7950	1.4101	0.4523	0.5086
					6%	0.5031	1.1819	2.4669	0.5077	0.4875
					10%	0.5146	1.7173	3.9562	0.4896	0.4579
300	200	2	122.388	3.015	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3994	0.4077	0.4782	0.4318
					1%	0.4654	0.4282	0.4709	0.4715	0.4524
					3%	0.4969	0.4983	0.6327	0.4523	0.4839
					6%	0.5031	0.6022	0.8861	0.5077	0.5042
					10%	0.5146	0.7446	1.2461	0.4896	0.5105
300	200	3	102.216	5.415	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3965	0.4013	0.4782	0.4293
					1%	0.4654	0.4142	0.4388	0.4715	0.4425
					3%	0.4969	0.4579	0.5350	0.4523	0.4671
					6%	0.5031	0.5235	0.6860	0.5077	0.4895
					10%	0.5146	0.6138	0.9013	0.4896	0.5042
300	200	4	94.434	7.815	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3953	0.3988	0.4782	0.4282
					1%	0.4654	0.4084	0.4259	0.4715	0.4380
					3%	0.4969	0.4411	0.4954	0.4523	0.4578
					6%	0.5031	0.4906	0.6048	0.5077	0.4785
					10%	0.5146	0.5591	0.7610	0.4896	0.4953

Table 4.8 Continued

300	200	5	90.308	10.215	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3947	0.3973	0.4782	0.4277
					1%	0.4654	0.4052	0.4188	0.4715	0.4354
					3%	0.4969	0.4318	0.4738	0.4523	0.4519
					6%	0.5031	0.4724	0.5605	0.5077	0.4706
					10%	0.5146	0.5287	0.6843	0.4896	0.4875
300	200	6	87.753	12.6150	0%	0.39204	0.39204	0.39204	0.4256	0.4256
					0.2%	0.4499	0.3943	0.3965	0.4782	0.4273
					1%	0.4654	0.4033	0.4143	0.4715	0.4337
					3%	0.4969	0.4259	0.4602	0.4523	0.4479
					6%	0.5031	0.4607	0.5325	0.5077	0.4646
					10%	0.5146	0.5092	0.6359	0.4896	0.4809

Table 4.9 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 2-D randomly distributed particles at 120°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	$E_{11}=E_{22}$ (GPa)	E_{33} (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6565	0.7084	0.4350	0.4225
					1%	0.7646	0.8081	1.0795	0.4631	0.4598
					3%	0.8283	1.1627	2.0199	0.4355	0.4721
					6%	0.8332	1.6907	3.4760	0.4334	0.4547
					10%	0.8664	2.4245	5.5244	0.4111	0.4300
300	200	2	122.388	3.015	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6263	0.6387	0.4350	0.4089
					1%	0.7646	0.6672	0.7303	0.4631	0.4259
					3%	0.8283	0.7665	0.9645	0.4355	0.4522
					6%	0.8332	0.9143	1.3308	0.4334	0.4691
					10%	0.8664	1.1171	1.8500	0.4111	0.4739

Table 4.9 Continued

300	200	3	102.216	5.415	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6224	0.6297	0.4350	0.4069
					1%	0.7646	0.6479	0.6851	0.4631	0.4180
					3%	0.8283	0.7111	0.8272	0.4355	0.4388
					6%	0.8332	0.8062	1.0500	0.4334	0.4575
					10%	0.8664	0.9371	1.3670	0.4111	0.4696
300	200	4	94.434	7.815	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6207	0.6260	0.4350	0.4060
					1%	0.7646	0.6398	0.6665	0.4631	0.4144
					3%	0.8283	0.6876	0.7704	0.4355	0.4312
					6%	0.8332	0.7602	0.9336	0.4334	0.4487
					10%	0.8664	0.8605	1.1663	0.4111	0.4627
300	200	5	90.308	10.215	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6198	0.6239	0.4350	0.4056
					1%	0.7646	0.6354	0.6562	0.4631	0.4122
					3%	0.8283	0.6745	0.7389	0.4355	0.4263
					6%	0.8332	0.7343	0.8691	0.4334	0.4422
					10%	0.8664	0.8173	1.0550	0.4111	0.4564
300	200	6	87.753	12.6150	0%	0.61595	0.61595	0.61595	0.4038	0.4038
					0.2%	0.7285	0.6193	0.6226	0.4350	0.4053
					1%	0.7646	0.6325	0.6496	0.4631	0.4108
					3%	0.8283	0.6661	0.7189	0.4355	0.4229
					6%	0.8332	0.7176	0.8280	0.4334	0.4372
					10%	0.8664	0.7895	0.9839	0.4111	0.4510

Table 4.10 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 2-D randomly distributed particles at RT

$E_{nanoclay}$ y (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	$E_{11}=E_{22}$ (GPa)	E_{33} (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2607	1.3384	0.4167	0.4086
					1%	1.5085	1.4898	1.8940	0.4056	0.4404
					3%	1.6278	2.0290	3.3022	0.3852	0.4577
					6%	1.6463	2.8298	5.4780	0.3722	0.4475
					10%	1.7426	3.9392	8.5241	0.3725	0.4275
300	200	2	122.388	3.015	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2171	1.2374	0.4167	0.3988
					1%	1.5085	1.2843	1.3880	0.4056	0.4128
					3%	1.6278	1.4488	1.7726	0.3852	0.4357
					6%	1.6463	1.6943	2.3725	0.3722	0.4520
					10%	1.7426	2.0310	3.2196	0.3725	0.4583
300	200	3	102.216	5.415	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2111	1.2236	0.4167	0.3973
					1%	1.5085	1.2548	1.3188	0.4056	0.4067
					3%	1.6278	1.3636	1.5621	0.3852	0.4246
					6%	1.6463	1.5279	1.9431	0.3722	0.4416
					10%	1.7426	1.7541	2.4834	0.3725	0.4533
300	200	4	94.434	7.815	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2085	1.2177	0.4167	0.3966
					1%	1.5085	1.2420	1.2889	0.4056	0.4038
					3%	1.6278	1.3262	1.4713	0.3852	0.4185
					6%	1.6463	1.4543	1.7574	0.3722	0.4341
					10%	1.7426	1.6316	2.1642	0.3725	0.4471

Table 4.10 Continued

300	200	5	90.308	10.215	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2070	1.2143	0.4167	0.3962
					1%	1.5085	1.2348	1.2719	0.4056	0.4020
					3%	1.6278	1.3047	1.4196	0.3852	0.4144
					6%	1.6463	1.4118	1.6515	0.3722	0.4286
					10%	1.7426	1.5607	1.9817	0.3725	0.4416
300	200	6	87.753	12.6150	0%	1.2001	1.2001	1.2001	0.3947	0.3947
					0.2%	1.3172	1.2060	1.2121	0.4167	0.3960
					1%	1.5085	1.2300	1.2609	0.4056	0.4008
					3%	1.6278	1.2906	1.3859	0.3852	0.4116
					6%	1.6463	1.3839	1.5824	0.3722	0.4244
					10%	1.7426	1.5141	1.8626	0.3725	0.4369

Table 4.11 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 2-D randomly distributed particles at -4°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	$E_{11}=E_{22}$ (GPa)	E_{33} (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.4325	3.5516	0.3307	0.3495
					1%	4.0082	3.7678	4.3765	0.3156	0.3659
					3%	4.1786	4.5775	6.4694	0.3114	0.3831
					6%	4.2123	5.7895	9.7001	0.3056	0.3866
					10%	4.3331	7.4697	14.2001	0.2928	0.3810
300	200	2	122.388	3.015	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3752	3.4115	0.3307	0.3457
					1%	4.0082	3.4897	3.6737	0.3156	0.3525
					3%	4.1786	3.7749	4.3419	0.3114	0.3653
					6%	4.2123	4.2068	5.3797	0.3056	0.3768
					10%	4.3331	4.8027	6.8353	0.2928	0.3841

Table 4.11 Continued

300	200	3	102.216	5.415	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3669	3.3915	0.3307	0.3451
					1%	4.0082	3.4487	3.5729	0.3156	0.3500
					3%	4.1786	3.6544	4.0359	0.3114	0.3600
					6%	4.2123	3.9684	4.7574	0.3056	0.3707
					10%	4.3331	4.4039	5.7734	0.2928	0.3794
300	200	4	94.434	7.815	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3631	3.3822	0.3307	0.3448
					1%	4.0082	3.4297	3.5262	0.3156	0.3487
					3%	4.1786	3.5981	3.8942	0.3114	0.3571
					6%	4.2123	3.8566	4.4688	0.3056	0.3668
					10%	4.3331	4.2166	5.2802	0.2928	0.3756
300	200	5	90.308	10.215	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3608	3.3765	0.3307	0.3447
					1%	4.0082	3.4182	3.4979	0.3156	0.3479
					3%	4.1786	3.5639	3.8083	0.3114	0.3552
					6%	4.2123	3.7884	4.2935	0.3056	0.3639
					10%	4.3331	4.1022	4.9802	0.2928	0.3725
300	200	6	87.753	12.6150	0%	3.34644	3.34644	3.34644	0.3438	0.3438
					0.2%	3.7128	3.3592	3.3727	0.3307	0.3445
					1%	4.0082	3.4104	3.4785	0.3156	0.3474
					3%	4.1786	3.5405	3.7494	0.3114	0.3538
					6%	4.2123	3.7417	4.1733	0.3056	0.3617
					10%	4.3331	4.0236	4.7742	0.2928	0.3699

Table 4.12 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 2-D randomly distributed particles at -65°F

$E_{nanoclay}$ y (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	$E_{11}=E_{22}$ (GPa)	E_{33} (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.9547	4.0781	0.3283	0.3480
					1%	4.6031	4.3048	4.9347	0.3193	0.3631
					3%	4.6944	5.1529	7.1079	0.3076	0.3803
					6%	4.9732	6.4218	10.4614	0.3039	0.3852
					10%	5.0920	8.1783	15.1285	0.2983	0.3810
300	200	2	122.388	3.015	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8957	3.9340	0.3283	0.3446
					1%	4.6031	4.0178	4.2117	0.3193	0.3508
					3%	4.6944	4.3225	4.9190	0.3076	0.3628
					6%	4.9732	4.7846	6.0168	0.3039	0.3739
					10%	5.0920	5.4224	7.5549	0.2983	0.3814
300	200	3	102.216	5.415	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8872	3.9134	0.3283	0.3441
					1%	4.6031	3.9757	4.1081	0.3193	0.3486
					3%	4.6944	4.1983	4.6049	0.3076	0.3580
					6%	4.9732	4.5387	5.3783	0.3039	0.3681
					10%	5.0920	5.0109	6.4662	0.2983	0.3766
300	200	4	94.434	7.815	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8832	3.9038	0.3283	0.3439
					1%	4.6031	3.9560	4.0599	0.3193	0.3475
					3%	4.6944	4.1400	4.4585	0.3076	0.3553
					6%	4.9732	4.4227	5.0803	0.3039	0.3645
					10%	5.0920	4.8166	5.9575	0.2983	0.3730

Table 4.12 Continued

300	200	5	90.308	10.215	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8808	3.8979	0.3283	0.3437
					1%	4.6031	3.9441	4.0304	0.3193	0.3468
					3%	4.6944	4.1043	4.3690	0.3076	0.3536
					6%	4.9732	4.3515	4.8979	0.3039	0.3619
					10%	5.0920	4.6971	5.6456	0.2983	0.3701
300	200	6	87.753	12.6150	0%	3.86509	3.86509	3.86509	0.3429	0.3429
					0.2%	4.2836	3.8792	3.8939	0.3283	0.3436
					1%	4.6031	3.9358	4.0101	0.3193	0.3463
					3%	4.6944	4.0797	4.3072	0.3076	0.3523
					6%	4.9732	4.3023	4.7718	0.3039	0.3598
					10%	5.0920	4.6144	5.4297	0.2983	0.3676

Table 4.13 Comparison of Mori-Tanaka calculations and experimental results of EPON 828/clay nanocomposites for 2-D randomly distributed particles at RT

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	$E_{11}=E_{22}$ (GPa)	E_{33} (GPa)	Poisson's Ratio (Experimental)	ν_{31}
300	200	6	87.7525	12.6150	0%	2.8177	2.8177	2.8177	0.3105	0.3105
					1%	2.9127	2.9142	3.0249	0.3238	0.3119
					3%	2.9840	3.1089	3.4486	0.3288	0.3399
					6%	3.0965	3.4082	4.1083	0.3340	0.4168
					10%	3.3427	3.8261	5.0367	0.3348	0.5236

4.1.1-e) 3-D Randomly Distributed Nanoclay Particles

Referring to [197], the effective bulk and shear modulus, Young's modulus and Poisson's ratio for the case of 3-D randomly distributed nanoclay particles can be expressed as:

$$p = \frac{p_2}{p_1} \quad (48)$$

$$q = \frac{q_2}{q_1} \quad (49)$$

$$p_1 = 1 + c \frac{[2(S_{1122} + S_{2222} + S_{2233} - 1)(a_3 + a_4) + (S_{1111} + 2S_{2211} - 1)(a_1 - 2a_2)]}{3a} \quad (50)$$

$$p_2 = \frac{[a_1 - 2(a_2 - a_3 - a_4)]}{3a} \quad (51)$$

$$q_1 = 1 - c \left\{ \frac{2}{5} \frac{2S_{1212} - 1}{2S_{1212} + \frac{\mu_0}{(\mu_1 - \mu_0)}} + \frac{1}{3} \frac{2S_{2323} - 1}{2S_{2323} + \frac{\mu_0}{(\mu_1 - \mu_0)}} - \frac{1}{15a} [(S_{1122} - S_{2233})(2a_3 - a_4 + a_5) \right. \\ \left. + 2(S_{1111} - S_{2211} - 1)(a_1 + a_2) + (S_{1122} - S_{2222} + 1)(2a_3 - a_4 - a_5a)] \right\} \quad (52)$$

$$q_2 = -\frac{2}{5} \frac{1}{2S_{1212} + \frac{\mu_m}{(\mu_p - \mu_m)}} - \frac{1}{3} \frac{1}{2S_{2323} + \frac{\mu_m}{(\mu_p - \mu_m)}} \\ + \frac{1}{15a} [2(a_1 + a_2 - a_3) + a_4 + a_5a] \quad (53)$$

$$\kappa = \frac{\kappa_m}{1 + cp} \quad (54)$$

$$\mu = \frac{\mu_m}{1 + cp} \quad (55)$$

$$E = \frac{3\mu}{1 + \frac{\mu}{3\kappa}} \quad (56)$$

$$\nu = \frac{1}{2} - \frac{E}{6K} \quad (57)$$

4.1.1-f) Results of Parametric Study for 3-D Randomly Distributed Nanoclay Particles

Again the number of flakes N was varied from 1 to 6 and the Young's modulus obtained experimentally for PP 3371 and epoxy were used in the Mori-Tanaka calculations. The comparison of Mori-Tanaka calculations and experimental results for 3-D randomly distributed particles is given in Tables 4.14 - 4.18 for PP 3371 and in Table 4.19 for EPON 828.

Table 4.14 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 3-D randomly distributed particles at 160°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E (GPa)	κ (GPa)	μ (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	0.39204	0.39204	0.8782	0.1375	0.4256	0.4256
					0.2%	0.4499	0.4284	0.8957	0.1508	0.4782	0.4203
					1%	0.4654	0.5734	0.9661	0.2046	0.4715	0.4011
					3%	0.4969	0.9353	1.1469	0.3428	0.4523	0.3641
					6%	0.5031	1.4874	1.4318	0.5605	0.5077	0.3269
					10%	0.5146	2.2590	1.8395	0.8720	0.4896	0.2953
300	200	2	122.388	3.015	0%	0.39204	0.39204	0.8782	0.1375	0.4256	0.4256
					0.2%	0.4499	0.4007	0.8828	0.1407	0.4782	0.4244
					1%	0.4654	0.4356	0.9013	0.1534	0.4715	0.4195
					3%	0.4969	0.5244	0.9488	0.1862	0.4523	0.4079
					6%	0.5031	0.6627	1.0238	0.2380	0.5077	0.3921
					10%	0.5146	0.8577	1.1312	0.3122	0.4896	0.3736
300	200	3	102.216	5.415	0%	0.39204	0.39204	0.8782	0.1375	0.4256	0.4256
					0.2%	0.4499	0.3972	0.8812	0.1394	0.4782	0.4249
					1%	0.4654	0.4183	0.8934	0.1471	0.4715	0.4220
					3%	0.4969	0.4720	0.9245	0.1668	0.4523	0.4149
					6%	0.5031	0.5560	0.9737	0.1979	0.5077	0.4048
					10%	0.5146	0.6753	1.0442	0.2425	0.4896	0.3922
300	200	4	94.434	7.815	0%	0.39204	0.39204	0.8782	0.1375	0.4256	0.4256
					0.2%	0.4499	0.3959	0.8806	0.1389	0.4782	0.4251
					1%	0.4654	0.4113	0.8902	0.1445	0.4715	0.4230
					3%	0.4969	0.4507	0.9148	0.1589	0.4523	0.4179
					6%	0.5031	0.5126	0.9536	0.1817	0.5077	0.4104
					10%	0.5146	0.6007	1.0094	0.2144	0.4896	0.4008

Table 4.14 Continued

300	200	5	90.308	10.215	0%	0.39204	0.39204	0.8782	0.1375	0.4256	0.4256
					0.2%	0.4499	0.3951	0.8803	0.1386	0.4782	0.4252
					1%	0.4654	0.4075	0.8885	0.1431	0.4715	0.4236
					3%	0.4969	0.4391	0.9095	0.1547	0.4523	0.4195
					6%	0.5031	0.4889	0.9428	0.1729	0.5077	0.4136
					10%	0.5146	0.5598	0.9905	0.1991	0.4896	0.4058
300	200	6	87.753	12.6150	0%	0.39204	0.39204	0.8782	0.1375	0.4256	0.4256
					0.2%	0.4499	0.3946	0.8800	0.1384	0.4782	0.4253
					1%	0.4654	0.4051	0.8874	0.1422	0.4715	0.4239
					3%	0.4969	0.4318	0.9062	0.1520	0.4523	0.4206
					6%	0.5031	0.4739	0.9359	0.1674	0.5077	0.4156
					10%	0.5146	0.5340	0.9786	0.1895	0.4896	0.4091

Table 4.15 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 3-D randomly distributed particles at 120°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E (GPa)	κ (GPa)	μ (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	0.61595	0.61595	1.0671	0.2194	0.4038	0.4038
					0.2%	0.7285	0.6656	1.0912	0.2380	0.4350	0.3983
					1%	0.7646	0.8633	1.1882	0.3130	0.4631	0.3789
					3%	0.8283	1.3579	1.4373	0.5057	0.4355	0.3425
					6%	0.8332	2.1149	1.8292	0.8089	0.4334	0.3073
					10%	0.8664	3.1755	2.3892	1.2419	0.4111	0.2785
300	200	2	122.388	3.015	0%	0.61595	0.61595	1.0671	0.2194	0.4038	0.4038
					0.2%	0.7285	0.6284	1.0736	0.2240	0.4350	0.4024
					1%	0.7646	0.6784	1.0996	0.2428	0.4631	0.3972
					3%	0.8283	0.8058	1.1664	0.2909	0.4355	0.3849
					6%	0.8332	1.0039	1.2718	0.3668	0.4334	0.3684
					10%	0.8664	1.2833	1.4225	0.4754	0.4111	0.3496

Table 4.15 Continued

300	200	3	102.216	5.415	0%	0.61595	0.61595	1.0671	0.2194	0.4038	0.4038
					0.2%	0.7285	0.6236	1.0714	0.2222	0.4350	0.4030
					1%	0.7646	0.6543	1.0884	0.2337	0.4631	0.3998
					3%	0.8283	0.7329	1.1322	0.2632	0.4355	0.3921
					6%	0.8332	0.8559	1.2012	0.3098	0.4334	0.3812
					10%	0.8664	1.0302	1.3000	0.3765	0.4111	0.3679
300	200	4	94.434	7.815	0%	0.61595	0.61595	1.0671	0.2194	0.4038	0.4038
					0.2%	0.7285	0.6216	1.0704	0.2215	0.4350	0.4032
					1%	0.7646	0.6444	1.0838	0.2300	0.4631	0.4009
					3%	0.8283	0.7028	1.1182	0.2519	0.4355	0.3952
					6%	0.8332	0.7943	1.1723	0.2863	0.4334	0.3871
					10%	0.8664	0.9245	1.2500	0.3358	0.4111	0.3767
300	200	5	90.308	10.215	0%	0.61595	0.61595	1.0671	0.2194	0.4038	0.4038
					0.2%	0.7285	0.6205	1.0699	0.2211	0.4350	0.4033
					1%	0.7646	0.6389	1.0813	0.2279	0.4631	0.4015
					3%	0.8283	0.6861	1.1105	0.2456	0.4355	0.3970
					6%	0.8332	0.7603	1.1565	0.2734	0.4334	0.3904
					10%	0.8664	0.8659	1.2225	0.3133	0.4111	0.3819
300	200	6	87.753	12.6150	0%	0.61595	0.61595	1.0671	0.2194	0.4038	0.4038
					0.2%	0.7285	0.6198	1.0696	0.2208	0.4350	0.4034
					1%	0.7646	0.6355	1.0797	0.2266	0.4631	0.4019
					3%	0.8283	0.6756	1.1056	0.2416	0.4355	0.3982
					6%	0.8332	0.7386	1.1465	0.2652	0.4334	0.3926
					10%	0.8664	0.8285	1.2051	0.2990	0.4111	0.3854

Table 4.16 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 3-D randomly distributed particles at room temperature

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E (GPa)	κ (GPa)	μ (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	1.2001	1.2001	1.8994	0.4302	0.3947	0.3947
					0.2%	1.3172	1.2741	1.9358	0.4582	0.4167	0.3903
					1%	1.5085	1.5699	2.0828	0.5711	0.4056	0.3744
					3%	1.6278	2.3124	2.4597	0.8607	0.3852	0.3433
					6%	1.6463	3.4504	3.0515	1.3154	0.3722	0.3115
					10%	1.7426	5.0419	3.8945	1.9630	0.3725	0.2842
300	200	2	122.388	3.015	0%	1.2001	1.2001	1.8994	0.4302	0.3947	0.3947
					0.2%	1.3172	1.2205	1.9100	0.4379	0.4167	0.3935
					1%	1.5085	1.3027	1.9529	0.4690	0.4056	0.3888
					3%	1.6278	1.5122	2.0629	0.5488	0.3852	0.3778
					6%	1.6463	1.8380	2.2360	0.6743	0.3722	0.3630
					10%	1.7426	2.2973	2.4831	0.8535	0.3725	0.3458
300	200	3	102.216	5.415	0%	1.2001	1.2001	1.8994	0.4302	0.3947	0.3947
					0.2%	1.3172	1.2132	1.9066	0.4352	0.4167	0.3940
					1%	1.5085	1.2660	1.9356	0.4551	0.4056	0.3910
					3%	1.6278	1.4009	2.0102	0.5062	0.3852	0.3838
					6%	1.6463	1.6119	2.1275	0.5867	0.3722	0.3737
					10%	1.7426	1.9107	2.2952	0.7018	0.3725	0.3612
300	200	4	94.434	7.815	0%	1.2001	1.2001	1.8994	0.4302	0.3947	0.3947
					0.2%	1.3172	1.2100	1.9052	0.4340	0.4167	0.3941
					1%	1.5085	1.2502	1.9283	0.4491	0.4056	0.3919
					3%	1.6278	1.3529	1.9878	0.4879	0.3852	0.3866
					6%	1.6463	1.5139	2.0815	0.5490	0.3722	0.3788
					10%	1.7426	1.7427	2.2155	0.6365	0.3725	0.3689

Table 4.16 Continued

300	200	5	90.308	10.215	0%	1.2001	1.2001	1.8994	0.4302	0.3947	0.3947
					0.2%	1.3172	1.2082	1.9043	0.4333	0.4167	0.3943
					1%	1.5085	1.2412	1.9242	0.4457	0.4056	0.3925
					3%	1.6278	1.3256	1.9752	0.4775	0.3852	0.3881
					6%	1.6463	1.4581	2.0556	0.5276	0.3722	0.3818
					10%	1.7426	1.6467	2.1706	0.5994	0.3725	0.3736
300	200	6	87.753	12.615 0	0%	1.2001	1.2001	1.8994	0.4302	0.3947	0.3947
					0.2%	1.3172	1.2071	1.9038	0.4328	0.4167	0.3943
					1%	1.5085	1.2354	1.9216	0.4435	0.4056	0.3929
					3%	1.6278	1.3079	1.9671	0.4707	0.3852	0.3892
					6%	1.6463	1.4218	2.0389	0.5137	0.3722	0.3838
					10%	1.7426	1.5842	2.1417	0.5754	0.3725	0.3767

Table 4.17 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 3-D randomly distributed particles at -4°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E (GPa)	κ (GPa)	μ (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	3.34644	3.34644	3.5707	1.2451	0.3438	0.3438
					0.2%	3.7128	3.4543	3.6263	1.2877	0.3307	0.3412
					1%	4.0082	3.8873	3.8507	1.4595	0.3156	0.3317
					3%	4.1786	4.9843	4.4251	1.8991	0.3114	0.3123
					6%	4.2123	6.6798	5.3244	2.5872	0.3056	0.2909
					10%	4.3331	9.0582	6.5993	3.5628	0.2928	0.2712
300	200	2	122.388	3.015	0%	3.34644	3.34644	3.5707	1.2451	0.3438	0.3438
					0.2%	3.7128	3.3818	3.5893	1.2591	0.3307	0.3430
					1%	4.0082	3.5240	3.6644	1.3152	0.3156	0.3397
					3%	4.1786	3.8871	3.8569	1.4591	0.3114	0.3320
					6%	4.2123	4.4527	4.1585	1.6847	0.3056	0.3215
					10%	4.3331	5.2507	4.5871	2.0053	0.2928	0.3092

Table 4.17 Continued

300	200	3	102.216	5.415	0%	3.34644	3.34644	3.5707	1.2451	0.3438	0.3438
					0.2%	3.7128	3.3714	3.5841	1.2549	0.3307	0.3432
					1%	4.0082	3.4718	3.6384	1.2945	0.3156	0.3410
					3%	4.1786	3.7286	3.7776	1.3960	0.3114	0.3355
					6%	4.2123	4.1299	3.9960	1.5552	0.3056	0.3277
					10%	4.3331	4.6978	4.3066	1.7819	0.2928	0.3182
300	200	4	94.434	7.815	0%	3.34644	3.34644	3.5707	1.2451	0.3438	0.3438
					0.2%	3.7128	3.3666	3.5818	1.2530	0.3307	0.3433
					1%	4.0082	3.4477	3.6267	1.2850	0.3156	0.3416
					3%	4.1786	3.6553	3.7418	1.3668	0.3114	0.3372
					6%	4.2123	3.9803	3.9226	1.4954	0.3056	0.3309
					10%	4.3331	4.4413	4.1800	1.6786	0.2928	0.3229
300	200	5	90.308	10.215	0%	3.34644	3.34644	3.5707	1.2451	0.3438	0.3438
					0.2%	3.7128	3.3637	3.5804	1.2519	0.3307	0.3434
					1%	4.0082	3.4331	3.6197	1.2792	0.3156	0.3419
					3%	4.1786	3.6110	3.7205	1.3492	0.3114	0.3382
					6%	4.2123	3.8898	3.8789	1.4592	0.3056	0.3329
					10%	4.3331	4.2858	4.1046	1.6161	0.2928	0.3260
300	200	6	87.753	12.6150	0%	3.34644	3.34644	3.5707	1.2451	0.3438	0.3438
					0.2%	3.7128	3.3617	3.5795	1.2511	0.3307	0.3435
					1%	4.0082	3.4232	3.6150	1.2752	0.3156	0.3422
					3%	4.1786	3.5808	3.7062	1.3371	0.3114	0.3390
					6%	4.2123	3.8280	3.8495	1.4345	0.3056	0.3343
					10%	4.3331	4.1795	4.0538	1.5734	0.2928	0.3282

Table 4.18 Comparison of Mori-Tanaka calculations and experimental results of PP 3371/clay nanocomposites for 3-D randomly distributed particles at -65°F

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	t (nm)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E (GPa)	κ (GPa)	μ (GPa)	Poisson's Ratio (Experimental)	ν
300	200	1	300	0.615	0%	3.86509	3.86509	4.1005	1.4391	0.3429	0.3429
					0.2%	4.2836	3.9772	4.1585	1.4834	0.3283	0.3406
					1%	4.6031	4.4276	4.3926	1.6620	0.3193	0.3320
					3%	4.6944	5.5692	4.9915	2.1191	0.3076	0.3140
					6%	4.9732	7.3344	5.9288	2.8344	0.3039	0.2938
					10%	5.0920	9.8096	7.2568	3.8478	0.2983	0.2747
300	200	2	122.388	3.015	0%	3.86509	3.86509	4.1005	1.4391	0.3429	0.3429
					0.2%	4.2836	3.9026	4.1203	1.4539	0.3283	0.3421
					1%	4.6031	4.0538	4.2003	1.5136	0.3193	0.3391
					3%	4.6944	4.4397	4.4053	1.6665	0.3076	0.3320
					6%	4.9732	5.0409	4.7264	1.9062	0.3039	0.3222
					10%	5.0920	5.8887	5.1822	2.2466	0.2983	0.3106
300	200	3	102.216	5.415	0%	3.86509	3.86509	4.1005	1.4391	0.3429	0.3429
					0.2%	4.2836	3.8920	4.1150	1.4497	0.3283	0.3424
					1%	4.6031	4.0002	4.1736	1.4923	0.3193	0.3403
					3%	4.6944	4.2770	4.3238	1.6017	0.3076	0.3351
					6%	4.9732	4.7094	4.5593	1.7733	0.3039	0.3278
					10%	5.0920	5.3212	4.8940	2.0174	0.2983	0.3188
300	200	4	94.434	7.815	0%	3.86509	3.86509	4.1005	1.4391	0.3429	0.3429
					0.2%	4.2836	3.8870	4.1126	1.4477	0.3283	0.3425
					1%	4.6031	3.9753	4.1615	1.4825	0.3193	0.3408
					3%	4.6944	4.2013	4.2868	1.5716	0.3076	0.3367
					6%	4.9732	4.5549	4.4834	1.7115	0.3039	0.3307
					10%	5.0920	5.0563	4.7632	1.9108	0.2983	0.3231

Table 4.18 Continued

300	200	5	90.308	10.215	0%	3.86509	3.86509	4.1005	1.4391	0.3429	0.3429
					0.2%	4.2836	3.8840	4.1111	1.4465	0.3283	0.3425
					1%	4.6031	3.9602	4.1542	1.4764	0.3193	0.3411
					3%	4.6944	4.1552	4.2647	1.5532	0.3076	0.3376
					6%	4.9732	4.4607	4.4380	1.6738	0.3039	0.3325
					10%	5.0920	4.8945	4.6849	1.8458	0.2983	0.3259
300	200	6	87.753	12.6150	0%	3.86509	3.86509	4.1005	1.4391	0.3429	0.3429
					0.2%	4.2836	3.8819	4.1102	1.4457	0.3283	0.3426
					1%	4.6031	3.9497	4.1493	1.4723	0.3193	0.3413
					3%	4.6944	4.1235	4.2496	1.5406	0.3076	0.3383
					6%	4.9732	4.3959	4.4072	1.6479	0.3039	0.3338
					10%	5.0920	4.7831	4.6318	1.8010	0.2983	0.3279

Table 4.19 Comparison of Mori-Tanaka calculations and experimental results of EPON 828/clay nanocomposites for 3-D randomly distributed particles at room temperature

E_{nanoclay} (GPa)	D (nm)	N	E_p (GPa)	E_{nanoclay} (GPa)	Nanoclay Rein.	Young's Modulus (Experimental) (GPa)	E (GPa)	κ (GPa)	μ (GPa)	Poisson's Ratio (Experimental)	ν
300	200	6	87.7525	12.6150	0%	2.8177	2.8177	2.4782	1.0750	0.3105	0.3105
					1%	2.9127	2.9363	2.5451	1.1227	0.3238	0.3077
					3%	2.9840	3.1794	2.6824	1.2205	0.3288	0.3025
					6%	3.0965	3.5594	2.8976	1.3740	0.3340	0.2953
					10%	3.3427	4.0978	3.2034	1.5922	0.3348	0.2868

4.1.2. Results Obtained From Various Mori-Tanaka Calculations

Before starting the Mori-Tanaka calculations, we first analyzed the electron microscopy images showing the distribution of nanoclay particles. The SEM images show that as the nanoclay reinforcement percentage increases more agglomeration of nanoclay particles occurs and bigger particles are evident. Based on this observation, we assumed a different N composition for each percentage. As a consequence, at high percentages larger N values and at low percentage smaller N values were used in the calculations.

Table 4.20 shows the composition of nanoclay particles depending on the reinforcement percentage.

Table 4.20 Nanoclay flakes number composition for each percentage

Nanoclay Reinforcement	Composition of Flakes Number
0.2%	100% N=1
1%	40% N=1 30% N=2 30% N=3
3%	40% N=2 30% N=3 30% N=4
6%	100% N=5
10%	100% N=6

Using this composition we recalculated the Mori-Tanaka results for the three formulations outlined above, namely “oriented particles”, “2-D randomly distributed particles” and “3-D randomly distributed particles”.

4.1.2-a) Comparison of Experimental Results with Mori-Tanaka Calculations

The Mori-Tanaka results for the three types of assumed distributions are compared with the experimental results. In Figures 4.2-4.11 the experimental and numerical results obtained for Young's modulus and Poisson's ratio are compared for the various nanoclay reinforcement percentages. As explained previously, the numerical results are obtained from Mori-Tanaka calculations for oriented, 2-D randomly distributed and 3-D randomly distributed particles.

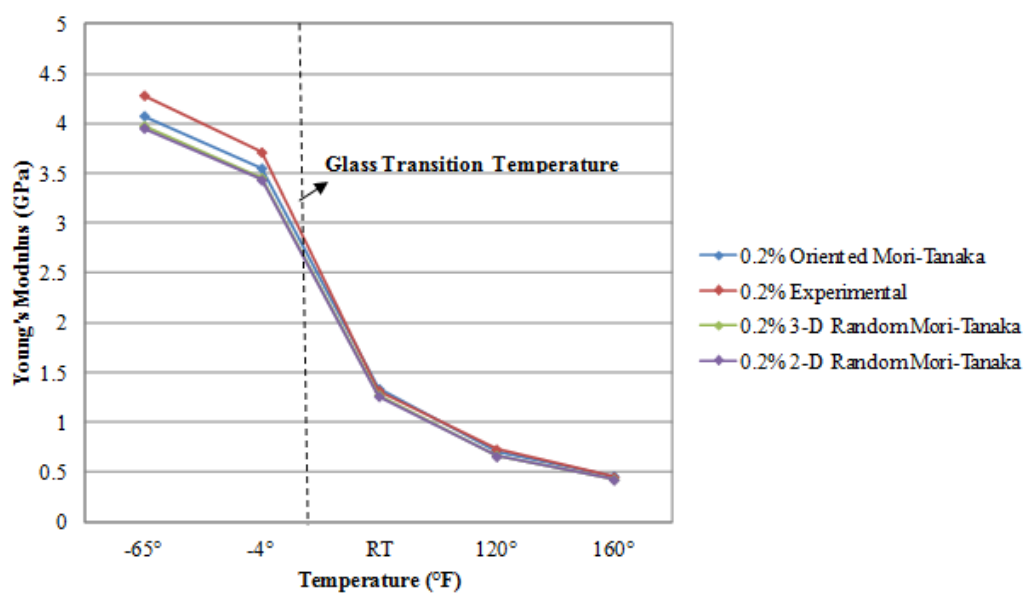


Figure 4.2 Comparison of Young's modulus obtained from Mori-Tanaka and experimental results for 0.2% nanoclay reinforced PP 3371 specimens at various temperatures

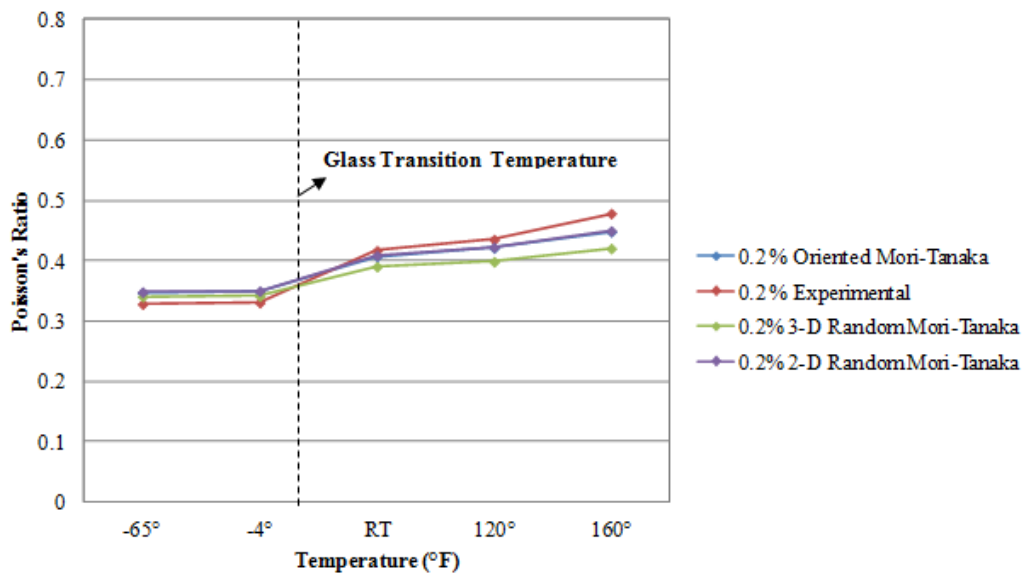


Figure 4.3 Comparison of Poisson's ratio obtained from Mori-Tanaka and experimental results for 0.2% nanoclay reinforced PP 3371 specimens at various temperatures

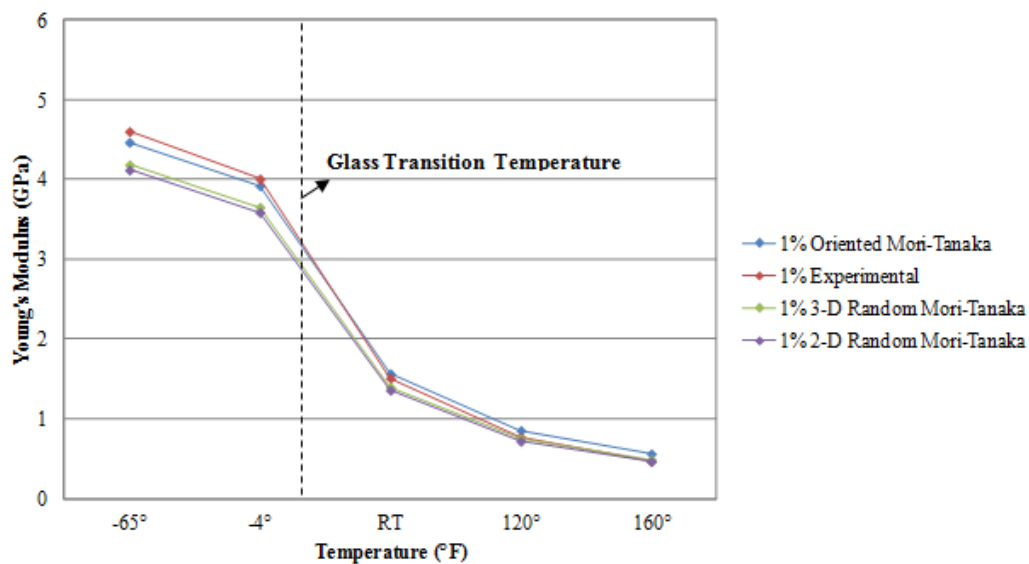


Figure 4.4 Comparison of Young's modulus obtained from Mori-Tanaka and experimental results for 1% nanoclay reinforced PP 3371 specimens at various temperatures

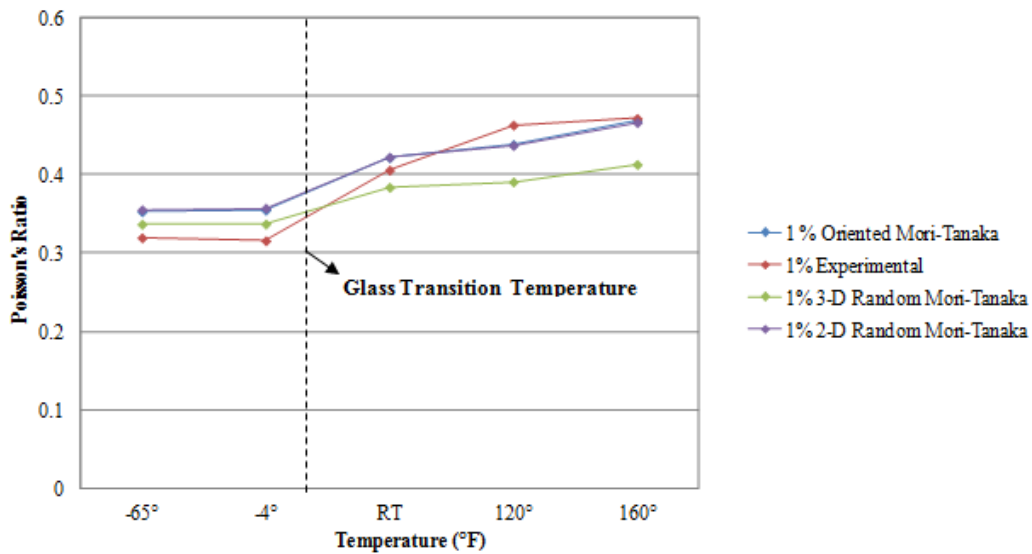


Figure 4.5 Comparison of Poisson's ratio obtained from Mori-Tanaka and experimental results for 1% nanoclay reinforced PP 3371 specimens at various temperatures

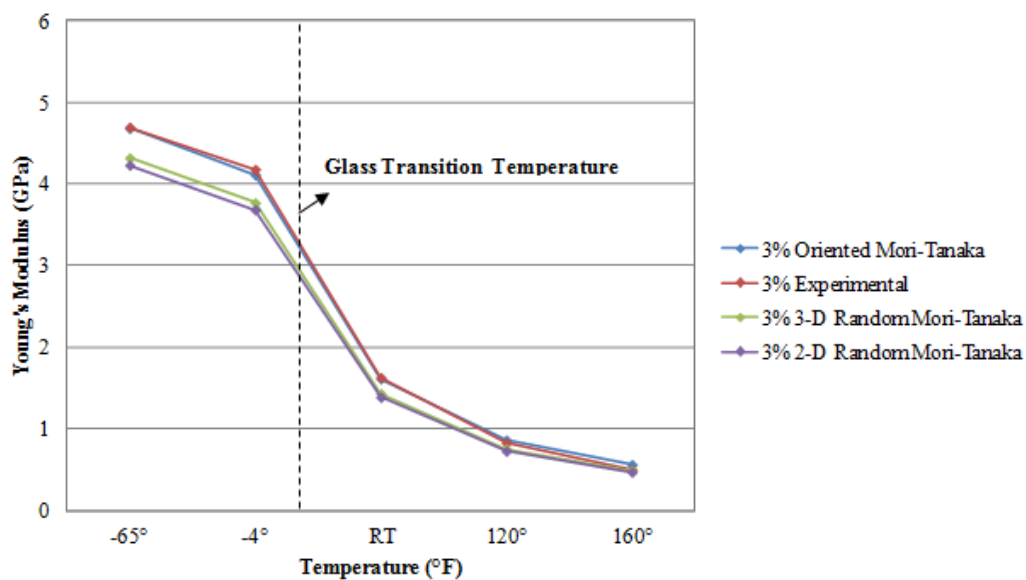


Figure 4.6 Comparison of Young's modulus obtained from Mori-Tanaka and experimental results for 3% nanoclay reinforced PP 3371 specimens at various temperatures

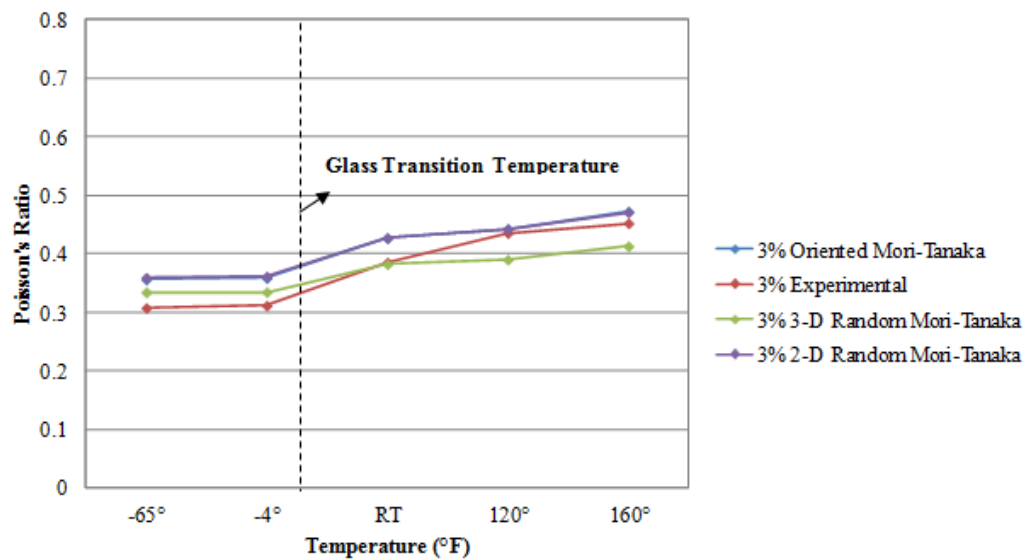


Figure 4.7 Comparison of Poisson's ratio obtained from Mori-Tanaka and experimental results for 3% nanoclay reinforced PP 3371 specimens at various temperatures

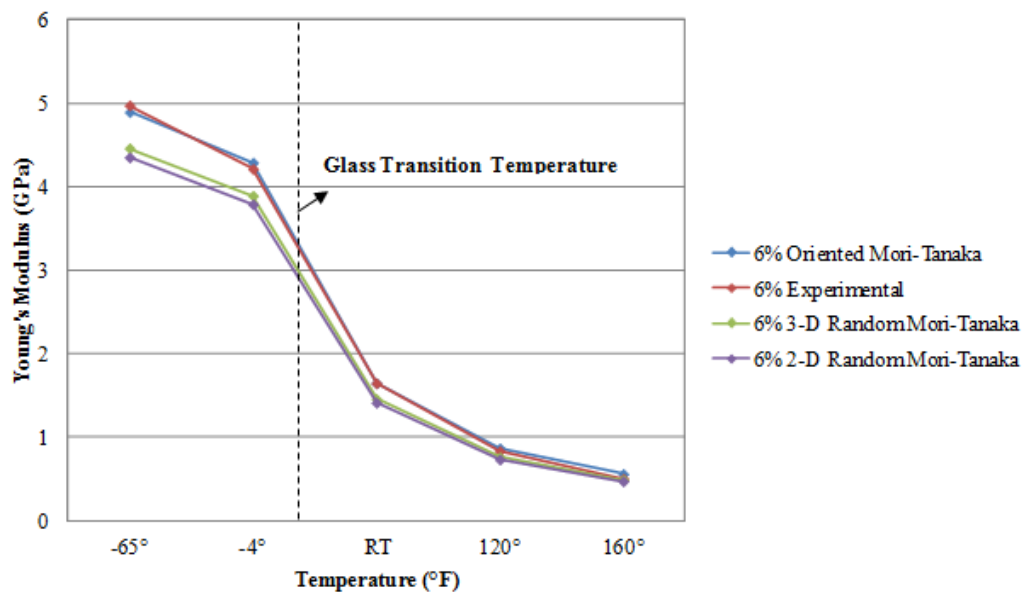


Figure 4.8 Comparison of Young's modulus obtained from Mori-Tanaka and experimental results for 6% nanoclay reinforced PP 3371 specimens at various temperatures

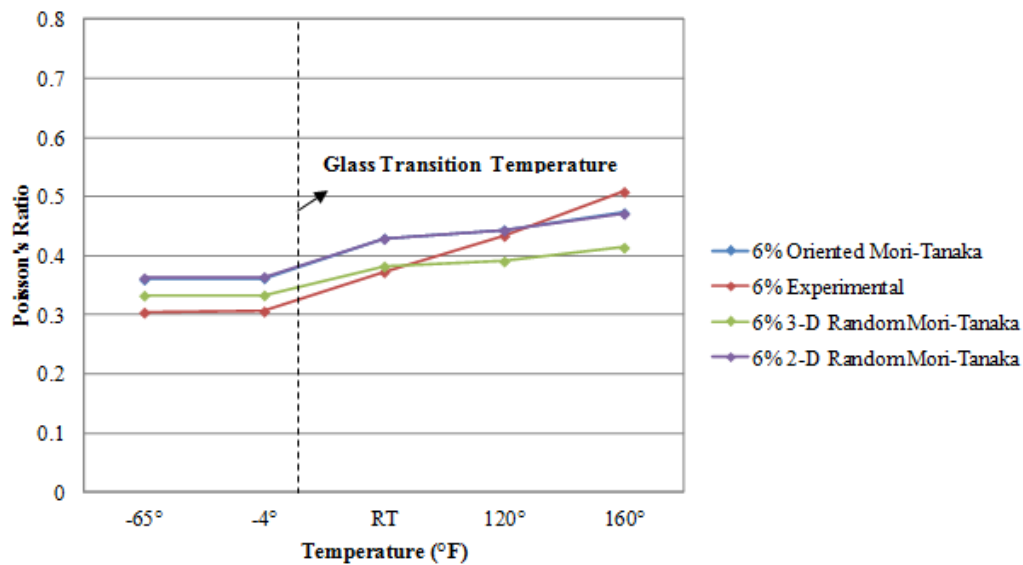


Figure 4.9 Comparison of Poisson's ratio obtained from Mori-Tanaka and experimental results for 6% nanoclay reinforced PP 3371 specimens at various temperatures

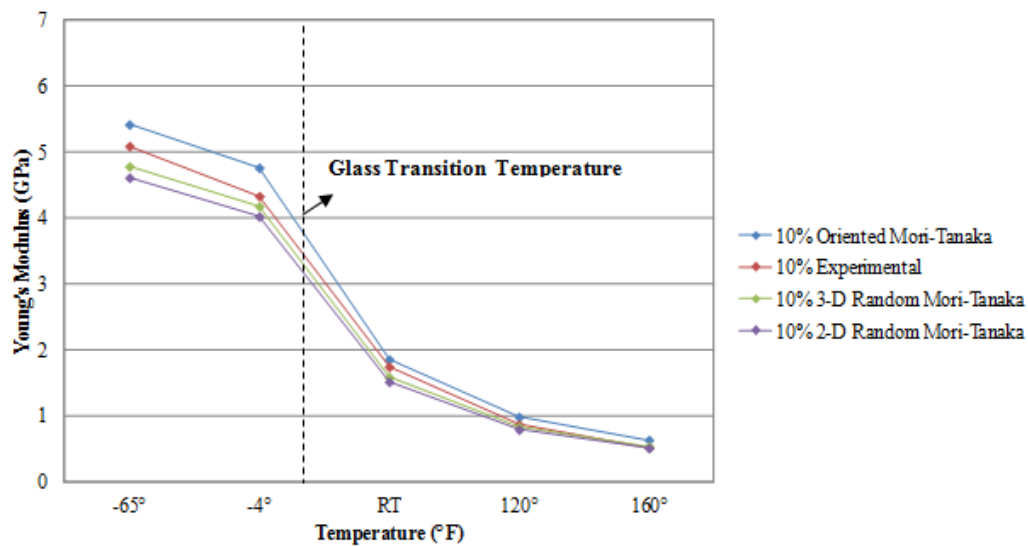


Figure 4.10 Comparison of Young's modulus obtained from Mori-Tanaka and experimental results for 10% nanoclay reinforced PP 3371 specimens at various temperatures

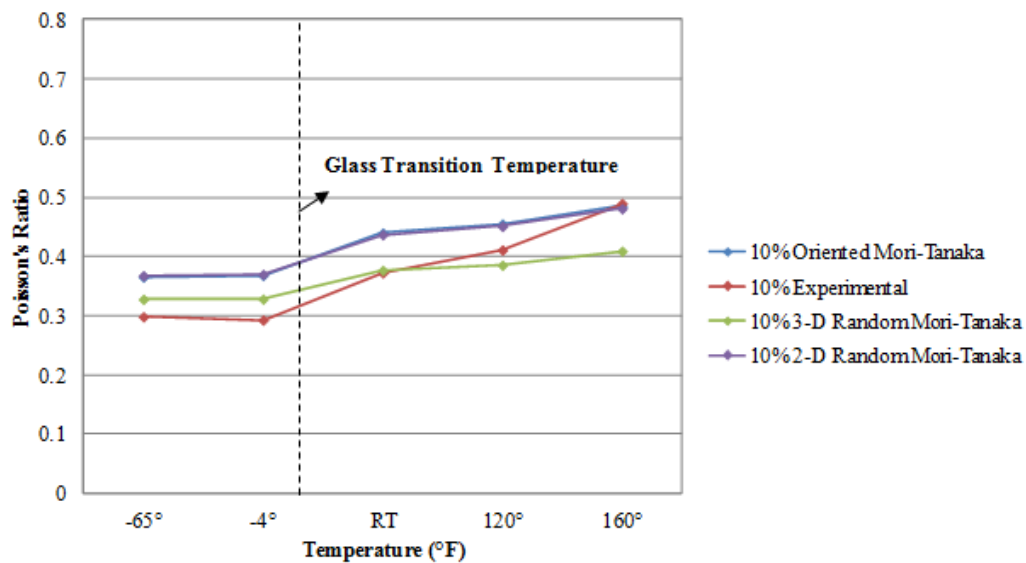


Figure 4.11 Comparison of Poisson's ratio obtained from Mori-Tanaka and experimental results for 10% nanoclay reinforced PP 3371 specimens at various temperatures

Figures 4.12-4.21 show the comparison of experimental and analytical results for each temperature as a function of the reinforcement percentage.

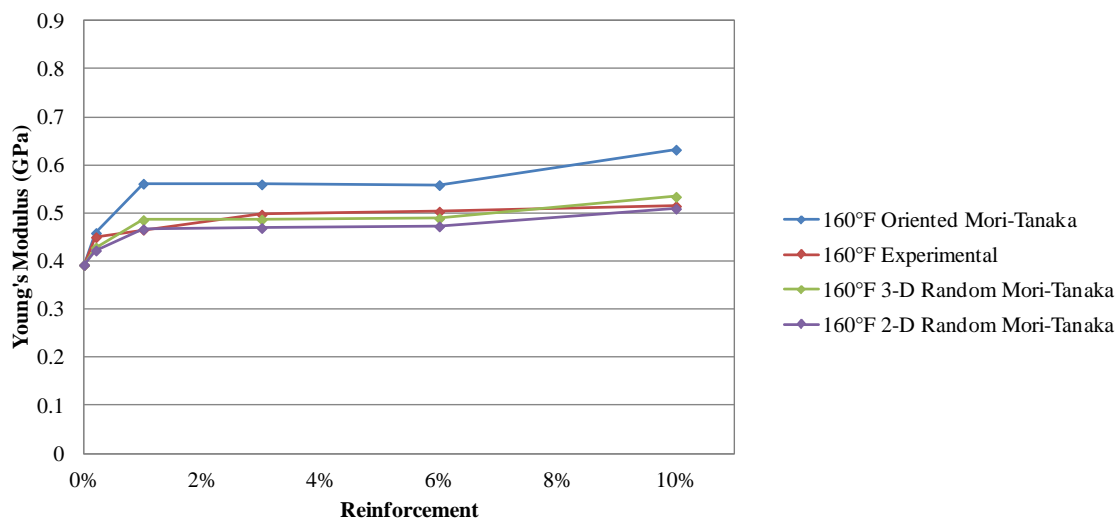


Figure 4.12 Comparison of Young's modulus obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at 160°F

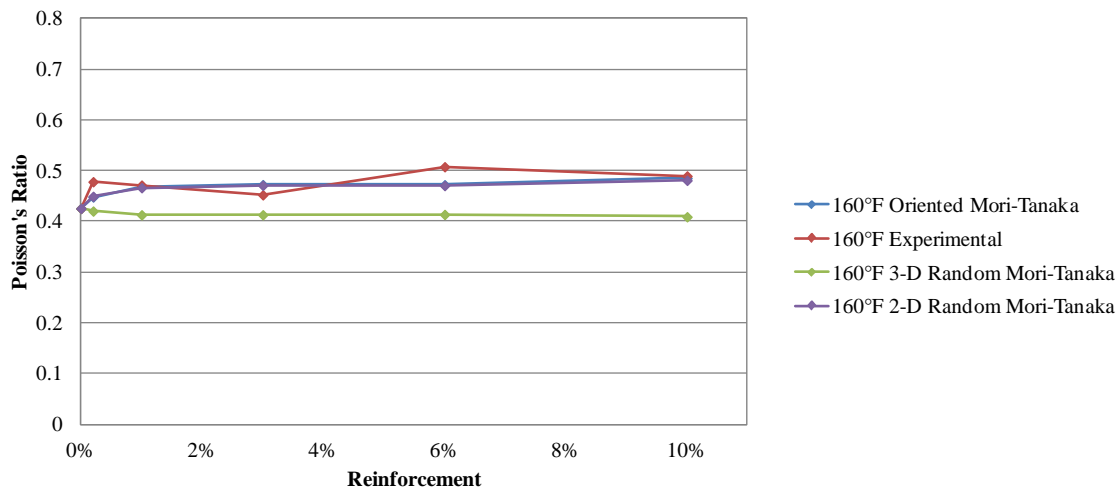


Figure 4.13 Comparison of Poisson's ratio obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at 160°F

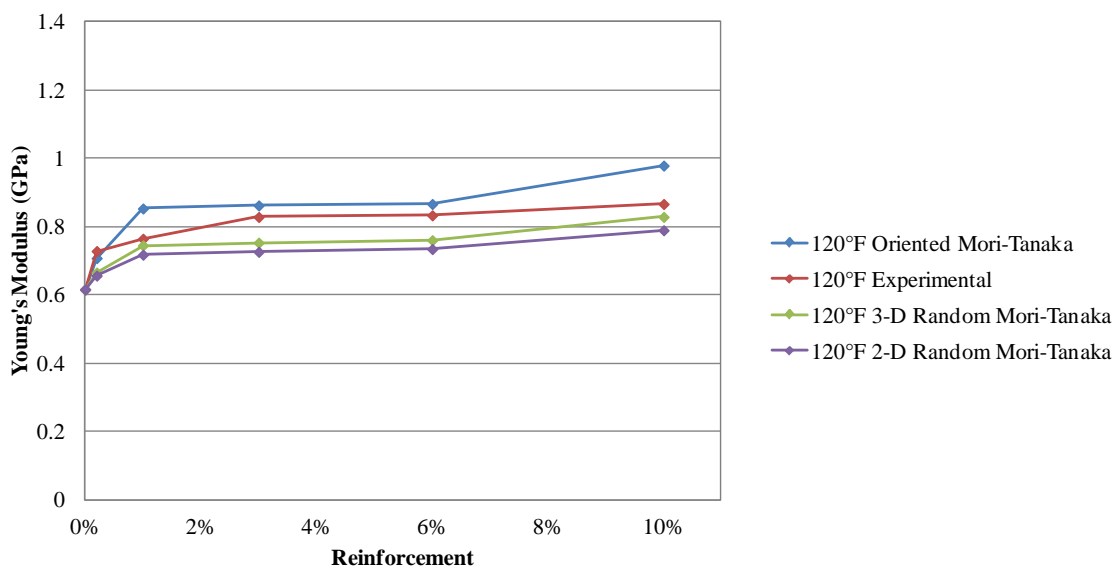


Figure 4.14 Comparison of Young's modulus obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at 120°F

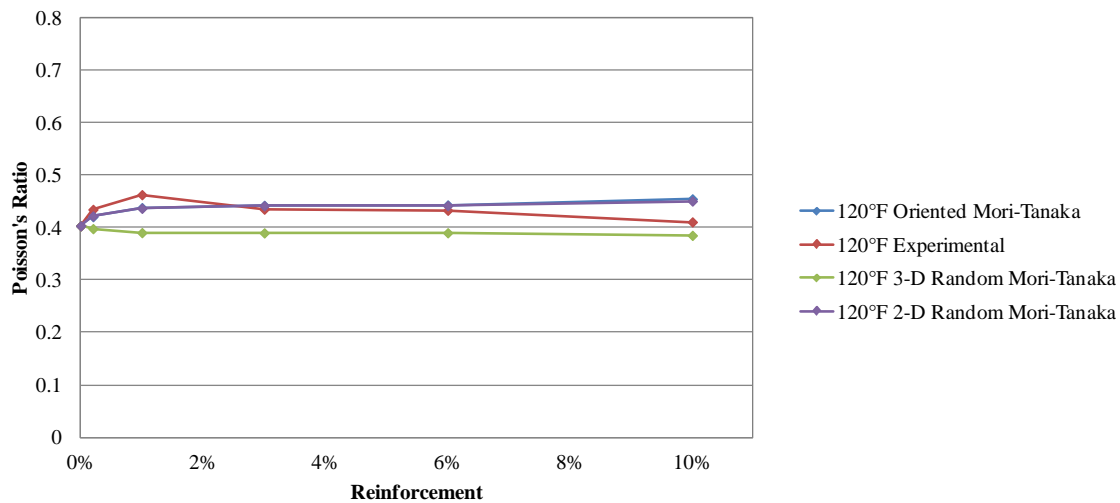


Figure 4.15 Comparison of Poisson's ratio obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at 120°F

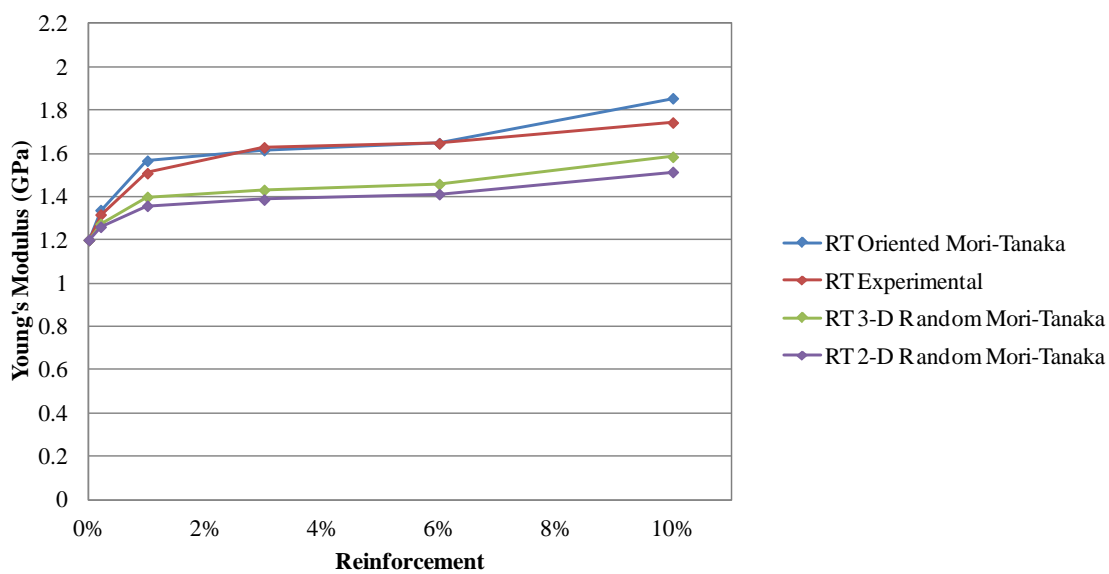


Figure 4.16 Comparison of Young's modulus obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at RT

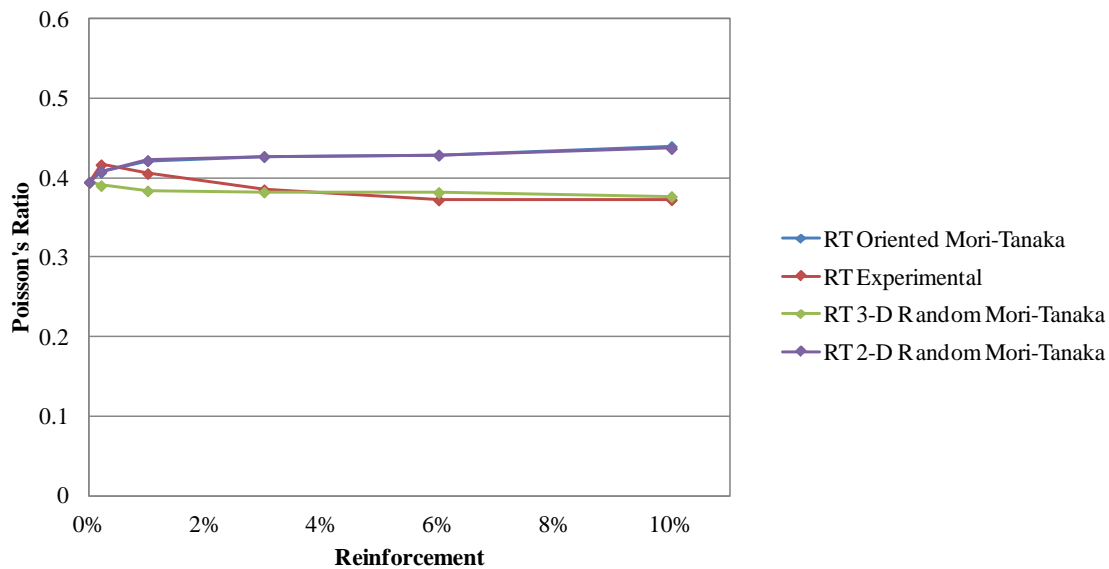


Figure 4.17 Comparison of Poisson's ratio obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at RT

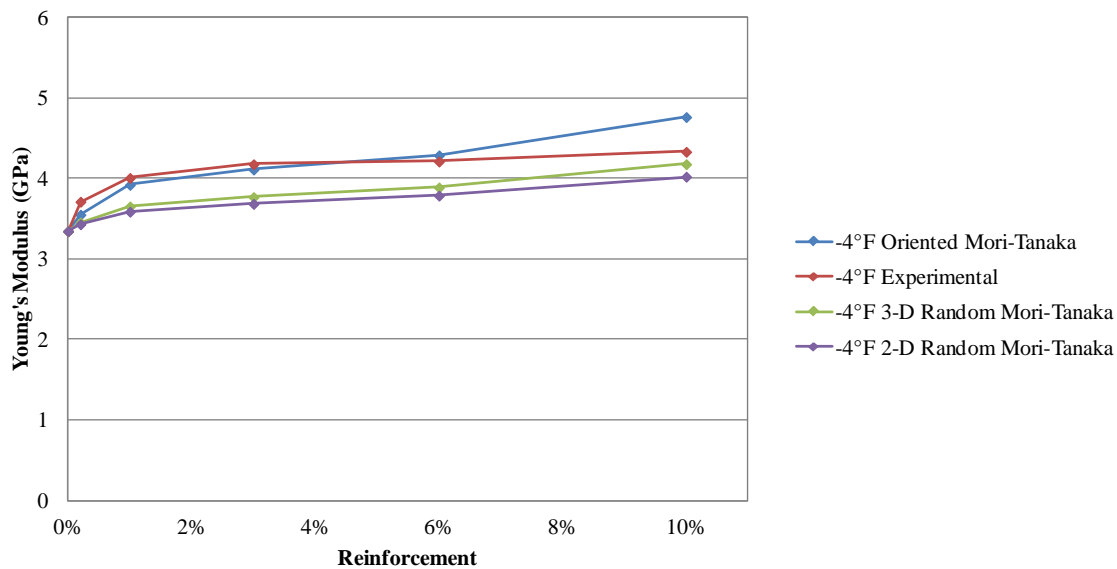


Figure 4.18 Comparison of Young's modulus obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at -4°F

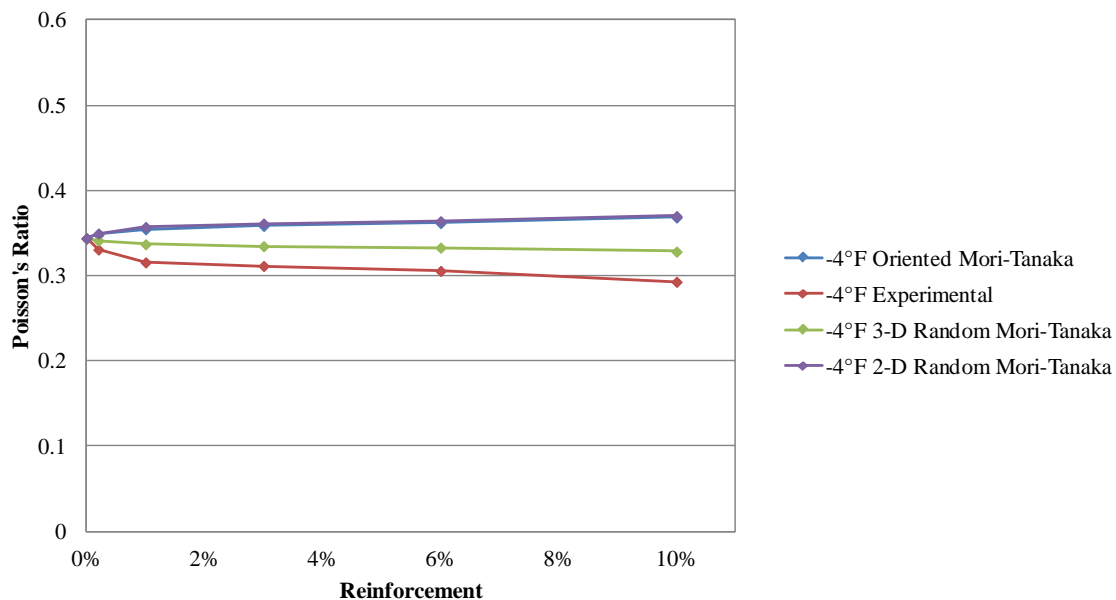


Figure 4.19 Comparison of Poisson's ratio obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at -4°F

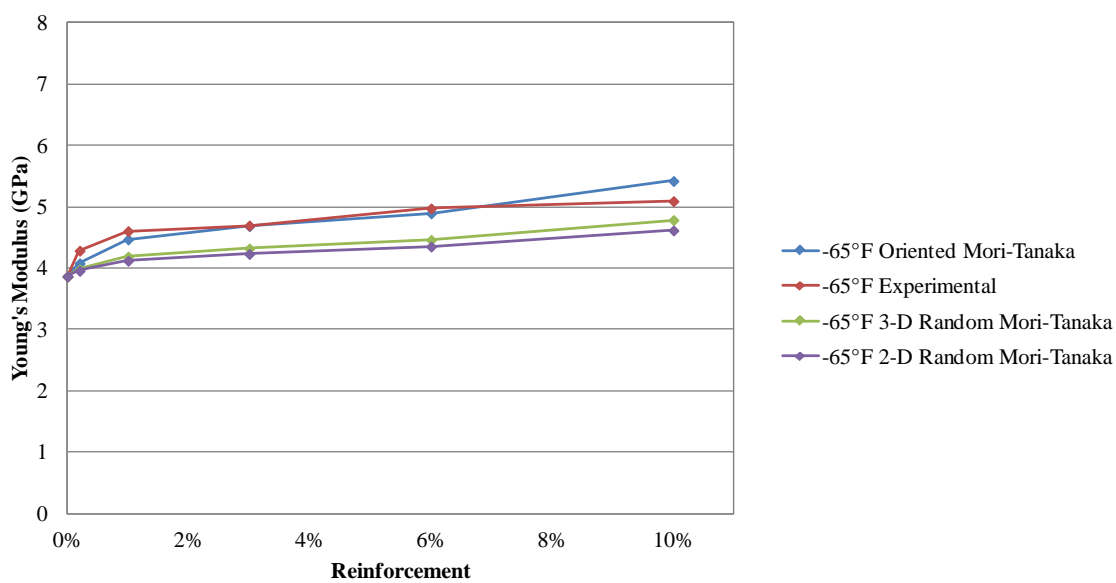


Figure 4.20 Comparison of Young's modulus obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at -65°F

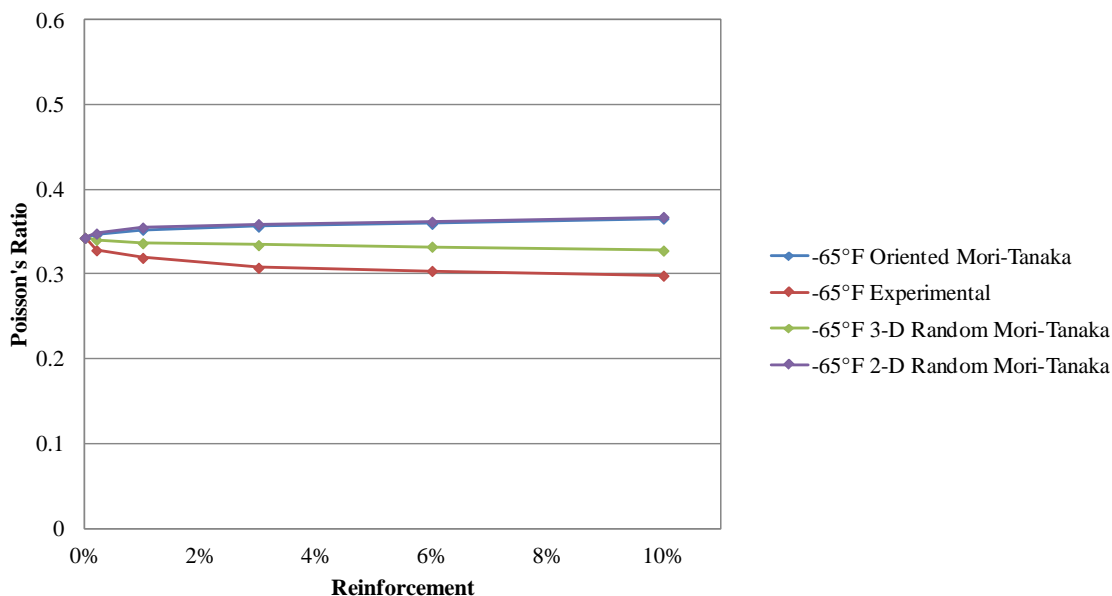


Figure 4.21 Comparison of Poisson's ratio obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at -65°F

As we mentioned before, the results indicate that the agreement between the experimental data and the modified Mori-Tanaka calculations in general is very good, except at higher temperatures and higher reinforcement percentages.

At high temperatures and high percentage values 3 different approaches of Mori Tanaka calculations do not provide a good match with experimental results. We believe that at high temperatures PP material behaves differently than at room or low temperatures and as a consequence the Mori Tanaka approach does not provide good results. This is also confirmed by the fact that for temperatures above the glass transition temperature of PP, there is a steep drop in the values of Young's modulus (Figures 4.2, 4.4, 4.6, 4.8 and 4.10). Also for high reinforcement percentages as was observed in the SEM pictures, the nanoclay particles do not disperse very well and there may be voids at the nanoclay/matrix interface. This may explain the divergence of results between experimental and calculated values at the higher percentages. To remedy the

situation, the Mori-Tanaka formulation was modified to take into account the effect of voids and temperature (see sections 4.2.3 and 4.2.4).

4.1.2-b) Comparison of Experimental Results with Mori-Tanaka Calculations for Epoxy Based Nanocomposites

Figures 4.22-4.23 show the comparison of Young's modulus and Poisson's ratio obtained from experiments and Mori-Tanaka calculations for EPON 828 epoxy nanoclay reinforced specimens at room temperature. For epoxy there is divergence between experimental and calculated results. This may be due to the fact that more agglomeration of nanoclay particles may occur because of the higher viscosity of epoxy relative to PP.

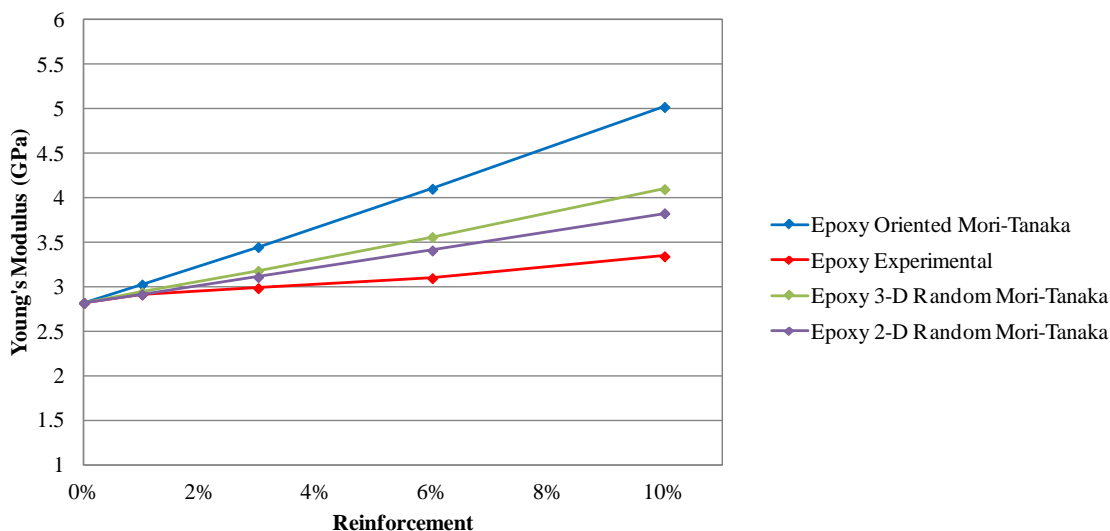


Figure 4.22 Comparison of Young's modulus obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at room temperature

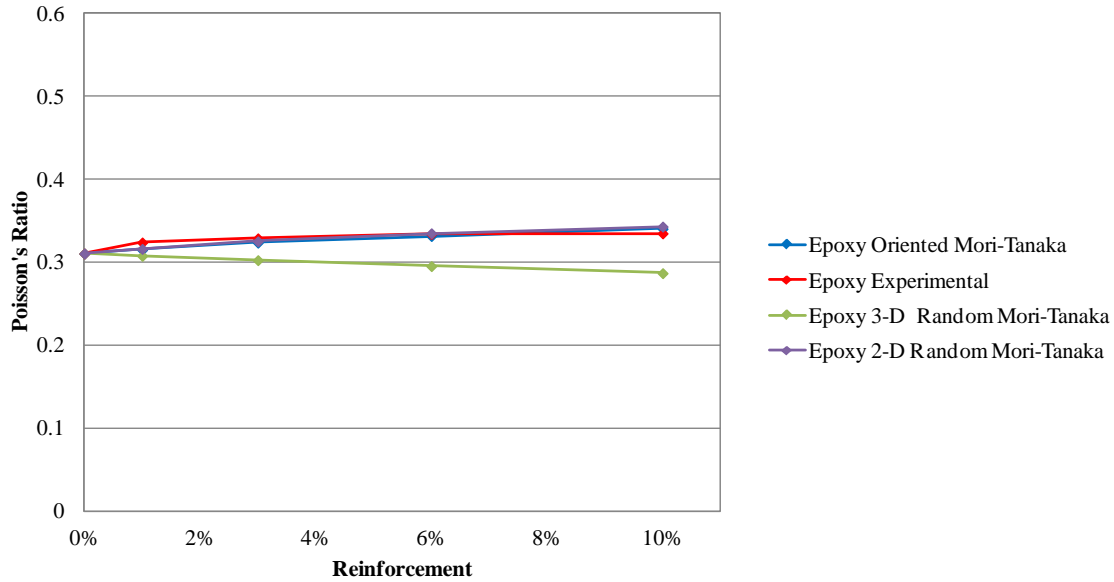


Figure 4.23 Comparison of Poisson's ratio obtained from experiments and oriented, 2-D and 3-D random Mori - Tanaka calculations at room temperature

4.1.3. Mori-Tanaka Calculations with Effect of Voids

When the SEM pictures are analyzed, it is observed that at high reinforcement percentages (contrary to low percentages) there may be voids between the particles and matrix. Because of this observation it was decided to modify the Mori Tanaka calculations by including the effect of voids.

We assume a parabolic distribution of voids as a function of reinforcement percentage since there are fewer voids at lower percentages than at higher percentages. With this assumption the volume fraction of particles can be calculated as follows:

Define:

V_d : max void percentage

w_p : nanoclay weight percentage

v_d : void fraction at each percentage

The parabolic void distribution can be expressed as:

$$v_d = Aw_p^2 \quad (58)$$

Assuming maximum void percentage occurs at $w_p=10\%$ the unknown A can be obtained from:

$$\begin{aligned} V_d &= A(0.1)^2 \\ \rightarrow A &= 100V_d \end{aligned}$$

Thus, Eq. (58) becomes

$$v_d = 100V_d (w_p)^2 \quad (59)$$

The total volume V and total weight are:

$$V = V_p + V_m + V_v \quad (60)$$

$$W = W_p + W_m \quad (61)$$

where,

V_p : total volume of particles

V_m : total volume of matrix

V_v : total volume of voids

W_p : total weight of particles

W_m : total weight of matrix

From,

$$V_p + V_m = \frac{W_p}{\gamma_p} + \frac{(W - W_p)}{\gamma_m} \quad (62)$$

$$V_v = Vv_d \quad (63)$$

where γ_p and γ_m are the specific weights of nanoclay and matrix respectively.

The total volume V is obtained as:

$$V = \frac{W_p}{\gamma_p} + \frac{(W - W_p)}{\gamma_m} + Vv_d$$

or

$$V = \frac{\left(\frac{W_p}{\gamma_p} + \frac{(W - W_p)}{\gamma_m} \right)}{(1 - v_d)} \quad (64)$$

With the new total volume V , we redefine the volume fraction c of particles in the presence of voids as:

$$c = \frac{\left(\frac{W_p}{\gamma_p} \right) (1 - v_d)}{\frac{W_p}{\gamma_p} + \frac{(1 - W_p)}{\gamma_0}} \quad (65)$$

In the ensuing sections, the Young's modulus and the Poisson's ratio are recalculated using the 3 different Mori Tanaka approaches with the new volume fraction c . For all cases the maximum void fraction V_d is assumed to be 6%.

4.1.3-a) Oriented Nanoclay Particles with Effect of Voids

Figures 4.24-4.33 show the comparison of experimental results and oriented Mori Tanaka calculations with and without voids at various temperatures.

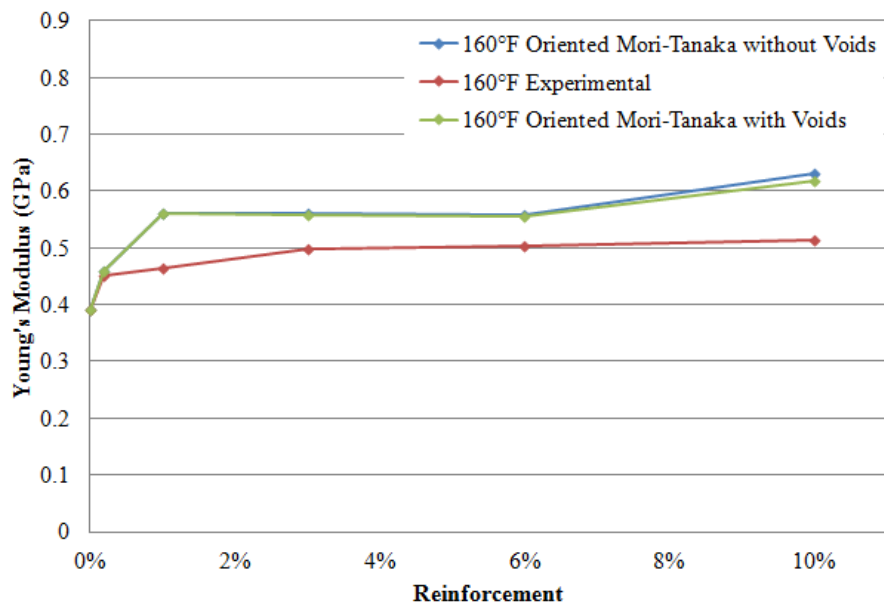


Figure 4.24 Comparison of Young's modulus obtained from experiments and oriented Mori - Tanaka calculations with and without voids at 160°F

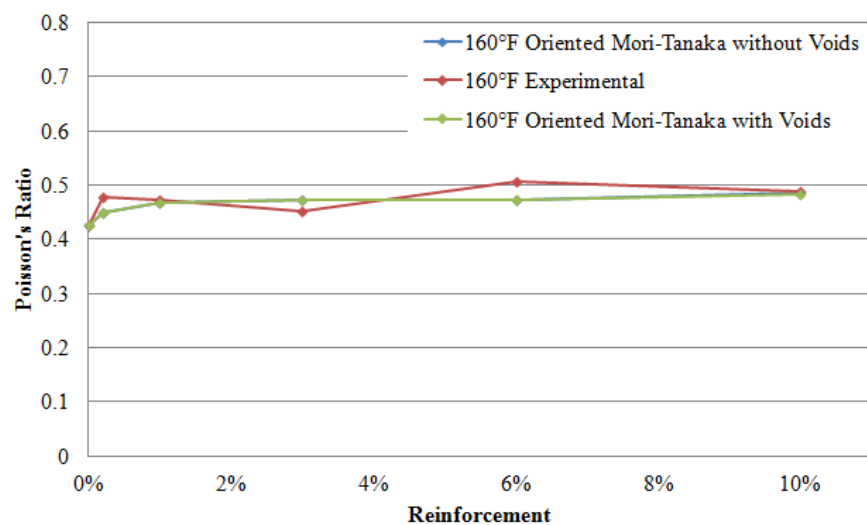


Figure 4.25 Comparison of Poisson's ratio obtained from experiments and oriented Mori - Tanaka calculations with and without voids at 160°F

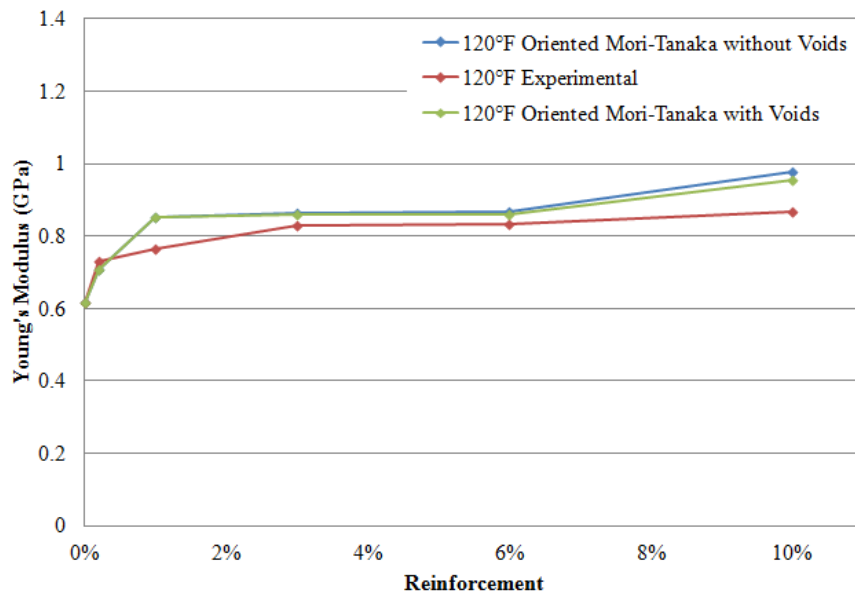


Figure 4.26 Comparison of Young's modulus obtained from experiments and oriented Mori - Tanaka calculations with and without voids at 120°F

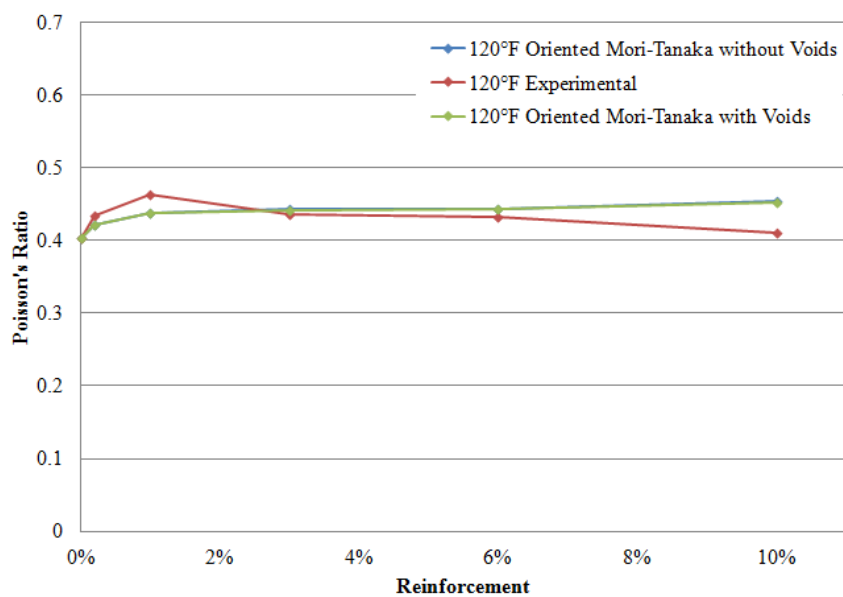


Figure 4.27 Comparison of Poisson's ratio obtained from experiments and oriented Mori - Tanaka calculations with and without voids at 120°F

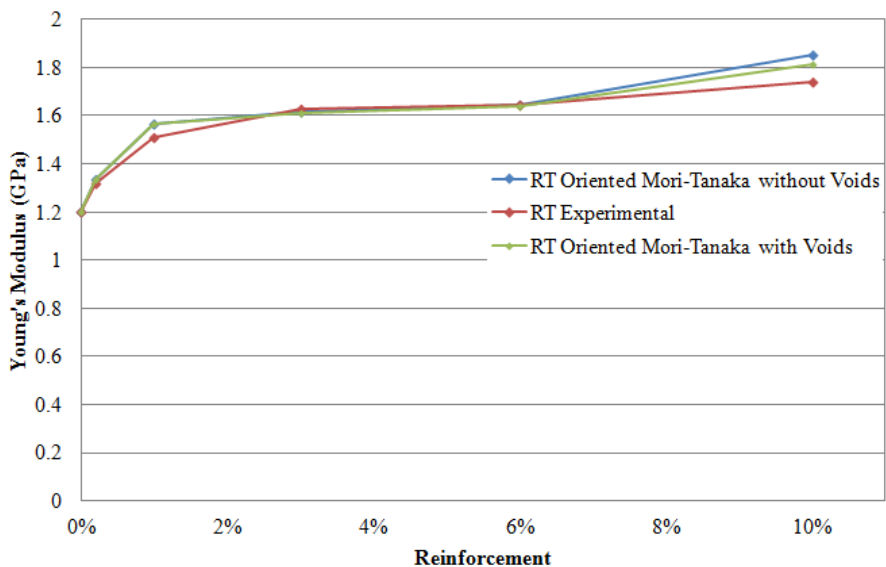


Figure 4.28 Comparison of Young's modulus obtained from experiments and oriented Mori - Tanaka calculations with and without voids at RT

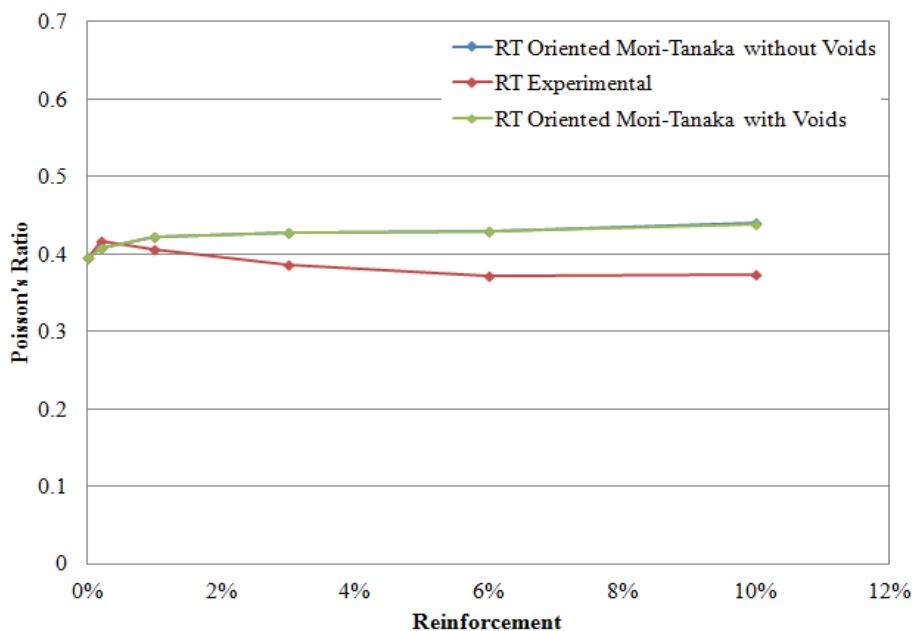


Figure 4.29 Comparison of Poisson's ratio obtained from experiments and oriented Mori - Tanaka calculations with and without voids at RT

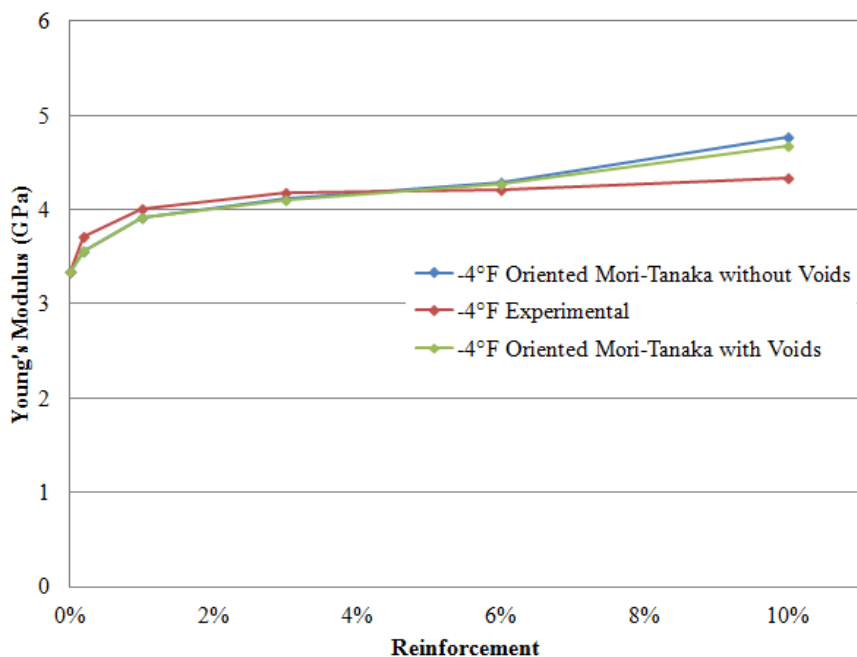


Figure 4.30 Comparison of Young's modulus obtained from experiments and oriented Mori - Tanaka calculations with and without voids at -4°F

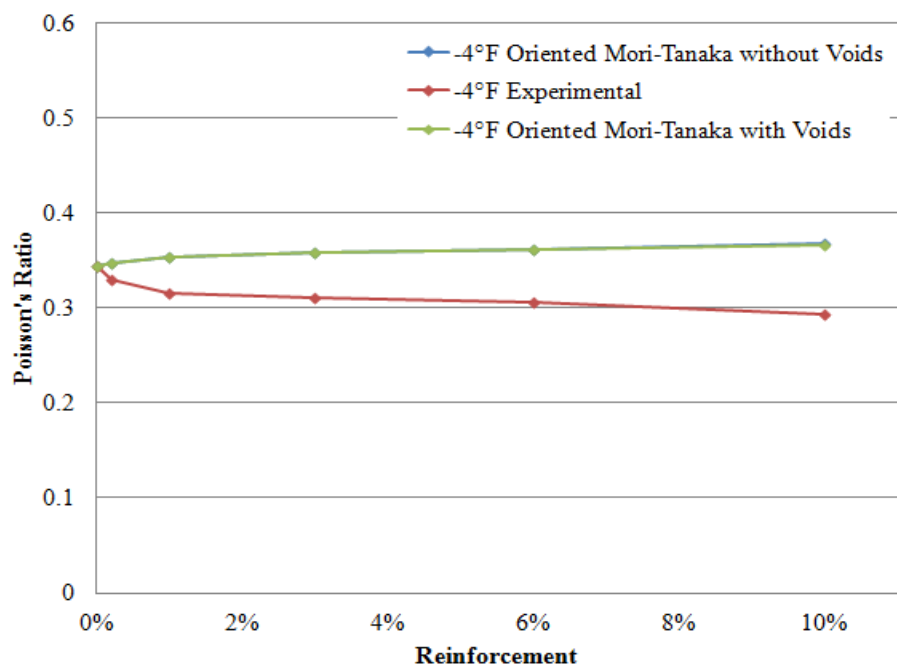


Figure 4.31 Comparison of Poisson's ratio obtained from experiments and oriented Mori - Tanaka calculations with and without voids at -4°F

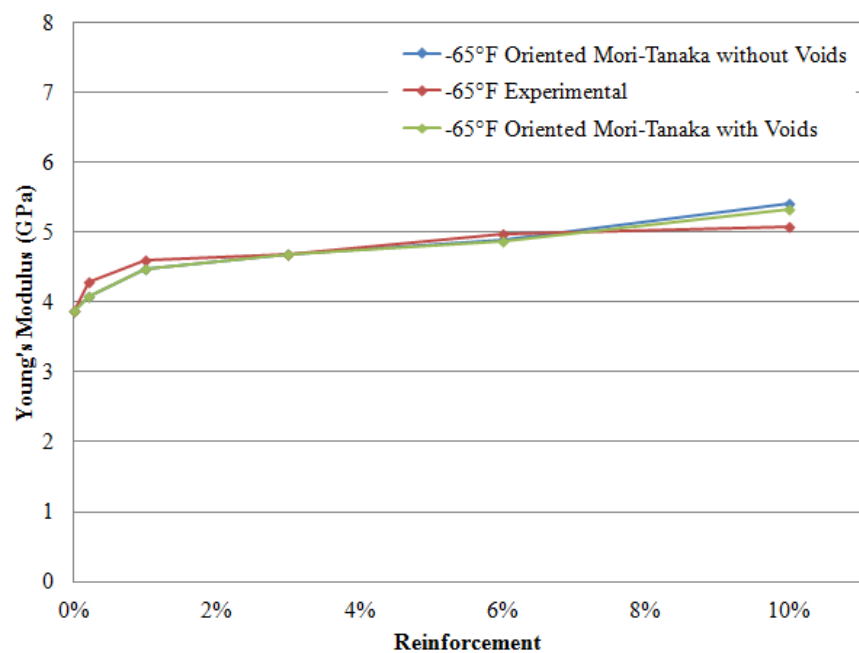


Figure 4.32 Comparison of Young's modulus obtained from experiments and oriented Mori - Tanaka calculations with and without voids at -65°F

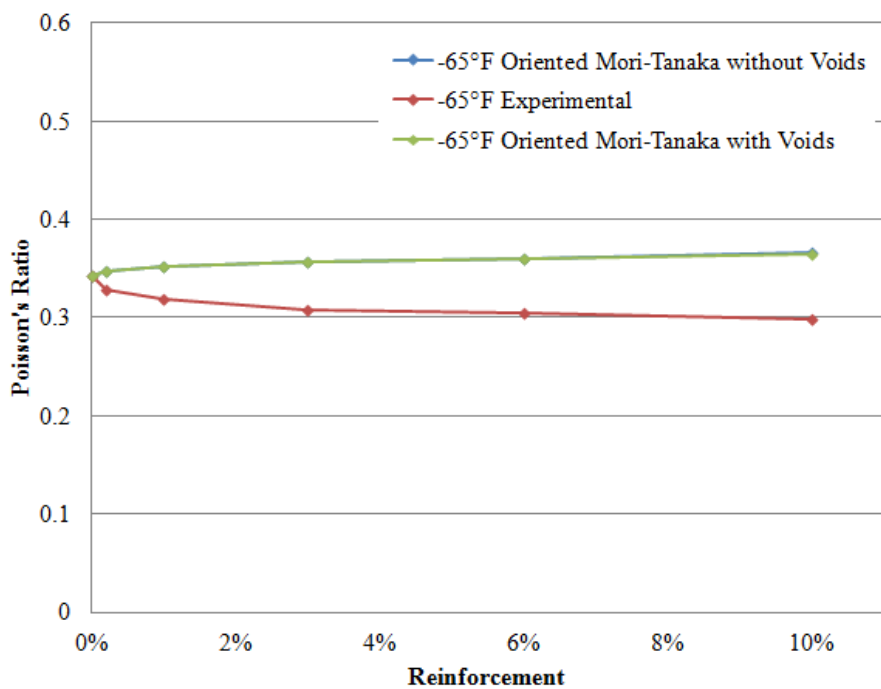


Figure 4.33 Comparison of Poisson's ratio obtained from experiments and oriented Mori - Tanaka calculations with and without voids at -65°F

As the results given in the previous Figures indicate the inclusion of voids in the calculations affects the value of Young's modulus mainly at 10% but it does not affect the Poisson's ratio significantly. The results obtained by including the effect of voids in general give a better match with experimental data.

4.1.3-b) 2-D Randomly Distributed Nanoclay Particles with Effect of Voids

We recalculated the Mori Tanaka results for the 2-D random cases by including the effect of voids and compared the results with experimental and 2-D random calculations without voids.

Figures 4.34-4.43 show these comparisons at each temperature.

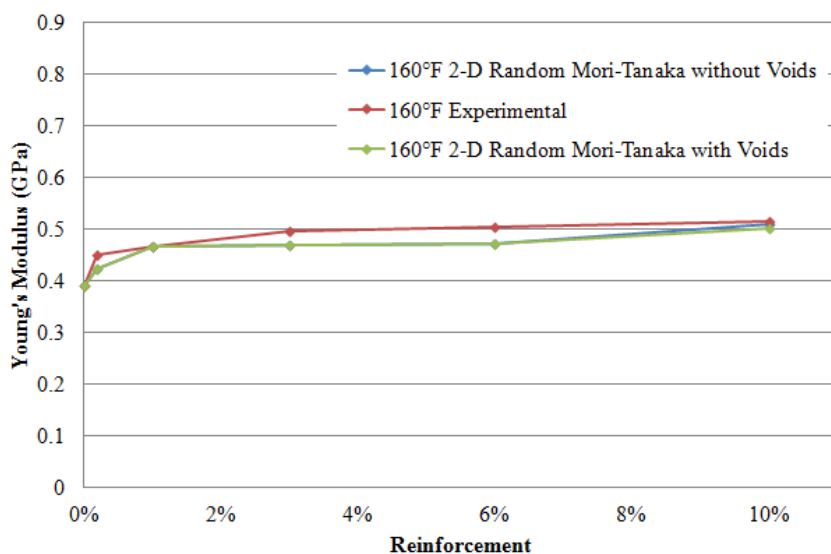


Figure 4.34 Comparison of Young's modulus obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at 160°F

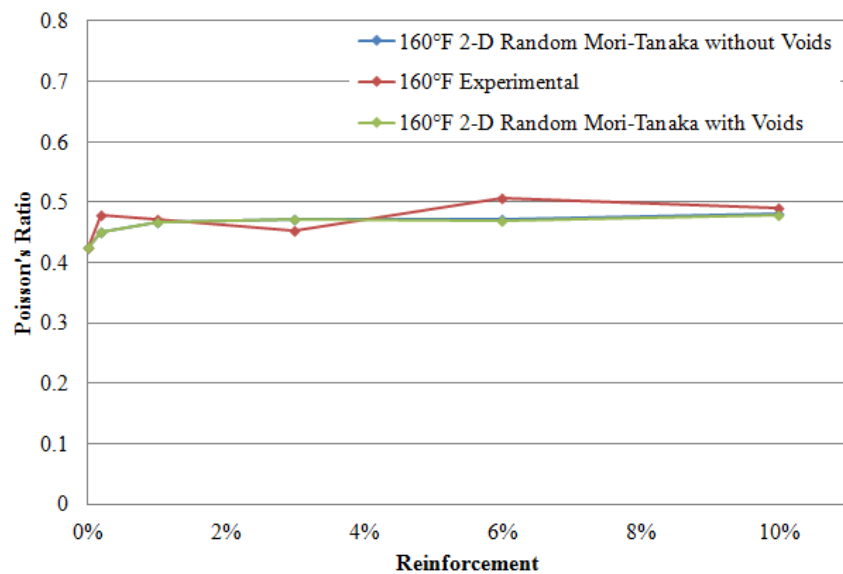


Figure 4.35 Comparison of Poisson's ratio obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at 160°F

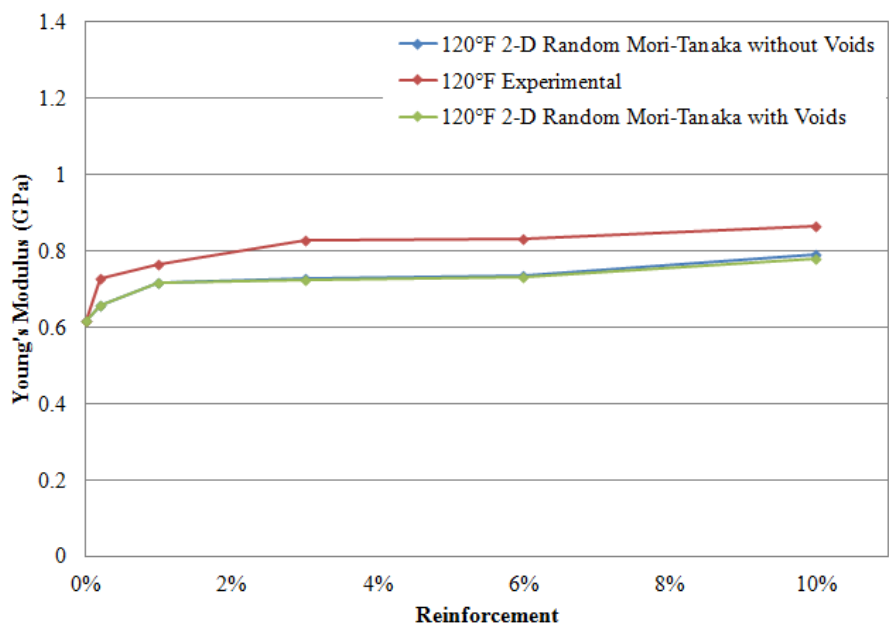


Figure 4.36 Comparison of Young's modulus obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at 120°F

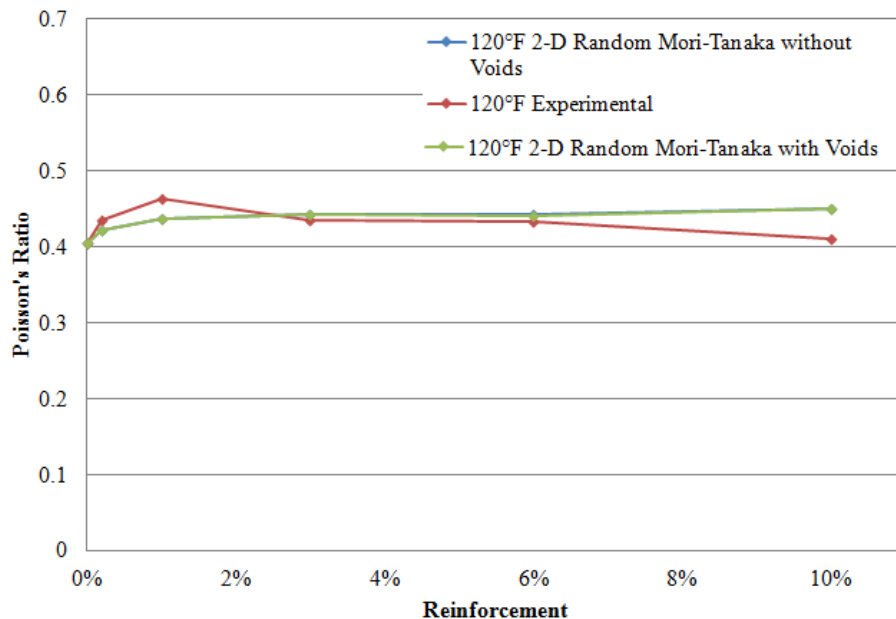


Figure 4.37 Comparison of Poisson's ratio obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at 120°F

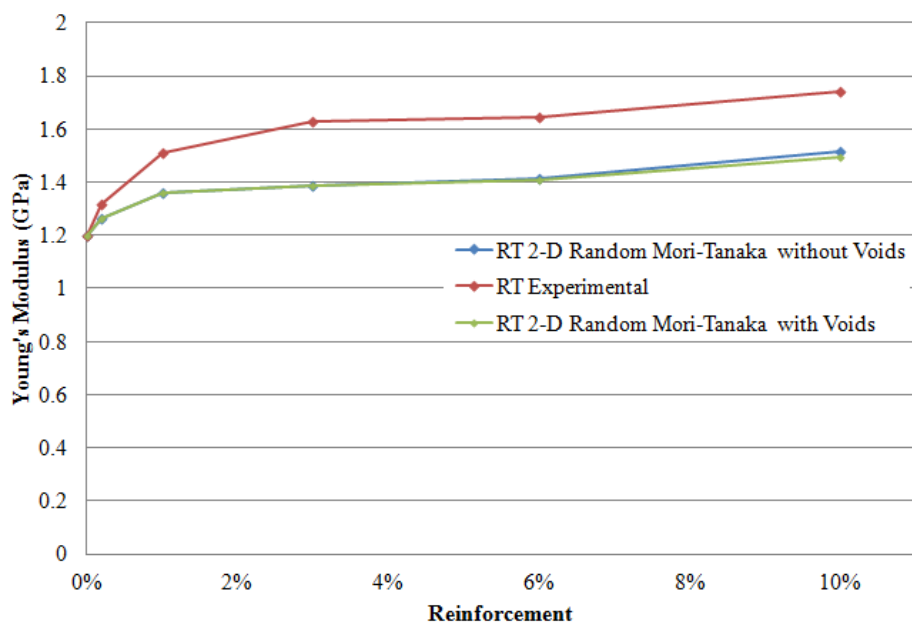


Figure 4.38 Comparison of Young's modulus obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at RT

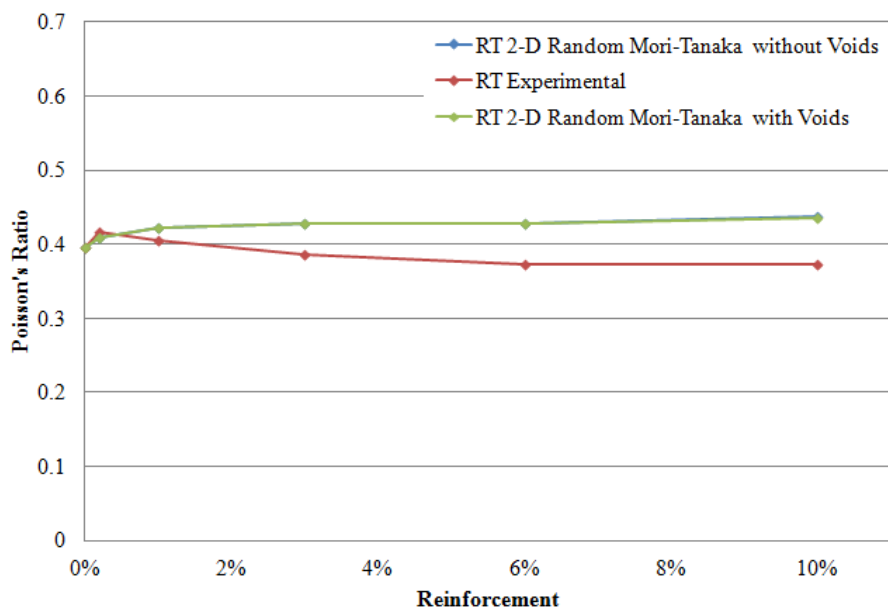


Figure 4.39 Comparison of Poisson's ratio obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at RT

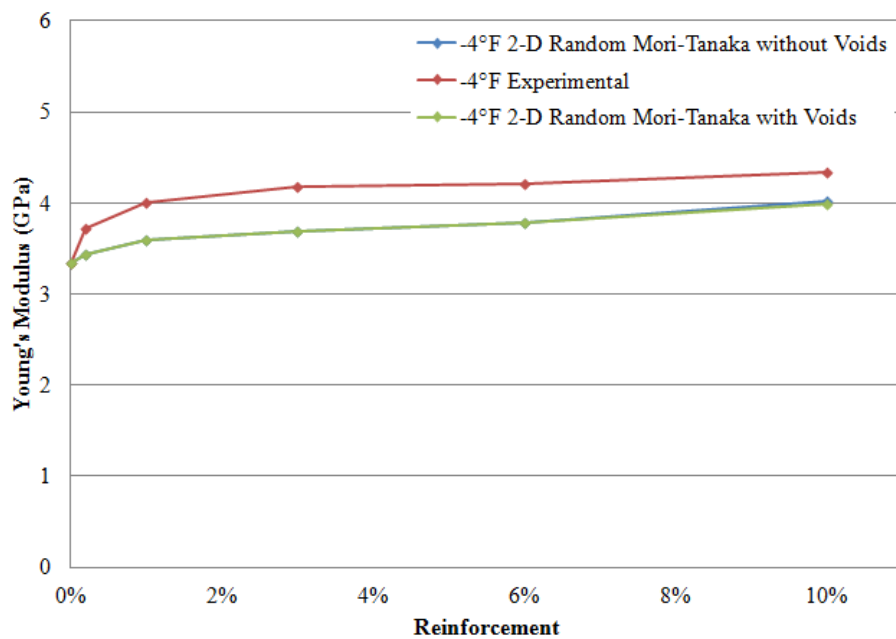


Figure 4.40 Comparison of Young's modulus obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at -4°F

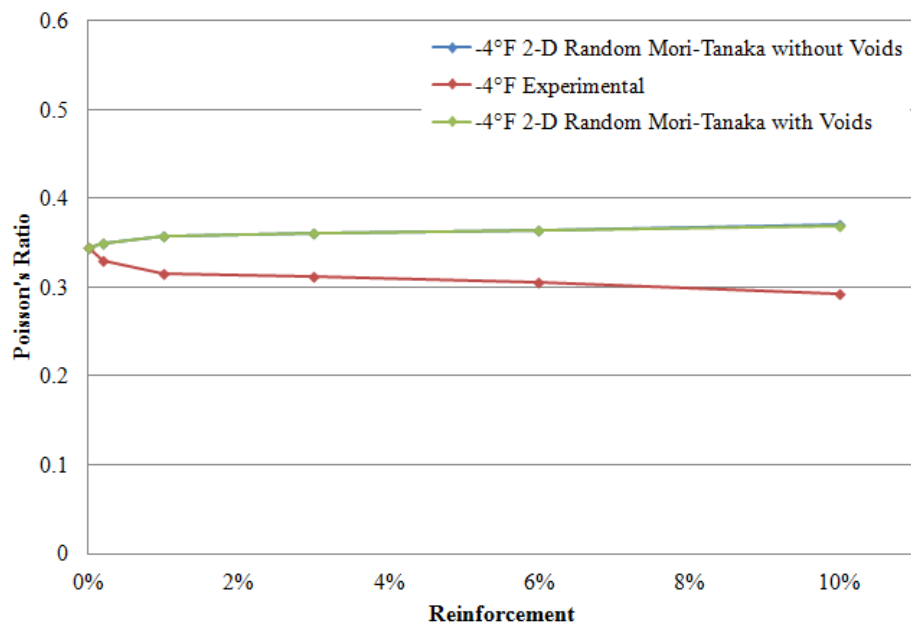


Figure 4.41 Comparison of Poisson's ratio obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at -4°F

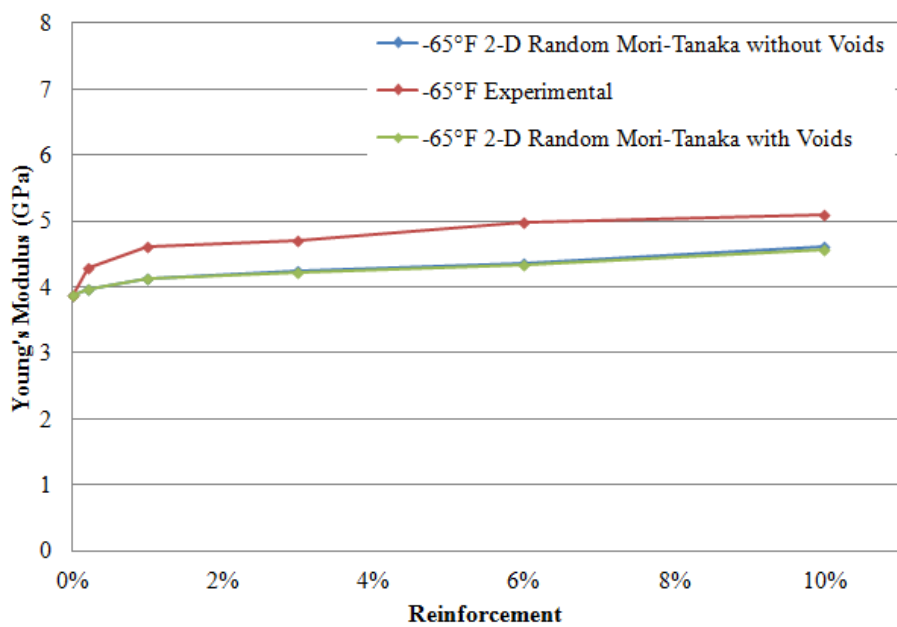


Figure 4.42 Comparison of Young's modulus obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at -65°F

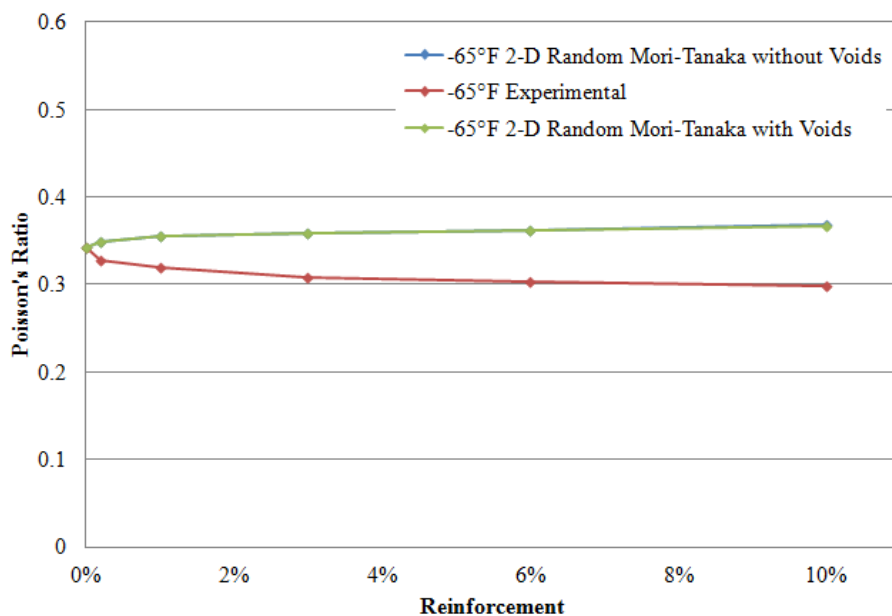


Figure 4.43 Comparison of Poisson's ratio obtained from experiments and 2-D random Mori - Tanaka calculations with and without voids at -65°F

In the Figures one may observe that, the Young's modulus values are lower than the experimental results and the Poisson's ratio values are not affected significantly. There is improvement in matching the experimental results for higher percentages of reinforcement at room and lower temperatures. At higher temperatures the improvement is not evident.

4.1.3-c) 3-D Randomly Distributed Nanoclay Particles with Effect of Voids

Finally, we performed Mori -Tanaka calculations for 3-D random cases by including the effect of voids and compared the results with experimental and 3D random calculations without voids. Figures 4.44-4.53 show the comparison at each temperature.

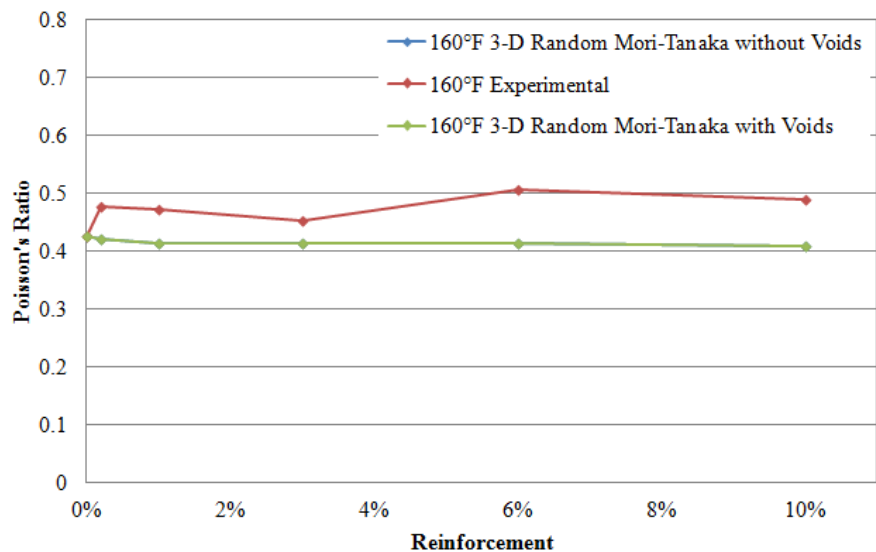


Figure 4.44 Comparison of Young's modulus obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at 160°F

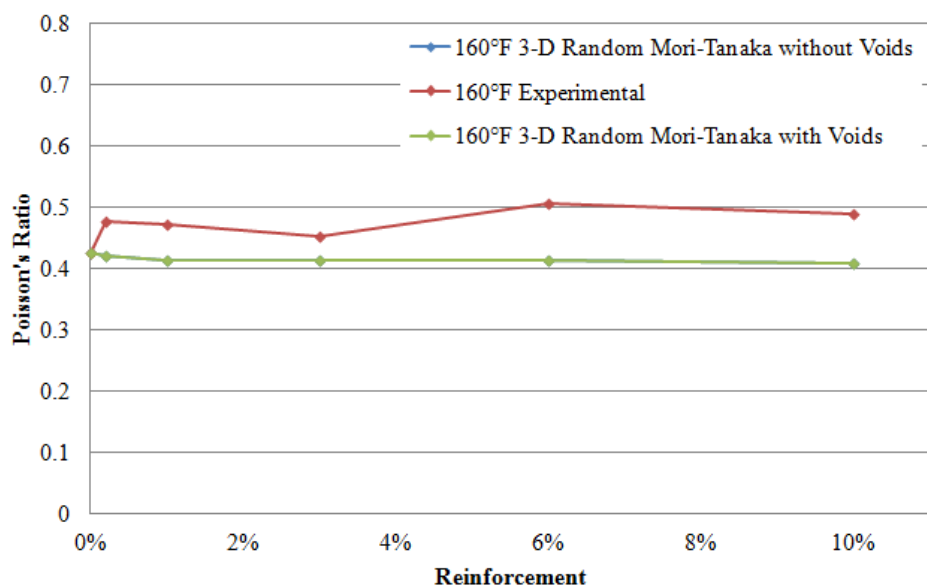


Figure 4.45 Comparison of Poisson's ratio obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at 160°F

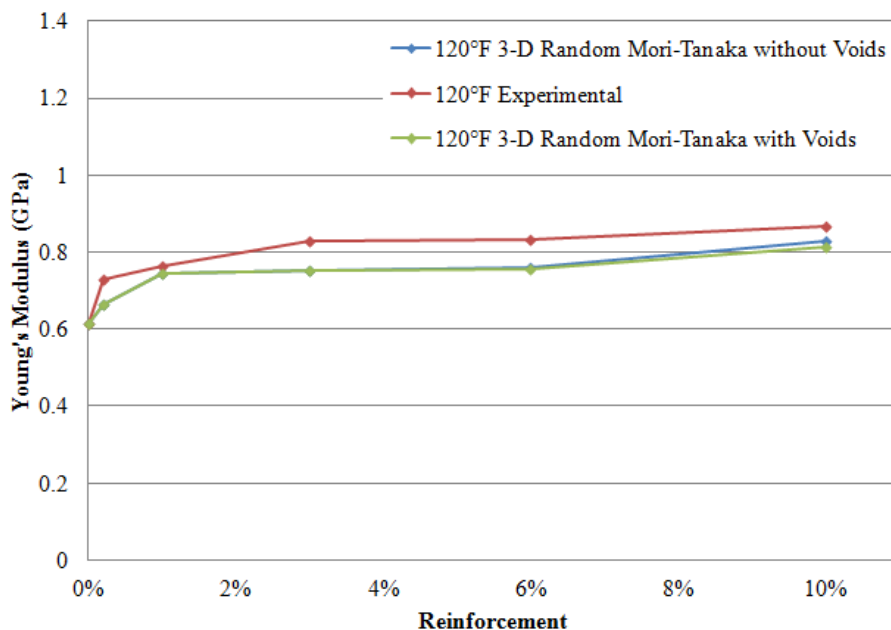


Figure 4.46 Comparison of Young's modulus obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at 120°F

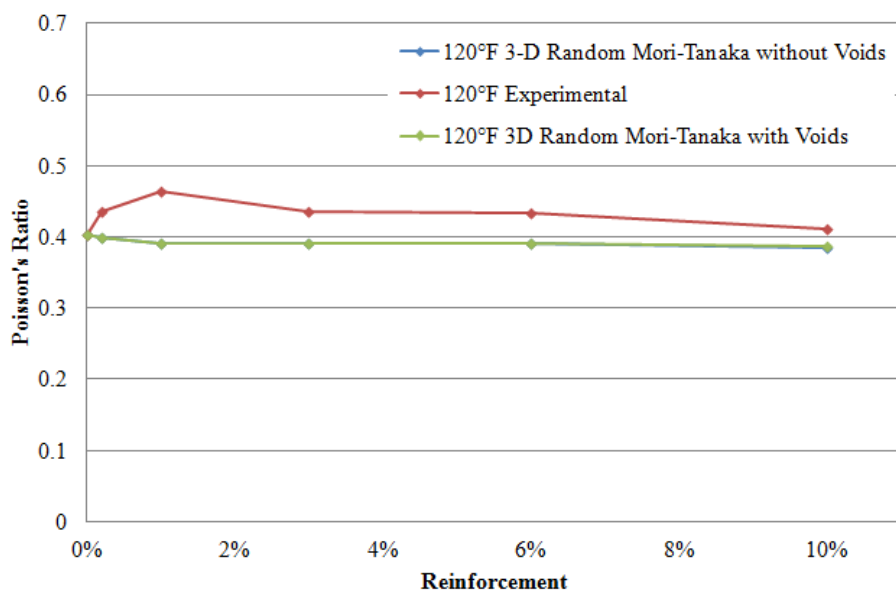


Figure 4.47 Comparison of Poisson's ratio obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at 120°F

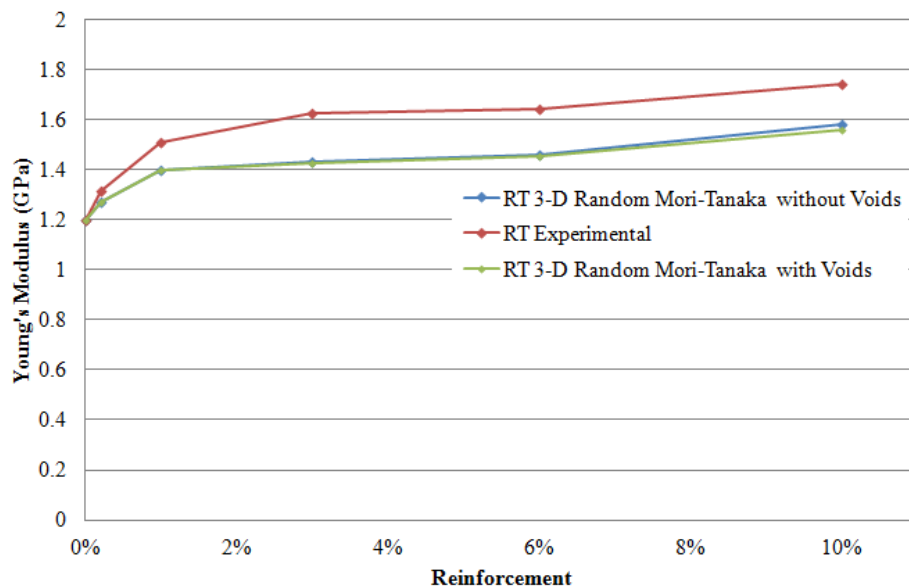


Figure 4.48 Comparison of Young's modulus obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at RT

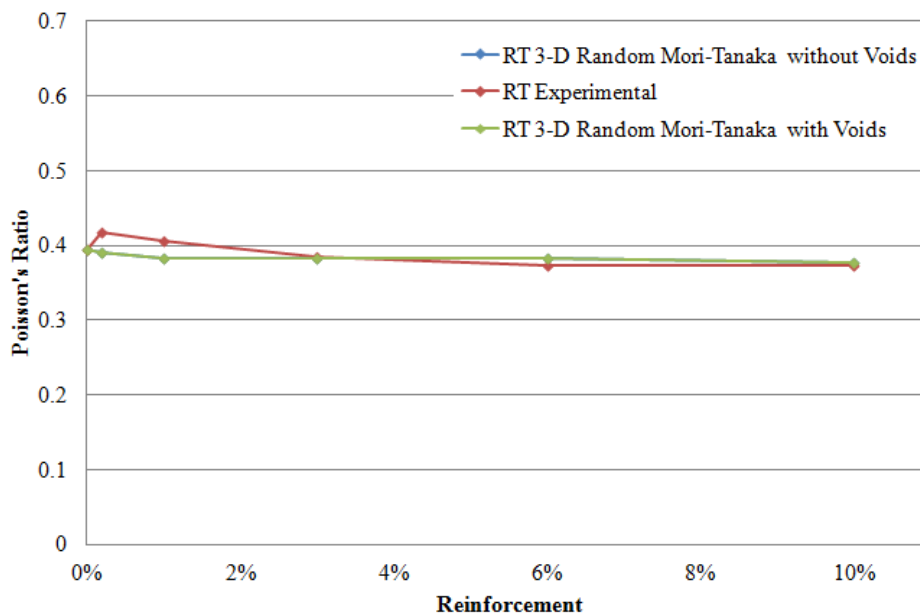


Figure 4.49 Comparison of Poisson's ratio obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at RT

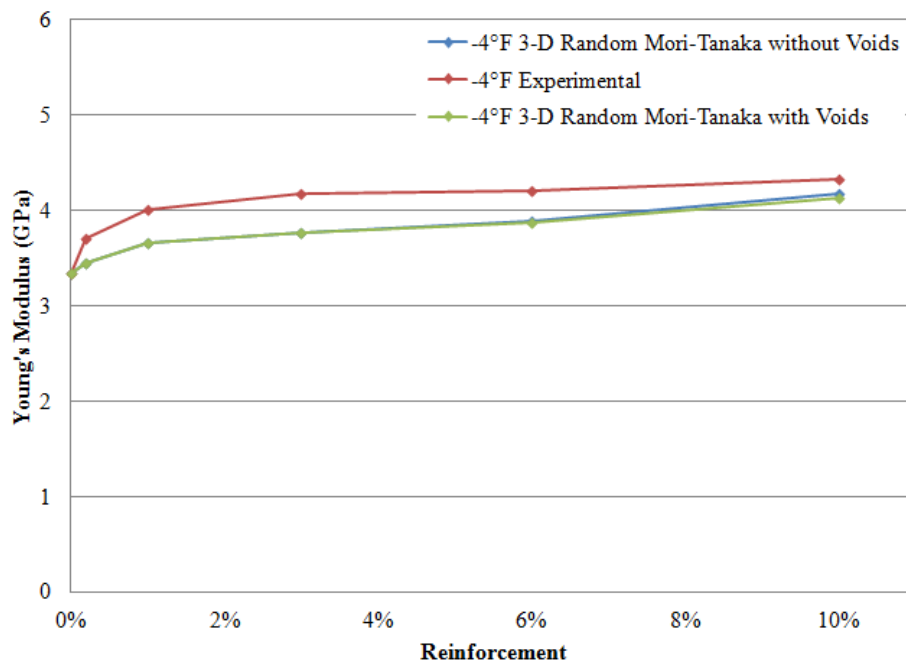


Figure 4.50 Comparison of Young's modulus obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at -4°F

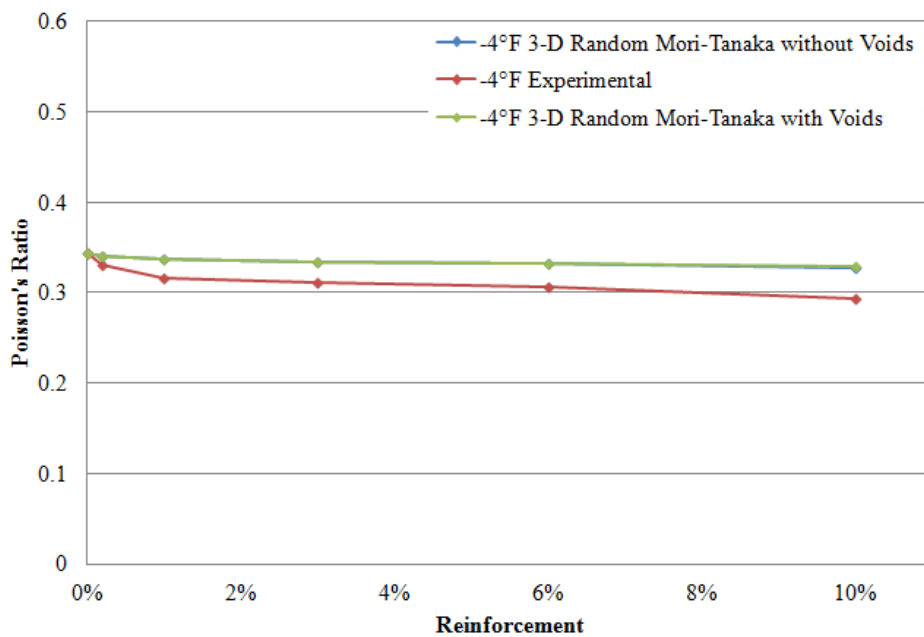


Figure 4.51 Comparison of Poisson's ratio obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at -4°F

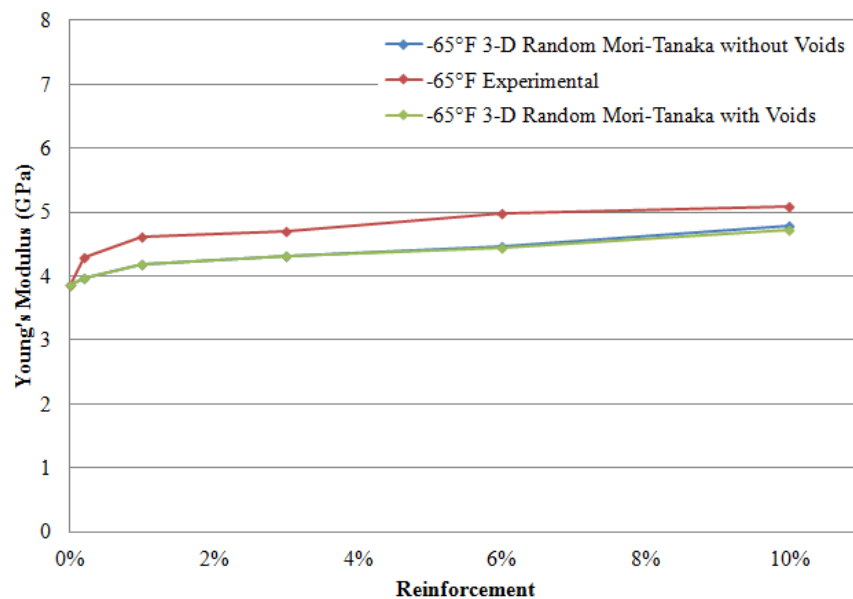


Figure 4.52 Comparison of Young's modulus obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at -65°F

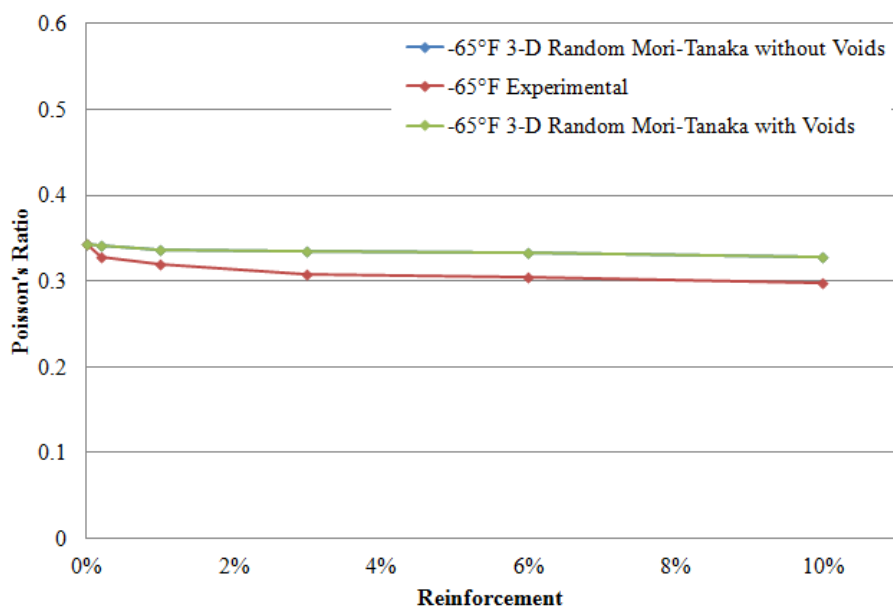


Figure 4.53 Comparison of Poisson's ratio obtained from experiments and 3-D random Mori - Tanaka calculations with and without voids at -65°F

In general, the 3- D random Mori - Tanaka calculations give lower Young's modulus values than the experimental results and the inclusion of voids does not materially change the results.

4.1.4 Modification of Mori-Tanaka Model with the Inclusion of Temperature Effects

The Figures in sections 4.1.2 show that at the higher temperatures the Mori-Tanaka results do not match the experimental results well.

Since nanoclay reinforced PP specimens don't show a pronounced linear material behavior at high temperatures and Mori-Tanaka calculations depend on linear material properties, we decided to include the temperature effect into the Mori-Tanaka calculations for 120°F and 160° by redefining the Young's modulus.

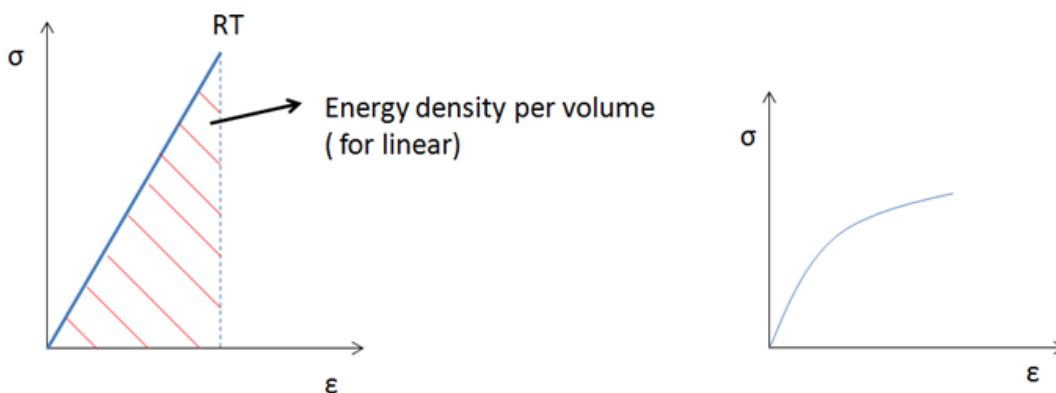


Figure 4.54 Stress vs. strain curves for linear and non-linear materials

At room temperature we have a clear linear behavior for low strains (or stresses) and we can calculate the energy density per volume. But when we perform the tensile tests at higher temperatures the linear portion of the curve is less pronounced and the curve is more-or-less nonlinear (Figure 4.54). Even though we did not do unloading during the tensile tests, we believe that at higher temperatures the material shows viscoelastic behavior.

Due to viscoelastic behavior, at high temperatures there is energy loss during unloading as shown in Figure 4.55. Considering the nonlinear behavior of the material, at high temperatures

we believe that it is more appropriate to use the secant rather than the tangent in calculating the Young's modulus.

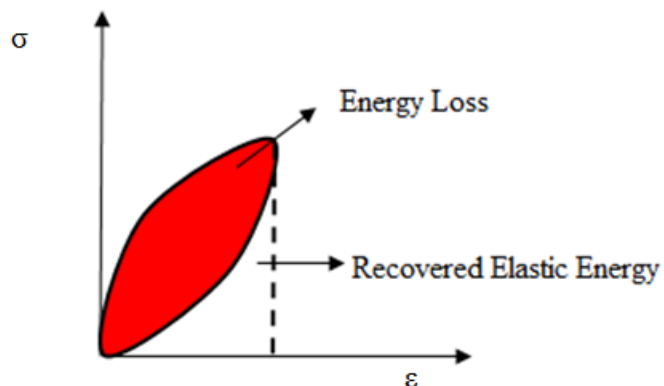


Figure 4.55 Stress vs. strain curves for viscoelastic behavior and lost energy.

To calculate the Young's modulus at 120°F and 160°F we assume that the elastic energy density (recoverable energy) in calculating the Young's modulus at room temperature is the same at high temperatures.

Figure 4.56 shows schematically the secant line and the elastic energy density using the stress-strain curve at 120°F and 160°F.

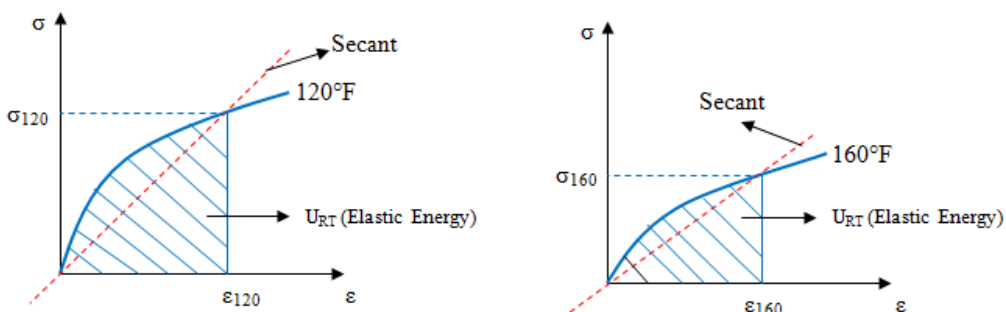


Figure 4.56 Calculation of secant Young's modulus from nonlinear material curve at 120°F and 160°F

First we calculate the strain energy density from the linear portion of the stress-strain curve at room temperature and then use this elastic energy to determine the strain (or stress) to be used in the calculation of the secant Young's modulus at higher temperatures.

Using this procedure we recalculated the Young's Modulus of neat PP specimens at 120°F and 160°F based on the secant line as (Figure 4.56): $E_{120} = \frac{\sigma_{120}}{\epsilon_{120}}$; $E_{160} = \frac{\sigma_{160}}{\epsilon_{160}}$ and we used these new values in the Mori-Tanaka calculations formulas given in Eqs. (1)-(6) and the composition assumed in Table 4.20 to calculate the elastic properties.

This temperature effect assumption was considered only for oriented particles in the Mori-Tanaka calculations.

In Figures 4.57, the Young's modulus values at 120°F obtained experimentally and with Mori-Tanaka calculations using the tangent or secant definitions are plotted vs. nanoclay reinforcement percentage. Table 4.21 also shows the same results and the comparison of experimental Young's modulus with Mori-Tanaka calculations depending on the tangent or secant assumptions.

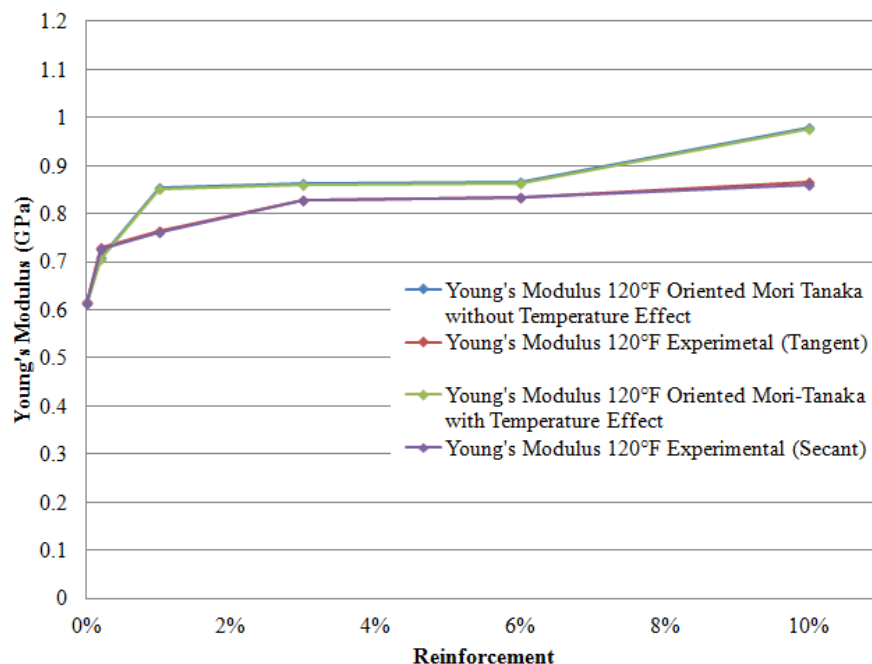


Figure 4.57 Comparison of experimental and calculated Young's Modulus values at 120°F

Table 4.21 Young's Modulus values based on tangent and secant calculations at 120°F

120°F	YM Exp. (GPa) (Tangent)	YM Mori- Tanaka Without Temperature Effect (GPa)	YM Exp (Secant) (GPa)	YM Mori- Tanaka With Temp. Effect (GPa)
0%	0.61595	0.61595	0.61389	0.61389
0.2%	0.7285	0.7079	0.72561	0.7057
1%	0.7646	0.85319	0.76208	0.85054
3%	0.8283	0.86217	0.82759	0.85941
6%	0.8332	0.8664	0.8328	0.8635
10%	0.8664	0.9789	0.85936	0.9757

In Figure 4.58 the experimental and Mori-Tanaka results at 160°F are compared using the tangent and secant calculations. The same values are also given in Table 4.22.

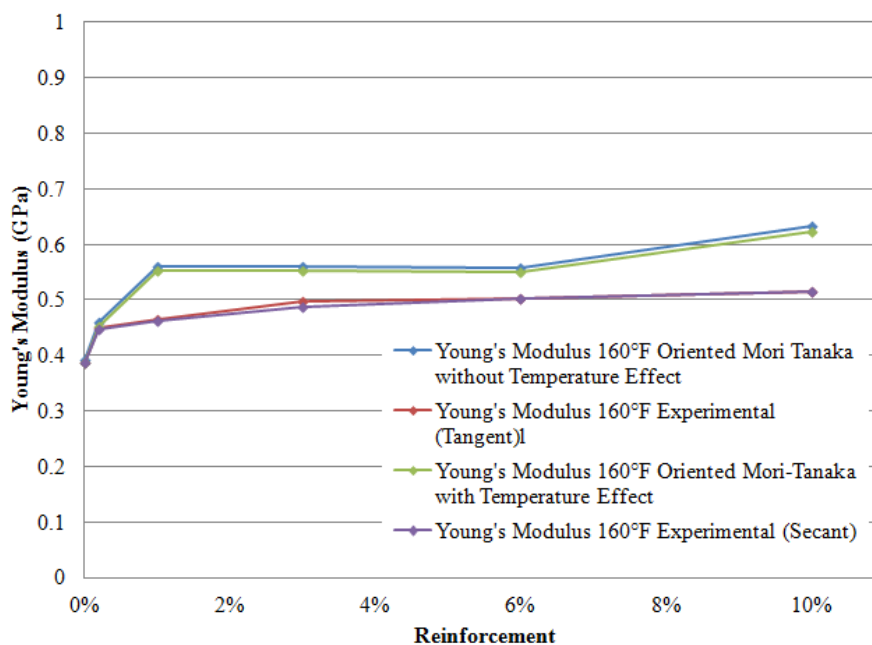


Figure 4.58 Comparison of experimental and calculated Young's Modulus values for at 160°F

Table 4.22 Young's Modulus values based on tangent and secant calculations at 160°F

160°F	YM Exp. (GPa) (Tangent)	YM Mori- Tanaka Without Temperature Effect(GPa)	YM Exp (Secant) (GPa)	YM Mori-Tanaka With Temp. Effect (GPa)
0%	0.39204	0.39204	0.38639	0.38639
0.2%	0.4499	0.4586	0.44771	0.4522
1%	0.4654	0.4654	0.46195	0.55355
3%	0.4969	0.4969	0.48835	0.55199
6%	0.5031	0.5583	0.50229	0.5504
10%	0.5146	0.632	0.51415	0.623

The results in Figures 4.57 and Figure 4.58 show the effect of temperature on the Mori-Tanaka calculations. At 120°F, the temperature effect is not significant, because all the results are close to each other. But at 160°, we can clearly see the temperature effect. Thus at high temperatures since nanoclay reinforced PP specimens don't behave linearly, to obtain more reasonable Mori-Tanaka results one may have to consider the temperature effect by calculating the Young's Modulus values based on the secant assumption.

4.2 3-D FINITE ELEMENT MODELING FOR TENSILE TESTING

A 3-D finite element model of the nanocomposite was developed using ABAQUS software. The 3-D model is based on the concept of representative volume of the material.

The representative volume has dimensions of (40 unit) \times (40 unit) \times (40 unit) and includes the matrix and oriented disk shaped particles (parallel to the loading direction). The number of particles n is adjusted to result in a fixed $\frac{t}{D}$ aspect ratio (used in the Mori-Tanaka calculations) and the desired nanoclay reinforcement percentage which varies from 0.2% to 10%.

To demonstrate how the various particle parameters are calculated, consider the case for 0.2% nanoclay reinforcement. For 0.2% reinforcement, the volume fraction of nanoclay, using Eq.(28) or Table 4.1, is 0.095%.

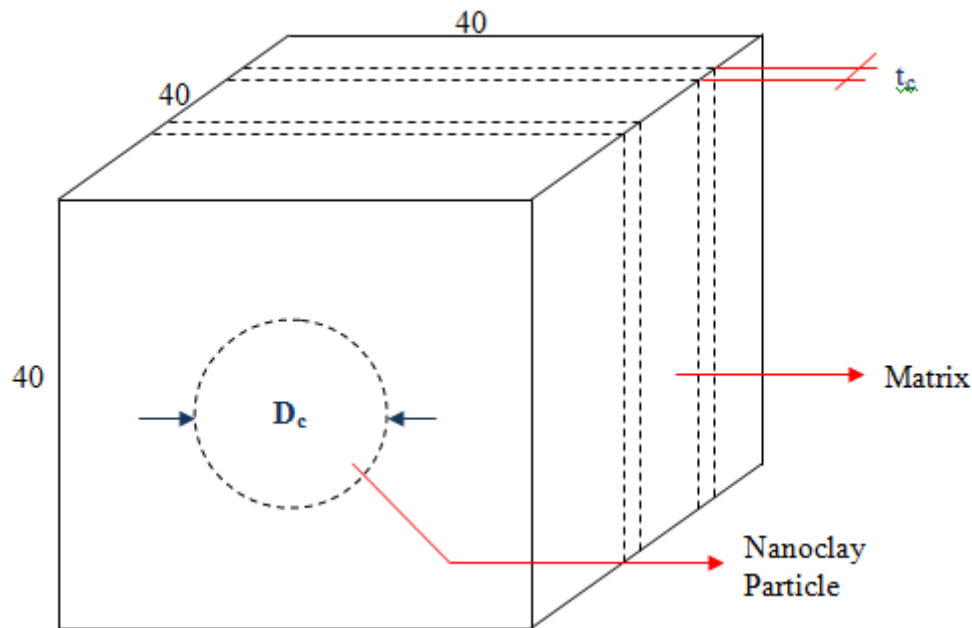


Figure 4.59 Representative Volume Element

In the previous section for 0.2% the disk diameter and particle thickness were assumed as:
 $D=200$ nm and $t=0.615$ nm.

Thus, $\frac{t}{D} = \frac{1}{325.20}$ or $D=325.20t$.

To maintain the same aspect ratio, we write

$$\frac{t_c}{D_c} = \frac{t}{D} = \frac{1}{325.20} \text{ or } D_c=325.20t_c$$

Then, assuming $n=2$ and the total volume as V , from the volume fraction equation t_c is found as:

$$\frac{n \frac{\pi D_c^2}{4} t_c}{V} = c \quad (66)$$

or

$$\frac{2\pi (325.20)^2 t_c^3}{4(40)^3} = 0.00095 \rightarrow t_c = 0.0715 \text{ unit and } D_c = 23.25 \text{ unit.}$$

For 6% and 10%, t_c and D_c are calculated similarly. For 1% and 3%, since we have particles of different thickness the calculation, though in principle the same, is slightly different. To illustrate this, consider the case of 1% reinforcement.

As explained in section 4.2.2 and Table 4.20, for 1% we use a combination of particles with different thickness. The composition used was 40% of $N=1$ particles, 30% of $N=2$ particles and 30% of $N=3$ particles. The volume fraction c corresponding to 1% is (Table 4.1) 0.48%.

Again, to maintain the same aspect ratios used previously, we calculate the particle thicknesses as follows:

- For N=1 particles, D=200 nm, t=0.615 nm or D=325.20t. If we assume 2 layers with particles of this geometry, then with $D_c=325.20t_c$ we obtain:

$$\frac{2\pi(325.20)^2 t_c^3}{4(40)^3} = 0.4(0.0048) \rightarrow t_c = 0.0904 \text{ unit and } D_c = 29.41 \text{ unit.}$$

- For N=2 particles, D=200 nm, t=3.015 nm or D=66.33t. If we assume 2 layers with particles of this geometry, then with $D_c=66.33t_c$ we obtain:

$$\frac{2\pi(66.33)^2 t_c^3}{4(40)^3} = 0.3(0.0048) \rightarrow t_c = 0.2371 \text{ unit and } D_c = 15.73 \text{ unit.}$$

- For N=3 particles, D=200 nm, t=5.415 nm or D=36.93t. If we assume 1 layer with particles of this geometry, then with $D_c=36.93t_c$ we obtain:

$$\frac{\pi(36.93)^2 t_c^3}{4(40)^3} = 0.3(0.0048) \rightarrow t_c = 0.4414 \text{ unit and } D_c = 16.30 \text{ unit.}$$

The calculations for 3% are similar to those presented above. The calculated particle parameters for each reinforcement percentage are given in Table 4.23.

Table 4.23 The particle parameters used in the finite element calculation for each reinforcement percentage

Reinforcement Percentage	n	t_c	D_c
0.2%	2	0.0715	23.25
1%	2	0.0904	29.41
	2	0.2371	15.73
	1	0.4414	16.30
3%	2	0.3764	24.97
	2	0.5053	18.66
	1	0.8131	20.81
6%	6	1.0125	19.82
10%	10	1.1747	18.62

To explain the methodology used in the finite element model, consider the case for 0.2% reinforcement. Here, we have 2 layers with nanoclay particles, in the PP matrix. Figure 4.60 shows the symmetric finite element model constructed for this case. Then the appropriate boundary conditions were applied. The bottom surface was constrained in y-direction and one point in the middle of the bottom surface was fixed to prevent translational displacement. The boundary conditions and loading are shown in Figure 4.60. For loading, a unit displacement was applied at the top surface.

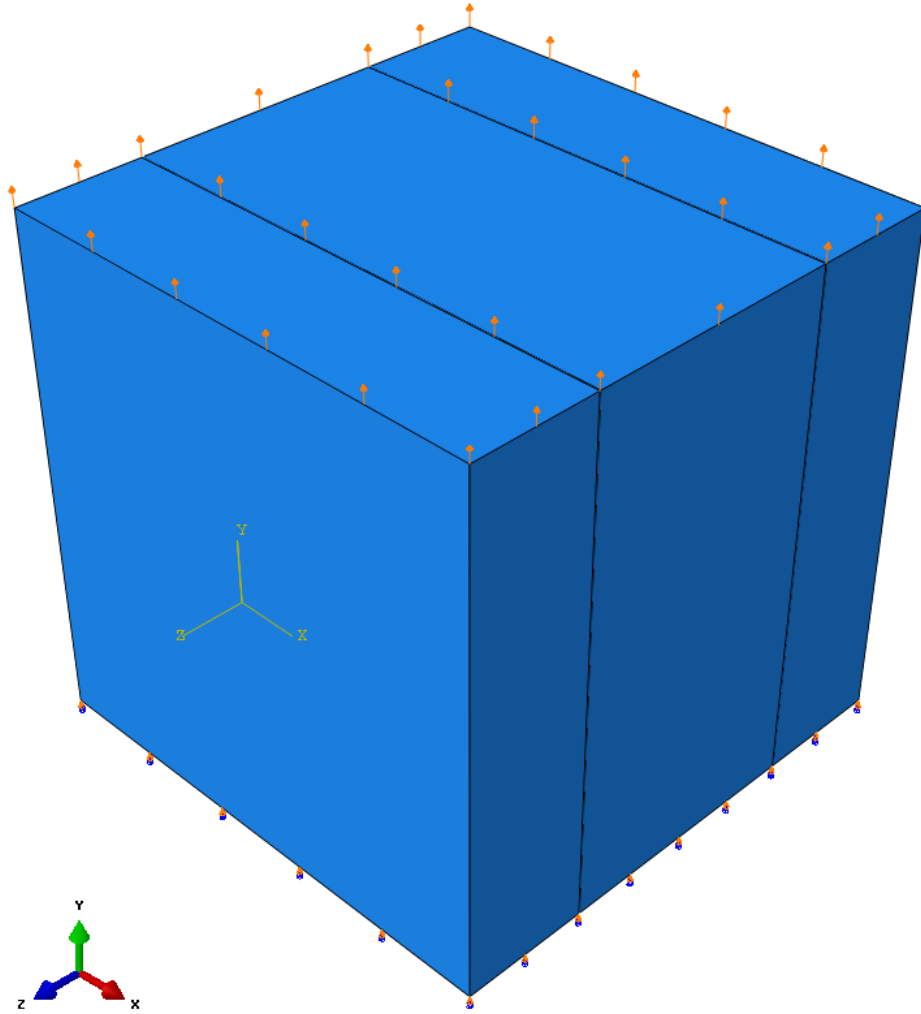


Figure 4.60 The boundary conditions and displacement loading for the 0.2% specimens

The 3-D finite element mesh is shown in Figure 4.61.

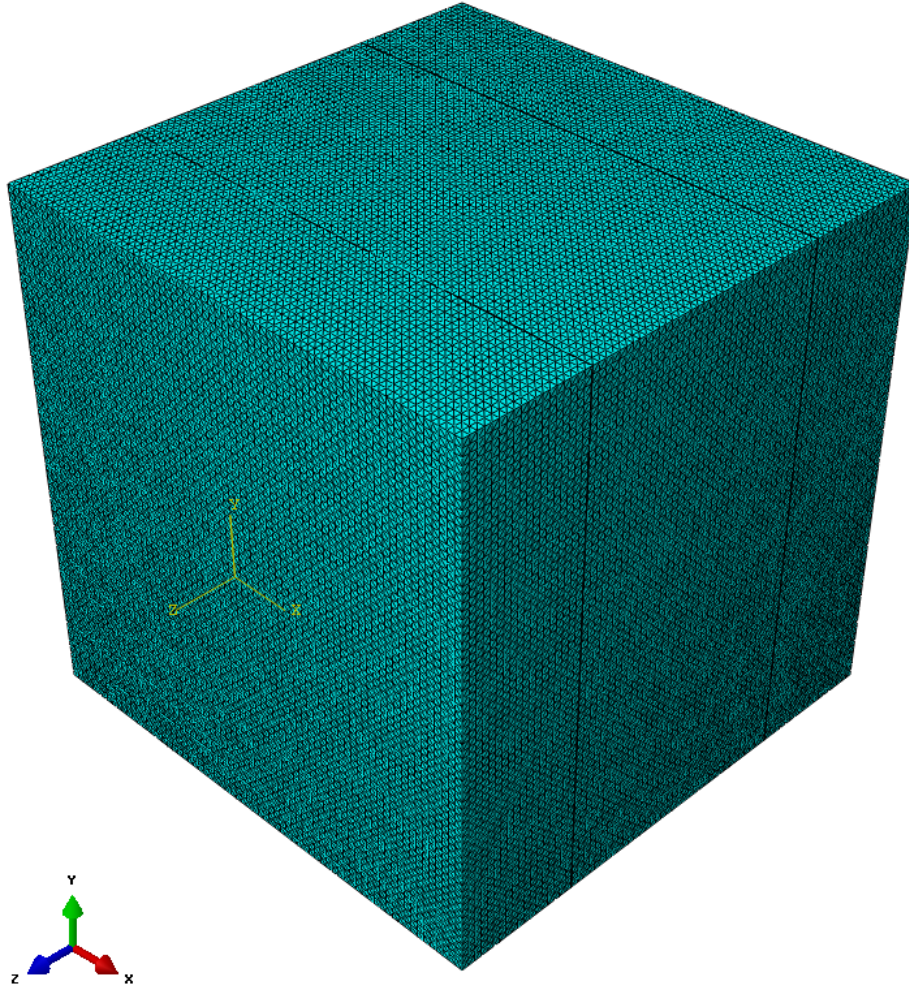


Figure 4.61 The mesh structure of the finite element model

Some of the 3-D FE results obtained for the 0.2% nanoclay reinforced specimens are shown in Figures 4.62- 4.65.

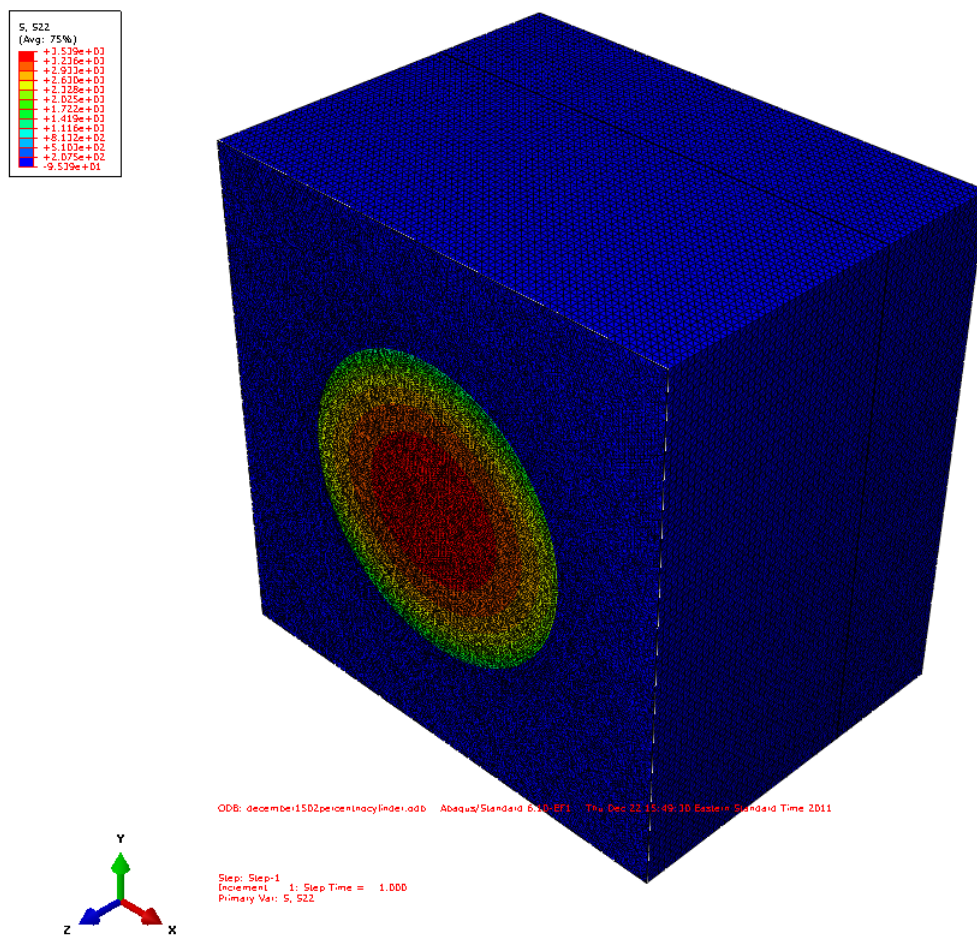


Figure 4.62 Distribution of stresses in y- direction (σ_{yy}) at the particle plane

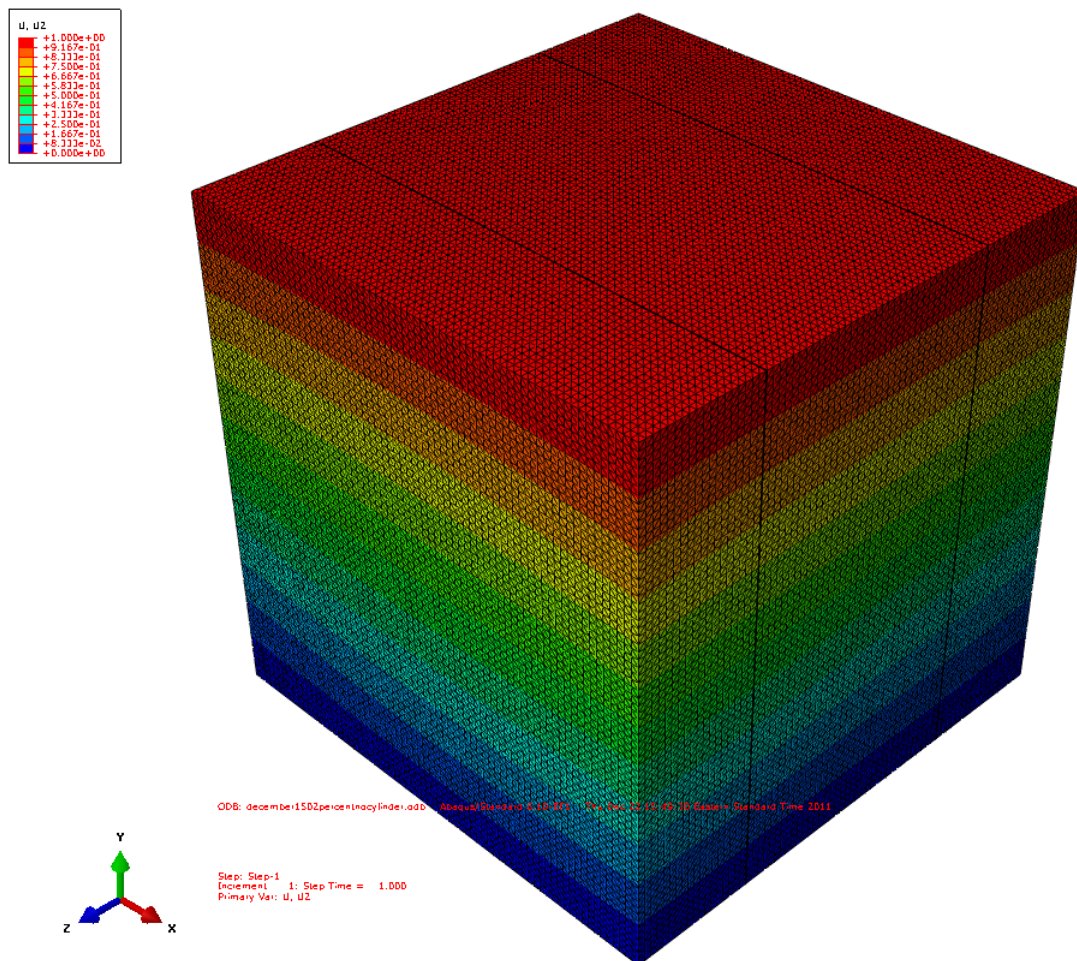


Figure 4.63 Distribution of y-component of displacement (U_{yy})

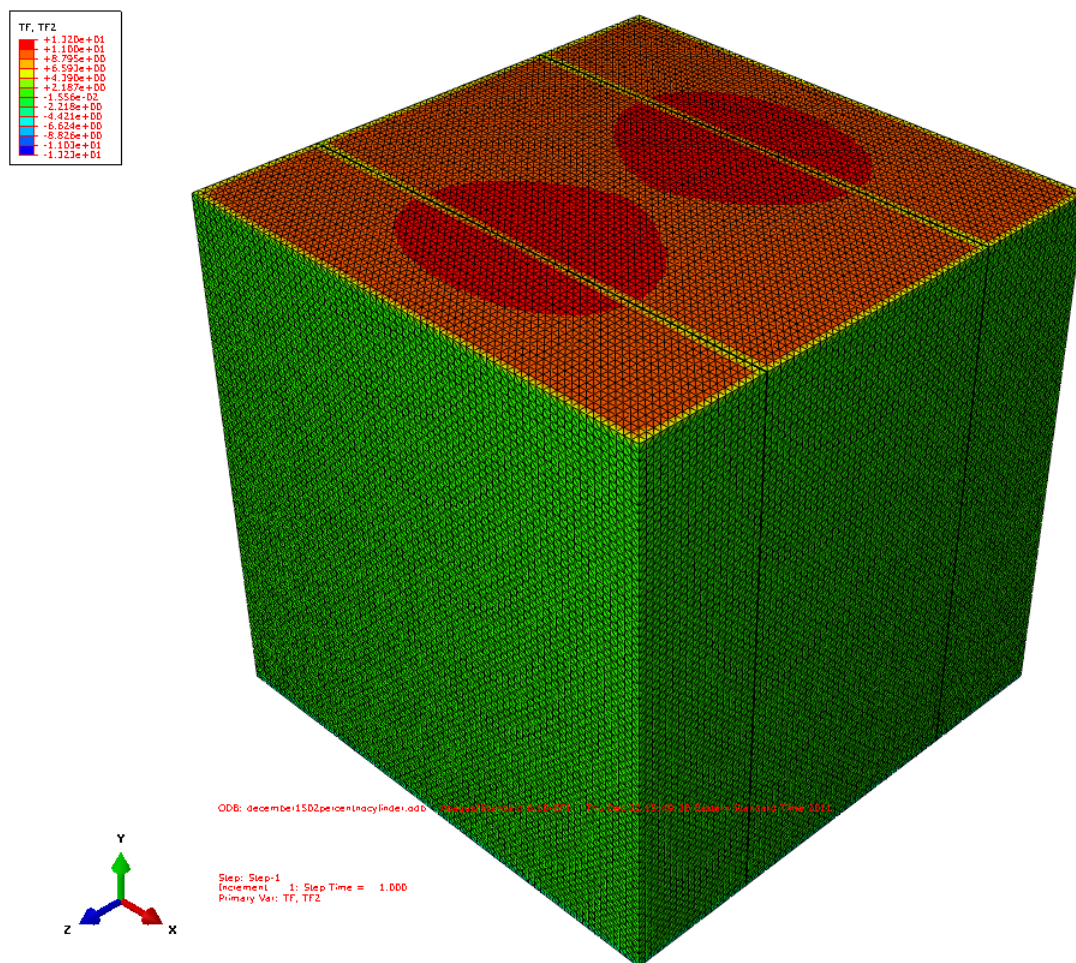


Figure 4.64 Total force values in y-direction at the nodal points of the top surface (TF_{yy})

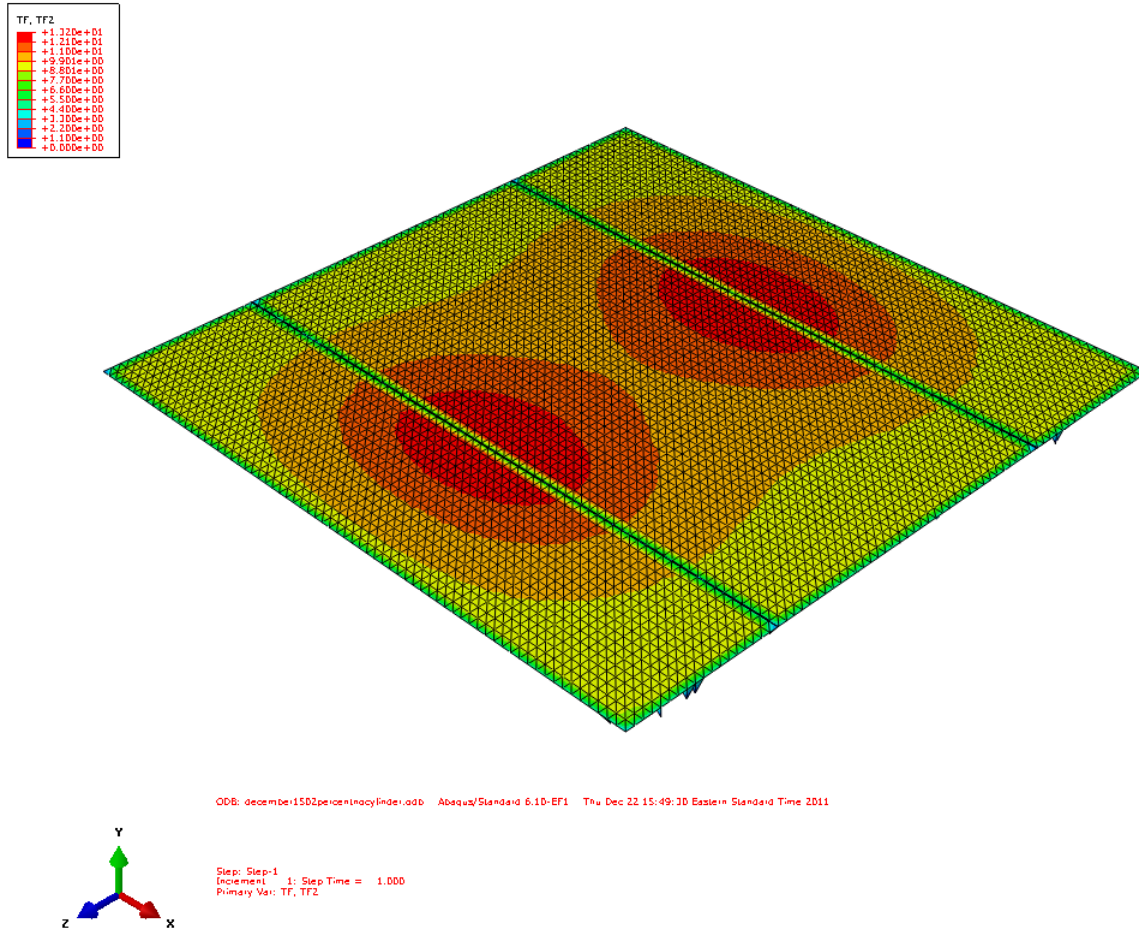


Figure 4.65 Total force values (TF_{yy}) at elements of the top surface

To determine the Young's modulus of the nanocomposite we use the total force values shown in Figure 4.65.

$$\text{Total force} = 52536.5 \text{ N}$$

$$\sigma_y = 32.8353 \text{ MPa}$$

$$\varepsilon_y = \frac{1}{40}$$

$$E = \frac{\sigma_y}{\varepsilon_y} = 1.313 \text{ GPa}$$

The Poisson's ratio is determined by using the ratio of the average strains ϵ_x and ϵ_y as:

$$\nu_{12} = -\frac{\epsilon_x}{\epsilon_y} = -\frac{(\Delta u)_{average}/40}{1/40}$$

$$\nu_{12} = 0.3845$$

The Young's modulus and the Poisson's ratio were similarly calculated using the finite element technique for the remaining percentages, namely 1%, 3%, 6% and 10% at -65°F, -4°F and RT.

The finite element results for Young's modulus and the Poisson's ratio are summarized in Tables 4.24 and 4.25 respectively and compared to those obtained from experiments and the Mori-Tanaka calculations for oriented particles.

Table 4.24 Comparison of E/E_m values obtained from experiments and the Mori-Tanaka calculations at various temperatures

Reinforcement Percentage of PP 3371	E/E_m			
	Temperature	Experimental	Mori-Tanaka Calculation (Oriented)	FEM
0.2%	-65°F (-54°C)	1.1083	1.0551	1.0489
	-4 °F (-20°C)	1.1095	1.0612	1.0539
	Room Temp.	1.0976	1.1148	1.0945
1%	-65°F (-54°C)	1.1909	1.1559	1.1390
	-4 °F (-20°C)	1.1977	1.1722	1.1521
	Room Temp.	1.2569	1.3048	1.2544
3%	-65°F (-54°C)	1.2146	1.2118	1.1757
	-4 °F (-20°C)	1.2487	1.2289	1.1877
	Room Temp.	1.3564	1.3458	1.2668
6%	-65°F (-54°C)	1.2867	1.2498	1.2066
	-4 °F (-20°C)	1.2587	1.2638	1.2162
	Room Temp.	1.3718	1.3443	1.2751
10%	-65°F (-54°C)	1.3174	1.4023	1.2452
	-4 °F (-20°C)	1.2948	1.4239	1.2533
	Room Temp.	1.4520	1.5455	1.3055

Table 4.25 Comparison of v_{12}/v_m values obtained from experiments and the Mori-Tanaka calculations at various temperatures v_{12}/v_m

Reinforcement Percentage of PP 3371	v_{12}/v_m			
	Temperature	Experimental	Mori-Tanaka Calculation (Oriented)	FEM
0.2%	-65°F (-54°C)	0.9574	1.0111	0.9798
	-4 °F (-20°C)	0.9618	1.0125	0.9869
	<i>Room Temp.</i>	1.0557	1.0317	0.9742
1%	-65°F (-54°C)	0.9311	1.0281	0.9813
	-4 °F (-20°C)	0.9179	1.0312	0.9805
	<i>Room Temp.</i>	1.0276	1.0681	0.9764
3%	-65°F (-54°C)	0.8970	1.0388	0.9725
	-4 °F (-20°C)	0.9057	1.0425	0.9724
	<i>Room Temp.</i>	0.9759	1.0813	0.9724
6%	-65°F (-54°C)	0.8863	1.0462	0.9956
	-4 °F (-20°C)	0.8888	1.0497	0.9974
	<i>Room Temp.</i>	0.9429	1.0815	1.0106
10%	-65°F (-54°C)	0.8699	1.0665	1.0382
	-4 °F (-20°C)	0.8516	1.0713	1.0418
	<i>Room Temp.</i>	0.9437	1.1140	1.0689

The results also are depicted in Figures 4.67-4.72.

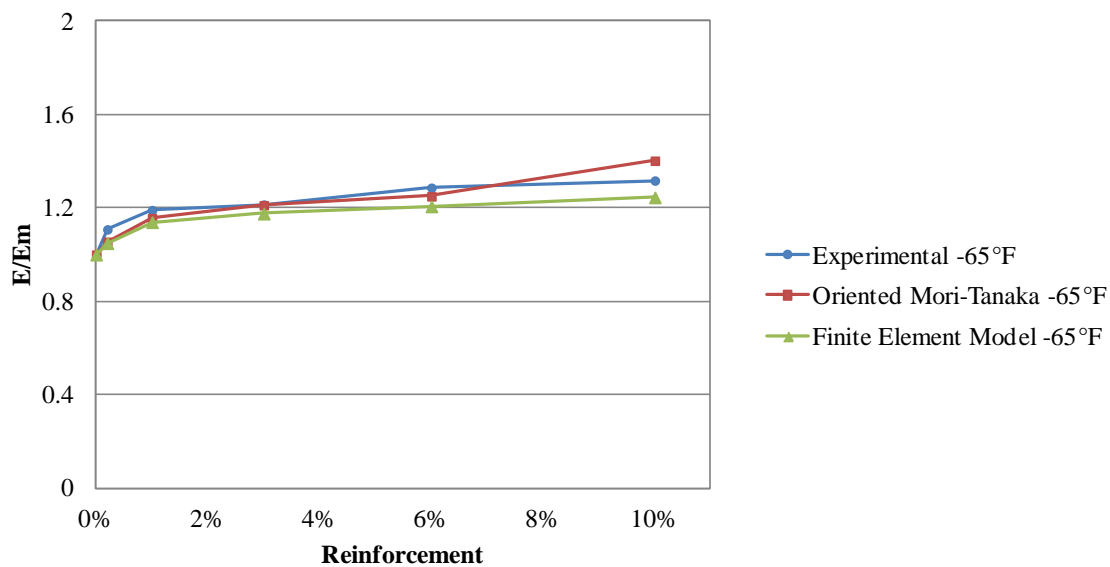


Figure 4.66 Comparison of E/E_m values obtained from the experimentally, Mori-Tanaka calculation and the finite element model at -65°F

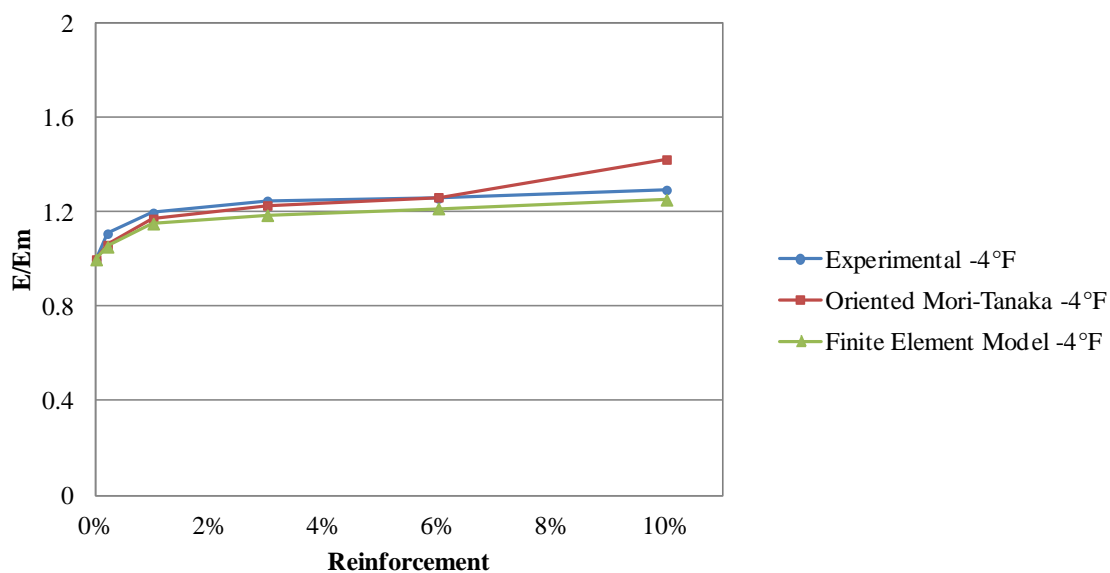


Figure 4.67 Comparison of E/E_m values obtained from the experimentally, Mori-Tanaka calculation and the finite element model at -4°F

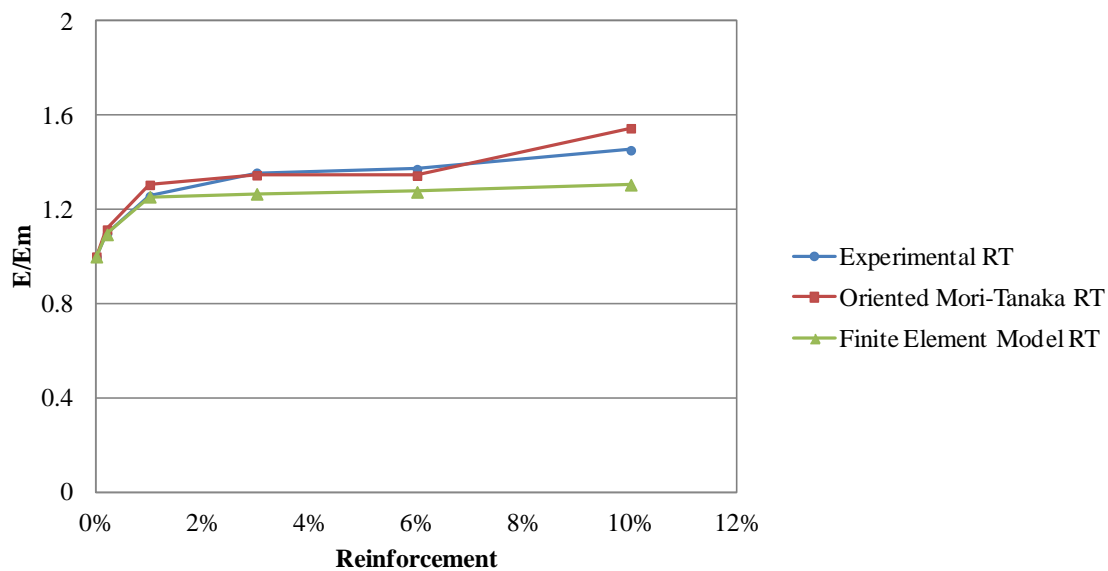


Figure 4.68 Comparison of E/E_m values obtained from the experimentally, Mori-Tanaka calculation and the finite element model at room temperature

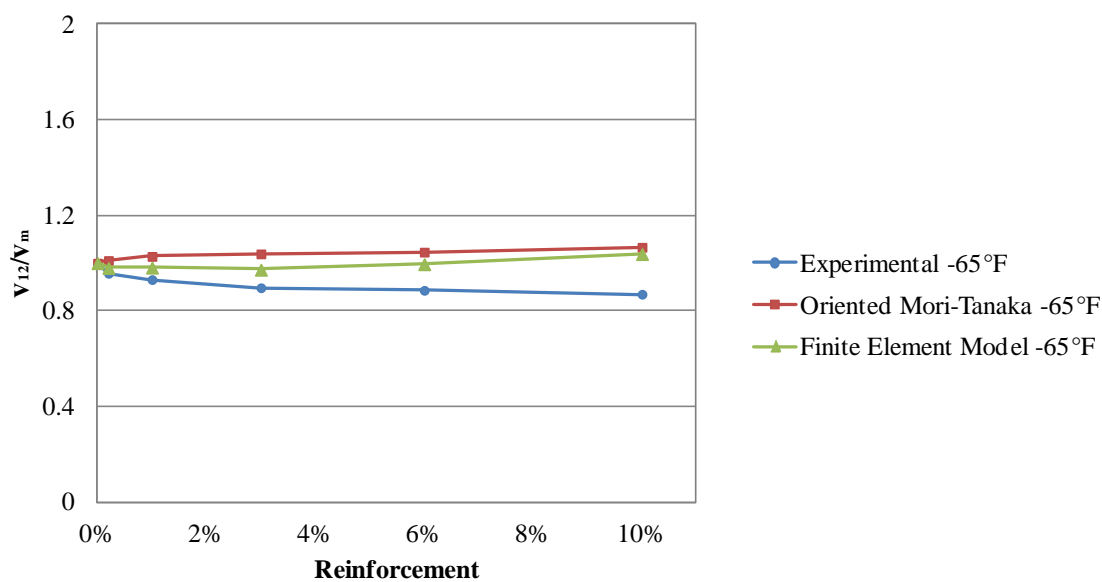


Figure 4.69 Comparison of V_{12}/V_m values obtained from the experimentally, Mori-Tanaka calculation and the finite element model at -65°F

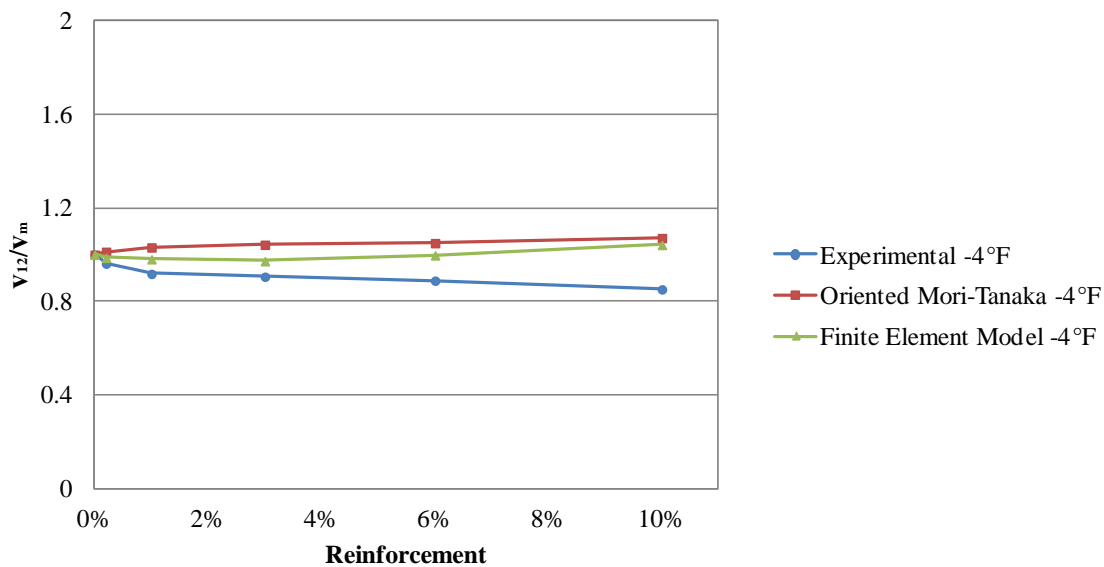


Figure 4.70 Comparison of V_{12}/V_m values obtained from the experimentally, Mori-Tanaka calculation and the finite element model at -4°F

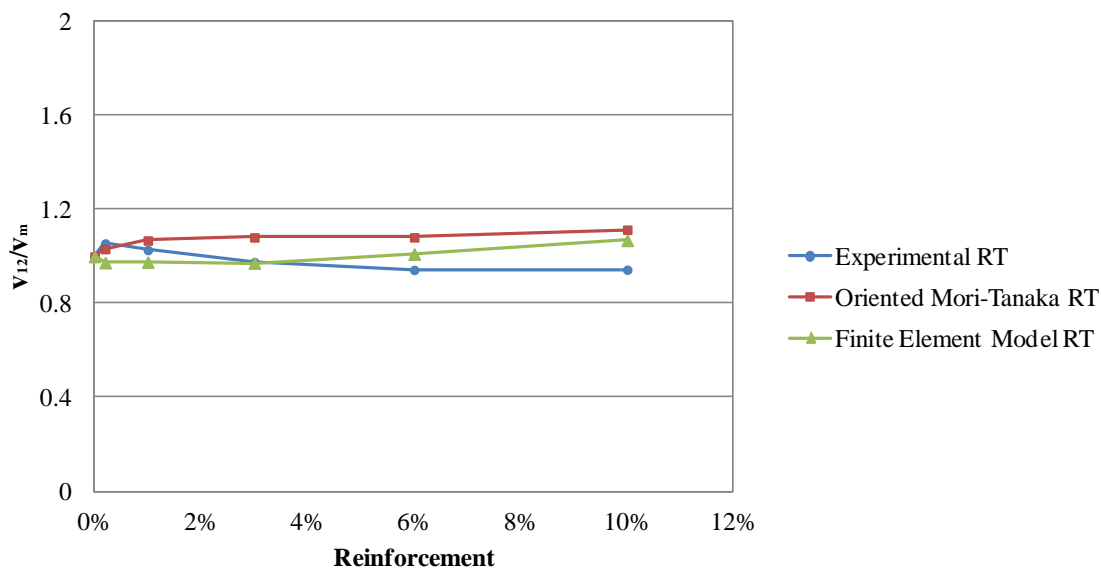


Figure 4.71 Comparison of V_{12}/V_m values obtained from the experimentally, Mori-Tanaka calculation and the finite element model at room temperature

The results given in Tables 4.24 and 4.25 and Figures 4.66-4.71 indicate that the finite element model may be a good predictive tool to determine the elastic properties of nanoclay reinforced polymers.

5. DISCUSSION AND CONCLUSIONS

An experimental and analytical study was conducted to study the mechanical properties of nanoclay reinforced polymer based composites. Also, a hierarchical multi-scale model to predict the properties of nanocomposites was constructed.

The experimental results were displayed as stress-strain curves at each reinforcement percentage and for various temperatures. Basic mechanical properties such as Young's modulus, Poisson's ratio and strength were calculated from the stress-strain curves and displayed graphically and tabular form. A careful analysis of the results indicates that both nanoclay reinforcement percentage and temperature have significant effect on the mechanical properties. For example an increase in nanoclay reinforcement improves both the stiffness and the strength of the material. The improvement in strength is especially more pronounced at lower percentages. Temperature has the reverse effect. At higher temperatures both the strength and stiffness of the material decrease, while at lower temperatures the material becomes stronger and stiffer. Also as at higher temperatures the material is highly ductile, but very brittle at lower temperatures. The plots of Young's modulus vs. temperature indicate that there is a drastic drop in the Young's modulus values as the temperature increases from -4°F to RT ($\sim 70^{\circ}\text{F}$). The reason for this drastic change is that PP has a glass transition temperature (GTT) of approximately 15°F and as the temperature is increased beyond the GTT, the material transitions from the glassy state resulting in the decrease observed in the value of the Young's modulus. Finally, it is also observed that the effect of temperature is more significant than that of nanoclay reinforcement.

The drop-weight test results were less revealing. It was observed that at RT as expected for 1% or higher reinforcement, the material became more brittle and most specimens were penetrated

upon impact. Temperature also had a significant effect on the impact behavior of the composite. At higher temperatures no specimen was penetrated even at 10% reinforcement. However, at lower temperatures all specimens were penetrated.

The epoxy based nanocomposite specimens had very different mechanical properties compared to PP based specimens. They had higher Young's modulus and very low failure strain which may be attributed to the brittle behavior of epoxy. It was also observed that the use of different PP resins resulted in significant change in the behavior of the nanocomposite.

Theoretical predictions using the Mori-Tanaka formulation and the finite element method (FEM) were obtained and compared with the experimental results. The predictions from the Mori-Tanaka calculations, in general, compared well with the experimental results, except at higher temperatures and higher reinforcement percentages. To improve the Mori-Tanaka calculations, the model was modified to include the effect of voids and temperature. The results obtained from the modified Mori-Tanaka calculations improved the match between the theoretical and experimental results.

For the finite element calculations, a representative volume element was constructed. Even though good match was obtained for low reinforcement percentages, the FE results were somewhat lower at higher percentages. This may be due to the approximate boundary conditions assumed in the FE calculations.

The following conclusions can be drawn:

- a. The experimental results indicate that for nanoclay reinforced PP specimens as the percentage of reinforcement increases the Young's modulus and the ultimate stress of the nanocomposite also increase.
- b. Temperature has a significant effect on the behavior of the nanocomposites. For PP based nanocomposites high temperature has a deleterious effect on the material properties. As the temperature increases, the Young's modulus and the ultimate tensile stress of the nanocomposites decrease. However, as the temperature decreases, the Young's modulus and ultimate tensile stress increase. At low temperatures the material becomes brittle and fails at relatively low strains.
- c. The type of resin used has significant effect on the properties of the nanocomposite.
- d. The results of drop weight impact tests indicate that the addition of nanoclay to PP made the specimens more brittle and most were penetrated at 1% or higher percentages at room temperature. At lower temperatures all specimens were penetrated. However, at higher temperatures no penetration was observed.
- e. With a proper choice of geometric and material properties for the constituents, the Mori-Tanaka formulation may be a good tool in predicting the Young's modulus. The Mori-Tanaka results, especially those obtained for oriented particles, matched well with the experiments, except at high temperatures and high reinforcement percentages.
- f. Inclusion of voids in the Mori-Tanaka formulation gave a somewhat better match with experimental data at higher reinforcement percentages.

- g. The effect of temperature was included by using the secant instead of the tangent calculation for the Young's modulus. As a consequence, better match with experiments was obtained at higher temperatures.
- h. The 3-D finite element model also provides a good estimate for the Young's modulus of the nanocomposite. As the results displayed in the figures show, the finite element results compare well with those obtained experimentally.

6. FUTURE WORK

The research outlined in this thesis opens new research venues on nanoclay reinforced polymers. The following future topics are suggested by the author:

- a. Detailed electron microscopy study of the nanocomposite to determine the dispersion patterns of nanoclay in the matrix. This will enable the research to construct better predictive models.
- b. Study of the effect of clay particles on fire retardation and toxicity.
- c. Addition of nanoclay in fabric or fiber reinforced polymers to design better armor materials.
- d. Develop a finite element model for randomly distributed particles.

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